

Supplementary material

# Strategies Regarding High-Temperature Applications w.r.t Strength, Toughness, and Fatigue Life for SA508 Alloy

Muhammad Raies Abdullah <sup>1</sup>, Cai Hongneng <sup>1</sup>, and Fang Liang <sup>1,2,\*</sup>

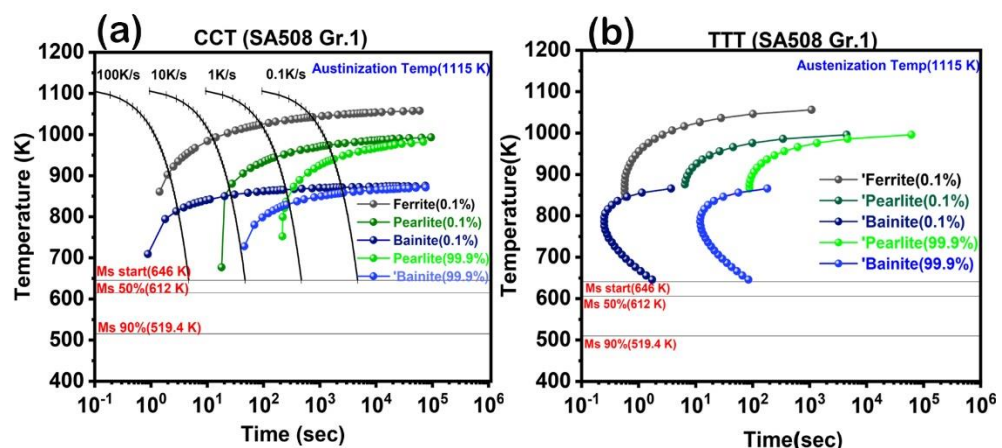
<sup>1</sup> Key Laboratory for Mechanical Behavior of Materials, Xi'an Jiaotong University, Xi'an 710049, China; hntsai@mail.xjtu.edu.cn (C.H.); raieskhan@stu.xjtu.edu.cn (M.R.A.)

<sup>2</sup> School of Mechanical & Electrical Engineering, Xiamen University Tan Kah Kee College, Zhangzhou 363105, China

\* Correspondence: fangl@xjtu.edu.cn; Tel.: +(86)-29-8266-5479

## 1. Simulation Conditions for Optimization

We use Jmat-Pro (version 7) and Thermo-Calc software (2019a) for simulations, so we some assumptions for these calculations. Thermocalc is working on Gibbs's free energy models [1] for equilibrium conditions. For mechanical calculations related to SA508 analysis, it was assumed at  $\gamma$ -FCC, grain size is 10–15  $\mu\text{m}$ , and calculations were performed at 650 K temperature. For stress-strain calculations, the strain rate was assumed to be 0.1 at 650 K temperature.

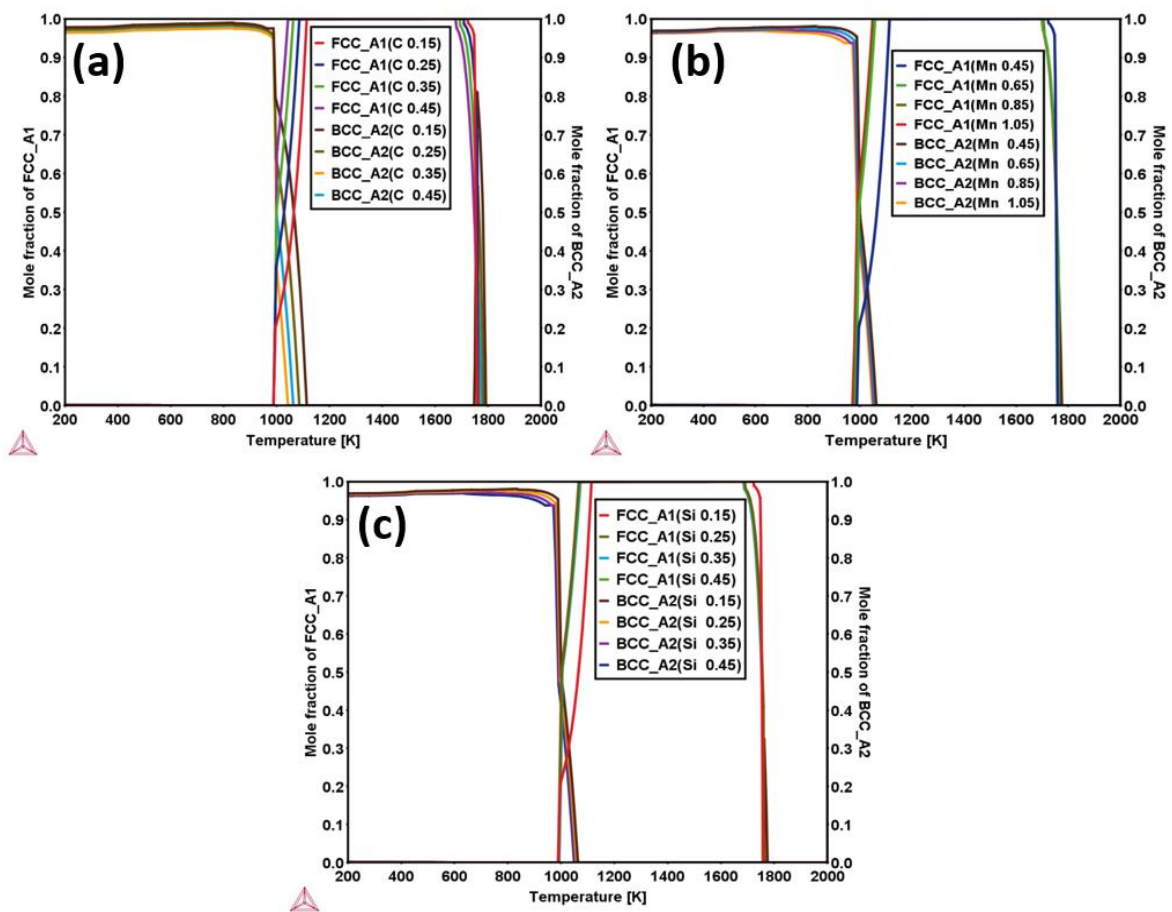


**Figure S1.** Jmat Pro calculation for SA508 grade steel for RPV (a) Continuous Cooling Transformation (CCT) for SA508 (b) Isothermal transformation (TTT) for SA508 steel.

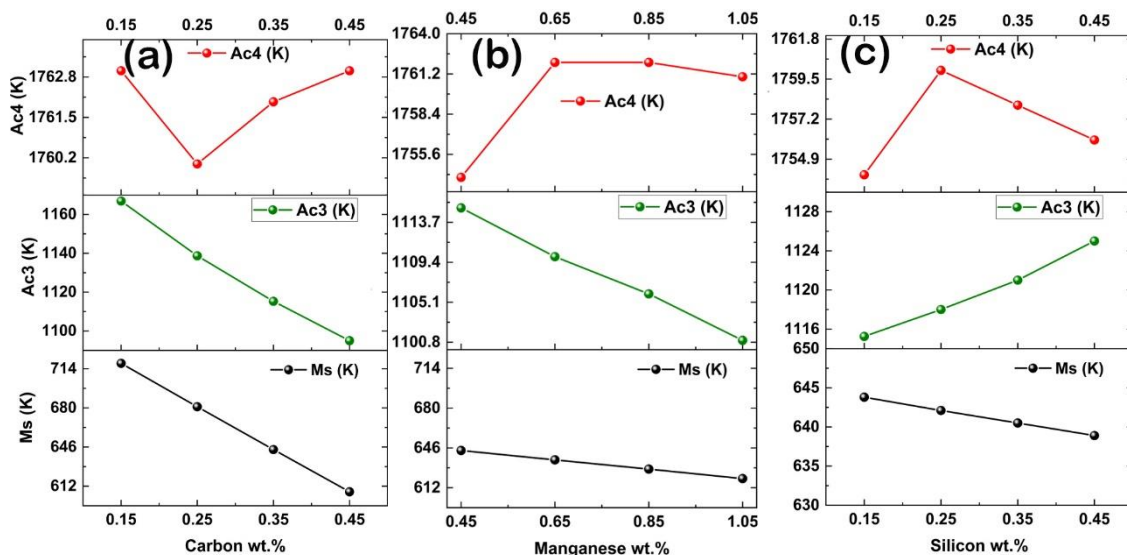
Creep is reflected to be a temperature-dependent process and satisfies the power-law equation [2].

$$\varepsilon = A \left( \frac{\sigma}{E} \right) \exp \left( \frac{-Q}{RT} \right) \quad (S1)$$

T is the temperature,  $\sigma$  is significant stress, E is the Modulus, and A is a material-dependent constant; however, R, Q, and n are gas constant, activation energy, and stress exponent. R, T, and A are constants or known, so only Q, n, E, and  $\sigma$  are responsible parameters that change the material's steady-state creep rate.



**Figure S2.**  $Ac_3$ (ferrite to Austenite transition temperature) and  $Ac_1$ (Austenite to Ferrite transition temperature) temperature calculations (a) FCC and BCC for MS1 samples; (b) FCC and BCC for MS2 samples; (c) FCC and BCC for MS3 samples.



**Figure S3.** Martensite temperature start (Ms),  $Ac_3$  and  $Ac_4$  temperatures w.r.t (a) MS1 (changing Carbon contents); (b) MS2 (changing Manganese contents); (c) MS3 (changing Silicon contents).

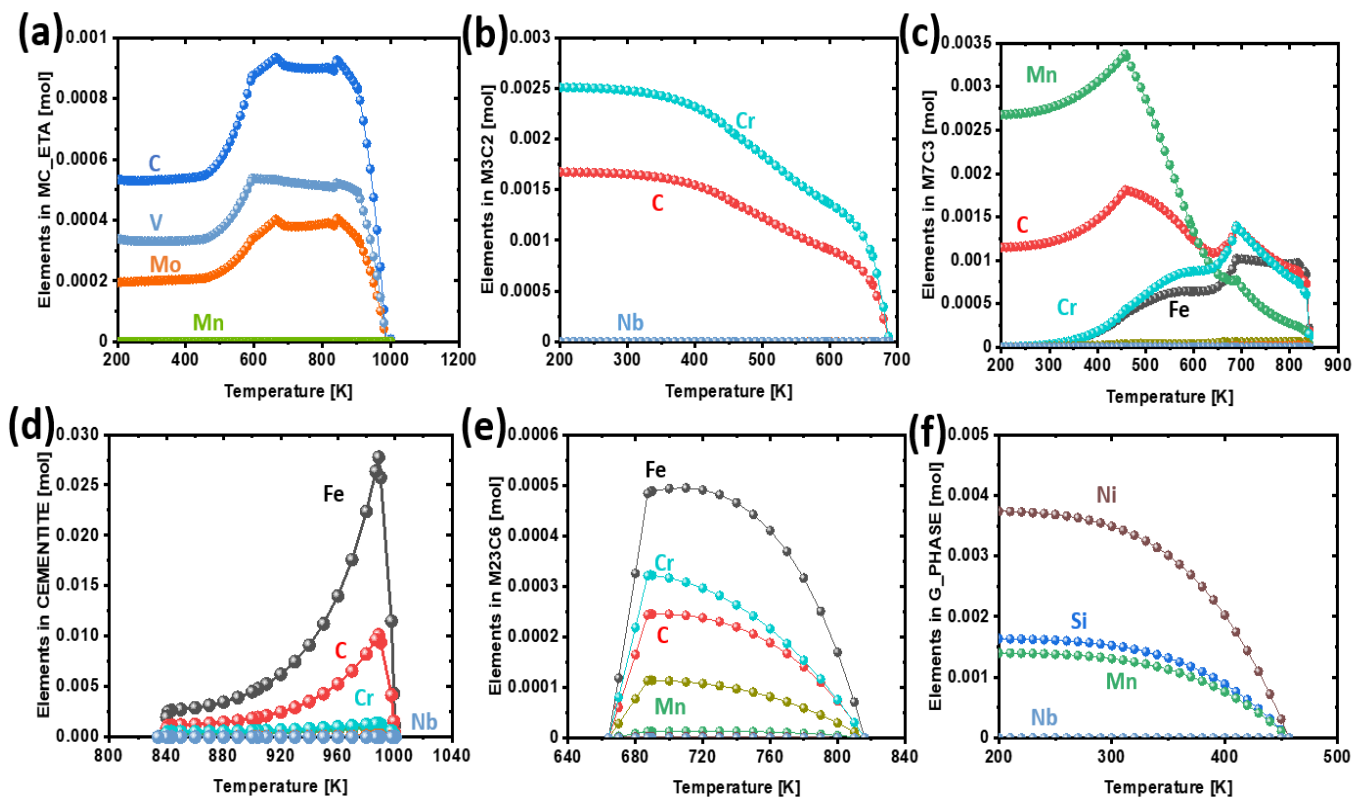


Figure S4. Composition of phases w.r.t elements (a) MC\_ETA; (b) M3C2; (c) M7C3; (d) Cementite; (e) M23C6; (f) G-phase.

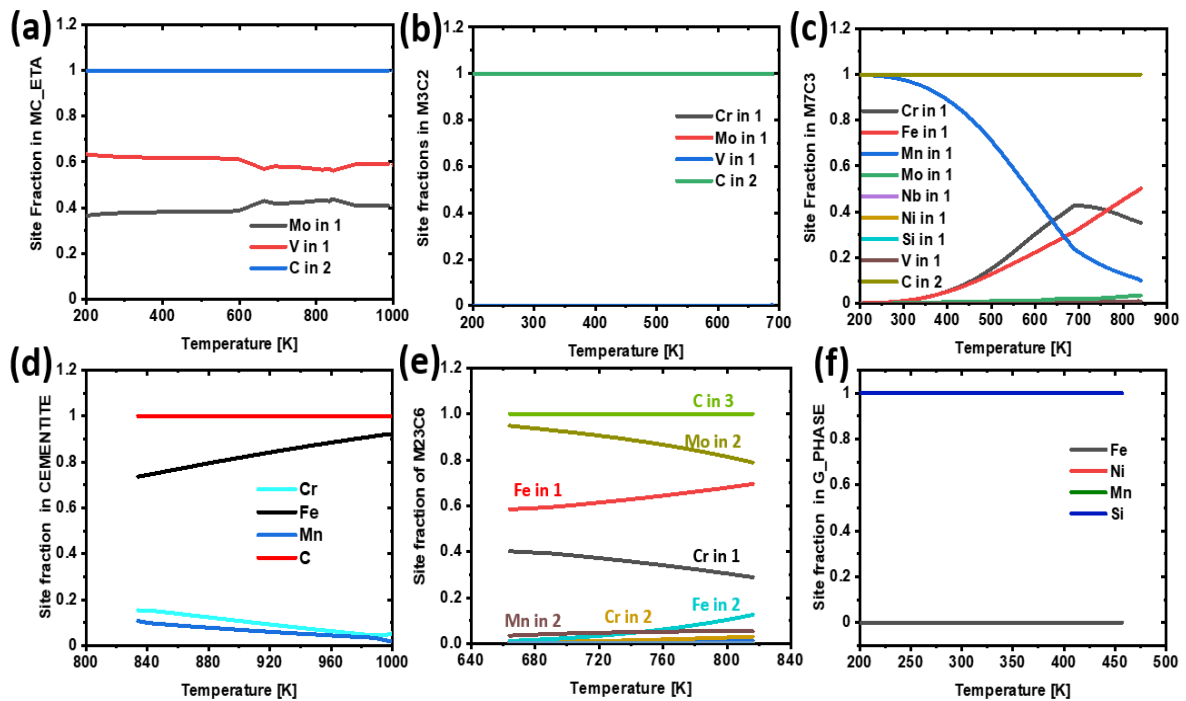


Figure S5. Site fractions of phases w.r.t elements (a) MC\_ETA, (b) M3C2, (c) M7C3, (d) Cementite, (e) M23C6, (f) G-phase.

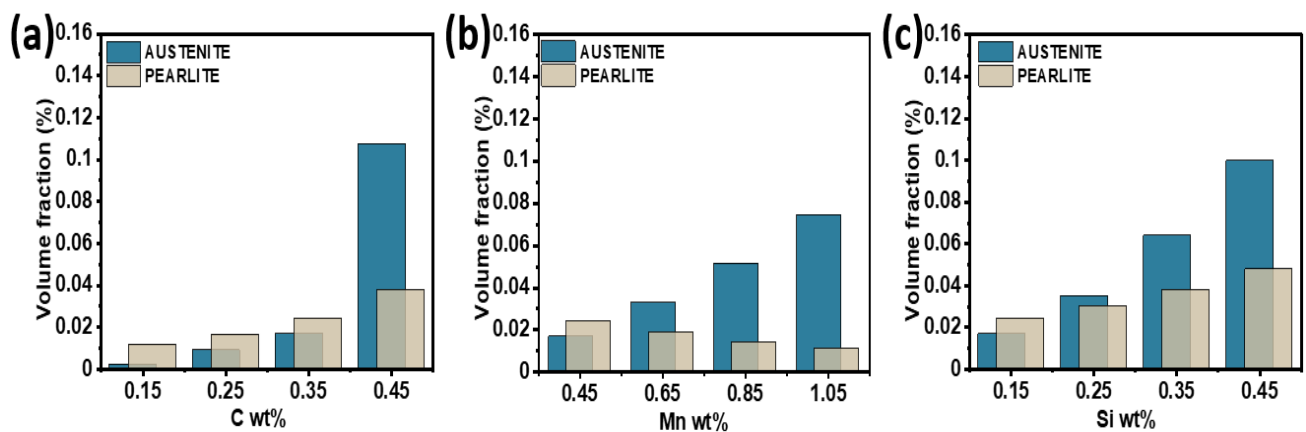


Figure S6. (a) Volume fraction Austenite and pearlite regarding Carbon; (b) Volume fraction Austenite and pearlite regarding Mn; (c) Volume fraction Austenite and pearlite regarding Silicon.

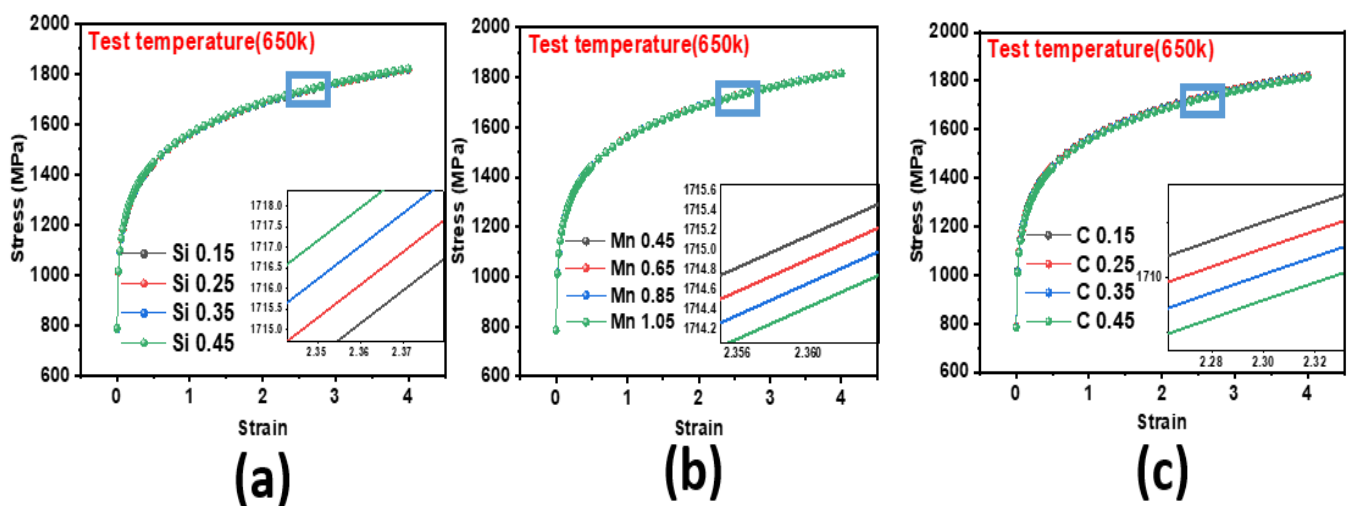


Figure S7. Stress-strain calculation for SA508 (a); Silicon, (b); Manganese; (c) Carbon.

## 2. Fatigue Equations

Fatigue is an effect of permanent microplastic deformation when a material is exposed to cyclic loading. This often occurs together with thermal and/or corrosive attack owing to the presence of oxygen and hydrogen, causing a phenomenon known as stress corrosion cracking [3].

$$\frac{\Delta\epsilon}{2} = \frac{\Delta\epsilon_e}{2} + \frac{\Delta\epsilon_p}{2} = \frac{\sigma_f'}{E}(2N)^b + \epsilon_f'(2N)^c \quad (S2)$$

In this equation  $\Delta\epsilon$ ,  $\sigma_f'$ ,  $\epsilon_f'$ ,  $E$ , and  $2N$  are strain amplitude, fracture stress, fracture strain, Young's Modulus and number of cycles respectively, where  $b(-0.1)$  and  $c(-0.9)$  are user defined constants [4].

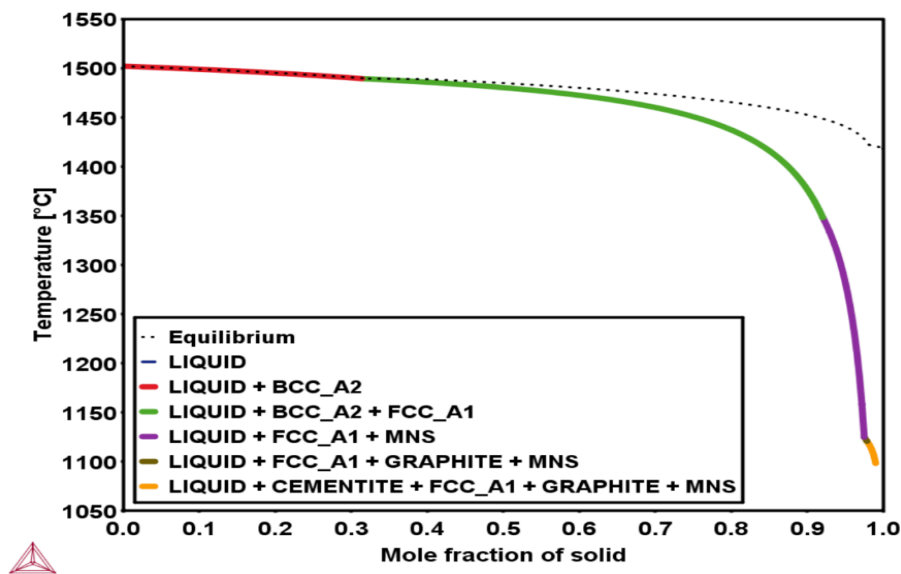


Figure S8. Scheil solidification diagram for the SUS304 composition and their effect on solidifications.

## References

1. Chen, H.L.; Mao, H.; Chen, Q. Database development and Calphad calculations for high entropy alloys: Challenges, strategies, and tips. *Mater. Chem. Phys.* **2018**, *210*, 279–290. <https://doi.org/10.1016/j.matchemphys.2017.07.082>.
2. Miodownik, A.P.; Li, X.; Saunders, N.; Schille, J.-P. Modelling of Creep in Nickel Based Superalloys, 6th Int. Charles Parsons Turbine Conf. (2003). Available online: <http://www.sentsoftware.co.uk/media/2555/creep.pdf>.
3. Raabe, D.; Tasan, C.C.; Olivetti, E.A. Strategies for improving the sustainability of structural metals. *Nature* **2019**, *575*, 64–74. <https://doi.org/10.1038/s41586-019-1702-5>.
4. Guo, Z.; Saunders, N.; Miodownik, P.; Schillé, J.P. Modeling material properties of lead-free solder alloys. *J. Electron. Mater.* **2008**, *37*, 23–31. <https://doi.org/10.1007/s11664-007-0218-1>.