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Electrospinning Mo-Doped Carbon Nanofibers as an Anode to Simultaneously Boost Bioelectrocatalysis and Extracellular Electron Transfer in Microbial Fuel Cells

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Abstract: The sluggish electron transfer at the interface of microorganisms and an electrode is a bottleneck of increasing the output power density of microbial fuel cells (MFCs). Mo-doped carbon nanofibers (Mo-CNFs) prepared with electrostatic spinning and high-temperature carbonization are used as an anode in MFCs here. Results clearly indicate that $Mo_{2}C$ nanoparticles uniformly anchored on carbon nanowire, and Mo-doped anodes could accelerate the electron transfer rate. The Mo-CNF II anode delivered a maximal power density of 1287.38 mW m $^{-2}$, which was twice that of the unmodified CNFs anode. This fantastic improvement mechanism is attributed to the fact that Mo doped on a unique nanofiber surface could enhance microbial colonization, electrocatalytic activity, and large reaction surface areas, which not only enable direct electron transfer, but also promote flavin-like mediated indirect electron transfer. This work provides new insights into the application of electrospinning technology in MFCs and the preparation of anode materials on a large scale.

Keywords: electrospinning; microbial fuel cells; Mo-doped carbon nanofibers; interface modification

Citation: Wu, X.; Li, X.; Shi, Z.; Wang, X.; Wang, Z.; Li, C.M. Electrospinning Mo-Doped Carbon Nanofibers as an Anode to Simultaneously Boost Bioelectrocatalysis and Extracellular Electron Transfer in Microbial Fuel Cells. *Materials* **2023**, *16*, 2479. <https://doi.org/10.3390/ma16062479>

Academic Editor: Enrico Negro

Received: 19 February 2023 Revised: 15 March 2023 Accepted: 15 March 2023 Published: 21 March 2023

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1. Introduction

Energy shortage, environmental pollution, and water scarcity are urgent challenges for humans, and microbial fuel cells (MFCs) are increasing attention as an ecofriendly and sustainable energy conversion technology [\[1,](#page-8-0)[2\]](#page-8-1). MFCs use electroactive bacteria as catalysts to convert chemical energy from organic substrates into electrical energy, and they are widely used in wastewater treatment, heavy-metal remediation, and other applications [\[3](#page-8-2)[,4\]](#page-8-3). However, the sluggish extracellular electron transfer (EET) rate of MFCs leads to their low actual output power density, which limits large-scale applications [\[5\]](#page-8-4). The surface properties and structure of anodes are critical for increasing the available effective surface for bacterial colonization and interfacial electrocatalytic activity, and thus for improving electron transfer to solid electrodes [\[6,](#page-8-5)[7\]](#page-8-6).

Carbon material is still the most suitable anode material, with almost all the basic requirements such as high electrical conductivity, biocompatibility, and low cost [\[8,](#page-8-7)[9\]](#page-8-8). Electrospinning is a direct and simple method for producing continuous nanofibers on a large scale [\[10\]](#page-8-9). Obtained fibers with electrospinning have the advantages of a small diameter and a large specific surface area [\[10\]](#page-8-9). After stabilization and carbonization, carbon nanofibers are obtained, which are widely used in many fields [\[11](#page-9-0)[–13\]](#page-9-1). Therefore, many researchers are studying the use of electrospinning technology to enhance the performance of MFCs, and the outcomes show that electrospinning has a broad application prospect in the preparation of efficient microbial fuel cell electrodes $[8,9,14]$ $[8,9,14]$ $[8,9,14]$. For example, a self-supporting electrode composed of iron cobalt bimetallic metal–organic frameworks (FeCo-MOF/CNFs) prepared with electrospinning dramatically enhanced the output power of MFCs [\[15\]](#page-9-3). Nitrogendoped carbon nanofibers anchored with iron nanoparticles (Fe/N-x@CNFs) were designed

as anode electrocatalysts with good electrocatalytic activity and biocompatibility using electrospinning and calcination processes [\[16\]](#page-9-4). Molybdenum carbides with excellent electrocatalytic activity and good biocompatibility were extensively investigated for MFC anode modification [\[17,](#page-9-5)[18\]](#page-9-6). For example, Zou et al. prepared molybdenum-carbide-hybridized graphene nanocomposite ($Mo₂CaG$) with electrostatic self-assembly and high-temperature carbonization [\[17\]](#page-9-5), synergistically promoting microbial biofilm growth and interfacial bioelectrocatalysis via Mo2C@G anodes for bioelectricity production. Molybdenum doping can improve the active site, electrocatalytic activity, and biocompatibility of carbon nanofibers. The preparation of Mo-doped carbon nanowires using electrostatic spinning improves the electrocatalytic activity and biocompatibility of the electrodes, and the nanowire structure adds more bacterial attachment sites. However, to the best of our knowledge, no one has prepared Mo-doped carbon nanowires using the electrostatic spinning technique for application in the anode of microbial fuel cells.

We prepared Mo2C-modified carbon nanowires as an anode for *Shewanella putrefaciens* CN32 (*S. putrefaciens* CN32) MFCs with a simple method. The material was specifically prepared via electrostatic spinning, followed by stabilization and carbonization. Electrochemical analysis confirmed that Mo-CNF II exhibited high electrocatalytic activity towards flavin-like electron mediators. The doping of $Mo₂C$ improves the bacteria-to-electrode interfacial electron transfer, which enhances the biocompatibility of the electrode surface, adds more active sites, and increases electrocatalytic activity. The effects of different amounts of Mo doping on enhanced EET during bioelectrocatalysis were compared. MFCs equipped with Mo-CNF II exhibited a great output power density of around 1287.38 mW·m⁻².

2. Materials and Methods

2.1. Preparation of Mo-Doped Carbon Nanofibers

The preparation process of Mo-doped carbon nanofibers is shown in Scheme [1.](#page-2-0) First, 1g of polyacrylonitrile (PAN) powder was weighed and added to N,N-dimethylformamide (DMF) to prepare a 10 wt % PAN solution, and a uniform solution was obtained via magnetic stirring for 24 h at room temperature. Next, molybdenum acetylacetone $(M_0O_2(a\alpha c)_2)$ was added and stirred for 24 h to obtain a green, homogeneous spinning precursor solution. Nanofiber membranes were fabricated with electrospinning. Electrospinning parameters were as follows: positive voltage, $8-10$ KV; negative voltage, -3 to -2 KV; pushing speed, \sim 0.0008–0.001 mm/s; distance between needle and collection drum, 16–18 cm; rotational speed of the drum receiver, 100 rpm; temperature and humidity, controlled at 25 \pm 3 °C and $35 \pm 5\%$, respectively. First, the fiber membrane was preoxidized in a muffle furnace at 280 °C for 2 h with a heating rate of 5 °C/min to stabilize the fiber structure. It was then carbonized at 900 °C for 2 h in an argon atmosphere with a heating rate of 2 °C/min. After carbonization, the morphology of the fiber membrane was basically unchanged, and Modoped carbon nanofibers were obtained after grinding. When the $MoO₂(acac)₂$ addition levels in the spinning precursor solution were 0, 5, 10, and 20 mM, the prepared Mo-doped carbon fibers were designated as CNFs and Mo-CNFs I–III, respectively. Transmission electron microscopy (TEM, JEOL JEM-2100HR, Tokyo, Japan) and scanning electron microscopy (SEM, JEOL JSM-7800F, Tokyo, Japan) were used to analyze the morphology and structure of the material. The surface properties of the material were characterized with X-ray photoelectron spectroscopy (XPS) and Raman spectroscopy. The crystal structure and crystal phase of the material was determined with powder X-ray diffraction (XRD).

Scheme 1. Preparation of Mo-doped carbon nanofibers.

2.2. Electrochemical Characterization

An electrochemical workstation (CHI660, Shanghai Chen Hua Instrument Co., Ltd., Shanghai, China) was used for all electrochemical testing. A three-electrode system was used in the experiment with a carbon cloth modified with nanomaterials as the working electrode, saturated calomel electrode (SCE) as the reference electrode, and a titanium sheet (2×2 cm) as the counter electrode. First, the experiment was conducted in a 0.1 M phosphate buffer saline (PBS) buffer with 2 μ M flavin mononucleotide (FMN). With the addition of FMN, cyclic voltammetry (CV), differential pulse voltammetry (DPV), and electrochemical impedance spectroscopy (EIS) experiments were conducted at regular intervals. When the peak current reached its maximum, CV tests with various sweep speeds were conducted. Subsequently, tests were performed in bacterial suspensions (*S. putrefaciens* CN32 suspended in M9 buffer (KH2PO4, NH4Cl, NaCl, Na2HPO4·12H2O) with lactic acid as the electron donor) by discharging at 0.2 V for 12 h and then testing CV, DPV, and EIS. The potential ranges for CV and DPV tests were −0.8 to 0.6 and −0.8 to 0 V, and the CV sweep rates were 1 and 30 mV s⁻¹, respectively. The DPV test's frequency was 1 Hz, the amplitude was 50 mV, and the potential step was 4 mV. All EIS experiments used a perturbation signal of 5 mV at -0.45 V and a frequency range of 0.01–100,000 Hz.

2.3. MFC Setup and Operation

In this work, *S. putrefaciens* CN32 was used as an electroactive microorganism, and the MFC's performance was tested using an H-type double-chamber apparatus. H-type MFCs consist of two counter-mouth glass vessels separated by a proton exchange membrane (Nafion 117, N117), with the anode being a carbon cloth coated with nanomaterials and the cathode being a carbon brush. The cathodic solution was a 0.01 mM PBS buffer containing 50 mM potassium ferricyanide ($K_3Fe[(CN)_6]$), and the anodic solution was an M9 buffer with suspended *S. putrefaciens* CN32 and lactic acid as the sole electron donor. The discharge curves of MFCs were tested with an external 2 k Ω resistor for stable operation at 30 °C. After three rounds of discharge testing, and when the output voltage was stable, the resistance was changed (from 80,000 to 500 Ω) via the external resistor box to test the power-density and polarization curves.

After three rounds of discharge testing, the anodic biofilm treatment was first fixed with paraformaldehyde followed by gradient dehydration with ethanol. The concentrations of ethanol were 30%, 40%, 50%, 60%, 70%, 80%, 90%, and 100%, and each concentration was dehydrated for 20 min. In this paper, several steps were taken to prepare the electrode: 2 mg of material was put into a mortar, and 30–40 µL of a diluted polytetrafluoroethylene (PTFE) emulsion was added and mixed thoroughly to form a paste that was evenly coated

on the front and back sides of the carbon cloth (1 \times 1 cm), and dried under vacuum at 110 \degree C for 3 h to obtain the required electrode for the experiment. The details of the above operations and bacterial culture are in our previous work [\[19](#page-9-7)[,20\]](#page-9-8).

3. Results and Discussion

The detailed structural and morphological characterization of the Mo-CNFs and CNFs was examined with SEM and TEM. Figure [1a](#page-3-0),b show that the CNFs were nanowires with a uniform diameter and a smooth surface, and Mo was uniformly distributed on the nanofiber surface in the form of nanoparticles in Mo-CNF II. There was no significant change in the fiber diameter before and after Mo doping, and the fiber diameter was around 400–500 nm. Figure [1c](#page-3-0),d show the TEM image and EDX element mappings of Mo-CNF II. The results are consistent with SEM, and distinctly indicate that Mo elements were uniformly and effectively distributed within the entire nanofiber. Mo doping on the nanofiber surface increased fiber surface roughness and the electrochemically active sites, promoting more bacterial adhesion to the anode. This greatly improved the efficiency of extracellular electron transfer between the electrodes and bacteria.

Figure 1. Characterization of Mo-CNFs and CNFs. SEM illustrations of (**a**) CNFs, (**b**) Mo-CNF II; (**c**) TEM image and (**d**) TEM element mappings of Mo-CNF II.

The surface properties and crystal structure of the materials were characterized using XPS, Raman spectroscopy, and XRD. In the XRD spectrum (Figure [2a](#page-4-0)), two broad peaks at 25◦ and 43◦ were found for CNFs that were identified as graphitized carbon peaks [\[21\]](#page-9-9). The positions and relative intensities of the characteristic diffraction peaks in the Mo-CNFs were in good agreement with the PDF cards (PDF #15-0457), indicating that Mo was present in the carbon nanowires in the form of $Mo₂C$. In addition, the characteristic peaks in Mo-CNF

I were not obvious, probably because the Mo content was too small. The Raman analysis of Mo-CNF and CNF electrodes is shown in Figure [2b](#page-4-0). The obvious peaks that existed at 1350 and 1580 cm⁻¹ for all electrodes corresponded to the D and G peaks of the carbon material, respectively. The $I_D:I_G$ value is a crucial parameter for determining the degree of lattice collapse and graphitization. Analyzing the statistics showed that the $I_D: I_G$ values of Mo-CNFs were all higher than those of CNFs (0.94). Furthermore, Mo-CNF II had the largest I_D:I_G value of 0.97. This indicates that there were more lattice defects in Mo-CNF II, which allowed for exposing more active centers and increasing electrocatalytic activity. The full XPS spectra in Figure [2c](#page-4-0) show that the surface of the Mo-CNFs contained four elements, namely, Mo (BE = 232.02 eV), N (BE = 398.03 eV), O (BE = 531.01 eV), and C (BE = 287.01 eV), compared to the surface of CNFs, demonstrating the effective doping of Mo into the carbon nanofibers [\[22\]](#page-9-10). Then, the 3d Mo spectrum (Figure [2d](#page-4-0)) showed that, in Mo-CNF II, Mo3d could be divided into four peaks corresponding to $MO_x3d_{3/2}$ (235 eV), $Mo²⁺3d_{3/2}$ (232.63 eV), $MoO_x3d_{5/2}$ (231.67 eV), and $Mo^{2+}3d_{3/2}$ (228.82 eV) [\[23,](#page-9-11)[24\]](#page-9-12). The high-resolution C1s, O1s, and N1s XPS spectra of Mo-CNF II are shown in Figures S1 and S2; the C1s spectrum shows that C1s could be divided into four peaks in Mo-CNF II corresponding to C-N (285.47 eV), C-O (286.62 eV), C=O (289.15 eV), and C-C (284.57 eV) [\[25\]](#page-9-13).

Figure 2. (**a**) XRD patterns of various Mo-CNFs and CNFs; (**b**) Raman image (insert is a histogram of ID:I^G values); (**c**) total XPS spectra of CNFs and different Mo-CNF composites; (**d**) high-resolution Mo 3d XPS spectrum of Mo-CNF II.

To comprehend the interfacial redox response of flavin media on the surface of various Mo-CNFs, we used a three-electrode system to perform the experiments in a PBS buffer containing 2 µM FMN. The CVs results show that all electrodes had oxidation peaks at the −0.45 V potential, which was the electrochemical response to the oxidation of flavin mediators on the electrode interface, as shown in Figure [3a](#page-6-0). In addition, the peak current of CV curves increased and then decreased with the increase in Mo content, showing a volcanic curve. Mo-CNF II exhibited the highest peak current density of 306.1 μ Acm⁻², which was much larger than that of the unmodified CNF electrode (84.2 µAcm−²). The results of the DPV curves (Figure [3b](#page-6-0)) are consistent with the CV curves. The maximal redox peak current of the CNF II electrode in the DPV curve was 0.91 mA cm⁻², which means that the surface of the Mo-CNF II electrodes had the largest number of electrochemically active sites, which is favorable for promoting flavin-mediated indirect electron transfer. Figure [3c](#page-6-0) shows the EIS results that indicate that the *R*_{ct} of Mo-CNF I was much lower than that of the CNFs (123.36 Ω), and Mo-CNF II had the lowest interfacial electron transfer impedance (5.59 Ω). In addition, the *R*_{ct} of Mo-CNFs I and III were 12.08 and 9.55 Ω , respectively. This demonstrates that the modification of electrode interfaces with $Mo₂C$ considerably impacted the improvement in electron transfer rate. The relationship between peak current and sweep speed for CV curves with different sweep speed levels is shown in Figure [3d](#page-6-0) and Figure S3. The redox peak current of the CNF electrode tended to be more linearly related to the sweep rate, indicating that the electrochemical process was surface-controlled. In contrast, the redox peak current of the Mo-CNF II electrode surface tended to be more linearly related to the square root of the sweep speed, showing that the electrochemical process was diffusion-controlled. The results show that the Mo-doped carbon nanowires transformed the interfacial electrochemical reaction from a surface-controlled process into a diffusion-controlled process, and promoted the rapid electrochemical reaction on the electrode surface.

Subsequently, the electrochemical properties of the Mo-CNFs and CNFs were evaluated in the *S. putrefaciens* CN32 cell suspension with 18 mM lactate, which served as the electron donor. First, electrodes were allowed to discharge at 0.2 V for 12 h, and the outcomes (Figure [4b](#page-7-0)) indicated that the discharge curves of the Mo-CNF anodes rose first and then remained stable. The Mo-CNF II anode had the maximal discharge current (422.60 µA cm⁻²), while the CNF discharge current was 165.80 µA cm⁻². Then, a CV test was conducted with a sweep speed of 1 mV s $^{-1}$, and the outcomes are depicted in Figure [4a](#page-7-0). The results show that all CVs had steady-state catalytic currents at −0.45 V. The −0.45 V position corresponded to the catalytic peak of the indirect electron transfer (IET) mediated by bacterial autocrine flavin-like electron mediators. Furthermore, Mo-CNF II had the highest catalytic current (0.75 mAcm−²), while that of CNFs was 0.22 mAcm−² . The DPV curves displayed redox peaks at the 0 and -0.45 V potentials, as displayed in Figure [4c](#page-7-0). The 0 V position corresponded to the catalytic peak of direct electron transfer (DET). The Mo-CNF II electrode had the maximal redox peak current density at both the 0 and −0.45 V potentials, which agreed with the results of the CVs. Figure [4d](#page-7-0) depicts the EIS results, which show that the $R_{\rm ct}$ of Mo-CNFs was much lower than that of CNFs (598.20 Ω), and that Mo-CNF II had the lowest interfacial electron transfer impedance (13.34 Ω). The results show that the $Mo₂C$ modification of the electrode interface could remarkably accelerate the electron transfer rate between bacteria and the electrode.

To assess the bioelectrocatalytic performance of the Mo-CNFs and CNFs, they were employed as anodes in H-type double-chamber MFCs inoculated with *S. putrefaciens* CN32 cell suspension. Figure [5a](#page-7-1) shows that the Mo-CNFs and CNFs MFCs were discharged with an external 2 kΩ resistor. The MFCs' output current density using the Mo-CNF II anode was stable at 0.20 mA cm $^{-2}$ in the third discharge cycle, while the CNFs' anode was 0.14 mA cm⁻², demonstrating that the Mo₂C-modified carbon nanofibers remarkably improved the bioelectrical output. After three rounds of discharge, the polarization and power-density curves of the anode were calculated by changing the resistance value of the external resistor (80,000–500 Ω) when the voltage was stable. Figure [5b](#page-7-1) shows that the maximal power

densities of Mo-CNFs I–III were 1019.70, 1287.38, and 986.05 mW m⁻², respectively, which were significantly greater than the output power density of the undoped CNFs (649.69 mW m⁻²). Moreover, Mo-CNF II had a maximal power density that was twice that of the CNFs. We compare the performance of the anode materials prepared in this work with previously reported anode materials in Table S1 (Supplementary Materials). Figure [5c](#page-7-1),d show the results of the characterization of Mo-CNF and CNF anode biofilms after three rounds of discharge. The results show that the Mo-CNF II anode was enriched with a large number of bacteria, forming a good biofilm, but the microorganisms on the surface of the CNF anode were scarce. The improved biocompatibility of molybdenum-doped carbon nanofibers can be attributed to the good biocompatibility of $Mo₂C$. Moreover, the presence of the Mo2C nanoparticles increased the number of active sites and promoted the adsorption of extracellular bacterial proteins. The aforementioned investigations demonstrate that the Mo-doped nanowires enhanced the anode's biocompatibility and raised its level of bioelectric output.

Figure 3. Electrochemical behavior of Mo-CNF and CNF electrodes in 0.1 M PBS containing 2 µM FMN. (a) CVs (30 mV·s⁻¹, insert: histogram of redox peak current density). (b) DPV curves. (**c**) Nyquist curves. (**d**) CV curves and peak current density at various scan rates of the Mo-CNF II electrode.

Figure 4. Different electrodes electrochemical behaviors in the *S. purefaciens* CN32 cell suspension. (**a**) CVs (1 mV s−¹). (**b**) Discharge, (**c**) DPV, and (**d**) Nyquist curves.

Figure 5. Performance comparison of different electrodes in MFCs. (**a**) Cyclic discharge curves. (**b**) Power density and polarization curves. SEM image of bacterial cells grown on (**c**) Mo-CNF II and (**d**) CNF anode after three-cycle discharge.

4. Conclusions

In this study, an $Mo₂C$ nanoparticle-modified nanowire was obtained via electrospinning and high-temperature carbonization. Then, it was employed as an MFC anode to boost the generation of bioelectricity. The Mo-CNF II anode in the MFCs achieved the maximal power density of 1287.38 mW m⁻², which was double that of the unmodified CNF anode. The doping of $Mo₂C$ improved the bacteria-to-electrode interfacial electron transfer, which enhanced the biocompatibility of the electrode surface, added more active sites, and accelerated the direct electron transfer process. Additionally, Mo-CNF II had a good response to FMN and promoted a flavin-like mediated indirect electron transfer. This work provides new insights into the application of electrostatic spinning technology in microbial fuel cells and the preparation of anode materials on a large scale.

Supplementary Materials: The following supporting information can be downloaded at: [https:](https://www.mdpi.com/article/10.3390/ma16062479/s1) [//www.mdpi.com/article/10.3390/ma16062479/s1,](https://www.mdpi.com/article/10.3390/ma16062479/s1) Figure S1. High-resolution C 1s XPS spectrum of Mo-CNFs II. Figure S2. High-resolution (a) N 1s and (b) O 1s XPS spectrum of Mo-CNFs II. Figure S3. CV curves and peak current density at different scan rates of CNF electrode. Table S1. Performance comparison of anode materials prepared in this work and previously reported anode materials. References [\[26](#page-9-14)[–31\]](#page-9-15) are cited in the Supplementary Materials.

Author Contributions: Conceptualization, Z.S.; Formal analysis, Z.S.; Investigation, Z.S., X.W. (Xiaohai Wang) and Z.W.; Data curation, X.W. (Xiaoshuai Wu) and X.L.; Writing—original draft, X.L.; Writing—review & editing, X.W. (Xiaoshuai Wu) and C.M.L.; Funding acquisition, X.W. (Xiaoshuai Wu). All authors have read and agreed to the published version of the manuscript.

Funding: We gratefully acknowledge the National Key Research and Development Program of China (project no. 2021YFA0910400), the Natural Science Research Foundation of Jiangsu Higher Education Institutions (grant no. 19KJB150038, 21KJB180018) and the Natural Science Foundation of Suzhou University of Science and Technology (XKQ2018014, XKZ2019011). This work was also funded by the Open Foundation of the Jiangsu Key Laboratory for Biochip and Medical diagnosis.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Informed consent was obtained from all subjects involved in the study.

Conflicts of Interest: The authors declare no conflict of interest.

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