

## Article

# Fatigue Damage Evaluation of Short Carbon Fiber Reinforced Plastics Based on Thermoelastic Temperature Change and Second Harmonic Components of Thermal Signal

Daiki Shiozawa <sup>1,\*</sup>, Takahide Sakagami <sup>1</sup>, Yu Nakamura <sup>1</sup>, Takato Tamashiro <sup>1</sup>, Shinichi Nonaka <sup>2</sup>, Kenichi Hamada <sup>2</sup> and Tomoaki Shinchi <sup>2</sup>

- <sup>1</sup> Department of Mechanical Engineering, Kobe University, Kobe 657-8501, Japan; sakagami@mech.kobe-u.ac.jp (T.S.); 169t348t@stu.kobe-u.ac.jp (Y.N.); 140t348t@stu.kobe-u.ac.jp (T.T.)
- <sup>2</sup> DIC Corporation, Tokyo 103-8233, Japan; shinichi-nonaka@mb.dic.co.jp (S.N.); kenichi-hamada@mb.dic.co.jp (K.H.); tomoaki-shinchi@mb.dic.co.jp (T.S.)
- \* Correspondence: shiozawa@mech.kobe-u.ac.jp; Tel.: +81-78-803-6303

**Abstract:** Short fiber reinforced plastics (SFRPs) have excellent moldability and productivity compared to continuous fiber composites. In this study, thermoelastic stress analysis (TSA) was applied to detect delamination defects in short carbon fiber reinforced plastics (SCFRPs). The thermoelastic temperature change  $\Delta T_E$ , phase of thermal signal  $\theta_E$ , and second harmonic temperature component  $\Delta T_D$  were measured. In the fatigue test of SCFRP, it was confirmed that changes in  $\Delta T_E$ ,  $\theta_E$ , and  $\Delta T_D$  appeared in the damaged regions. A staircase-like stress level test for a SCFRP specimen was conducted to investigate the generation mechanism of the  $\Delta T_D$ . The distortion of the temperature change appeared at the maximum tension stress of the sinusoidal load—and when the stress level decreased, the temperature change returned to the original sinusoidal waveform.  $\Delta T_D$  changed according to the change in the maximum stress during the staircase-like stress level test, and a large value of  $\Delta T_D$  was observed in the final ruptured region. A distortion of the temperature change and  $\Delta T_D$  was considered to be caused by the change in stress sharing condition between the fiber and resin due to delamination damage. Therefore,  $\Delta T_D$  can be applied to the detection of delamination defects and the evaluation of damage propagation.

**Keywords:** nondestructive evaluation; thermoelastic stress analysis; phase analysis; infrared camera; short carbon fiber reinforced plastics



**Citation:** Shiozawa, D.; Sakagami, T.; Nakamura, Y.; Tamashiro, T.; Nonaka, S.; Hamada, K.; Shinchi, T. Fatigue Damage Evaluation of Short Carbon Fiber Reinforced Plastics Based on Thermoelastic Temperature Change and Second Harmonic Components of Thermal Signal. *Materials* **2021**, *14*, 4941. <https://doi.org/10.3390/ma14174941>

Academic Editors: Davide Palumbo and Rosa De Finis

Received: 12 July 2021

Accepted: 23 August 2021

Published: 30 August 2021

**Publisher's Note:** MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



**Copyright:** © 2021 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

## 1. Introduction

Short fiber reinforced plastics (SFRPs) have received much attention due to their excellent moldability and productivity, as compared to continuous fiber composites. The mechanical properties of an SFRP depend on the fiber length and orientation distribution. A comprehensive literature review of SFRPs was published by Mortazavian and Fatemi [1]. The effects of loading conditions, microstructure, and environmental factors on the lifetime of SFRPs were investigated. Belmonte et al. [2,3] presented the influence of the fiber volume fraction on the damage mechanism in a short glass fiber. An effective nondestructive technique is required to detect and measure the various types of damage during fatigue fracture. Fragoudakis [4] discussed the effect of the fiber orientation around geometric discontinuities in glass fiber reinforced plastic (GFRP) and presented an important topic for the manufacturing and design against failure of GFRP laminated structures. Nondestructive evaluation techniques using infrared thermography (thermographic NDT) have been effectively employed for the detection of delamination in FRPs. The thermographic NDT technique based on the thermal insulation effect of the delamination defect was applied to the NDT of composite materials. Avdelidis et al. [5,6] reviewed NDT techniques using a transient temperature distribution for CFRP. Chatterjee et al. [7] compared the defect

detection performance of transient thermography, pulse lock-in, and frequency-modulated thermography. Maldague et al. [8] developed pulse-phase infrared thermography for composite materials.

Thermoelastic stress analysis (TSA) is a well-known, non-contact, full-field technique that provides stress maps of a component subjected to dynamic loading [9–13]. Thermoelasticity for orthotropic materials has been studied by many researchers, and the TSA technique has been employed as a powerful tool for evaluating the impact or fatigue damage in composite materials and structures [14–18]. Krstulovic-Opara et al. [19] applied a combination of TSA and pulse heating thermography to detect damage, including fiber fractures and delamination. Emery and Dulieu-Barton [20] applied TSA to fatigue damage evaluation in laminated glass fiber epoxy materials, and demonstrated that TSA could analyze complicated fatigue damages, such as fiber breakage, matrix cracking, and delamination. Uenoya and Fujii [21] developed a thermoelastic damage analysis (TDA) for early damage detection in plain-woven CFRP.

A thermal signal analysis approach was developed to assess the different harmonic components of the thermal signal related to both the thermoelastic and dissipated heat sources. Palumbo et al. [22] showed that the amplitude of the second harmonics of the thermal signal of GFRP laminates was representative of the intrinsic dissipations, and it can be used to detect and monitor the damage. De Finis et al. [23,24] showed that the second harmonic of the thermal signal and damage state parameter of quasi-isotropic CFRP exhibited a linear relationship with a good correlation, and the second harmonic of the thermal signal could be used for real-time damage monitoring.

The author reported that a delamination defect affected the phase shift of the thermoelastic temperature change on the surface of SCFRPs, and that the behavior of the phase shift differs depending on the orientation state of the fiber [25]. This phenomenon is considered to be caused by the difference in the thermoelastic modulus of the fiber and resin. The thermoelastic constant  $k$  of the carbon fiber has a negative value. Therefore, the phase shift of the thermoelastic temperature change occurs, corresponding to the stress sharing condition between the fiber and resin due to the delamination defect. Damage evaluation can also be assessed using the thermoelastic signals and phase shifts of thermal signals on SCFRPs. In this study, TSA was applied to detect fatigue damage in SCFRPs. The effectiveness of the parameters obtained by TSA, such as the thermoelastic temperature change  $\Delta T_E$ , phase of thermal signal  $\theta_E$ , and second harmonic temperature component  $\Delta T_D$ , were investigated. A staircase-like stress level test was performed to investigate the effect of the magnitude of the cyclic loading on the evaluation of the fatigue damage based on these parameters.

## 2. Thermoelastic Stress Analysis Using Infrared Thermography

Dynamic stress changes cause a very small temperature change under adiabatic conditions in a solid. This phenomenon is known as the thermoelastic effect, and is described by Lord Kelvin's equation, which relates the temperature change ( $\Delta T_E$ ) to the changes in the sum of the principal stresses ( $\Delta\sigma$ ) under cyclic variable loading as follows:

$$\Delta T_E = \frac{\alpha}{\rho C_p} T \Delta\sigma = -k T \Delta\sigma \quad (1)$$

Here,  $\alpha$  is the coefficient of thermal expansion,  $\rho$  is the mass density,  $C_p$  is the specific heat at constant pressure, and  $T$  is the absolute temperature. The coefficient  $k$  is called the thermoelastic constant. The change in the sum of the principal stresses ( $\Delta\sigma$ ) was obtained by measuring the temperature change ( $\Delta T_E$ ) using infrared thermography [9–13].

Because the thermoelastic temperature change is very small and sometimes hidden by the thermal noise of the infrared camera, lock-in infrared thermography using reference signals synchronized with the stress changes is commonly employed to improve the accuracy of stress measurements [26]. The TSA technique uses a lock-in algorithm with

a reference-loading signal extracted from the load cell or strain gauge to improve the signal-to-noise ratio.

Because the magnitude of the thermoelastic temperature change was minor, the lock-in thermography method was applied. In this method, a high temperature resolution is realized by performing correlation signal processing with a reference signal, such as a load signal. First, the input reference signal generates the digital data  $\sin(t)$  and  $\cos(t)$  of sine and cosine waves with the same frequency by post-processing at the PC. The correlation coefficient was calculated between the measured temperature data  $T(t)$  and the reference signals  $\sin(t)$  and  $\cos(t)$ , respectively, as follows:

$$\Delta T_{E,\sin} = \frac{2}{N} \sum_{t=1}^N T(t) \sin\left(2\pi t \frac{f_{\text{load}}}{f_{\text{meas.}}}\right) \quad (2)$$

$$\Delta T_{E,\cos} = \frac{2}{N} \sum_{t=1}^N T(t) \cos\left(2\pi t \frac{f_{\text{load}}}{f_{\text{meas.}}}\right) \quad (3)$$

Here,  $f_{\text{mean}}$  is the sampling frequency,  $f_{\text{load}}$  is the loading frequency,  $\Delta T_{E,\sin}$  and  $\Delta T_{E,\cos}$  are the amplitudes of the signal synchronized with the reference signal, respectively, and the amplitude of the signal is synchronized with the opposite phase of the reference signal. In addition, the measurement signal was integrated and averaged to obtain a high-precision amplitude. Because the  $\Delta T_{E,\cos}$  component appears when the thermoelastic temperature change deviates from the phase of the load signal, the absolute value  $\Delta T_E$  and the phase delay  $\theta_E$  of the amplitude of the temperature change can be obtained using the following equation:

$$\Delta T_E = \sqrt{\Delta T_{E,\sin}^2 + \Delta T_{E,\cos}^2} \quad (4)$$

$$\theta_E = \tan^{-1}\left(-\frac{\Delta T_{E,\cos}}{\Delta T_{E,\sin}}\right) \quad (5)$$

The second harmonic component of the thermal signal is calculated using the following equation:  $\Delta T_{D,\sin}$  and  $\Delta T_{D,\cos}$  are obtained as the amplitude of the thermal signal  $T(t)$  synchronized with  $\sin(2t)$  and  $\cos(2t)$  with a frequency which is double that of the reference signal [26].

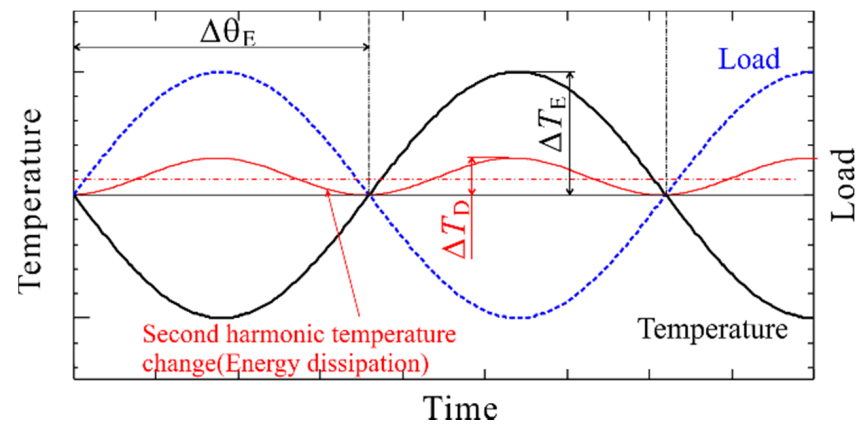
$$\Delta T_{D,\sin} = \frac{2}{N} \sum_{t=1}^N T(t) \sin\left(4\pi t \frac{f_{\text{load}}}{f_{\text{meas.}}}\right) \quad (6)$$

$$\Delta T_{D,\cos} = \frac{2}{N} \sum_{t=1}^N T(t) \cos\left(4\pi t \frac{f_{\text{load}}}{f_{\text{meas.}}}\right) \quad (7)$$

In metallic materials, the temperature rise due to irreversible energy dissipation occurs at the maximum tensile stress and at the maximum compressive stress during one sinusoidal loading. Therefore, the temperature change due to energy dissipation can be obtained as a component having double the frequency of the load signal. In this study, the magnitude of the second harmonic temperature component  $T_D$  is defined as the temperature change range due to the energy dissipation, and is calculated as follows:

$$T_D = 2\sqrt{\Delta T_{D,\sin}^2 + \Delta T_{D,\cos}^2} \quad (8)$$

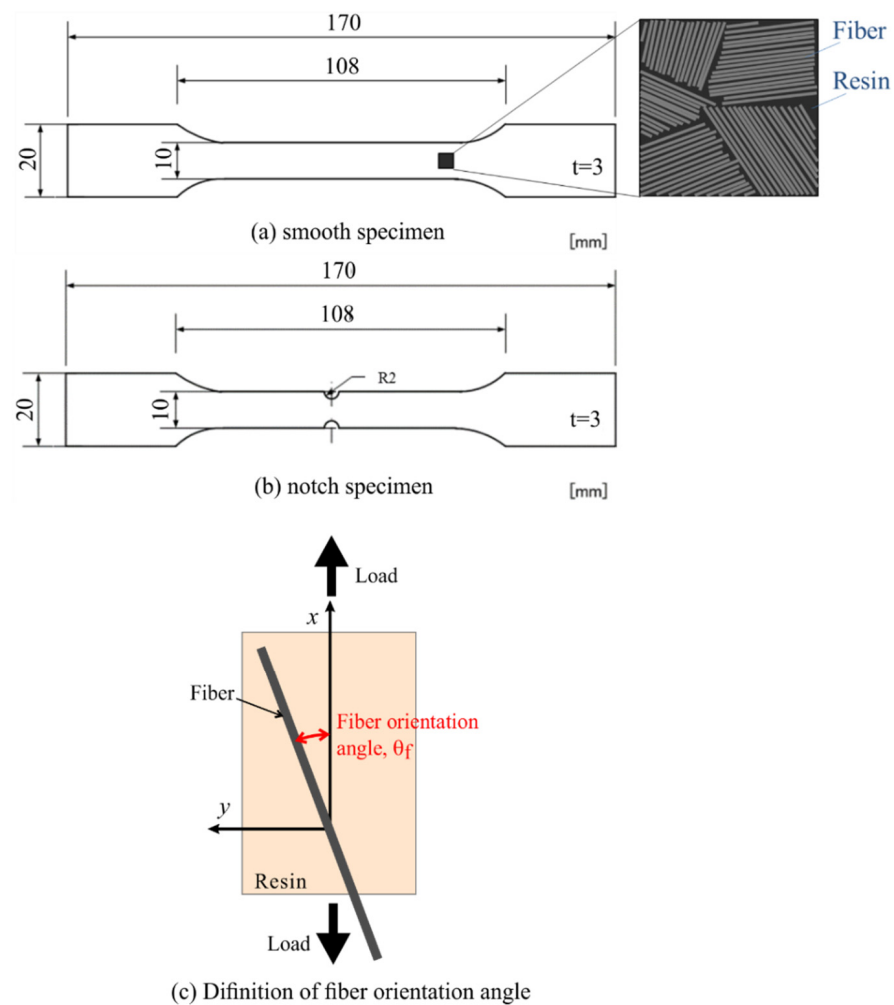
A schematic illustration of the thermoelastic temperature change is shown in Figure 1. The thermoelastic temperature change in the material with a positive value of  $k$  shows an opposite phase waveform against the loading waveform. The phase difference was defined as the difference in phase between the thermoelastic temperature change and the loading signal, as shown in Figure 1. For a material with a positive  $k$  value, the phase difference  $\Delta\theta_E$  is  $180^\circ$ .



**Figure 1.** Schematic illustration of thermoelastic temperature change and phase of thermal signal.

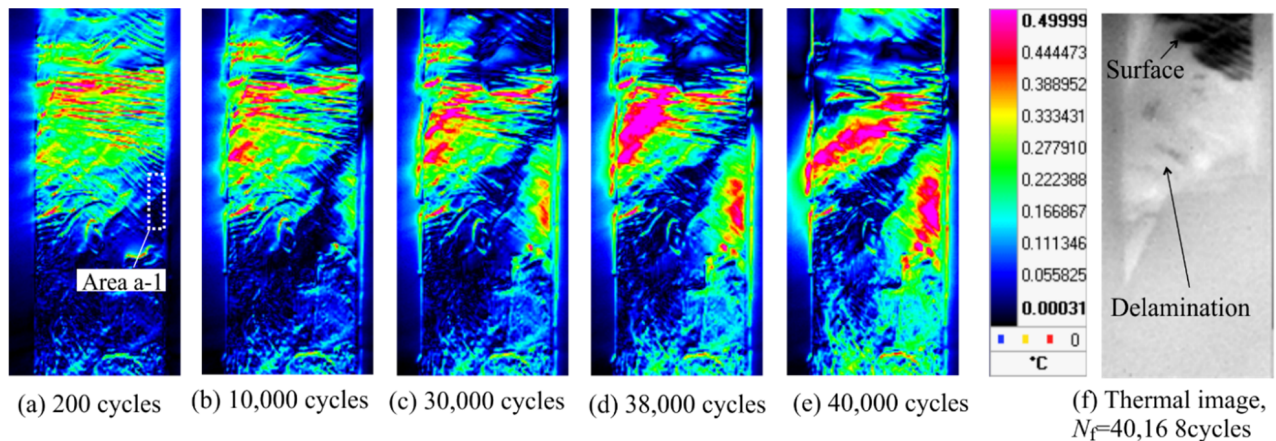
### 3. Experimental Setup

The configurations of the CFRP specimens employed in this study are shown in Figure 2. The specimens were cut from laminated short fiber CFRP sheets with vinyl ester resin and 25.4 mm long carbon fiber bundles. Each bundle was composed of 12,000 short carbon fibers. The mass contents (wt %) of the resin and fiber were 67 and 33, respectively. The specimen had circular notches with a radius of 2 mm.



**Figure 2.** Configurations of the employed short carbon fiber reinforced plastic specimen.

The fiber orientation angle  $\varphi_f$  is defined as shown in Figure 3.  $\varphi_f$  is equal to  $0^\circ$  when the fiber bundle is oriented parallel to the loading axis. The distribution of the fiber orientation angle  $\varphi_f$  was measured after the TSA measurement. The thin surface layer of the resin was removed by polishing to expose the carbon fiber bundles, and an optical image of the surface carbon fiber was captured using a digital camera. The fiber orientation angle  $\theta_f$  was determined using an image processing program developed by Enomae [27].



**Figure 3.** Change in the distribution of thermoelastic temperature change.

Cyclic-axis sinusoidal waveform loading with a frequency  $f$  of 7 Hz and a stress ratio of  $R = 0.1$  was applied to the specimen by an electrohydraulic fatigue testing machine. When the loading frequency is small, thermal diffusion occurs. So accurate thermoelastic stress analysis (TSA) cannot be performed. It was found from preliminary experimental result that the effect of heat diffusion was large at  $f = 1$  Hz, and a loading frequency of 3 Hz or higher was desirable. On the other hand, at  $f = 9$  Hz and above, second harmonic thermal component caused by the fatigue testing machine was observed. Therefore, the load frequency was set to 7 Hz. Microscopic visible images of the specimen surface and side surface were obtained using an optical microscope. The temperature change on the specimen surface was measured by infrared thermography with an MCT array detector (FLIR Systems Inc., Croissy Beaubourg, France (Wilsonville, OR, USA), SC7500). The specifications and settings of the infrared camera are listed in Table 1.

**Table 1.** Specifications and setting of employed infrared camera.

Infrared Detector	MCT
Detectable wavelength	7.7–9.3 $\mu\text{m}$
Number of detectors	320 $\times$ 256
Temperature resolution (NETD)	25 mK
Framing rate	373 Hz
Time of data acquisition	10 s

## 4. Experimental Results

### 4.1. Effect of Damage Generation on Thermoelastic Stress, Phase and Second Harmonic Components

Stress-amplitude constant fatigue tests at  $\sigma_{\max} = 100$  MPa,  $R = 0.1$  were conducted on smooth specimens (Specimen A). The specimens were broken at  $N_f = 40,168$  cycles. Figure 3 shows the distribution of the thermoelastic temperature change immediately after the start of the fatigue test and immediately before the fracture, and the infrared image after the rupture, as shown in Figure 3f. In Figure 3f, the white area in the infrared images indicates where the final rupture occurred and the fracture surface appeared. It can be observed from Figure 3 that the area where a high  $\Delta T_E$  appears coincides with the fracture area.

Figure 4 shows the changes in  $\Delta T_E$  and  $\Delta \theta_E$  in the rectangular area in Figure 3 (Area a-1), where changes in  $\Delta T_E$  were observed.  $\Delta T_E$  and  $\Delta \theta_E$  in Figure 4 indicate the average value in the rectangle (range of 5 pixels  $\times$  15 pixels). It was found that  $\Delta \theta_E$  changed from 180° to 0° at  $N = 1500$  cycles, and  $\Delta T_E$  increased after the change of  $\Delta \theta_E$  with an increase in the number of cycles. It is considered that the thermoelastic temperature change in the fiber was mainly measured because the fiber orientation angle  $\theta_f$  in this area was approximately 0°, and the fiber tended to share stress. Further, and the stress sharing condition between fibers and resin changed owing to the delamination.

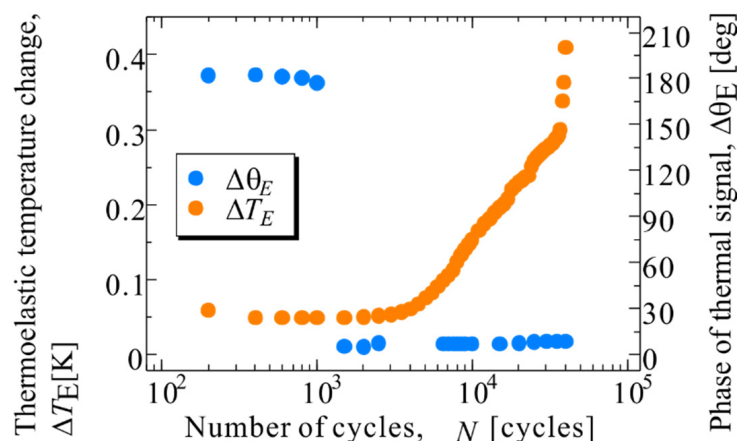


Figure 4. Change in the thermoelastic temperature change and phase of thermal signal in Area a-1.

Figure 5 shows the result of analyzing the second harmonic temperature component  $\Delta T_D$  for the same specimen shown in Figure 3. It was found that the area showing a high  $\Delta T_D$  expands with an increase in the number of cycles. The areas showing a high  $\Delta T_D$  coincided with the areas where fracture occurred. The load signal and thermal signal at the point showing the high  $\Delta T_D$  surrounded by white lines (Area a-2) are shown in Figure 6. As shown in Figure 6, the temperature change is a sinusoidal wave with the opposite phase to the load signal at  $N = 38,000$  cycles; whereas the distortion of temperature change, which shows a constant value without a temperature drop, is observed at the maximum tensile load at  $N = 40,000$  cycles. Delamination occurred in this evaluation area. Therefore, it is considered that this distortion in the temperature signal is related to the initiation and progression of delamination.

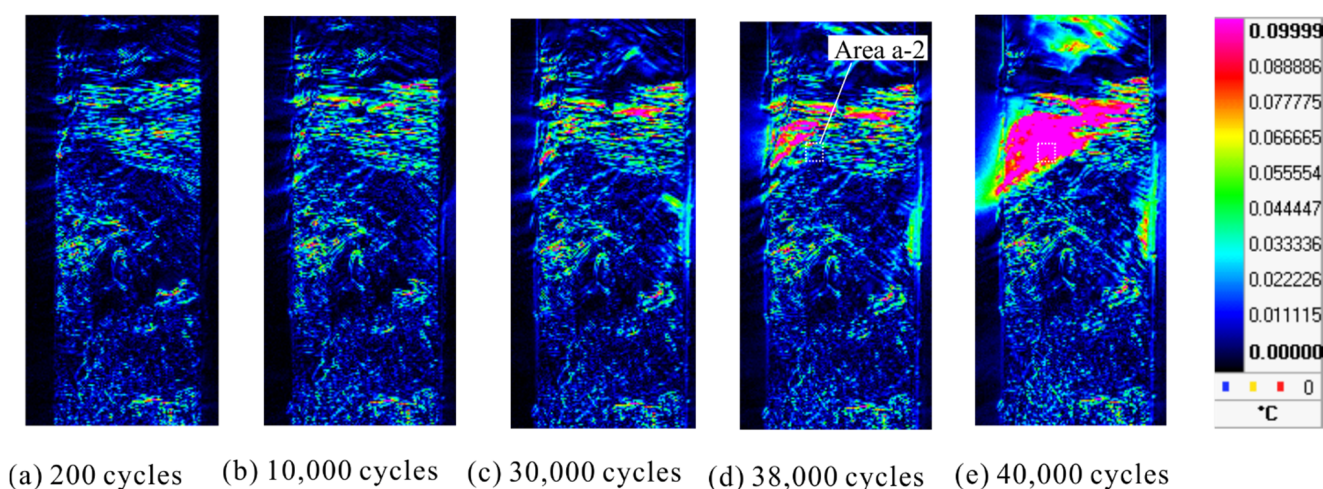


Figure 5. Change in the second harmonic temperature components.

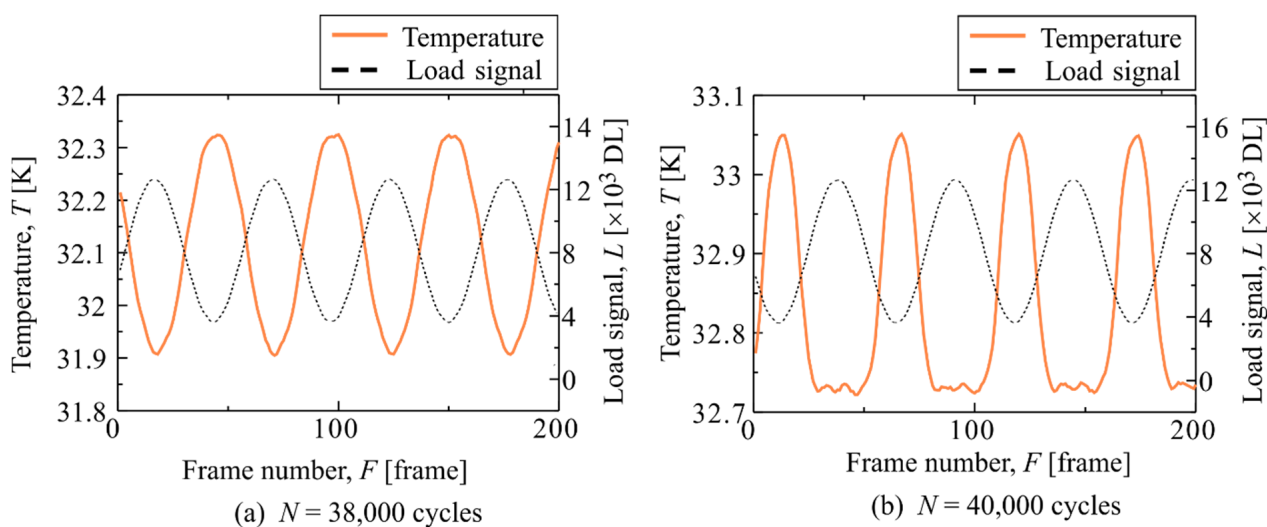


Figure 6. Temperature change and load signal in Area a-2.

The constant stress amplitude fatigue test for another smooth specimen (Specimen B) was conducted at  $\sigma_{\max} = 180$  MPa and  $R = 0.1$ . This specimen was broken at  $N_f = 4.42 \times 10^4$  cycles. The distribution of the second-harmonic temperature component is shown in Figure 7. It was found that some regions with high  $\Delta T_D$  appeared locally. Figure 8 shows the load and temperature signals at two points (Area b-1 and b-2) in Figure 7. In these regions, the load and temperature signals at  $N = 200$  cycles are in phase, and the temperature change in Area b-1 is slightly distorted at the maximum tensile load, compared with that at the minimum tensile load. From Figure 8b, the temperature waveform at  $N = 4000$  cycles (life ratio  $N/N_f = 0.1$ ) is further distorted than that at  $N = 200$  cycles, and no temperature rise occurs above a certain stress magnitude. It was clarified that the distortion of temperature change appears at the maximum tensile load in both cases where the temperature signal is in the opposite phase and in phase with respect to the load signal. The change in the second-harmonic temperature component reflects the distortion of temperature change. It has been reported that the second harmonic component of the thermal signal is caused by the plastic deformation in metallic materials. The distortion of the temperature change and second harmonic temperature component in SCFRPs may be caused by the change in load sharing conditions between the fiber and resin due to delamination damage, the energy dissipation due to the plastic deformation in the resin, and/or the viscoelasticity of resin.

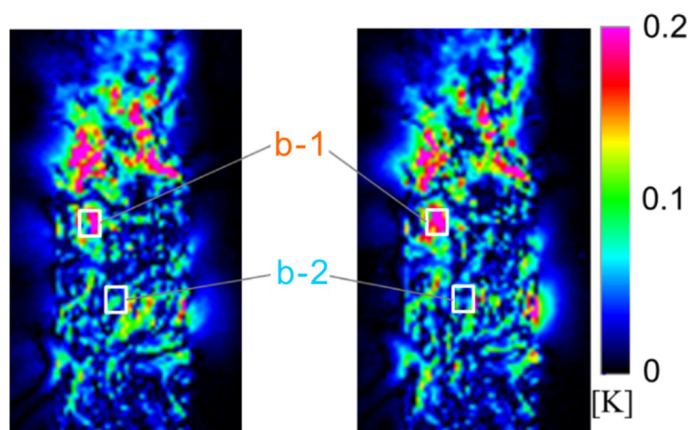
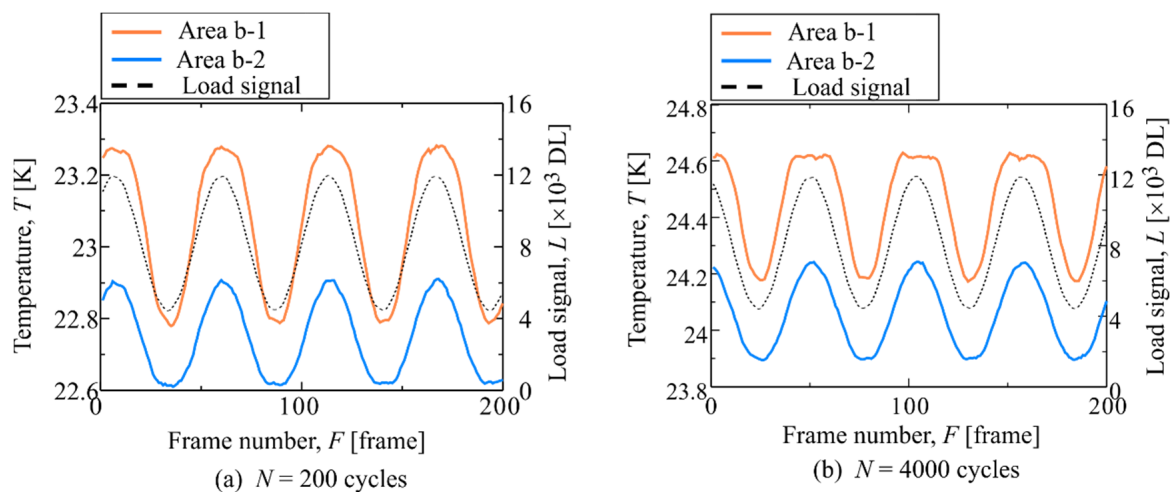


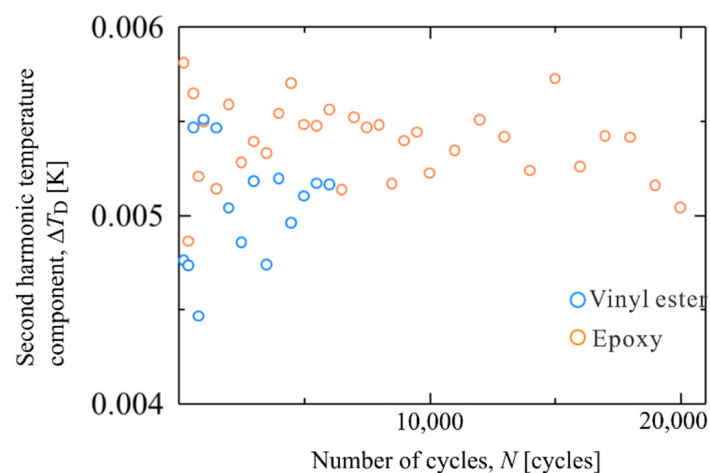
Figure 7. Distribution of second harmonic temperature component for Specimen B.



**Figure 8.** Thermal and load signal at Area b-1 and Area b-2, measured at (a)  $N = 200$  cycles, (b)  $N = 4000$  cycles.

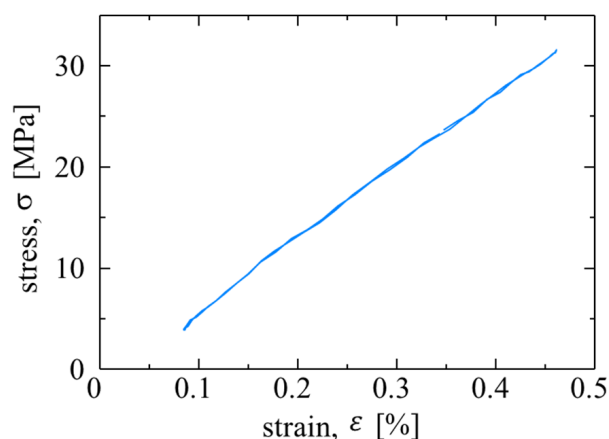
#### 4.2. Second Harmonic Temperature Components in Resin Specimens

The constant stress amplitude fatigue tests were conducted resin specimens to investigate the second-harmonic temperature component on the resin. The loading conditions were as follows: maximum stress  $\sigma_{\max} = 32$  MPa, stress ratio  $R = 0.1$ , the repetitive load frequency  $f = 7$  Hz, and the stress waveform was sinusoidal. Fatigue tests were conducted on epoxy and vinyl ester specimens. The shape of specimen is smooth specimen. A strain gauge was set on the surface, opposite to the infrared measurement. The epoxy specimens were broken at  $N_f = 2.06 \times 10^4$  cycles, and the vinyl ester specimens were broken at  $N_f = 6.33 \times 10^3$  cycles. Figure 9 shows the change in the second-harmonic temperature component  $\Delta T_D$  for the epoxy and vinyl ester specimens. As shown in Figure 9, no change in  $\Delta T_D$  was observed during the fatigue test. Figure 10 shows the hysteresis loop of the strain and stress for the vinyl ester specimen. In this stress amplitude, the stress and strain exhibit linear behavior, and the viscoelastic deformation is very small. Because the value of  $\Delta T_D$  for epoxy and vinyl ester specimens is smaller than that for SCFRPs, it is considered that the  $\Delta T_D$  observed for SCFRP specimens is not due to energy dissipation from viscoelasticity and plastic deformation of the resin.



**Figure 9.** Second harmonic temperature component for epoxy and vinyl ester specimen.

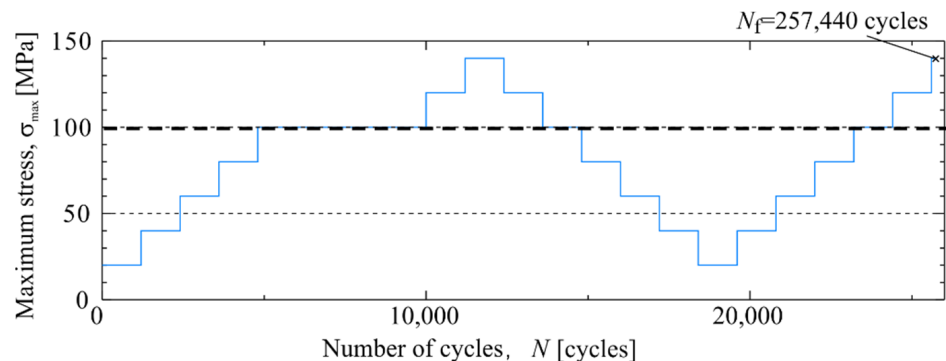




**Figure 10.** Hysteresis loop of strain and stress for vinyl ester specimen.

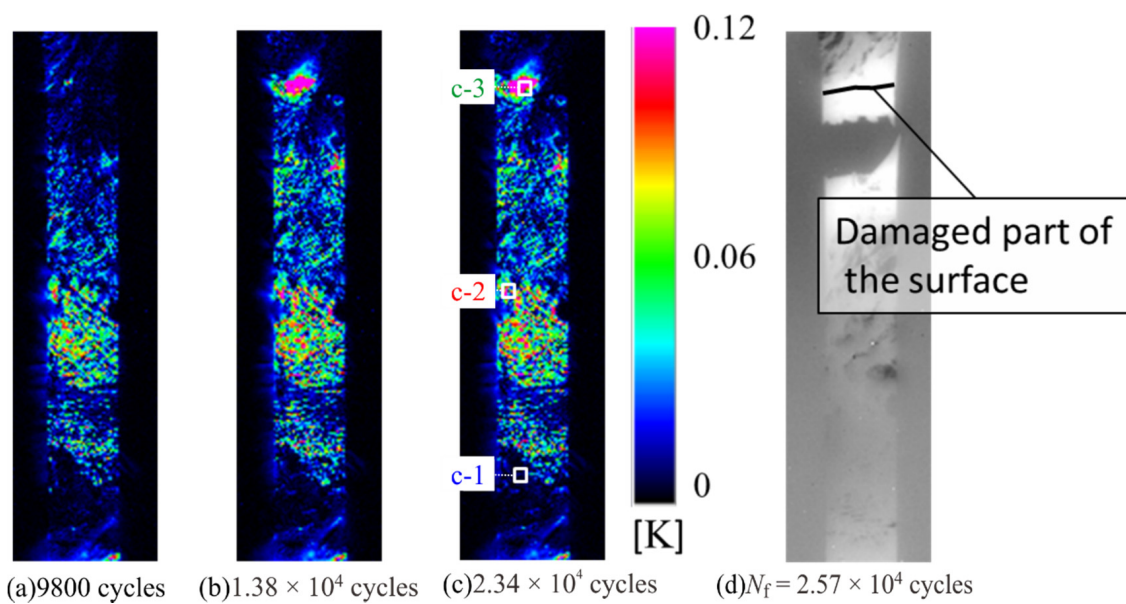
#### 4.3. Relation between the Second Harmonic Temperature Components and Maximum Loading Stress in SCFRP

The distortion of the temperature change occurred at the maximum tension load during cyclic loading. Then, the relationship between the second harmonic temperature component and the magnitude of the maximum stress was investigated. A SCFRP notched specimen (Specimen C) was subjected to staircase-like stress level tests. Figure 11 shows a schematic diagram of the change in the applied maximum stress during the staircase-like stress level test. The specimens were broken at  $N_f = 2.57 \times 10^4$  cycles. The loading conditions applied were a stress ratio  $R = 0.1$ , and a load frequency  $f = 7$  Hz.



**Figure 11.** Change of maximum stress applied to the SCFRP specimen during the staircase-like stress level tests.

Figure 12 shows the results of the second harmonic components of the thermal signal and the infrared images at the breaking point, and when loaded with  $\sigma_{\max} = 100$  MPa. As shown in Figure 12, the magnitude of  $\Delta T_D$  increased with the number of cycles, and the location where a particularly large increase in  $\Delta T_D$  was observed coincides with the location where the fracture occurred. The temperature change and the second harmonic temperature component were investigated for each of the following three evaluation areas: Area c-1: no change in  $\Delta T_D$  appeared; Area c-2: the high value of  $\Delta T_D$  was measured from the beginning but did not change after the test; Area c-3: the location where the fracture occurred and a high value of  $\Delta T_D$  was measured during the test.



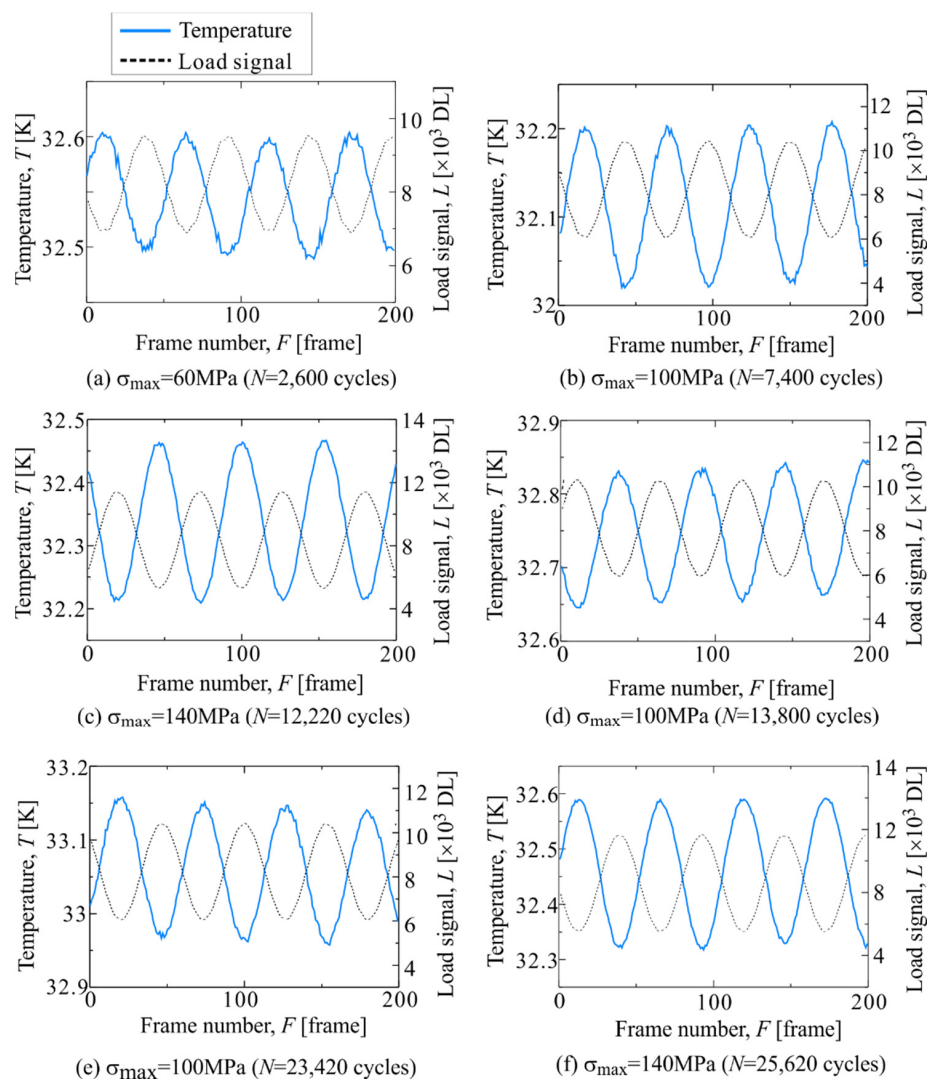
**Figure 12.** Second harmonic temperature component measured at different timings with  $\sigma_{\max} = 100$  MPa, measured at (a)  $N = 9800$  cycles, (b)  $N = 1.38 \times 10^4$  cycles, (c)  $2.34 \times 10^4$  cycles, and (d) infrared image at  $N_f = 2.57 \times 10^4$  cycles.

#### 4.3.1. Area c-1

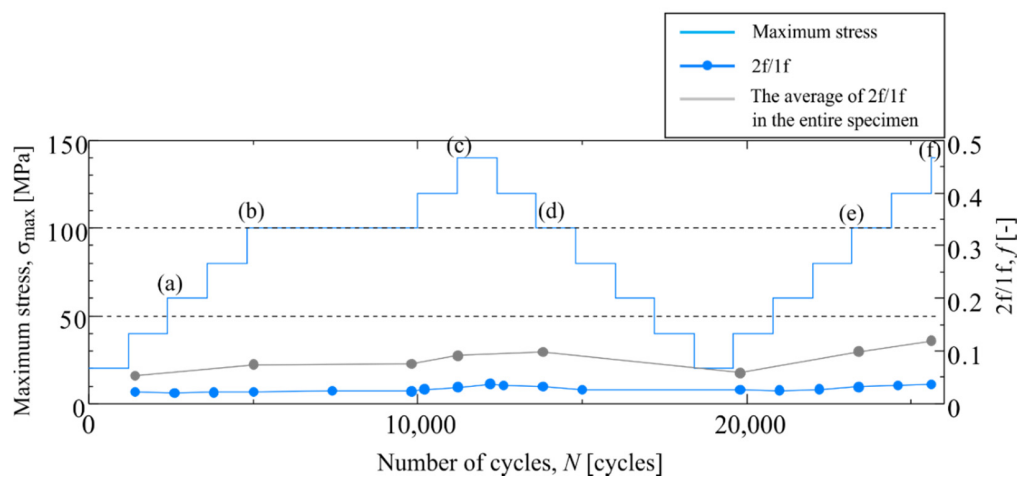
Figure 13 shows the temperature change in Area c-1, where  $\Delta T_D$  did not change. A sinusoidal temperature change was observed regardless of the increase or decrease in the magnitude of the cyclic load. To normalize the magnitude of  $\Delta T_D$  during the test, the ratio of  $\Delta T_D$  to the thermoelastic temperature change  $\Delta T_E$  was calculated ( $2f/1f$ ). Figure 14 shows the change in  $2f/1f$  during the tests. The change in the average value of  $2f/1f$  across the specimen, and the change in the maximum stress of the cyclic load are plotted in Figure 14. It was found that no change in  $2f/1f$  in Area c-1 was observed, and the magnitude of  $2f/1f$  in Area c-1 was very small as compared to the average value of the whole specimen.

#### 4.3.2. Area c-2

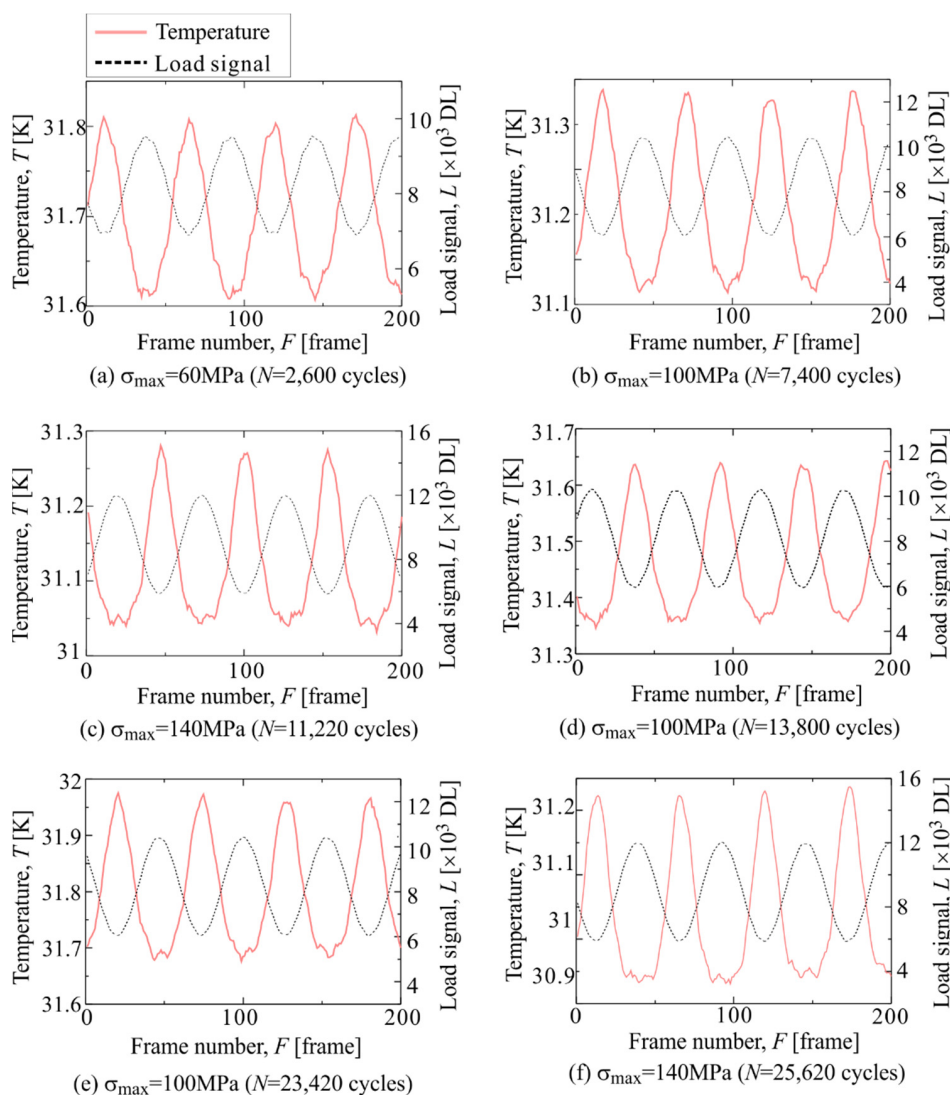
Figure 15 shows the temperature change in Area c-2, where a high  $\Delta T_D$  appeared from the beginning of the staircase-like stress level test. As shown in Figure 15, the temperature waveform is distorted at the maximum tension in the initial round of the staircase-like stress level test. When the maximum stress increases, the distortion of the temperature change at the maximum tension increases. The waveform returns to that in the initial round of the fatigue test when the maximum stress decreases. Figure 16 shows the changes in  $2f/1f$  in Area c-2 and the entire surface. The change in  $2f/1f$  is similar to the changes in the maximum stress in the staircase-like stress level test. When the stress amplitude is increased,  $2f/1f$  also increases, and when the stress amplitude is decreased, the value of  $2f/1f$  also returns to its original state. Even when the stress amplitude is increased again, the same value of  $2f/1f$  is shown for the same stress amplitude.



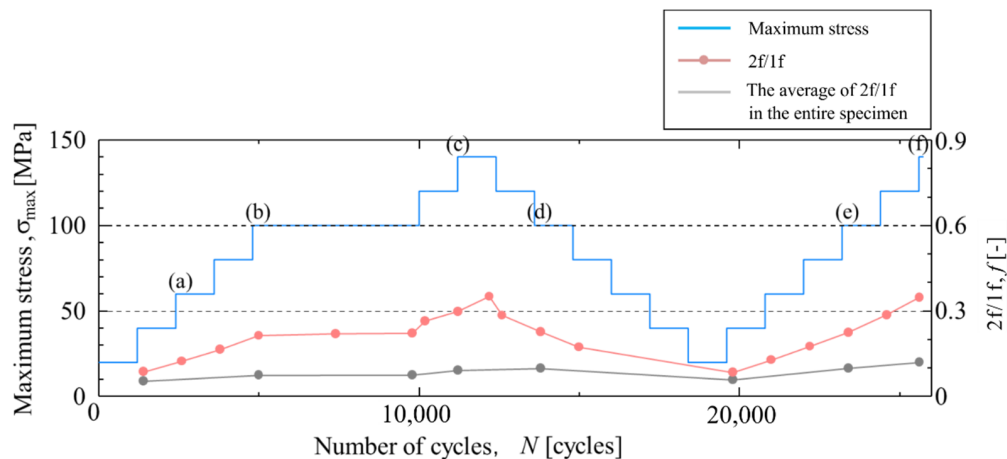
**Figure 13.** Change in temperature and load signal in Area c-1 measured at (a)  $\sigma_{\max} = 60$  MPa and  $N = 2600$  cycles, (b)  $\sigma_{\max} = 100$  MPa and  $N = 7400$  cycles, (c)  $\sigma_{\max} = 140$  MPa and  $N = 11,200$  cycles, (d)  $\sigma_{\max} = 100$  MPa and  $N = 13,800$  cycles, (e)  $\sigma_{\max} = 100$  MPa and  $N = 23,420$  cycles and (f)  $\sigma_{\max} = 140$  MPa and  $N = 25,620$  cycles.



**Figure 14.** Change in  $2f/1f$  in staircase-like stress level test in Area c-1 The letters (a)–(f) indicates the measurement timing shown in Figure 13.



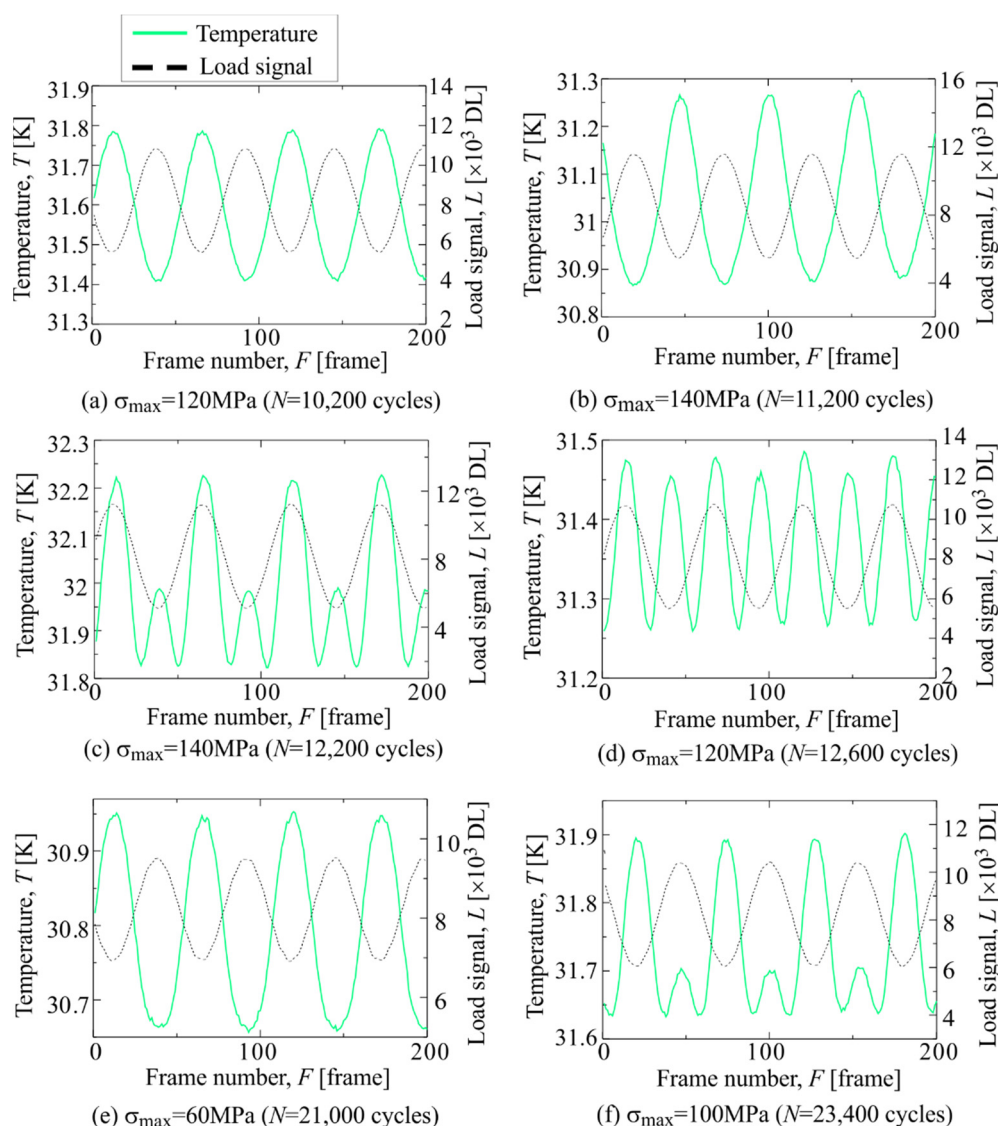
**Figure 15.** Change in mean temperature and load signal in Area c-2 measured at (a)  $\sigma_{\max} = 60$  MPa and  $N = 2600$  cycles, (b)  $\sigma_{\max} = 100$  MPa and  $N = 7400$  cycles, (c)  $\sigma_{\max} = 140$  MPa and  $N = 11,200$  cycles, (d)  $\sigma_{\max} = 100$  MPa and  $N = 13,800$  cycles, (e)  $\sigma_{\max} = 100$  MPa and  $N = 23,420$  cycles and (f)  $\sigma_{\max} = 140$  MPa and  $N = 25,620$  cycles.



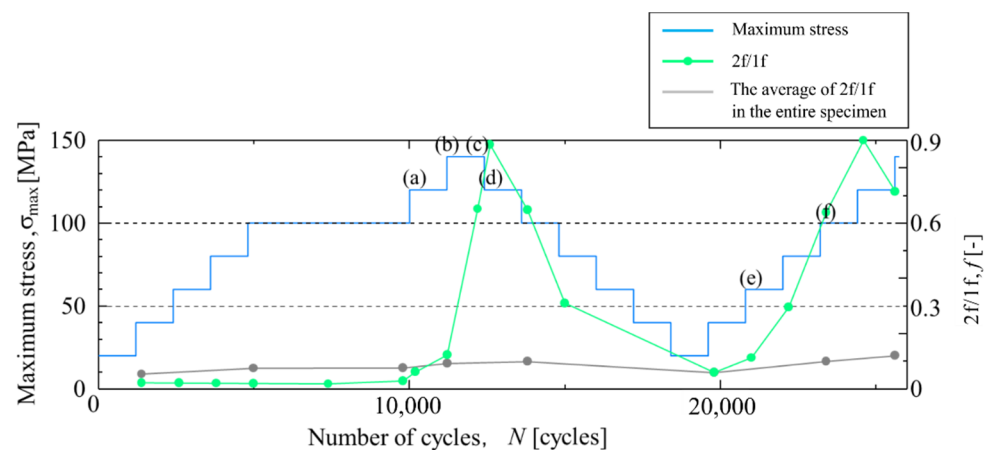
**Figure 16.** Change in  $2f/1f$  in staircase-like stress level test in Area c-2. The letters (a)–(f) indicates the measurement timing shown in Figure 15.

## 4.3.3. Area c-3

The temperature change and load signal in Area c-3 are shown in Figure 17. As shown in Figure 17, the temperature change at  $\sigma_{\max} = 140$  MPa and  $N = 11,200$  cycles shows a sinusoidal waveform. On the other hand, the temperature change at  $\sigma_{\max} = 140$  MPa and  $N = 12,200$  cycles shows a waveform in phase with the load signal, and the temperature rise appears even at the minimum tensile load. When the maximum stress is reduced to 60 MPa, the temperature change returns to the initial waveform. When the maximum stress increased again, the temperature increase at the maximum stress reappeared. The change in  $2f/1f$  in Area c-3 is shown in Figure 18. It was found that the value of  $2f/1f$  in Area c-3 rapidly increased from  $N = 12,200$  cycles, and then the  $2f/1f$  changed depending on the maximum tensile stress.



**Figure 17.** Change in temperature and load signal in Area c-3 measured at (a)  $\sigma_{\max} = 120$  MPa and  $N = 10,200$  cycles, (b)  $\sigma_{\max} = 140$  MPa and  $N = 11,200$  cycles, (c)  $\sigma_{\max} = 140$  MPa and  $N = 12,200$  cycles, (d)  $\sigma_{\max} = 120$  MPa and  $N = 12,600$  cycles, (e)  $\sigma_{\max} = 60$  MPa and  $N = 21,000$  cycles and (f)  $\sigma_{\max} = 100$  MPa and  $N = 23,400$  cycles.



**Figure 18.** Change in  $2f/1f$  in staircase-like stress level test in Area c-3. The letters (a)–(f) indicates the measurement timing shown in Figure 17.

#### 4.4. Discussion

The staircase-like stress level tests for SCFRP showed different temperature waveforms in the three assessment areas (Area c-1, Area c-2, and Area c-3).

In Area c-1, a sinusoidal thermal signal was observed during the fatigue test. It is considered that no damage occurred in Area c-1 through the fatigue tests. In Area c-2, a distortion of temperature change was observed at the maximum tension from the early stage of the fatigue test, and the  $2f/1f$  value showed a similar change according to the increase and decrease of the maximum stress in the fatigue test. In Area c-3, no change in  $2f/1f$  was observed in the initial fatigue test, but after  $\sigma_{\max} = 120$  MPa, the  $2f/1f$  value changed according to the maximum stress, similar to that in Area c-2. Because Area c-3 coincides with the area where the fracture of the specimen occurred, it is considered that delamination initiated when the change in  $2f/1f$  occurred.

The characteristic temperature change appeared at the maximum tension in both Area c-2 and c-3; and in particular, a temperature rise in phase with the load signal appeared in Area c-3. It is considered that the temperature drop did not occur in Area c-2 because the stress above a certain stress level was not transmitted to the resin due to delamination defects. The thermoelastic constant of carbon fiber has a negative value, so the thermoelastic temperature change of the fiber shows an in-phase change with the load signal. In Area c-3, the thermoelastic temperature change in the fiber was observed mainly due to the stress transfer to the fiber without the stress being shared by the resin caused by the delamination above a certain stress level. This phenomenon is caused by delamination and fiber orientation [25].

The  $\Delta T_D$  in Area c-3 was larger than that in Area c-2. Although initial defects might have been present in Area c-2, the final fracture occurred in Area c-3, where a larger  $\Delta T_D$  was observed. Therefore, the value of  $\Delta T_D$  may correspond to the degree of delamination damage and the propagation rate of damage. When the stress amplitude decreased, it returned to its original value. This phenomenon indicates that the effect of stress sharing condition between the resins and fibers due to delamination was reduced. Under the condition of stress amplitude where the value of  $\Delta T_D$  is small, it is expected that the delamination does not propagate, or the propagation rate of the delamination is slow.

The second harmonic temperature component in metals is considered to be due to energy dissipation caused by plastic deformation [28,29], whereas the second harmonic temperature component in SCFRP is considered to be due to the stress sharing condition between the fibers and resins caused by delamination damage. Because the relationship between fatigue damage and the second harmonic temperature component was found, it is considered that the fatigue damage evaluation method using second harmonic components can be applied to damage detection, the evaluation of propagation rate for defect in SCFRP like energy release rate and the further life prediction.

## 5. Conclusions

In this study, the temperature change under cyclic loading was measured for short-fiber-reinforced SCFRP composites. The thermoelastic temperature change  $\Delta T_E$ , phase of thermal signal  $\theta_E$ , and second harmonic temperature components  $\Delta T_D$  were also evaluated. In the fatigue test of SCFRPs, it was confirmed that changes in  $\Delta T_E$ ,  $\theta_E$ , and  $\Delta T_D$  appeared in the damaged regions. To investigate the generation mechanism of the  $\Delta T_D$ , fatigue tests were conducted on the resin specimen, and the change in the  $\Delta T_D$  was examined. Because the  $\Delta T_D$  for the resin specimen was very small compared to that of the SCFRP specimen, it was considered that the  $\Delta T_D$  of SCFRP was generated by a different factor from the energy dissipation by the plastic deformation and the viscoelastic deformation of the resin. In addition, a staircase-like stress level test for SCFRPs was carried out, and the relationship between the maximum stress in the cyclic load and the  $\Delta T_D$  was examined. A distortion of temperature waveform was observed at the maximum tension, and when the stress level decreased, the temperature waveform returned to the original sinusoidal waveform.  $\Delta T_D$  changed according to the change in the maximum stress during the staircase-like stress level test, and a large value of  $\Delta T_D$  was observed in the final rupture region. A distortion of temperature change and  $\Delta T_D$  were considered to be caused by the change in the stress sharing condition between the fiber and resin due to delamination damage. Therefore,  $\Delta T_D$  can be applied to the detection of delamination defects, and the evaluation of damage propagation.

**Author Contributions:** The work presented in this paper is a collaboration between all authors. S.N., K.H. and T.S. (Tomoaki Shinchi) prepared the specimen and analyzed the structure of the specimen. Y.N. and T.T. acquired and processed the experimental data. D.S. and T.S. (Takahide Sakagami) analyses and contributed to writing the paper. The manuscript has been discussed by all the authors. All authors have read and agreed to the published version of the manuscript.

**Funding:** This research received no external funding.

**Institutional Review Board Statement:** Not applicable.

**Informed Consent Statement:** Not applicable.

**Acknowledgments:** The authors would like to acknowledge that this research was partly supported by a Grant-in-Aid for Scientific Research from the Japan Society for the Promotion of Science (B: 16K14119 and B: 17H03146).

**Conflicts of Interest:** The authors declare no conflict of interest.

## References

1. Mortazavian, S.; Fatemi, A. Fatigue behavior and modeling of short fiber reinforced polymer composites: A literature review. *Int. J. Fatigue* **2015**, *70*, 297–321. [[CrossRef](#)]
2. Belmonte, E.; Monte, D.M.; Hoffmann, C.-J.; Quaresimin, M. Damage mechanisms in a short glass fiber reinforced polyamide under fatigue loading. *Int. J. Fatigue* **2017**, *94*, 145–157. [[CrossRef](#)]
3. Belmonte, E.; Monte, D.M.; Hoffmann, C.-J.; Quaresimin, M. Damage initiation and evolution in short fiber reinforced polyamide under fatigue loading: Influence of fiber volume fraction. *Compos. Part B Eng.* **2017**, *113*, 331–341. [[CrossRef](#)]
4. Fragoudakis, R. A numerical approach to determine fiber orientations around geometric discontinuities in designing against failure of GFRP laminates. *Int. J. Struct. Integr.* **2019**, *10*, 371–379. [[CrossRef](#)]
5. Avdelidis, N.P.; Almond, D.P.; Dobbins, A.; Hawtin, B.C.; Ibarra-Castaneda, C.; Maldague, X. Aircraft composites assessment by means of transient thermal NDT. *Prog. Aerosp. Sci.* **2004**, *40*, 143–162. [[CrossRef](#)]
6. Avdelidis, N.P.; Gan, T.-H. Non-destructive evaluation (NDE) of Composites: Infrared (IR) thermography of wind turbine blades. In *Non-Destructive Evaluation (NDE) of Polymer Matrix Composites*; Karbhari, V.M., Ed.; Woodhead Publishing Limited: Cambridge, UK, 2013; pp. 634–648. ISBN 978-0-85709-344-8.
7. Chatterjee, K.; Tuli, S.; Pickering, S.G.; Almond, D.P. A comparison of the pulsed, lock-in and frequency modulated thermography nondestructive evaluation techniques. *NDT E Int.* **2011**, *44*, 655–667. [[CrossRef](#)]
8. Maldague, X.; Marinetti, S. Pulse phase infrared thermography. *J. Appl. Phys.* **1996**, *79*, 2694–2698. [[CrossRef](#)]
9. Greene, R.J.; Patterson, E.A.; Rowlands, R.E. Thermoelastic Stress Analysis. In *Springer Handbook of Experimental Solid Mechanics*; Sharpe, W.N., Jr., Ed.; Springer Science + Business Media, LLC: New York, NY, USA, 2008; pp. 743–767; ISBN 978-0-387-26883-5.

10. Dulieu-Barton, J.M.; Stanley, P. Development and applications of thermoelastic stress analysis. *J. Strain Anal. Eng. Des.* **1998**, *33*, 93–104. [[CrossRef](#)]
11. Dulieu-Barton, J.M. Introduction to thermoelastic stress analysis. *Strain* **1999**, *35*, 35–39. [[CrossRef](#)]
12. Pitarresi, G.; Patterson, E.A. A review of the general theory of thermoelastic stress analysis. *J. Strain Anal. Eng. Des.* **2003**, *38*, 405–417. [[CrossRef](#)]
13. Sakagami, T. Remote nondestructive evaluation technique using infrared thermography for fatigue cracks in steel bridges. *Fatigue Fract. Engng. Mater. Struct.* **2015**, *38*, 755–779. [[CrossRef](#)]
14. Lin, S.T.; Rowlands, R.E. Thermoelastic stress analysis of orthotropic composites. *Exp. Mech.* **1995**, *35*, 257–265. [[CrossRef](#)]
15. Dulieu-Barton, J.M.; Stanley, P. Applications of thermoelastic stress analysis to composite materials. *Strain* **1999**, *35*, 41–48. [[CrossRef](#)]
16. Emery, T.R.; Dulieu-Barton, J.M.; Earl, J.S.; Cunningham, P.R. A generalised approach to the calibration of orthotropic materials for thermoelastic stress analysis. *Compos. Sci. Technol.* **2008**, *68*, 743–752. [[CrossRef](#)]
17. Palumbo, D.; De Finis, R.; Demelio, G.P.; Galietti, U. Study of damage evolution in composite materials based on the Theroelastic Phase Analysis (TPA) method. *Compos. Part B Eng.* **2017**, *117*, 49–60. [[CrossRef](#)]
18. Sfarra, S.; Perilli, S.; Ambrosini, D.; Paoletti, D.; Nardi, I.; de Rubeis, T.; Santulli, C. A proposal of a new material for greenhouses on the basis of numerical, optical, thermal and mechanical approaches. *Constr. Build. Mater.* **2017**, *155*, 332–347. [[CrossRef](#)]
19. Krstulovic-Opara, L.; Klarin, B.; Neves, P.; Domazet, Z. Thermal image and thermoelastic stress analysis of impact damage of composite materials. *Eng. Fail. Anal.* **2011**, *18*, 713–719. [[CrossRef](#)]
20. Emery, T.R.; Dulieu-Barton, J.M. Thermoelastic Stress Analysis of damage mechanisms in composite materials. *Compos. Part A* **2010**, *41*, 1729–1742. [[CrossRef](#)]
21. Uenoya, T.; Fujii, T. Damage characterization of woven fabric composite materials by thermoelastic analysis. *J. Soc. Mater. Sci. Jpn.* **2000**, *49*, 941–947. [[CrossRef](#)]
22. Palumbo, D.; De Finis, R.; Demelio, G.P.; Galietti, U. A new rapid thermographic method to assess the fatigue limit in GFRP composites. *Compos. Part B Eng.* **2016**, *103*, 60–67. [[CrossRef](#)]
23. De Finis, R.; Palumbo, D.; Galietti, U. Fatigue damage analysis of composite materials using thermography-based techniques. *Procedia Struct. Integr.* **2019**, *18*, 781–791. [[CrossRef](#)]
24. De Finis, R.; Palumbo, D. Estimation of the dissipative heat sources related to the total energy input of CFRP composite by using the second amplified harmonic of the thermal signal. *Materials* **2020**, *13*, 2820. [[CrossRef](#)] [[PubMed](#)]
25. Shiozawa, D.; Sakagami, T.; Nakamura, Y.; Nonaka, S.; Hamada, K. Fatigue damage evaluation of short carbon fiber reinforced plastics based on phase information of thermoelastic temperature change. *Sensors* **2017**, *17*, 2824. [[CrossRef](#)] [[PubMed](#)]
26. Bremond, P.; Potet, P. Lock-in thermography: A tool to analyze and locate thermomechanical mechanism in materials and structure. *Proc. SPIE* **2001**, *4360*, 560–566.
27. Enomae, T. FiberOri8single03.exe. Available online: <http://www.enomae.com/FiberOri/index.htm> (accessed on 4 December 2017).
28. La Rosa, G.; Risitano, A. Thermographic methodology for rapid determination of the fatigue limit of materials and mechanical components. *Int. J. Fatigue* **2000**, *22*, 65–73. [[CrossRef](#)]
29. Shiozawa, D.; Inagawa, T.; Washio, T.; Sakagami, T. Accuracy improvement in dissipated energy measurement by using phase information. *Meas. Sci. Technol.* **2017**, *28*, 4. [[CrossRef](#)]