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# Fabrication of WS<sub>2</sub>/WSe<sub>2</sub> Z-Scheme Nano-Heterostructure for Efficient Photocatalytic Hydrogen Production and Removal of Congo Red under Visible Light

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Abstract: In this study, a novel tungsten disulfide/tungsten diselenide (WS<sub>2</sub>/WSe<sub>2</sub>) heterojunction photocatalyst by a facile hydrothermal process with great capable photocatalytic efficiency for hydrogen evolution from water and organic compound removal was discussed. The WS<sub>2</sub>/WSe<sub>2</sub> heterojunction photocatalyst to form heterojunctions to inhibit the quick recombination rate of photo-response holes and electrons is reflected to be a useful method to enhance the capability of photocatalysis hydrogen production. The hydrogen production rate of the WS<sub>2</sub>/WSe<sub>2</sub> photocatalyst approach is 3856.7 µmol/g/h, which is 12 and 11 folds the efficiency of bare WS<sub>2</sub> and WSe<sub>2</sub>, respectively. Moreover, the excellent photocatalytic performance for Congo Red (CR) removal (92.4%) was 2.4 and 2.1 times higher than those of bare  $WS_2$  and  $WSe_2$ , respectively. The great photocatalytic efficiency was owing to the capable electrons and holes separation of  $WS_2/WSe_2$  and the construction of Z-scheme heterostructure, which possessed vigorous photocatalytic oxidation and reduction potentials. The novel one-dimensional structure of WS<sub>2</sub>/WSe<sub>2</sub> heterojunction shortens the transport pathway of photo-induced electrons and holes. This work provided an insight to the pathway of interfacial separation and transferring for induced charge carriers, which can refer to the interfacial engineering of developed nanocomposite photocatalysts. It possessed great capable photocatalytic efficiency of hydrogen production and organic dye removal. This study offers an insight to the route of interfacial migration and separation for induced charge carriers to generating clean hydrogen energy and solve environmental pollution issue.

Keywords: Z-scheme heterojunction; hydrogen production; photocatalytic; visible light

## 1. Introduction

Owing to the consecutive development of broad industrialization, the global economy has made significant evolution. For these causes, it is essential to develop suitable methods to obtain energy from renewable sources such as solar energy. In addition, solar energy is a pure and abundant source, gaining increased attention over the last 20 years [1]. Hydrogen could be adopted as a clean replacement to fossil fuel, which presents high energy content and lesser environmental pollution [2]. While solar energy is a renewable resource, photocatalytic decomposition of water to generate hydrogen is a promising route to make clean energy and solve the environmental issues [3,4]. On the other hand, the poor capability of hydrogen evolution, complex fabrication procedure, and high cost of the photocatalysts deeply limit its utilizations. Therefore, great photo-induced capability, no secondary pollution, and low-cost process for decomposing water (H<sub>2</sub>O) into hydrogen (H<sub>2</sub>), photocatalytic decomposition are the highlights of the present research [5]. In addition, a probable photocatalytic hydrogen evolution route essential wide and vigorous light absorption to generate more amount of charge carriers, and useful electrons and holes



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**Copyright:** © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). separation efficiency as well as proper energy band gaps to induced the response of reducing  $H_2O$  to  $H_2$ .

In photocatalysis, the primary points affecting the photocatalytic capability include the visible light application potential, the migration, and separation performance of photogenerated electrons-holes [6]. To develop the photocatalytic process efficiency, many strategies were presented, such as fabricating nano-scale structures, varied morphologies, and synthesizing with semiconductor or co-catalyst semiconductor. It has been noted in photocatalysts that it is critical for a single component to own better photocatalysis capability to benefit the functional need because of its confined high recombination rate of photoexcited electron and hole pairs and light absorption efficiency. Z-scheme photocatalytic activities can apply renewable solar light energy to produce clean hydrogen and environmental pollution treatment, which is reflected to be a probable method to solve environmental and energy issues [7]. Therefore, the formation of heterostructures is proposed because they can combine the benefits of each material and improve the separation rate of photoexcited carries [8]. It is also indicated that the contact surface area of heterojunction interface provides enhancement significantly to the photoresponse capability in the route of photocatalytic redox reaction [9].

Transition metal tungsten sulfides have been broadly researched as excellent semiconductor photocatalysts [10,11]. Because of their outstanding photoelectric capability and particular band gap properties, they possess better potential in the field of photocatalytic activity. As favorable photocatalysts, WS<sub>2</sub> and WSe<sub>2</sub>, have been usually studied for photocatalytic hydrogen production owing to their appropriate band gap and proper energy band gap position. Then, the inactive activities of photoinduced charge carries separation confined the photocatalytic hydrogen energy performance of bare WS<sub>2</sub> and WSe<sub>2</sub> component [12]. On the other hand, the rapid recombination of photoexcited charge carriers of bare WS<sub>2</sub> and WSe<sub>2</sub> seriously suppressed its efficient utilization in the environment. Although, no elaborate research has been conducted on the heterostructures of WS<sub>2</sub> and WSe<sub>2</sub> for its potential in photocatalytic capabilities. Accordingly, modulation of the interface charge delivery toward Z-scheme route for excellent redox efficacies and operation of interfacial match for rapid removal of photoinduced electron-holes pairs has become a crucial requirement.

In this case, two necessities are essential to quickly separate the electron-hole pairs on the reduced material and the oxidized material. Moreover, the compact interface of the heterojunction can also enhance the removal of photoinduced charge carriers, improve the lifetime of photoresponse carriers, and advance the reduction of  $H^+$  on the photocatalyst surface. Based on the above studies, the formation of  $WS_2/WSe_2$  heterojunction photocatalyst could be favorable for the powerful separation and transfer of photoexcited charge carriers and for improving the photocatalytic performance. The photocatalytic activity of  $WS_2/WSe_2$  heterojunction photocatalyst was systematically studied by photocatalytic hydrogen evolution and CR removal under visible light irradiation. Our research study will open a new route in designing novel nanocomposites with great photocatalytic performance toward the photo-response hydrogen production rate and the removal of organic environmental pollutants.

## 2. Results

#### 2.1. Microstructure Characterization

Figure 1 displays the XRD patterns of the WS<sub>2</sub>, WSe<sub>2</sub>, WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20 samples recorded by combining various amounts of WSe<sub>2</sub>. As exhibited in the results, with the addition of WSe<sub>2</sub>, the diffraction peak position of the WS<sub>2</sub>/WSe<sub>2</sub> does not obviously change, and no other impurity peaks appear. As the counting of WSe<sub>2</sub> is increased from 5 to 20 mg, the XRD analysis of WS<sub>2</sub>/WSe<sub>2</sub>-X (X = 5, 10, and 20) become increasingly shifted within the range of 13.5° to 14.3° corresponding to (002) planes of WS<sub>2</sub> and WSe<sub>2</sub> (JCPDS 84-1398 and 96-901-2194) [13,14]. The major characteristic peaks of the (002) plane can be observed in the patterns of WS<sub>2</sub>/WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and

WS<sub>2</sub>/WSe<sub>2</sub>-20. In particular, contrast with the XRD analysis of WSe<sub>2</sub>, the overall diffraction intensities of  $WS_2/WSe_2$  heterojunctions were altered to lower peaks (see Figure 1 inset). With the increase of  $WSe_2$  content, the position of the characteristic peaks of the (002) plane in the  $WS_2/WSe_2-5$ ,  $WS_2/WSe_2-10$ , and  $WS_2/WSe_2-20$  patterns gradually shifts toward low 2 theta degree, which proves that the WS<sub>2</sub>/WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20 heterojunctions have been successfully constructed. This result also indicates the successful chemical combination of WSe<sub>2</sub> and WS<sub>2</sub> in WS<sub>2</sub>/WSe<sub>2</sub> heterostructure. The results exhibited that the facile hydrothermal process could achieve a heterostructure without the destruction of the lattice structure. UV–Vis was conducted to analyze the optical properties of the samples. Figure 1b displays that the absorption edge of  $WS_2/WSe_2-X$  ( $WS_2/WSe_2-5$ ,  $WS_2/WSe_2-10$ , and  $WS_2/WSe_2-20$ ) is about 450~550 nm, indicating that the  $WS_2/WSe_2-X$ impacts well the visible light range. The absorption edge of  $WS_2/WSe_2-20$  heterojunction has a redshift compared to  $WS_2/WSe_2$ -5 due to the interaction between  $WS_2$  and  $WSe_2$ . It means the generated WS<sub>2</sub>/WSe<sub>2</sub>-20 nanocomposite displayed the enhanced absorption ability of visible light, which was favorable to generate more photoexcited electrons and holes for beginning redox responses. The band gap (Eg) energies of WS<sub>2</sub>/WSe<sub>2</sub>-X are estimated by the Tauc equation  $((\alpha h v)^2 = A (hv - Eg)^n)$  [15]. Figure 1b inset reveals that the Eg of WS<sub>2</sub>/WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20 are 2.56, 2.66 and 2.74 eV, respectively.



**Figure 1.** (a) XRD patterns of the as-prepared samples; (b) UV–vis spectra of WS<sub>2</sub>/WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20 samples; plots of  $(\alpha h \upsilon)^2$  versus h $\upsilon$  for the bandgap energies of the photocatalysts in the inset.

The morphology and microstructure of the WS2/WSe2-5, WS2/WSe2-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20 samples were also measured by SEM and TEM. Figure 2a,b shows the SEM image of pure  $WS_2$  and  $WSe_2$  samples. It is obvious that the bulk  $WS_2$  and  $WSe_2$ materials are severely agglomerated in the range of average width  $20-50 \mu m$ . As shown in Figure 2c, the WS<sub>2</sub>/WSe<sub>2</sub>-20 has a particle size of nearly  $3-5 \mu m$ , and sphere-like morphology on its surface are developed by the rough stacking of particles. The rough organization of particles contribute not only to volume of space but also to endow support for the firm attaching of WS<sub>2</sub> and WSe<sub>2</sub> samples. Noticeably, compared with WS<sub>2</sub> and WSe<sub>2</sub>, the surface of WS<sub>2</sub>/WSe<sub>2</sub> reveals obviously loose, which will be profitable for WS<sub>2</sub>/WSe<sub>2</sub>-20 to serve a better specific surface area and will be easier to combine with other materials. The TEM images of  $WS_2/WSe_2$ -20 composites are shown in Figure 2d–i, it could be apparently observed that the  $WS_2$  was combined by a  $WSe_2$  with an evident interface between two materials, implying the presence of a heterojunction structure. In addition, Figure 2e exhibited the HRTEM image of WS<sub>2</sub>/WSe<sub>2</sub>-20 samples, and the interlayer distances of 0.28 and 0.34 nm were corresponding to the (002) lattice plane of  $WS_2$  and (110) lattice planes of WSe2, respectively. The elemental of W, S, and Se were smoothly dispersed on the entire composite in EDS mapping (Figure 2f-i), suggesting the presence of WS<sub>2</sub> and WSe<sub>2</sub> in nanocomposites. Moreover, it can be apparently noted that the areas of the elemental distribution of WS<sub>2</sub> with sphere shape were smaller than that of WSe<sub>2</sub> sample. It also suggested the WSe<sub>2</sub> nanoparticles had adhered to the surface of WS<sub>2</sub>, confirming the heterojunction with WS<sub>2</sub>/WSe<sub>2</sub> was well fabricated. This heterostructure would mainly expand the contact surface area between WSe<sub>2</sub> and WS<sub>2</sub>, thus improving the absorption ability of small molecules (organic pollutants and water molecules) [16,17]. The results reveal that improving the photocatalytic activity of the composite accelerates the separation of charge carriers. Therefore, we confirmed the formation of the WS<sub>2</sub>/WSe<sub>2</sub>-20 heterostructure.



**Figure 2.** SEM images of (**a**) WS<sub>2</sub>, (**b**) WSe<sub>2</sub>, and (**c**) WS<sub>2</sub>/WSe<sub>2</sub>-20, TEM images of (**d**) WS<sub>2</sub>/WSe<sub>2</sub>-20; HRTEM images (**e**) and element mapping images (**f**–**i**) of WS<sub>2</sub>/WSe<sub>2</sub>-20.

X-ray photoelectron spectroscopy (XPS) was performed to check the chemical composition and surface element status in the as-prepared  $WS_2/WSe_2-20$  sample. Figure 3 displays the high-resolution XPS spectra of different elements for WSSe-3 sample. As displayed in Figure 3a, the feature peaks of W could be divided into three peaks, among which the feature peaks at 32.1, 34.2, and 37.5 eV refer to the W 4f 7/2, W 4f 5/2, and W 5p 3/2, respectively. In Figure 3b, a high-resolution spectrum of the S 2p peak can be into two peaks of S 2p3/2 and S 2p1/2 spin–orbit doublets at 162.8 and 161.7 eV, implying that S<sup>2-</sup> existed in the WS<sub>2</sub>/WSe<sub>2</sub>-20. Then, two divided feature peaks of Se 3d in WS<sub>2</sub>/WSe<sub>2</sub>-20 examined at 53.8 eV and 54.6 eV can be referred to Se 3d5/2 and Se 3d3/2 (Figure 3c), displaying the chemical status of Se element was  $Se^{2+}$ , relative to the former report [13,14]. For the above XPS test events of the as-fabricated photocatalyst, it exhibited that the nanocomposite was well prepared by a facile hydrothermal process and there was a vigorous interaction between WS<sub>2</sub> and WSe<sub>2</sub>, which was in good agreement with the XRD results. To check the BET-specific surface area of the nanocomposites, N2 adsorption-desorption tests were achieved (Table 1). The pore structure and specific surface area of the samples analyze the size of the exposed active area during the photocatalytic activity and have a definite effect

on the photocatalytic efficiency of water splitting to generate hydrogen energy. The specific surface areas of WSe<sub>2</sub>, WS<sub>2</sub>, and WS<sub>2</sub>/WSe<sub>2</sub>-20 are 33.6, 23.3, and 68.6 m<sup>2</sup> g<sup>-1</sup>, respectively, displaying that the specific surface area of WS<sub>2</sub>/WSe<sub>2</sub>-20 increases significantly after combining with WSe<sub>2</sub>. Moreover, compared with WS<sub>2</sub>/WSe<sub>2</sub>-20 ( $0.347 \text{ cm}^3 \text{ g}^{-1}$ ), the average pore diameter (9.5 nm) of WS<sub>2</sub>/WSe<sub>2</sub>-20 is improved. On the other hand, it provides to effectively separating the photoexcited charge carries, thus will also demonstrate more active sites for promoting photocatalytic hydrogen evolution capability [18].



Figure 3. XPS survey spectra of  $WS_2/WSe_2$ . High-resolution XPS spectra of the (a) W 4f, (b) S 2p, and (c) Se 3d.

| Samples                               | Specific Surface<br>Area (m²/g, BET) <sup>a</sup> | Total Pore Volume<br>(cm <sup>3</sup> /g, BET) <sup>b</sup> | Average Pore<br>Diameter (nm, BJH) <sup>b</sup> |
|---------------------------------------|---|---|---|
| WSe <sub>2</sub>                      | 33.6±5  | $0.182{\pm}0.02$  | $38.4{\pm}5$                                    |
| $WS_2$                                | $23.3 \pm 5$                                      | $0.113 {\pm} 0.02$  | $48.5 \pm 5$                                    |
| WS <sub>2</sub> /WSe <sub>2</sub> -20 | $68.6 \pm 5$                                      | $0.347 {\pm} 0.02$  | 9.5±3   |

**Table 1.** The physical adsorption factors of the photocatalysts of  $WS_2$ ,  $WSe_2$ , and  $WS_2/WSe_2$ -20 were measured three times.

<sup>a</sup> Received from BET analysis. <sup>b</sup> Relative pressure (P/P<sub>0</sub>) was 0.99.

#### 2.2. Hydrogen Production Behavior

As displayed in Figure 4a,b, the hydrogen production rate of  $WS_2/WSe_2-20$  enhanced by 13 and 12 times of WS<sub>2</sub> and WSe<sub>2</sub>, respectively. The total production of H<sub>2</sub> followed the order:  $WS_2/WSe_2-20$  (3856.7 µmol g<sup>-1</sup>) >  $WS_2/WSe_2-10$  (3546.2 µmol g<sup>-1</sup>) >  $WS_2/WSe_2-5$  $(2743.5 \ \mu mol \ g^{-1}) > WSe_2 \ (334.6 \ \mu mol \ g^{-1}) > WS_2 \ (315.4 \ \mu mol \ g^{-1}).$  Adding excessive photocatalyst, owing to the WS2/WSe2-20 nanocomposites aggregation, suggests a reduced surface area of WS<sub>2</sub>/WSe<sub>2</sub>-20 and thus hindering the transfer ability of the charge carries [19]. However, all  $WS_2/WSe_2$  heterojunction photocatalyst shows improved photocatalytic hydrogen evolution efficiency, implying that the electron localization and boost interaction at the WS<sub>2</sub>/WSe<sub>2</sub>-20 sample interface capably accelerate charge transfer and separation ability. Moreover, excessive nanocomposites may result in recombination for electron-hole pairs, hence decreasing the photocatalytic efficiency. By contrast, a slight amount of photocatalyst will cause the interface between the WS2 and WSe2 too restricted to separate and transport the photoexcited electrons and holes pairs, which cannot mainly suppress the recombination rate of photoexcited charge carriers. As exhibited in Figure 4c, with the adding of  $WS_2/WSe_2$ -20 heterostructures, the amount of hydrogen production enhances gradually until the amount of  $WS_2/WSe_2-20$  nanocomposites approaches 30 mg. Moreover, the hydrogen production rate is gradually suppressed with increasing WS<sub>2</sub>/WSe<sub>2</sub>-20 nanocomposites. This reveals that the water molecules in the solution could be adsorbed by the photocatalysts and mainly employed when the amount of WS2/WSe2-20 nanocomposites is suitable. A mass of WS<sub>2</sub>/WSe<sub>2</sub>-20 nanocomposites over 30 mg results in excess photocatalyst stock in the mixture solution, inhibiting efficient charge carries transport and photon adsorption of the nanocomposite and hence decreasing the amount of photons on the sample surface and suggesting decreased hydrogen evolution efficiency. To investigate the reusability and the chemical stability of  $WS_2/WSe_2$ -20 nanocomposite in hydrogen evolution rate, cycling test was carried out at the same situations. As displayed in Figure 4d, the number of hydrogen generated in the fourth cycle was obviously decreased compared to the original case. The hydrogen production rate endured at nearly 96.2% compared to the original matter. When 20 mg WS<sub>2</sub>/WSe<sub>2</sub>-20 nanocomposites were increased again, the number of hydrogen production rate was evidently increased, implying that the photocatalysis of  $WS_2/WSe_2-20$  nanocomposites was the primary factor for the increase in hydrogen production. At the end of the cycling performance analysis, the XRD analysis of the sample was conducted again. As displayed in Figure 4e, the crystal structure and morphology of WS<sub>2</sub>/WS<sub>2</sub>-20 do not change obviously before and after the cycling test, which further demonstrates the excellent chemical stability and repeatability of the nanocomposites in photocatalytic hydrogen evolution efficiency. Therefore, the hydrogen evolution rate of the WS<sub>2</sub> mixtures is 315.4 lmol  $g^{-1} h^{-1}$ , which is significantly lower than the WS<sub>2</sub>/WSe<sub>2</sub>-X heterojunctions. The above results implied that the as-prepared  $WS_2/WSe_2$ -20 nanocomposite can be applied as a potential hydrogen generating photocatalysts.



**Figure 4.** H<sub>2</sub> evolution of (**a**) WS<sub>2</sub>, WSe<sub>2</sub>, and WS<sub>2</sub>/WSe<sub>2</sub>-20 and H<sub>2</sub> production comparison of (**b**) WS<sub>2</sub>/WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20. (**c**) H<sub>2</sub> production with various usage amounts of WS<sub>2</sub>/WSe<sub>2</sub>-20. (**d**) H<sub>2</sub> production of the cycling performance, (**e**) XRD analysis of WS<sub>2</sub>/WSe<sub>2</sub>-20 before and after the hydrogen production activity.

## 2.3. CR Degradation Behavior

It can be noted from Figure 5a that the feature absorption intensity of CR at 346 and 497 nm apparently decreased with increasing irradiation time, confirming that CR was practically degraded by the  $WS_2/WSe_2$ -20 nanocomposite. The above photocatalytic results revealed that decreased preparation of  $WS_2/WSe_2$ -20 nanocomposite accelerated the separation rate and space charge transfer distance of photoexcited charge carries, thus improving the photodegradation performance [20]. To further check the photodegradation performance of the photocatalyst, the photodegradation of the CR pollutant was conducted. Figure 5b displays the adsorption equilibrium test in dark for 20 min, the adsorption number of CR via all

photocatalysts is negligible. The blank test shows that the self-photocatalytic of CR could be ignored, indicating that the property of CR is stable in the photocatalytic activity. WS<sub>2</sub> and WSe<sub>2</sub> have poor capability for CR removal with visible light irradiation, which are 38.6%, and 46.5% within 60 min, respectively. Noticeably, WS<sub>2</sub>/WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20 approaches the better photocatalytic capability for CR removal, which is 82.5%, 88.3%, and 92.4%. Corresponding to photocatalytic hydrogen evolution, the operation series of photocatalytic MB removal is as follows: WS<sub>2</sub>/WSe<sub>2</sub>-20 > WS<sub>2</sub>/WSe<sub>2</sub>-10 > WS<sub>2</sub>/WSe<sub>2</sub>-5 > WSe<sub>2</sub> > WS<sub>2</sub>. Moreover, the CR degradation performance of WS<sub>2</sub>/WSe<sub>2</sub>-20 is 2.4 and 2 times that of WS<sub>2</sub> and WSe<sub>2</sub> samples, respectively. Figure 5c displays the cycling experiment of WS<sub>2</sub>/WSe<sub>2</sub>-20 for photo degradation CR pollutant. The photodegradation efficiency decreases slightly from 92.4% to 82.1% after six repeated cycling tests, which verify that WS<sub>2</sub>/WSe<sub>2</sub>-20 has excellent stability and repeatability in the photo degradation CR process.



Figure 5. Cont.



**Figure 5.** (a) CR spectra adsorption changes during the photocatalytic process, (b) Photodegradation curves of CR, (c) cycling test for  $WS_2/WSe_2-20$  sample, (d) reactive species trapping tests, and (e) DMPO spin-trapping ESR spectra of DMPO- $O_2^-$ . (three repeat tests to estimate error bars), (f) electrochemical impedance spectroscopy of the as-fabricated photocatalysts, (g) photoluminescence spectroscopy (PL), and (h) transient photocurrent curves of WS<sub>2</sub>, WSe<sub>2</sub>, and WS<sub>2</sub>/WSe<sub>2</sub>-20.

To further confirm the active species of photocatalyst, the influences of various capture agents during the photocatalytic activity were proved. The capture agents are isopropanol (IPA) for OH, benzoquinone (BQ) for  $O_2^-$ , and EDTA-2Na for h<sup>+</sup>. When IPA is adopted into the photocatalytic activity, the photocatalytic reaction slightly diminishes, implying that OH has weak effect on CR removal in Figure 5d. On the other hand, when BQ or EDTA-2Na is adopted into this test, the photocatalytic efficiency decreases obviously, implying that  $h^+$  and  $O_2^-$  are the major active species [21]. Furthermore, the ESR analysis (Figure 5e) indicates that  $WS_2/WSe_2-20$  has the DMPO- $O_2^-$  adduct signal, which also verifies that  $O_2^-$  exists in the photocatalytic activity. Electrochemical impedance spectroscopy (EIS) was performed to check the transportation and charge carries separation rate of nanocomposite surface [22]. In Figure 5f, the Nyquist semicircle radius of  $WS_2/WSe_2-20$  is significantly smaller than that of pure WS<sub>2</sub> and WSe<sub>2</sub>, verifying that the transfer resistance of photoexcited electrons and holes in WS<sub>2</sub>/WSe<sub>2</sub>-20 is the smallest among these catalysts. It was indicated that the WS<sub>2</sub>/WSe<sub>2</sub>-20 improved the charge transfer ability for composite. To further study the charge carries transfer and separation capability of the nanocomposites, a PL measurement was conducted. Figure 5g exhibits the PL spectra of WS<sub>2</sub>, WSe<sub>2</sub>, and WS<sub>2</sub>/WS<sub>2</sub>-20 under 420 nm excitation. Compared with WS<sub>2</sub> and WS<sub>2</sub>, the PL peak intensity of WS<sub>2</sub>/WSe<sub>2</sub>-20 decreases significantly. The PL intensity decreased in the order of WS<sub>2</sub>/WSe<sub>2</sub>-20 < WSe<sub>2</sub> < WS<sub>2</sub>, clearly describing that WS<sub>2</sub>/WSe<sub>2</sub>-20 sample presented the better the electrons and holes separating capability, which was in good agreement with above EIS analysis results. The result demonstrated that the  $WS_2/WSe_2$ -20 nanocomposite can mainly transfer photoexcited electrons and holes to inhibit their recombination ability. As shown in Figure 5h, these results demonstrate that the bare WS<sub>2</sub> and WSe<sub>2</sub> present a poor transient photocurrent response with visible light irradiation for several on–off cycles. This low photocurrent response of bare components can be attributed to the quick recombination rate of the photoexcited charge carriers. WS<sub>2</sub>/WSe<sub>2</sub>-20 exhibits excellent photocurrent intensity than bare WS<sub>2</sub> and WSe<sub>2</sub>, which implies enhanced charge carrier separation and migration in the heterojunction photocatalyst. All above results apparently verify that the photoexcited electrons and holes in WS<sub>2</sub>/WSe<sub>2</sub>-20 could be effectively transferred and separated.

Accordingly, the above results of different data analysis and previous investigations, the probable photoexcited charge carries transport paths for the enhanced photocatalytic efficiency are recommended in Figure 6. According to the prior study, the bandgap value of the made WSe<sub>2</sub> sample was 1.76 eV, and the bandgap value of WS<sub>2</sub> was 1.92 eV [23,24]. The VB-XPS analysis was also concluded to establish the VB levels of WSe<sub>2</sub> and WS<sub>2</sub>, and the results are displayed in Figure 6a–c. Obviously, the VB levels of  $WSe_2$  and  $WS_2$ were calculated to be 2.03 and 1.28 V, respectively. Furthermore, the CB potential of WSe<sub>2</sub> and WS<sub>2</sub> were calculated by the formula (Eg =  $E_{VB} - E_{CB}$ ), which were 0.27 eV and -0.64 eV, respectively. Both WS<sub>2</sub> and WSe<sub>2</sub> were induced and excited charge carries on their valence bands (VB) and conduction bands (CB) under visible light irradiation conditions owing to their narrow band energies. Accordingly, the photo-induced charge carries were transferred from VB and CB of  $WS_2$  to the consistent positions of  $WSe_2$ due to band gap energy difference. Because of the nanocomposites, the confined charge carries with high energy enhanced in the heterostructure can be transported rapidly to the photocatalyst surface across the tunneling influence to contribute to the chemical redox activity. Therefore, some section of the photoexcited electrons and holes could also directly contribute to the redox reaction owing to the nanocomposite developed in heterojunction  $WS_2/WS_2-20$  nanomaterials. This result was similar to the previous studies [25–27]. The photoexcited charges collected on surface of photocatalyst can directly contribute to the reduction of  $H_2O$  to  $H_2$  under the hydrogen production process. Then the nanocomposite was the main active site for reaction. For the photodegradation process of CR, the electrons on the surface of photocatalyst respond to  $O_2$  to generate  $O_2^-$  and  $H^+$  is reduced to  $H_2$ , which was the primary active site for degradation of CR and hydrogen production. At the same time, the holes created as a result of directly oxidation on CR organic molecule cause  $H_2O$  to generate OH. This is in good agreement with the ESR trapping results. The possible mechanism for the enhanced photocatalytic activity from above experiment analysis were as follows: (1) The extended specific surface area was profitable for creating more active species and better photoactivity; (2) the improved visible light absorption range can improve the light capture efficiency to promote photocatalytic application; (3) the synthesized heterostructure with preferred interface contact offered better transfer pathway for charge carries, which can promote photoexcited electrons and holes separation capability to enhance the photocatalytic efficiency.



**Figure 6.** UPS spectrum (**a**)  $WSe_2$ , (**b**)  $WS_2$ , and (**c**) band structure diagram of  $WS_2/WSe_2$ , (**d**) Schematic diagram of the proposed Z-Scheme photoelectron mechanism of  $WS_2/WSe_2$  photocatalytic system under visible light illumination.

## 3. Materials and Methods

## 3.1. Preparation of Materials

Sodium tungstate dehydrate (Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O,  $\geq$  99%), the L-cysteine (C<sub>3</sub>H<sub>7</sub>NO<sub>2</sub>S,  $\geq$  97%), tungstate selenium powder (WSe<sub>2</sub>,  $\geq$  99.5%), hydrochloric acid (HCl,  $\geq$  37%), Congo Red (CR), and ethanol (C<sub>2</sub>H<sub>5</sub>OH,  $\geq$  99.7%) were obtained from Sigma-Aldrich (St. Louis, MO, USA). Deionized (DI) water (resistivity  $\geq$  18.0 M $\Omega$ ) was used as experimental water source in further experimental procedures. All the reagents were applied without any further purification.

## 3.2. Synthesis

The water-soluble WS<sub>2</sub>/WSe<sub>2</sub> heterojunction photocatalyst was synthesized by a facile hydrothermal process. First, 65 mg of Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O was added to 20 mL of DI water with stirred for 10 min. Afterwards, 0.1 M HCl was used to adjust the pH to 6.8. Next, 25 mg of L-cysteine, tungstate selenium powder and 50 mL of water were added into the mixture solution by stirring for 15 min. The mixed solution was transferred to a Teflon-lined

autoclave after vigorous stirring for 20 min and kept at 180 °C for 36 h in an oven. After the reaction was completed, the precipitates were centrifuged for 20 min at the 10,000 rpm. The resulting product was washed with alcohol, deionized water, and acetone several times. The washed precipitates were dried in a vacuum drying oven (overnight). Eventually, a series of WS<sub>2</sub>/WSe<sub>2</sub> nanocomposite with diverse WSe<sub>2</sub> mass ratios of 5 wt%, 10 wt% and 20 wt% was arranged, and served as WS<sub>2</sub>/WSe<sub>2</sub>-5, WS<sub>2</sub>/WSe<sub>2</sub>-10, and WS<sub>2</sub>/WSe<sub>2</sub>-20, respectively. As displayed in Scheme 1, the fabrication of WS<sub>2</sub>/WSe<sub>2</sub> heterojunction was described by the hydrothermal process.



Scheme 1. The rough synthetic route of WS<sub>2</sub>/WSe<sub>2</sub> composites.

#### 3.3. Characterization

The field emission scanning electron microscopy (SEM) and transmission electron microscopy (TEM) were carried out to observe the microstructure and morphology of the samples. The chemical composition of the samples carried out by X-ray photoelectron spectroscopy (Thermo Fisher Scientific, Waltham, MA, USA) was also recorded. Brunauer–Emmett–Teller (BET) method was used to measure the pore volume, surface area, and pore distribution. The crystal phase and structure of the as-fabricated samples were analyzed with a powder X-ray diffractometer (Bruker D8 Advance) applying Cu Ka radiation (k = 0.154168 nm). The optical spectra were recorded by a UV–Vis reflectance spectrophotometer (Hitachi U-4100) and photoluminescence (PL). Photochemical property was determined on an electrochemical workstation with a 0.2-M Na<sub>2</sub>SO<sub>4</sub> solution as the electrolyte. Electron spin resonance (ESR) signals of the radicals captured were performed with visible light activity on a Bruker A300 spectrometer.

#### 3.4. Photocatalytic Hydrogen Evolution Tests

The hydrogen production test was checked by quantitatively and periodically extracting gas from a reactor with visible light irradiation. Driving nitrogen to replace the air in the reactor was sufficient to assure atmospheric pressure in the reactor. In this study, 10 mg of the photocatalyst was added to a mixture solution (50 mL) including 30 mmol Na<sub>2</sub>S and 20 mmol Na<sub>2</sub>SO<sub>3</sub> as a sacrificial agent. The mixture solution was evacuated to extract air entirely before activities. The pumping light source adopted was a 300 W Xe lamp with  $\lambda > 420$  nm filter. The mixture solution was stirred consecutively under the reaction. The amount of hydrogen production during light activity was determined using a gas chromatography (GC-7890, TCD) coupled with a thermal conductivity detector to measure the evolution of H<sub>2</sub>.

#### 3.5. Photocatalytic Activity Experiments

The photocatalytic capability of the prepared nanocomposite samples was used by their removal efficiency of an organic compound (CR) in aqueous solution by applying 300 W Xe lamp as a light source. Before photoreactions, 10 mg as-prepared sample was dispersed in 100 mL (10 ppm) CR solution and the reaction solution was continuously stirred for 30 min under darkness to achieve the adsorption–desorption balance between organic compound and photocatalysts surface. Accordingly, the reaction solution was irradiated with a Xe lamp source, and then 5 mL suspension was taken out at the same

conditions. After removing the photocatalyst by centrifugation, the concentration of CR was determined by the UV–Vis spectrophotometer. Moreover, trapping tests were further performed to photoexcited active species by using particular volume of scavengers 1,4-benzoquinone (BQ), EDTA-2Na, and 2-propanol (IPA) in the reaction system.

### 4. Conclusions

In conclusion, a novel Z-scheme WS<sub>2</sub>/WSe<sub>2</sub> heterojunction photocatalyst were synthesized by the hydrothermal process, which leads to the formation of intimated interfacial contact and heterojunction for energy conversion and environmental remediation. This work demonstrated that the as-fabricated  $WS_2/WSe_2$  proved beneficial to the migration of photoexcited charge carries, and can further enhance its photocatalytic hydrogen production activity and CR degradation with visible light irradiation. In addition, the photocatalytic efficiency of  $WS_2/WSe_2$ -20 rises to 3856.7 µmol g<sup>-1</sup> h<sup>-1</sup> (the H<sub>2</sub> production rate) and 92.4% (photocatalytic efficiency of CR degradation), which were 12 and 2.4 times higher than that of bare  $WS_2$ , respectively. The outstanding improvement of photocatalytic system might be due to the nanocomposite structure and the heterojunction effect between  $WS_2$ and WSe<sub>2</sub>. Moreover, the WS<sub>2</sub>/WSe<sub>2</sub>-20 nanocomposite exhibited better reusability and chemical stability. This study not only shows the improved light absorption efficiency of the prepared material, but also shows more photocatalytic active sites which shortens the electrons transport path. This study provides valuable reference to Z-scheme heterojunction photocatalysts in water splitting field, and the further exploration on photocatalytic mechanism for potential sustainable energy production in the future.

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