

## Article

# Factors Influencing the CO<sub>2</sub> Corrosion Pattern of Oil–Water Mixed Transmission Pipeline during High Water Content Period

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**Abstract:** After the oil field enters the high water content period, the oil–water mixed fluid in the mixing system will gradually change into the water-in-oil mixed fluid, while the dissolved CO<sub>2</sub> causes the pH value of the mixed fluid to decrease. There is also a certain amount of bacteria in the output fluid, with many factors leading to the intensification in the corrosion of the oil–water mixed system pipeline in the high water content period. To clarify its corrosion law, through the mixed transmission pipeline material, 20# carbon steel, in high water conditions under the action of different single factor dynamic corrosion rate experiments, along with the use of the SPSS method, were used to determine the corrosion of the main control factors. The results show that in the high water content period, the corrosion rate of the mixed pipeline 20# steel gradually increases with the increase in temperature pressure, CO<sub>2</sub> partial pressure, SRB content, Ca<sup>2+</sup> + Mg<sup>2+</sup> content, and Cl<sup>−</sup> content. The corrosion rate with the CO<sub>2</sub> partial pressure and SRB content changes show a strong multiplicative power relationship; with Ca<sup>2+</sup> + Mg<sup>2+</sup> content, Cl<sup>−</sup> content changes show a logarithmic relationship, the relationship degree R<sup>2</sup> is above 0.98. Through SPSS data analysis software combined with experimental data for correlation degree analysis, it is concluded that the correlation magnitude relationship between each factor and corrosion rate is CO<sub>2</sub> partial pressure > SRB content > Cl<sup>−</sup> content > Ca<sup>2+</sup> + Mg<sup>2+</sup> content > temperature pressure, which provides a theoretical basis for the corrosion protection of an oil gathering pipeline.

**Keywords:** high water content; oil–water mixed transmission system; CO<sub>2</sub> partial pressure; corrosion law; correlation analysis



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## 1. Introduction

The water content of oil and gas wells gradually increases in the middle and late stages of extraction due to the influence of water injection development [1,2]. Different oil–water mixed fluid ratios on metal corrosion are very complex, according to relevant statistics showing that the water content rate and the number of corrosion perforations occurring in the pipeline is a positive correlation change; when the water content rate is low, the number of corrosion perforations in the pipeline accounts for about 20% of the total accidents; in the water content rate higher than 60% of the region, the proportion of possession accidents occurring is nearly 50% [3].

Water content affects pipeline corrosion mainly because when the crude oil–water content is low (less than 30%), the oil–water mixed fluid will form an oil-in-water type mixture, thus inhibiting the contact between water and carbon steel, and the tendency of corrosion is small, and uniform corrosion mainly occurs; with the increase in water content (higher than 50%), water can be separated from the emulsion, the oil–water mixed

fluid will gradually become an oil-in-water type mixed fluid, or in the form of the free water phase, at this time the collection system is in the water-wetted environment, the environment of CO<sub>2</sub> dissolved in the water phase to produce a higher concentration of carbonic acid, which promotes the electrochemical corrosion of carbon steel, thereby increasing corrosion [4–6]. The increase in water content makes the corrosion environment complex; the presence of a small amount of CO<sub>2</sub> in the associated gas, the presence of a certain amount of bacteria in the water system, and Ca<sup>2+</sup>, Mg<sup>2+</sup>, Cl<sup>−</sup> contained in the water can lead to the corrosion of the gathering system pipeline affected by different factors [7]. In a complex corrosive environment, it is of great significance to establish the main control factors affecting corrosion for pipeline corrosion protection, and at present, it is more current to analyze the main control factors of corrosion by using SPSS correlation, gray correlation, Pearson correlation coefficient, etc. Fang [8] used the SPSS program to analyze the correlation between various factors and corrosion in oil field wastewater, which showed that in the mineralization degree of less than 20,000 mg/L, dissolved salts, dissolved oxygen, and bacteria are the main factors causing corrosion; Cheng [9] et al. concluded that CO<sub>2</sub> partial pressure and SRB content are the main controlling factors affecting the corrosion failure of shale gas wellbore by the gray correlation method; Tianli [10] et al. concluded that the main controlling factors of corrosion in the Yuanba gas surface gathering system are H<sub>2</sub>S partial pressure, gas-liquid ratio and CO<sub>2</sub> partial pressure; Yanshuang [11] et al. determined that the main controlling factors of flow corrosion in gathering pipelines are liquid holding rate and gas flow rate by using the Pearson correlation coefficient.

To clarify the corrosion law and main control factors of the gathering pipeline at high water content, 20# steel used in the gathering system of HH oilfield was selected as the test sample, and the dynamic corrosion evaluation experiment was carried out by using a high temperature and high pressure dynamic corrosion scaling evaluation instrument to study the influence of different temperature and pressure, CO<sub>2</sub> content, SRB content, Ca<sup>2+</sup>+Mg<sup>2+</sup> content, and Cl<sup>−</sup> content on the corrosion law of 20# steel at a high water content. SEM electron microscope scanning technology was used to microscopically analyze the specimens after the experiment, to further clarify the corrosion law of an oil gathering pipeline in a high water content area, and to determine the main factors of corrosion by SPSS data analysis software, so as to provide a theoretical basis for the protection of a high water content oil gathering pipeline.

## 2. Materials and Methods

The original 20# steel was processed into the size of 50 mm × 10 mm × 3 mm corrosion test piece, with 6 mm holes on the end for hanging, the use of not less than a 600–1200 purpose sandpaper on the test piece step by step grinding, degreasing, and dehydration with acetone and anhydrous ethanol, wrapped with filter paper, air-dried in a dry box, and weighed for use. The 20# steel chemical composition table is shown in Table 1.

**Table 1.** Chemical composition of the sample.

Element	C	Si	Mn	P	S	Cr	Ni	Cu	Fe
Elemental Content, %	0.200	0.210	0.410	0.014	0.005	0.060	0.050	0.165	Residuals

**Experimental instruments:** The main instruments used in the experiment are SA-1 type dynamic corrosion scaling evaluation instrument (Jiangsu UNIPAC TECHNOLOGY Company, China), SU8010 cold field emission scanning electron microscope (Hitachi Corporation, Ibaraki, Japan). **Other experimental instruments:** vacuum pump, analytical balance, CO<sub>2</sub> gas cylinder, N<sub>2</sub> gas cylinder, and other auxiliary devices and tools; the chemical reagents involved in the experiment (unless otherwise noted), should meet the national standards of chemical purity.

**Water sample compounding:** According to the results of ionic composition analysis of formation water samples in the reservoir output fluid (as shown in Table 2), water

sample compounding was carried out in the laboratory to carry out experiments, in which water samples containing bacteria were prepared using simulated compound water and water samples containing bacteria to form experimental water samples with different SRB contents, which were determined by the secondary repetitive bacterial count method of the dilution method, and the compounded water samples were filtered and treated with 45 µm filter membrane for use, and the oil samples were in the laboratory. The oil sample was laboratory simulated oil (the physical properties of the oil are shown in Table 3); the gas sample was industrial pure CO<sub>2</sub> gas with 99.999% purity; industrial pure N<sub>2</sub> gas with 99.9% purity.

**Table 2.** Analysis results of ion composition of the produced water samples.

Ionic Content, mg/L							Mineralization, mg/L
K <sup>+</sup> + Na <sup>+</sup>	Ca <sup>2+</sup>	Mg <sup>2+</sup>	Ba <sup>2+</sup> + Sr <sup>2+</sup>	Cl <sup>-</sup>	HCO <sub>3</sub> <sup>-</sup>	SO <sub>4</sub> <sup>2+</sup>	99,383.4
24,385.5	10,588	446.6	2520.6	49,250.5	26.8	47.9	

**Table 3.** Basic physical properties of the experimental oil.

Density, g/cm <sup>3</sup>	Viscosity, mPa·s	Sulfur Content, %
0.835~0.869	3.53~15.8	0.07~0.09

The experimental process refers to the nationally recommended standard GB/T 16545-2015 “corrosion of metals and alloys: removal of corrosion products on corrosion specimens” and JB/T 6073-92 “metal cladding laboratory full immersion corrosion test” in the loss of weight method, the specimen hanging in the dynamic corrosion instrument to simulate various working conditions on the corrosion of the specimen.

Before the experiment, the specimens were wiped with filter paper, put into petroleum ether vessels with a boiling range of 333~363 k, and then put into anhydrous ethanol for 5 min to further degrease and dehydrate after removing the surface grease of the hanging pieces with degreasing cotton balls. After removing the grease, the hanging pieces were put on the filter paper, blown dry with cold air, and then wrapped with filter paper, placed in a desiccator, and weighed before weight loss after one hour, accurate to 1 mg. The experiments were carried out in SA-1 type dynamic corrosion and scaling evaluation apparatus, the experimental solution and the experimental specimen were put into the reaction kettle first, and the kettle was evacuated by vacuum pump for 30 min, and then the CO<sub>2</sub> gas was introduced into the reaction kettle according to the experimental partial pressure requirement. Then, CO<sub>2</sub> gas was injected into the reactor according to the partial pressure requirement, and finally, nitrogen gas was injected to reach the total pressure requirement and the speed was adjusted to 125 r/min.

After the experiment, the experimental solution in the reaction kettle was discharged, and then the test piece was removed from the reaction kettle, further decontaminated and dehydrated with petroleum ether and anhydrous ethanol, and the surface corrosion products were removed with hydrochloric acid cleaning solution, rinsed with water and the test piece was immediately put into anhydrous ethanol for cleaning and dehydration, and then weighed after being blown dry with cold air. The weight loss of the test piece before and after the experiment was recorded; each group of experiments obtains three test pieces for parallel samples, using Equation (1) to calculate the uniform corrosion rate of the test piece and obtaining the average of the three test pieces.

$$v = \frac{87600 \times (m_1 - m_2)}{S \cdot t \cdot \rho} \quad (1)$$

where,  $v$ : corrosion rate, mm/a;  $m_1$ : the mass of the test piece before the experiment, g;  $m_2$ : the mass of the test piece after cleaning, g;  $S$ : specimen exposure area, cm<sup>2</sup>;  $t$ : experimental time, h;  $\rho$ : specimen density, g/cm<sup>3</sup>.

To clarify the dynamic corrosion law of 20# steel of the mixed pipeline of the gathering system under the condition of high water content, according to the actual working conditions of the mixed pipeline of the gathering system of HH oil field, the designed dynamic corrosion experiment scheme is shown in Table 4.

**Table 4.** Experimental conditions.

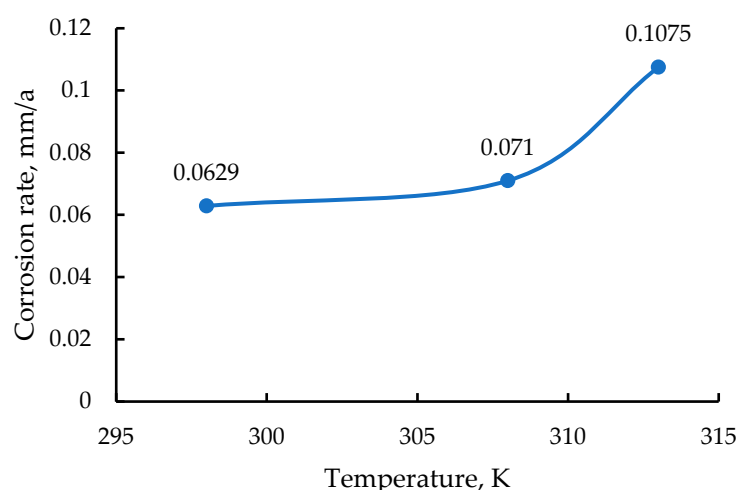
Temperature + Pressure	Rotational Speed (r/min)	CO <sub>2</sub> Partial Pressure (MPa)	Water Content (%)	Bacterial Content (SRB) (pcs/mL)	Ca <sup>2+</sup> Content (mg/L)	Cl <sup>-</sup> Content (mg/L)	Soaking Time (h)
298 K + 0.5 MPa	125	0.05	90	0	7700	40,000	72
308 K + 2.5 MPa		0.1		60	9500	45,000	
313 K + 3.5 MPa		0.17		120	11,000	49,250	
				300	12,500	55,000	
				600	14,000	60,000	

### 3. Results

#### 3.1. Effect of Temperature Pressure on Corrosion

Measurement conditions: 90% water content, 0MPa CO<sub>2</sub> partial pressure, 100,000 mg/L total mineralization, 11,000 mg/L Ca<sup>2+</sup> + Mg<sup>2+</sup> content, 49,250 mg/L Cl<sup>-</sup> content, temperature pressure designed as 298 K + 0.5 MPa, 308 K + 2.5 MPa and 313 K + 3.5 MPa respectively.

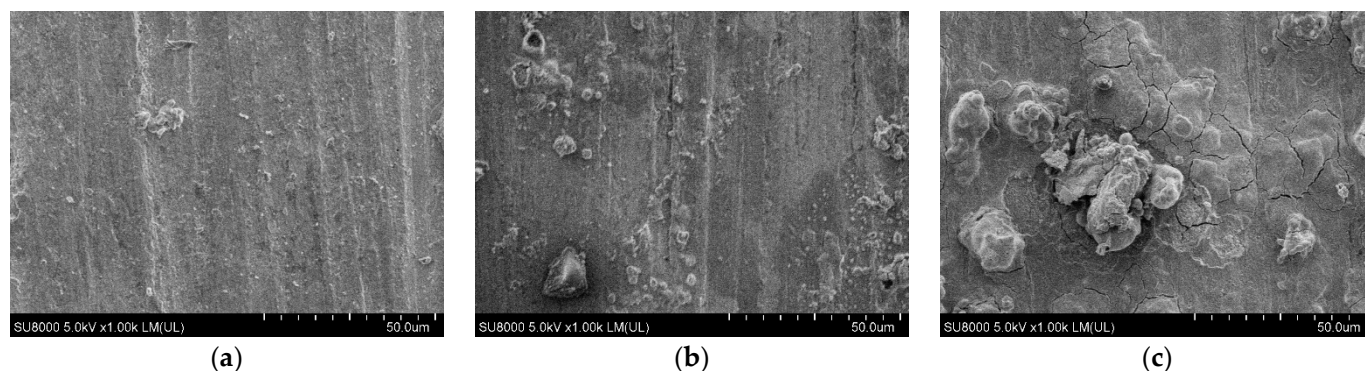
As seen in Figure 1, with the increase in temperature and pressure, the corrosion rate of 20# steel gradually increased. The main reasons are that on the one hand, with the increase in temperature, the diffusion coefficient of substances in water increases, and more dissolved oxygen diffuses to the cathodic area of the metal surface, accelerating corrosion; at the same time, the electrolyte resistance and the polarization resistance of the anodic reaction decrease with the increase in temperature, promoting the corrosion of the metal specimen. On the other hand, the change in temperature affects the deposition rate of corrosion products and the coverage, denseness, and bond strength of the corrosion product film, which in turn affects the protectiveness of the metal [12,13].



**Figure 1.** Variation curve of dynamic corrosion rate of 20# steel with temperature and pressure.

SEM scanning electron microscopy was used to observe the microscopic morphology of the hanging pieces after corrosion at different temperatures, and the results are shown in Figure 2. It can be seen from the figure, that 20# in the 298 K + 0.5 MPa test piece surface is cleaner, there are fewer corrosion products, and a small number of local clusters of corrosion products, but this does not quite affect the body; the main reason being that, at a lower temperature, the 20# steel electrochemical reaction rate is low, the degree of corrosion is relatively light. With the increase in temperature and pressure, corrosion products

gradually increased; a 308 K + 2.5 MPa corrosion product increase is not too obvious but at 313 K + 3.5 MPa, whereby the corrosion products increased significantly, indicating the higher temperature and pressure with the corrosion of the 20# steel influence [14].

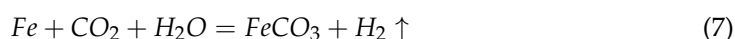
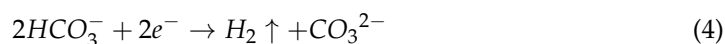


**Figure 2.** Microscopic corrosion morphology of hanging piece under different temperatures and pressure. (a) 298 K + 0.5 MPa; (b) 308 K + 2.5 MPa; (c) 313K + 3.5 MPa.

### 3.2. Effect of CO<sub>2</sub> Partial Pressure on Corrosion

Test conditions: temperature and pressure are 308 K and 2.5 MPa, water content is 90%, bacteria content is 0 pcs/mL, total mineralization is 100,000 mg/L, Ca<sup>2+</sup> + Mg<sup>2+</sup> content is 11,000 mg/L, Cl<sup>-</sup> content is 49,250 mg/L, CO<sub>2</sub> partial pressure is designed as 0 MPa, 0.05 MPa, 0.10 MPa and 0.17 MPa respectively. Through the above experimental conditions in order to carry out dynamic corrosion experiments on 20# steel, a clearly different CO<sub>2</sub> partial pressure on the corrosion rate of the law, CO<sub>2</sub> partial pressure, and 20# steel corrosion rate relationship curve is shown in Figure 2.

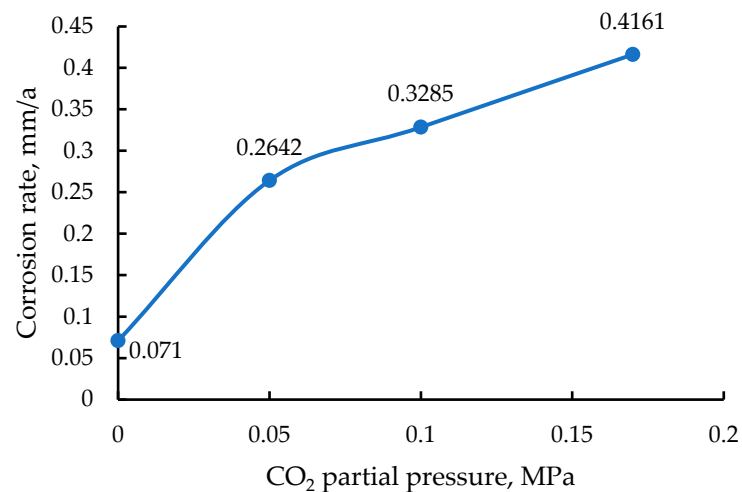
Figure 3 shows that the 20# steel corrosion rate with the CO<sub>2</sub> partial pressure increases the trend of change, mainly with the increase in CO<sub>2</sub> partial pressure, the corrosion rate gradually increases, and contains CO<sub>2</sub>, compared to the conditions without CO<sub>2</sub> which increase very quickly, indicating the presence of CO<sub>2</sub> on the corrosion rate and the impact is greater. The main reason is that CO<sub>2</sub> will produce carbonic acid when it meets water, and carbonic acid promotes the electrochemical corrosion of metals. As the partial pressure increases, more CO<sub>2</sub> combines with the water molecules in the solution, which increases the H<sub>2</sub>CO<sub>3</sub> in the solution and intensifies the ionization of H<sub>2</sub>CO<sub>3</sub> at the same time, and the H<sup>+</sup> in the solution increases, which decreases the pH of the solution and accelerates the cathodic reaction in the electrochemical reaction, leading to increased corrosion, and the reaction equation is shown in Equations (2)–(7).



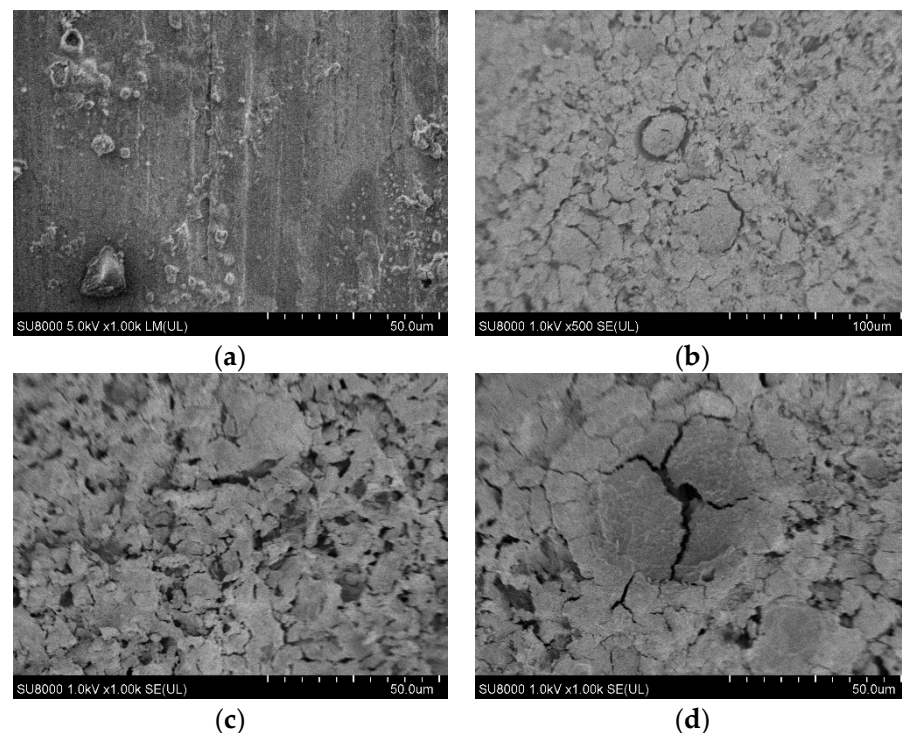
The corrosion morphology under a different partial pressure of CO<sub>2</sub> is shown in Figure 4. Comparing the corrosion morphology results under a different partial pressure of CO<sub>2</sub>, it can be seen that with the increase in partial pressure of CO<sub>2</sub> from 0 MPa to 0.1 MPa, the corrosion gradually intensifies and the corrosion product thickens, but the corrosion product film is relatively intact. This is because as the partial pressure of CO<sub>2</sub> increases, the solubility of CO<sub>2</sub> in solution also increases, the content of CO<sub>2</sub> series of ions in solution increases, the corrosion reaction is accelerated, and the content of Fe<sup>2+</sup> in solution increases.



The ions of the CO<sub>2</sub> series combined with the production of deposition on the surface of the specimen promote the deposition of corrosion products film more quickly, forming a dense corrosion product film, thus inhibiting the corrosion reaction. When the partial pressure of CO<sub>2</sub> increases from 0.1 MPa to 0.17 MPa, the corrosion product film is disrupted and local corrosion begins to occur and the corrosion rate increases. This is because, with the increasing partial pressure of CO<sub>2</sub>, CO<sub>2</sub> dissolves in water and forms carbonic acid, and the pH solution decreases will form an acidic solution, which will increase the corrosion rate of the material and destroy the corrosion product film [15–17].



**Figure 3.** Variation curve of 20# steel dynamic corrosion rate with CO<sub>2</sub> partial pressure.



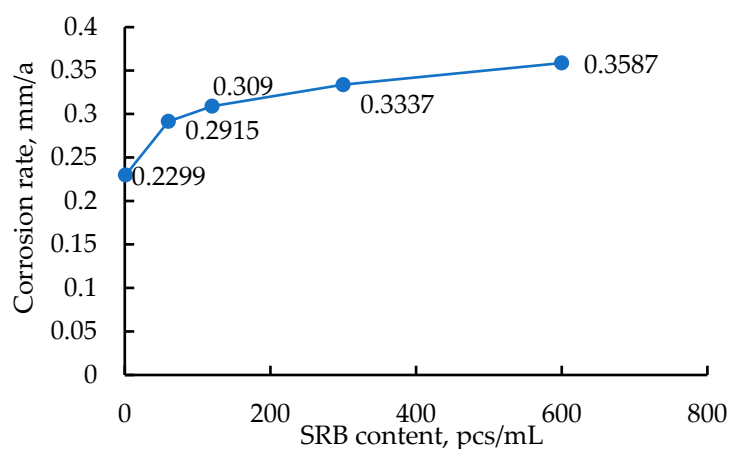
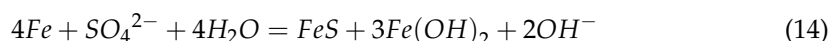
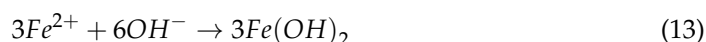
**Figure 4.** Microscopic corrosion morphology of hanging piece under different CO<sub>2</sub> partial pressure. (a) CO<sub>2</sub> partial pressure 0 MPa; (b) CO<sub>2</sub> partial pressure 0.05 MPa; (c) CO<sub>2</sub> partial pressure 0.1 MPa; (d) CO<sub>2</sub> partial pressure 0.17 MPa.

### 3.3. Effect of SRB Content on Corrosion

Test conditions: temperature and pressure were 40 °C and 3.5 MPa, water content was 80%, CO<sub>2</sub> partial pressure was 0.05 MPa, total mineralization was 100,000 mg/L, Ca<sup>2+</sup> +

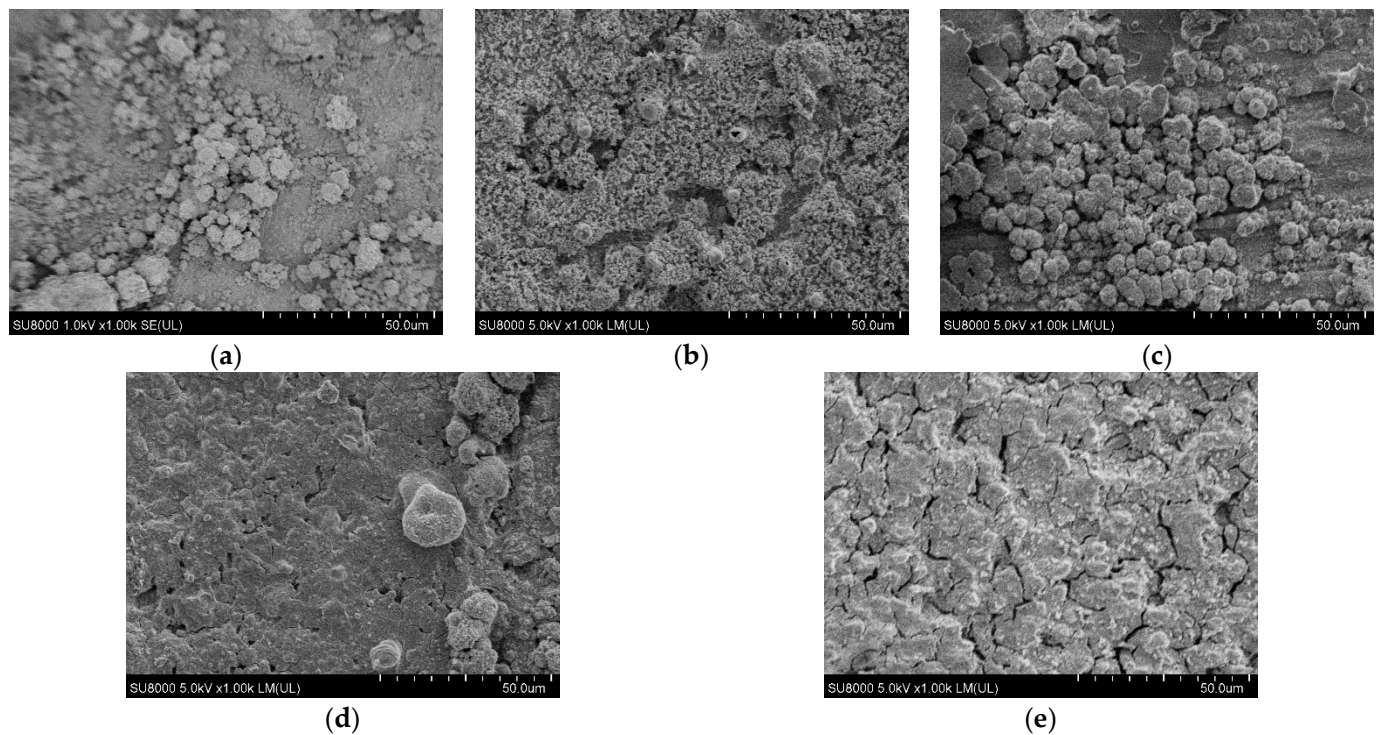
$Mg^{2+}$  content was 11,000 mg/L,  $Cl^{-}$  content was 49,250 mg/L, and bacterial content was designed to be 0, 60, 120, 300, and 600 pcs/mL, respectively.

To clarify the SRB content on the corrosion rate of the law, dynamic corrosion experiments were carried out through the above experimental conditions; the experimental results are shown in Figure 5. From the results, it can be seen that the corrosion rate of 20# steel mainly shows a gradual increase in corrosion rate with the increase in SRB content, while the increasing trend of corrosion rate gradually slows down with the increase in SRB content. The current mechanisms for the influence of SRB on metal corrosion includes the cathodic polarization, concentration cell mechanism, and metabolite mechanisms [18], of which the more recognized cathodic depolarization theory believes that in the oxygen-deficient environment, the SRB attached to the surface of carbon steel will produce depolarization, so that the  $SO_4^{2-}$  oxidation in the medium absorbs hydrogen atoms so that the corrosion rate of carbon steel increases and the reaction equation is shown in Equations (8)–(14).



**Figure 5.** Variation curve of dynamic corrosion rate of 20# steel with SRB content.

The corrosion morphology under different SRB content is shown in Figure 6. When the SRB content is 0, due to the presence of  $CO_2$  partial pressure, the surface of the test piece formed irregular granular-like corrosion products, and with the SRB content continuing to increase, more corrosion products gather on the surface of the test piece, indicating that under the conditions of the presence of  $CO_2$ , with the increase in SRB content, the corrosion rate accelerates and the product film formation faster. However, there is a local surface corrosion product thickening phenomenon, the corrosion product film cracks exist, easily caused by the intrusion of corrosive media and cause corrosion.



**Figure 6.** Microscopic corrosion morphology of hanging piece under different SRB content. (a) 0 pcs/mL; (b) 60 pcs/mL; (c) 120 pcs/mL; (d) 300 pcs/mL; (e) 600 pcs/mL.

### 3.4. Effect of $\text{Ca}^{2+} + \text{Mg}^{2+}$ Content on Corrosion

Test conditions: temperature and pressure of 313 K + 3.5 MPa, water content of 90%,  $\text{CO}_2$  partial pressure of 0MPa, total mineralization of 100,000 mg/L,  $\text{Cl}^-$  content of 49,250 mg/L,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content designed for 7700, 9500, 11,000, 12,500, and 14,000 mg/L respectively.

In order to carry out dynamic corrosion experiments, through the above experimental conditions on 20# steel, there was clearly different  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content on the corrosion rate of the influence of the law; the experimental results are shown in Figure 7. The results show that the dynamic corrosion rate of 20# steel with  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content increases gradually, and in the  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content of 14,000 mg/L corrosion rate reached a maximum of 0.1235 mm/a. With a  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content less than 12,500 mg/L, the 20# steel corrosion rate rises faster, when  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content is greater than 125,00 mg/L, the 20# steel dynamic corrosion rate continues to increase, but the increase in the amount slows down. The main reason is that on the one hand, the medium,  $\text{CO}_3^{2-}$  and  $\text{Ca}^{2+}/\text{Mg}^{2+}$  reaction  $\text{CaCO}_3/\text{MgCO}_3$  deposited on the surface of the test piece, and the deposition layer is not complete and dense, but increased the unevenness of the metal surface, and may become a pitting source and induce pitting and local corrosion of the metal, thereby increasing the corrosion rate. On the other hand, the increase in carbonate deposition on the metal surface, the increase in corrosion microcells, and corrosion products below the corrosion rate increase.

The corrosion morphology under different  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content is shown in Figure 8. When the  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content is 7700 mg/L (as shown in Figure 8a), there are irregular lumpy products attached to the surface of the test piece; at this time, the corrosion products may be calcium and magnesium chloride deposits, but the metal body is still visible, indicating that the corrosion rate is not high under this condition; when the  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content is 9500 mg/L (as shown in Figure 8b shown), the specimen surface becomes pitting corrosion, and the reason may be the solution of  $\text{Cl}^-$  broke through the deposition of  $\text{CaCO}_3/\text{MgCO}_3$  deposited on the metal surface, resulting in the specimen pitting corrosion, making the corrosion rate rise; with  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content increasing, the specimen surface cluster



corrosion products and the analysis of possible components should be calcium magnesium chloride deposition, the emergence of calcified deposits cannot block the medium on the specimen corrosion effect, and easily cause the metal local electrode potential differences, leading to electrochemical corrosion and resulting in increased corrosion under the scale. When  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content increased to 14,000 mg/L, corrosion products increases, but the corrosion products are loose, and there are many cracks that were not dense enough to complete. When the corrosion medium has  $\text{CO}_2$ , bacteria or other factors that exist, it will lead to increased corrosion, resulting in localized corrosion, or even pitting corrosion [19].

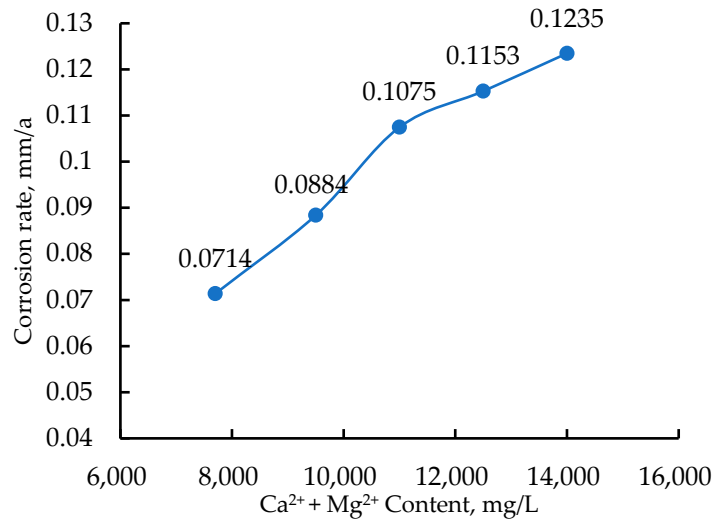


Figure 7. Variation curve of dynamic corrosion rate of 20# steel with  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content.

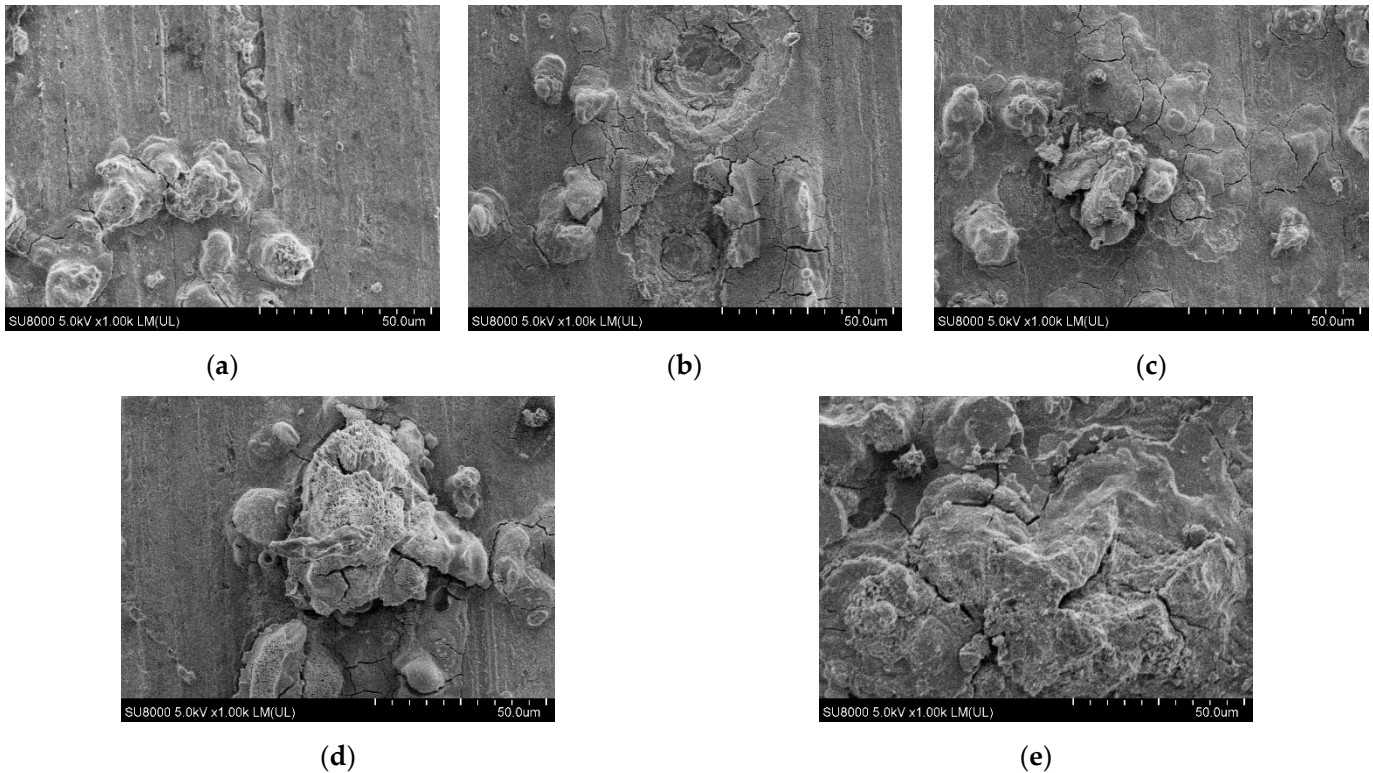
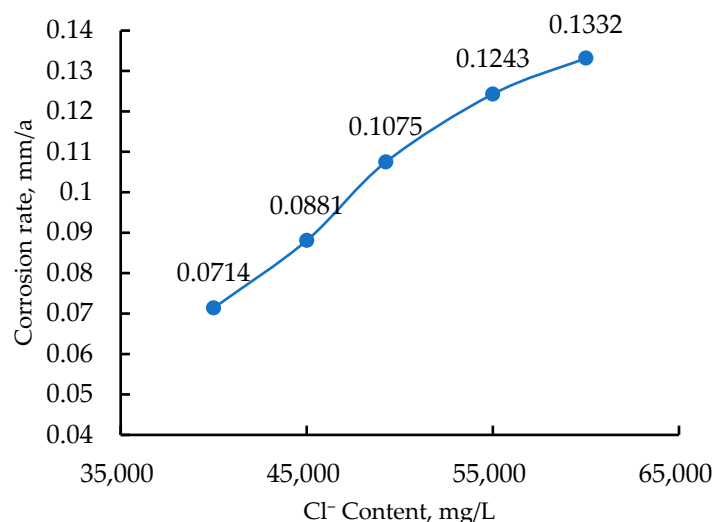


Figure 8. Microscopic corrosion morphology of hanging piece under different  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content. (a)  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content 7700 mg/L; (b)  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content 9500 mg/L; (c)  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content 11,000 mg/L; (d)  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content 12,500 mg/L; (e)  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content 14,000 mg/L.

### 3.5. Effect of $\text{Cl}^-$ Content

Test conditions: temperature and pressure of 313 K+3.5 MPa, water content of 90%,  $\text{CO}_2$  partial pressure of 0 MPa, total mineralization of 100,000 mg/L,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content of 11,000 mg/L, and  $\text{Cl}^-$  content of 40,000, 45,000, 49,250, 55,000, 60,000 mg/L designed, respectively.

To carry out dynamic corrosion experiments, through the above experimental conditions in the 20# steel, clear different  $\text{Cl}^-$  content on the corrosion rate of the influence of the law, as can be seen in Figure 9, the corrosion rate with the  $\text{Cl}^-$  content and rising is shown. Among them is the  $\text{Cl}^-$  content ranging from 45,000 mg/L to 50,000 mg/L, 20# steel uniform corrosion rate change is the largest when the  $\text{Cl}^-$  content is from 55,000 mg/L to 60,000 mg/L, 20# steel uniform corrosion rate change is the smallest. Combined with the corrosion morphology, the results were analyzed as follows: as can be seen from Figure 10 with the  $\text{Cl}^-$  content of 40,000 mg/L, the test piece surface corrosion products are less and the surface is relatively flat; when the  $\text{Cl}^-$  content rises to 45,000 mg/L, the test piece of corrosion products film is destroyed; the main reason may be the  $\text{Cl}^-$  ion radius is a very small penetrating force and can destroy the test piece of the protective film, so that the corrosion rate increased; with the  $\text{Cl}^-$  content continuing to rise, the corrosion products on the surface of the test piece also increase sharply, and there is a cluster of corrosion products, at this time under the influence of  $\text{Cl}^-$ , the corrosion product film cracks, resulting in  $\text{Cl}^-$  which can penetrate and easily cause pitting; when the  $\text{Cl}^-$  content ranges from 55,000 mg/L to 60,000 mg/L, the corrosion products on the surface increases and the corrosion product film thickens, and to a certain extent will obstruct  $\text{Cl}^-$ , so its corrosion rate rise is weakened [20–22].

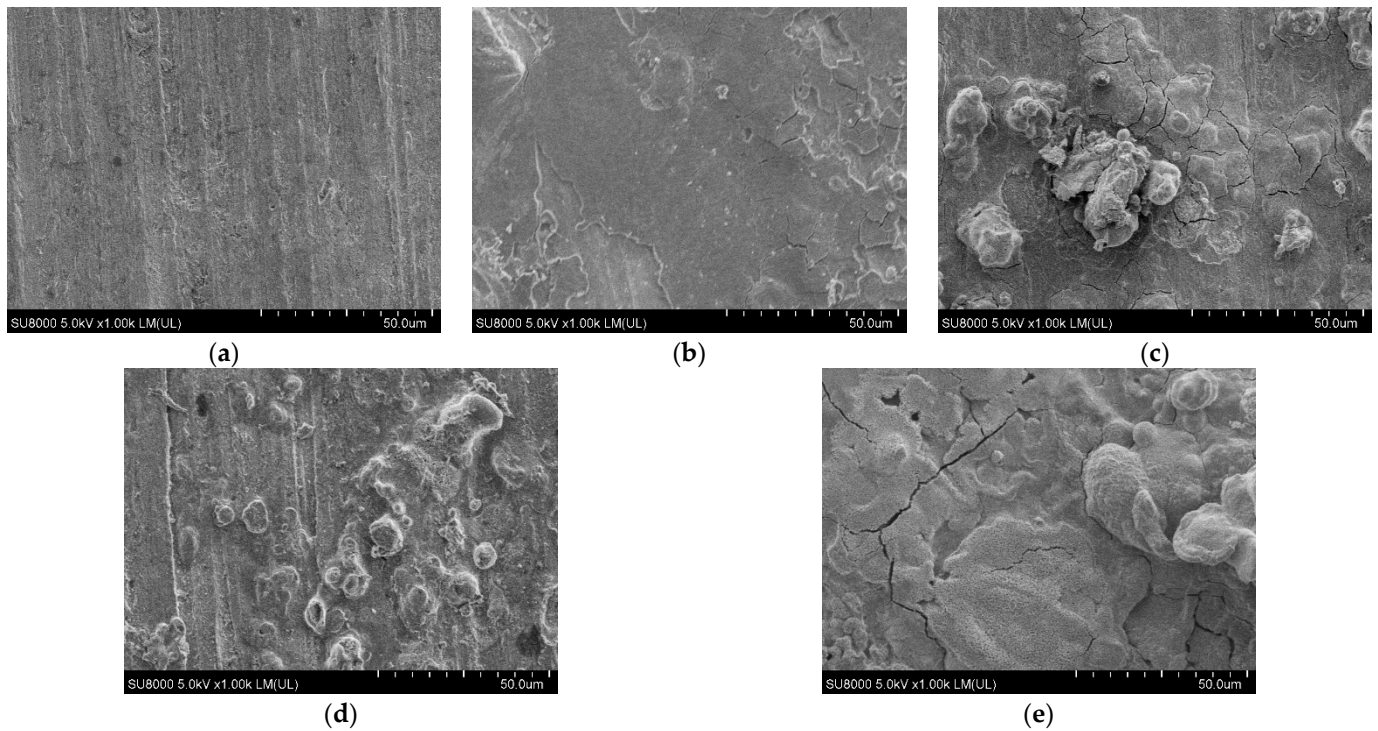


**Figure 9.** Variation curve of dynamic corrosion rate of 20# steel with  $\text{Cl}^-$  content.

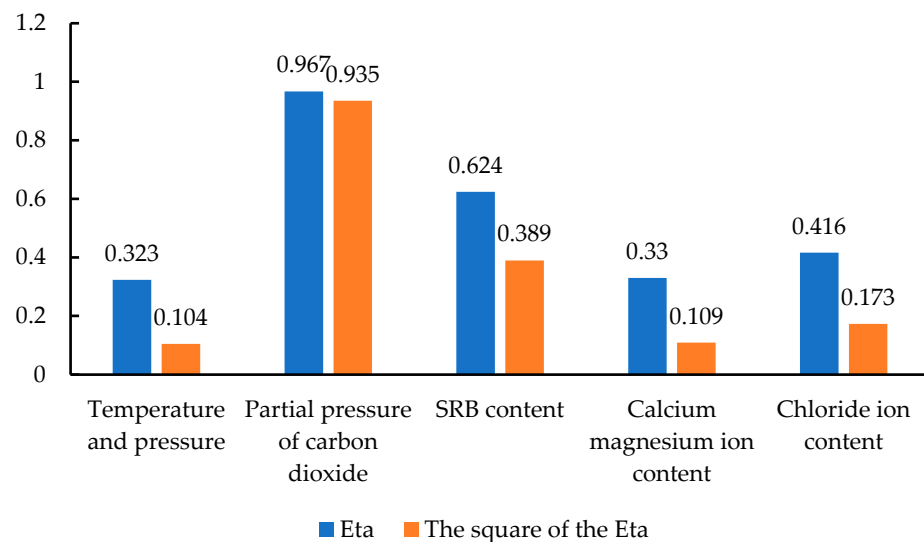
### 3.6. Analysis of Main Control Factors

To establish the main control factors influencing the corrosion of the oil–water mixed transmission system in the high water content period of the HH oil field, five influencing factors: temperature pressure,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content,  $\text{CO}_2$  partial pressure,  $\text{Cl}^-$  content, and SRB content were set as independent variables, and the corrosion rate of the test piece was set as the dependent variable, and Eta correlation analysis was performed using the mean comparison of SPSS; the operation results are shown in Figure 11. Eta squared represents “the proportion of the variance of the dependent variable explained by the difference between groups”. Eta squared is less than 0.06, indicating that the correlation between the variables is weak; this degree does not reach statistical significance, that is, the results are not statistically significant, cannot determine whether the difference in the mean value is due to random error; when Eta squared is greater than 0.06 and less than 0.16, this indicates

a moderate correlation between the variables; when Eta squared is greater than 0.16, it indicates a strong correlation between the variables.



**Figure 10.** Microscopic corrosion morphology of hanging piece under different  $\text{Cl}^-$  content. (a)  $\text{Cl}^-$  content 40,000 mg/L; (b)  $\text{Cl}^-$  content 45,000 mg/L; (c)  $\text{Cl}^-$  content 49,250 mg/L; (d)  $\text{Cl}^-$  content 55,000 mg/L; (e)  $\text{Cl}^-$  content 60,000 mg/L.



**Figure 11.** SPSS data analysis results.

From the results of the Eta correlation analysis, it can be seen that the correlation (square of Eta) between temperature pressure,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content,  $\text{CO}_2$  partial pressure,  $\text{Cl}^-$  content, SRB content, and corrosion rate of 20# steel is greater than 0.06, which means that all five factors have a statistically significant effect on the corrosion rate and show a strong correlation with the corrosion rate. Among them,  $\text{CO}_2$  partial pressure,  $\text{Cl}^-$  content, SRB content, and the corrosion rate of 20# steel have a strong correlation, temperature pressure,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content, and the corrosion rate of 20# steel have a moderate correlation.

The relationship between the magnitude of correlation (Eta value) of the five influencing factors and the corrosion rate is  $\text{CO}_2$  partial pressure > SRB content >  $\text{Cl}^-$  content >  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content > temperature pressure, indicating that  $\text{CO}_2$  partial pressure and SRB content are the main controlling factors affecting the corrosion rate of the pipeline when the HH oil field gathering pipeline is working under high water content conditions.

#### 4. Conclusions

- (1) The corrosion rate of HH oil field gathering pipeline in the high water content period showed a positive correlation with temperature pressure,  $\text{CO}_2$  partial pressure, SRB content,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content, and  $\text{Cl}^-$  content. The corrosion rate reached the maximum of 0.4697mm/a at the temperature and pressure of 313 K + 3.5 MPa, 0 bacteria/mL, total mineralization of 100,000 mg/L,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content of 11,000 mg/L,  $\text{Cl}^-$  content of 49,250 mg/L, and  $\text{CO}_2$  partial pressure of 0.17MPa.
- (2) In high water conditions, with the temperature pressure,  $\text{CO}_2$  partial pressure, SRB content,  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content,  $\text{Cl}^-$  content increases, 20# steel corrosion products are gradually increased when the presence of  $\text{CO}_2$ , the formed corrosion products film is looser, and the existence of gaps, cannot effectively prevent the occurrence of corrosion.
- (3) A 20# steel gathering pipeline in the high water content period for oil–water mixing shows the following impact of factors on the size of corrosion:  $\text{CO}_2$  partial pressure > SRB content >  $\text{Cl}^-$  content >  $\text{Ca}^{2+} + \text{Mg}^{2+}$  content > temperature pressure, where the partial pressure of  $\text{CO}_2$  for the control of the corrosion rate of the main control factors, and Eta square value of up to 0.934. Therefore, the effect of  $\text{CO}_2$  partial pressure should be considered first in the corrosion problem of high water-bearing catchment system containing  $\text{CO}_2$ .

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