

Article

# The Effects of MnO<sub>2</sub> Addition on the Physical Properties of Pb(Ni<sub>1/3</sub>Nb<sub>2/3</sub>)O<sub>3</sub>-Pb(Zr,Ti)O<sub>3</sub>-Pb(Mg<sub>1/2</sub>W<sub>1/2</sub>)O<sub>3</sub>-BiFeO<sub>3</sub> Ceramics

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**Abstract:** In this paper, for the application to multi-layer piezoelectric devices capable of being used in piezoelectric speakers, Pb(Ni<sub>1/3</sub>Nb<sub>2/3</sub>)O<sub>3</sub>-Pb(Zr,Ti)O<sub>3</sub>-BiFeO<sub>3</sub> ceramics substituted with Pb(Mg<sub>1/2</sub>W<sub>1/2</sub>)O<sub>3</sub> were manufactured according to MnO<sub>2</sub> addition, and their physical properties were studied. At non-doped MnO<sub>2</sub> added specimen, the maximum values of piezoelectric properties were shown, respectively: the  $\epsilon_r$  of 2182,  $d_{33}$  of 513 pC/N, and  $k_p$  of 0.634. When taking into consideration the low dielectric constant and high  $d_{33}$  in case of increasing the numbers of multilayer in ceramics, the  $x = 0.2$  composition ceramics was suitable for the device application such as speaker using low-temperature sintering multilayer piezoelectric actuators.

**Keywords:** low temperature sintering ceramics;  $d_{33}$ ;  $k_p$ ; piezoelectric properties



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## 1. Introduction

Recently, the regulations on hazardous substances have been strengthened. In particular, the manufacture and disposal of environmental pollutants have been severely restricted across all the electrical and electronic products. In the field of electronic ceramics, PZT system ceramics sintered above 1200 °C can cause in general the environmental pollution caused by PbO, which exhibits the rapid volatilization characteristics at around 1000 °C, because the PZT ceramics are composed of PbO over 60–70% [1–4]. Accordingly, in order to solve these problems, it is required to develop the low temperature sintering ceramics with high physical properties which can restrain the deteriorations of dielectric and piezoelectric properties in case of sintering at low temperature. Pb(Ni<sub>1/3</sub>Nb<sub>2/3</sub>)O<sub>3</sub>-Pb(Zr,Ti)O<sub>3</sub> ternary ceramics widely used as a compositions for piezoelectric speakers due to its large piezoelectric  $d_{33}$  constant can enhance piezoelectric properties through increasing the  $k_p$  and  $d_{33}$  by the substitution of PMW or PZN instead of PNN [5–8]. Here, it is possible to sinter the ceramics at the temperature below 920 °C through the addition of sintering aids of ZnO, CuO, Li<sub>2</sub>CO<sub>3</sub>, and CaCO<sub>3</sub>.

Piezoelectric speakers have high sound pressure levels at high-frequency ranges, while they have low sound pressure levels at low-frequency ranges because of a solid state device. To increase the sound pressure at low frequencies, the frequency constant must be lowered, the number of stacked actuators must be increased in order to improve the generation power. Moreover, the piezoelectric  $d_{33}$  constant must be increased to increase the displacement. The composition ceramics applied to the LG G-8 smartphone has a rather high dielectric constant [1]. Such a high dielectric constant ( $\epsilon_r$ ) has caused a problem that a high power consumption is accompanied by an increase in capacitance (C) when the piezoelectric speaker is manufactured. In particular, when the piezoelectric speaker is made as a multi-layer type, the capacitance C value further can increase in proportion to the number of stacks, and the use of a material with a high dielectric constant can cause restrictions on the number of stacks because capacitance C is proportional to dielectric constant ( $\epsilon_r$ ). Therefore, to lower the dielectric constant of the ceramics is required. And also, for price competitiveness, pure Ag or Ag rich-Pd electrodes must be utilized [9–11]. Since the sintering temperature of the composition applied to the LG G-8 is currently

950 °C, Ag/Pd = 90/10 is used as an internal electrode. Therefore, the simultaneous firing process with Ag/Pd internal electrode should be applied during the lamination process for manufacturing the multilayer device. However, in this process, in order to use inexpensive pure Ag/Pd = 95/05 as an electrode, the sintering temperature of the ceramic must be 920 °C or less.

In addition, when the applied voltage is increased when driving the multilayer actuator, the sound pressure can increase, so it is good to increase the applied voltage while increasing the number of layers.

However, when the coercive electric field  $E_c$  is small, a de-poling problem in the multilayer actuator may occur, and the piezoelectric performance can be decreased.

Therefore, in order to prevent these phenomena, there is urgently required to develop the composition ceramics with a coercive electric field of 11 kV/cm or more.

For manufacturing the ceramics sintered at low temperature for multilayer piezoelectric device application,  $\text{Pb}(\text{Ni}_{1/3}\text{Nb}_{2/3})\text{O}_3$ - $\text{Pb}(\text{Zr,Ti})\text{O}_3$ - $\text{BiFeO}_3$  composition ceramics substituted with  $\text{Pb}(\text{Mg}_{1/2}\text{W}_{1/2})\text{O}_3$  with the piezoelectric  $d_{33}$  constant was selected and  $\text{MnO}_2$  was added, and their physical properties were studied.

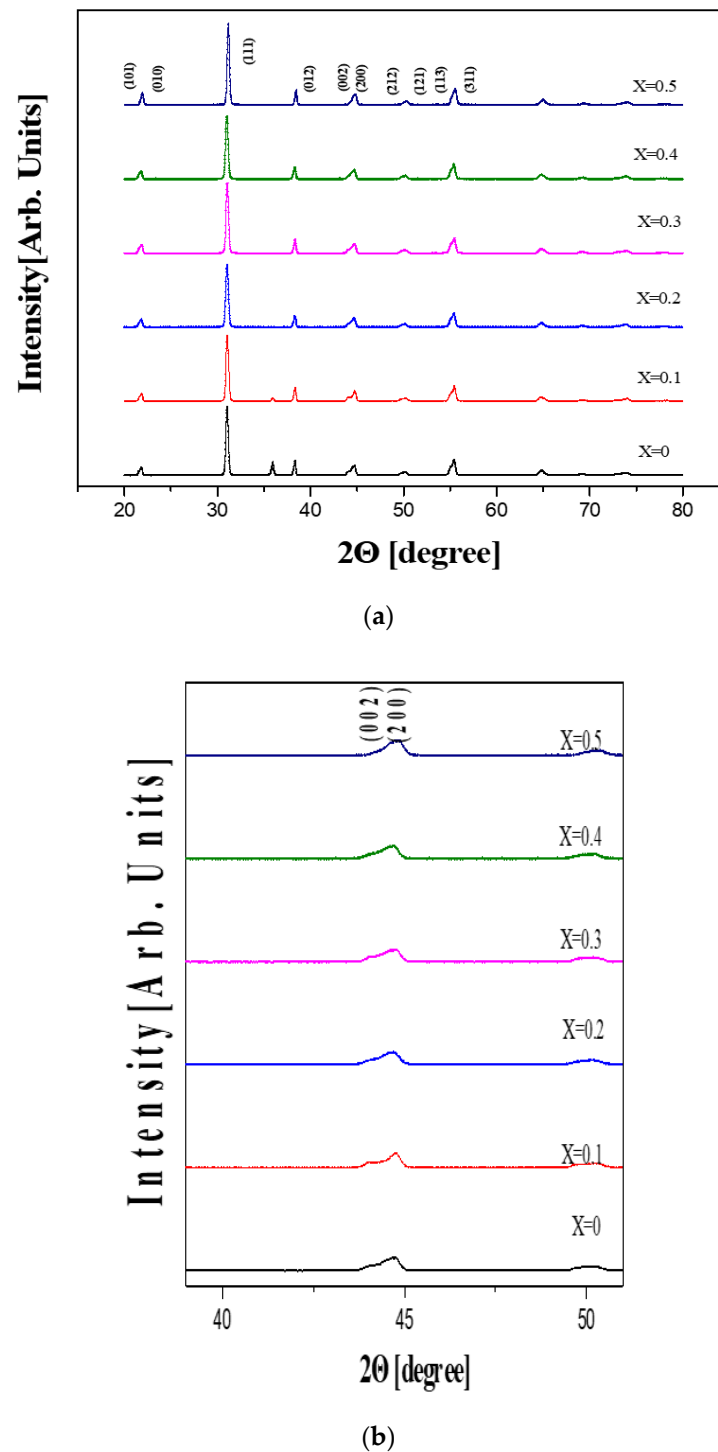
## 2. Experimental

The samples were fabricated using a conventional mixed oxide process. The composition ceramics used in this experiment were as follows;  $0.99[0.03\text{Pb}(\text{Mg}_{1/2}\text{W}_{1/2})\text{O}_3 - 0.09\text{Pb}(\text{Ni}_{1/3}\text{Nb}_{2/3})\text{O}_3 - 0.88\text{Pb}(\text{Zr}_{0.5}\text{Ti}_{0.5})\text{O}_3] + 0.01 \text{ BiFeO}_3 + 0.2 \text{ wt}\% \text{ Li}_2\text{CO}_3 + 0.25\text{wt}\% \text{ CaCO}_3 + x \text{ wt}\% \text{ MnO}_2$  ( $x = 0 \sim 0.5$ ).

The raw materials like  $\text{PbO}$ ,  $\text{ZrO}_2$ ,  $\text{TiO}_2$ ,  $\text{NiO}$ ,  $\text{MnO}_2$ ,  $\text{Nb}_2\text{O}_5$ ,  $\text{MgO}$ , and  $\text{WO}_3$  for the above composition were weighted by mole ratio and the powders were ball-milled for 24 h. After drying, they were calcined at 850 °C for 2 h. And then, sintering aids  $\text{CaCO}_3$  and  $\text{Li}_2\text{CO}_3$  were added, ball-milled, and dried. A 5% polyvinyl alcohol was added to the ceramic powders. The powders were molded by the pressure of 1 ton/cm<sup>2</sup> in a mold which has a diameter of 17 millimeters, burned out at 600 °C for 3 h, and then sintered at 920 °C for 2 h. The density was measured using Archimedes method. For measuring the physical properties, the samples were polished and then electrodeposited with Ag paste. The poling was performed at 120 °C in a silicon oil bath by applying DC Electric Fields of 30 kV/cm for 30 min. The microstructure and crystal structure of samples were measured using a scanning electron microscope (SEM: Model Hitachi, S-2400) and X-ray diffraction (XRD: Rigaku, D/MAX-2500H), respectively. For investigating the dielectric properties, capacitance  $C$  was measured at 1 kHz using an ANDO AG-4034 LCR meter. For the purpose of investigating the piezoelectric characteristics, by using an impedance analyzer (Agilent 4294A),  $f_r$  and  $f_a$  were measured according to IRE standard, and then  $k_p$  and  $Q_m$  were calculated.

## 3. Results and Discussion

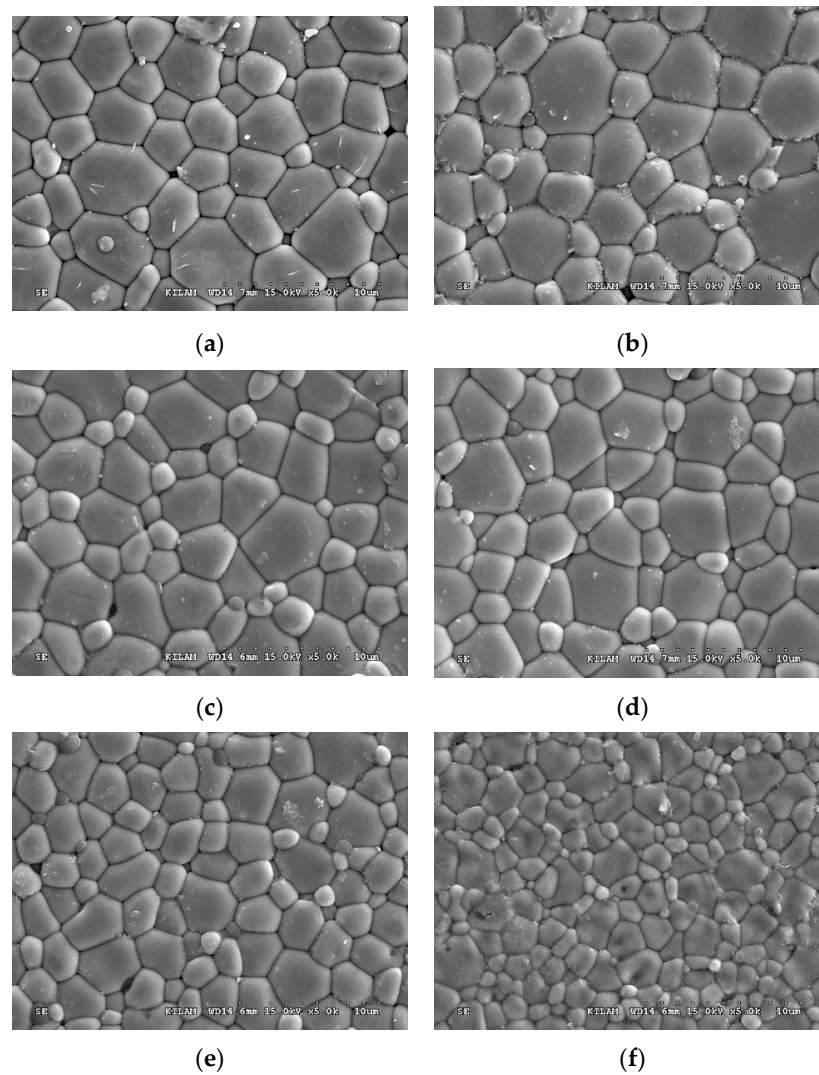
Figure 1 presents X-ray diffraction patterns of samples sintered at 920 °C with  $\text{MnO}_2$  addition. Pure perovskite phases were presented at all samples and no secondary phases are observed in the measurement range of XRD except for  $x = 0$  and 0.1. As shown in Fig.1b, the ceramic specimens possess a weak rhombohedral-tetragonal (R-T) phase coexistence from (002) and (200) peak along with the rhombohedral (200) peak between 40° and 50° from  $x = 0$  to  $x = 0.3$ . Above  $x = 0.4$ , rhombohedral phase appeared. Figure 2 presents the microstructure of samples sintered at 920 °C according to the amount of  $\text{MnO}_2$  addition. As can be seen in Figure 2, the average grain sizes gradually increased up to 6.13  $\mu\text{m}$  at 0.1 wt%  $\text{MnO}_2$  with increasing  $\text{MnO}_2$  addition and then reduced over 0.2 wt%  $\text{MnO}_2$  addition.



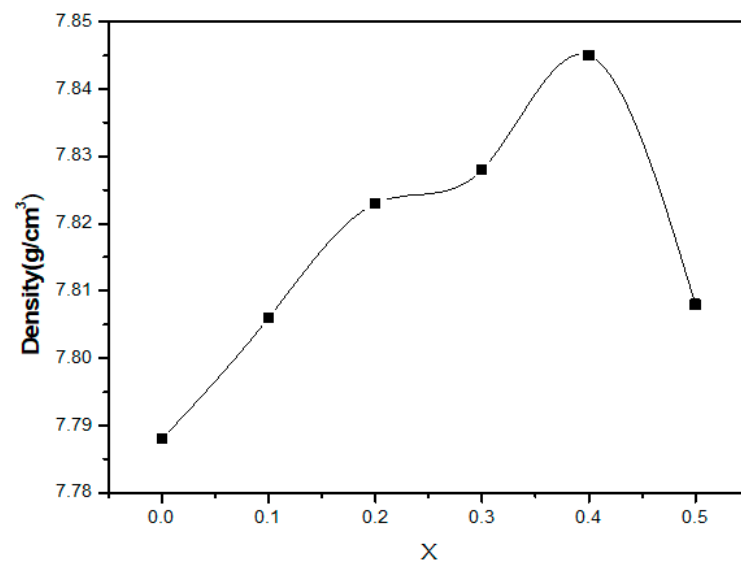
**Figure 1.** XRD pattern with the amount of MnO<sub>2</sub> addition (a) Wide range (b) Narrow range.

Figure 3 shows density with MnO<sub>2</sub> addition. The highest density of samples was increased up to 7.845 g/cm<sup>3</sup> and 0.4 wt% MnO<sub>2</sub> addition. Thereafter, the density was decreased due to the over addition of MnO<sub>2</sub>. These phenomena can be explained by the fact that the Mn<sup>3+</sup> ion can facilitate the particle diffusion by playing the role of acceptor dopant. Subsequently, the densification of specimens was enhanced [1].

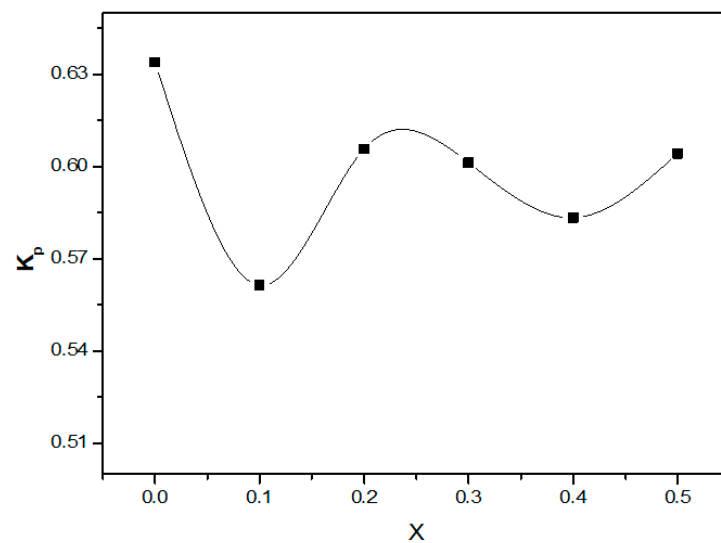
Figure 4 shows  $k_p$  according to MnO<sub>2</sub> addition. The  $k_p$  of samples increased according to the increase of the amount of MnO<sub>2</sub> addition. The  $k_p$  of samples showed the highest value of 0.634 at non doped MnO<sub>2</sub> addition and then decreased due to acceptor dopant Mn<sup>3+</sup> ion.



**Figure 2.** The SEM micrographs of the PMW-PNN-PZT with MnO<sub>2</sub> addition (a)  $x = 0$  wt% (b)  $x = 0.1$  wt% (c)  $x = 0.2$  wt% (d)  $x = 0.3$  wt% (e)  $x = 0.4$  wt% (f)  $x = 0.5$  wt%.

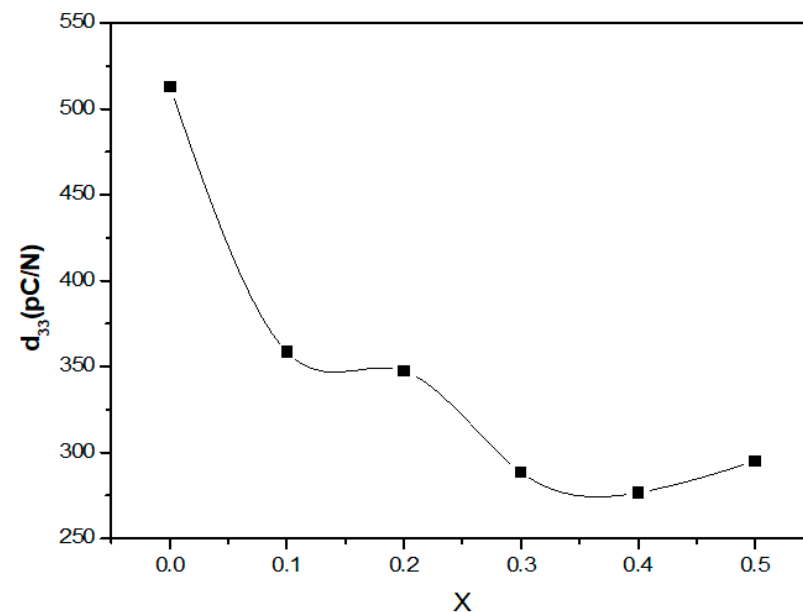


**Figure 3.** Density with MnO<sub>2</sub> addition.



**Figure 4.** Electromechanical coupling factor ( $k_p$ ) with  $MnO_2$  addition.

Figure 5 presents  $d_{33}$  according to  $MnO_2$  addition. The  $d_{33}$  of samples decreased according to the increase of the amount of  $MnO_2$  addition. The  $d_{33}$  of samples showed the highest value of 513 (pC/N) at non doped  $MnO_2$  addition. It is considered that the  $Mn^{3+}$  ion can play a role of acceptor dopant by the substitution of  $Zr^{4+}$  and  $Ti^{4+}$  ion sites.



**Figure 5.** Piezoelectric constant ( $d_{33}$ ) with  $MnO_2$  addition.

Figure 6 presents  $Q_m$  according to  $MnO_2$  addition. The behavior of  $Q_m$  showed opposite trends with  $k_p$ . That is, as the amount of  $MnO_2$  addition increased, the value of  $Q_m$  increased because the substitution of  $Mn^{3+}$  ion for  $Zr^{4+}$ ,  $Ti^{4+}$  ion sites can increase the oxygen vacancy of the ceramics.

Figure 7 presents  $\epsilon_r$  according to  $MnO_2$  addition. The  $\epsilon_r$  of specimens decreased according to the increase of the amount of  $MnO_2$  addition and was nearly consistent with the trends of piezoelectric constant  $d_{33}$ .

Figure 8 presents the temperature dependence of dielectric constant ( $\epsilon_r$ ) according to  $MnO_2$  addition. The Curie of specimens slowly decreased with the increase of the amount of  $MnO_2$  addition.

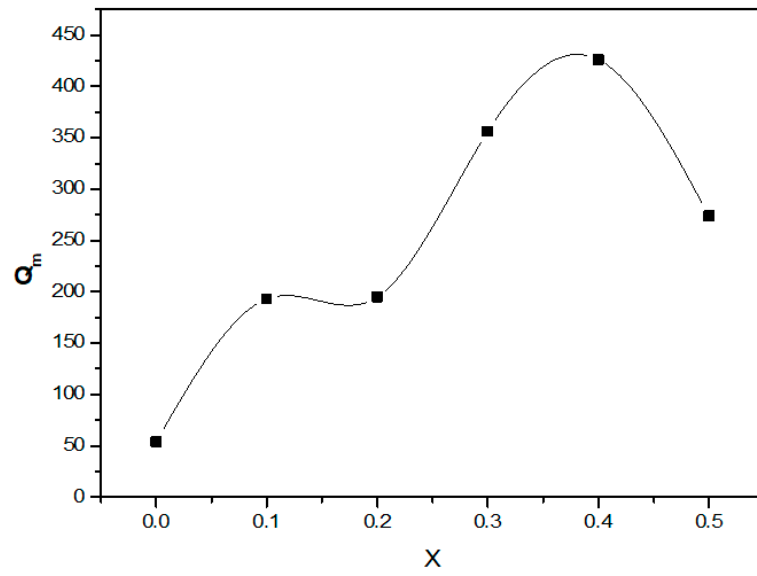


Figure 6. Mechanical quality factor ( $Q_m$ ) with MnO<sub>2</sub> addition.

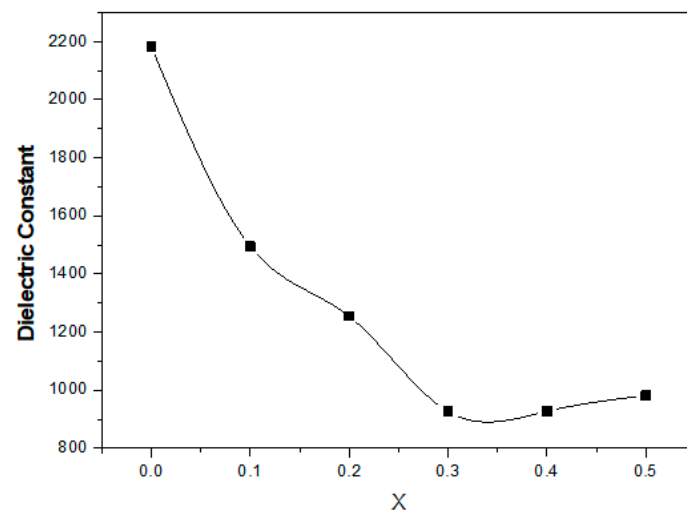


Figure 7. Dielectric constant ( $\epsilon_r$ ) with MnO<sub>2</sub> addition.

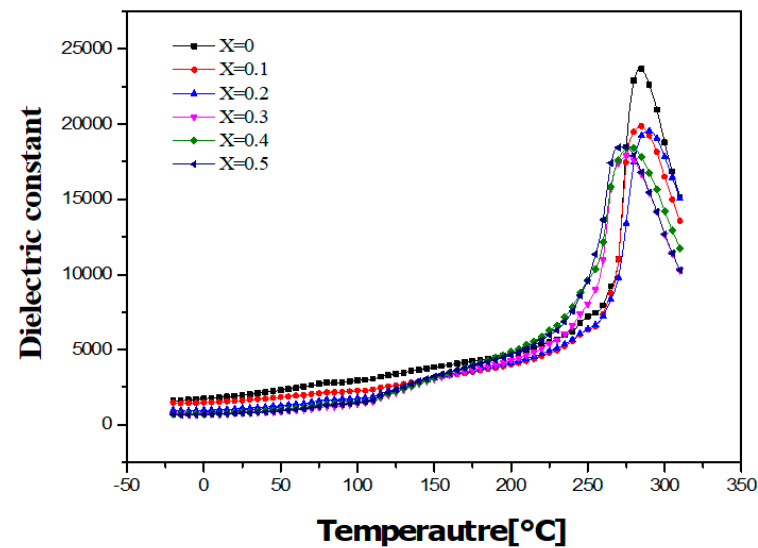


Figure 8. Temperature dependence of dielectric constant ( $\epsilon_r$ ) with MnO<sub>2</sub> addition.

Taking into consideration the physical properties of  $\epsilon_r = 1250$ ,  $d_{33} = 347$  pC/N, and  $k_p = 0.605$  were suitable for the device application such as speaker using low-temperature sintering multilayer piezoelectric actuators [1].

Table 1 shows the physical properties of specimens manufactured according to the amount of  $\text{MnO}_2$  addition.

**Table 1.** Physical properties with  $\text{MnO}_2$  addition.

X	Density [g/cm <sup>3</sup> ]	$k_p$	Dielectric Constant	$d_{33}$ [pC/N]	$g_{33}$ [mV·m/N]	$Q_m$	$T_c$ (°C)
0	7.788	0.634	2182	513	26.55	54	285
0.1	7.806	0.561	1495	358	27.04	193	285
0.2	7.823	0.605	1250	347	31.25	195	290
0.3	7.828	0.601	926	288	35.12	356	280
0.4	7.845	0.583	927	276	33.60	426	275
0.5	7.808	0.604	982	295	33.90	274	275

#### 4. Conclusions

For the application to multi-layer piezoelectric devices capable of being used in piezoelectric speakers,  $\text{Pb}(\text{Ni}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{-Pb}(\text{Zr,Ti})\text{O}_3\text{-BiFeO}_3$  ceramics substituted with  $\text{Pb}(\text{Mg}_{1/2}\text{W}_{1/2})\text{O}_3$  were manufactured according to  $\text{MnO}_2$  addition, and their physical properties were studied. The results obtained from the experiments are as follows:

The ceramic composition showed weak rhombohedral-tetragonal (R-T) phase coexistence from  $x = 0$  to  $x = 0.3$ .

At non-doped  $\text{MnO}_2$  added specimen, the maximum values of piezoelectric properties were shown, respectively: the  $\epsilon_r$  of 2182,  $d_{33}$  of 513 pC/N, and  $k_p$  of 0.634. In case of increasing the numbers of multilayer in ceramics.

At  $x = 0.2$  compositions, the  $\epsilon_r$  of 1250,  $d_{33}$  of 347 pC/N, and  $k_p$  of 0.605 were suitable for the device application such as speaker using low-temperature sintering multilayer piezoelectric actuators.

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**Conflicts of Interest:** The authors declare no conflict of interest.

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