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# **Dephosphorization in Double Slag Converter Steelmaking Process at Different Temperatures by Industrial Experiments**

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**Abstract:** In the present work, the effect of dephosphorization endpoint temperature on the dephosphorization of hot metal was studied for the double slag converter steelmaking process under the conditions of low temperature and low basicity by the industrial experiments. In the temperature range of 1350–1450 ℃, with an increasing dephosphorization endpoint temperature, the dephosphorization ratio and phosphorus distribution ratio first increase and then decrease. The phosphorus content in hot metal first decreases and then increases at the end of dephosphorization. At the dephosphorization temperature range of 1385–1410 ◦C, the dephosphorization ratio is higher than 55%, the  $P_2O_5$  content in the dephosphorization slag is 3.93–4.17%, the log $L_P$  value is 1.76–2.09, the value of  $P_{\rm CO}^{\rm P-C}$  of the selective oxidation reaction of carbon and phosphorus is 53–80 Pa, and the  $a_{FeO}$  value is 0.284–0.312. The path of phosphorus in hot metal entering the P-rich phase of dephosphorization slag can be reasonably inferred as: hot metal  $\rightarrow$  Fe-rich phase  $\rightarrow$  P-rich phase. Under the present industrial experimental conditions, the dephosphorization and rephosphorization reactions are in dynamic equilibrium at 1413 ◦C. Considering the experimental results and thermodynamic calculation results of industrial experiments by the double slag dephosphorization process, the optimal temperature range for intermediate deslagging is about 1400–1420 ◦C.

**Keywords:** double slag converter steelmaking process; hot metal dephosphorization; dephosphorization endpoint temperature; dephosphorization ratio; phosphorus distribution ratio; optimum temperature of intermediate deslagging

# **1. Introduction**

The double slag converter steelmaking process developed by Nippon Steel in 2001 is the most representative, which is called Multi-Refining Converter (MURC) [\[1\]](#page-21-0). In this process, the desiliconization and dephosphorization operations are performed in the converter firstly, and the dephosphorization slag is poured out by tilting furnace. Then, the decarburization operation is performed in the same converter and the decarburization slag is left in the converter for reusing in the next charge.

Some scholars have carried out researches on the double slag converter steelmaking process by industrial experiments. Yang Wang et al. [\[2\]](#page-21-1). conducted the double slag converter steelmaking process by using a low oxygen lance height in the dephosphorization stage of the industrial converter. The stirring of molten steel is strengthened, and the phosphorus content is reduced at the end of decarburization from an average value of 0.018% to an average value of 0.011%, and the dephosphorization ratio is increased by more than 6%. He Wu et al. [\[3\]](#page-21-2). conducted the industrial experiments and showed that the average dephosphorization ratio of 62.2% and the average phosphorus content of 0.048% can be achieved under the process conditions of the lower temperature of 1330–1350  $\degree$ C, the bottom blowing stirring intensity of 0.03–0.04  $m^3/t$ ·min, the medium slag basicity of 2.0–3.0



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and the high iron oxide content of 20–25%. Our previous experimental results [\[4\]](#page-21-3) showed that with increasing the basicity of dephosphorization slag, the phosphorus distribution ratio and dephosphorization ratio increase, and the area fraction of phosphorus rich phase in dephosphorization slag increases with the increasing area fraction of iron-rich phase.

In the hot metal dephosphorization reaction, the slag is in the state of solid–liquid multiphase coexistence. While lime dissolves,  $2CaO·SiO<sub>2</sub> (C<sub>2</sub>S)$  is formed [\[5\]](#page-21-4).  $C<sub>2</sub>S$  tends to combine with  $3CaO \cdot P_2O_5 (C_3P)$  in liquid slag to form solid solution  $2CaO \cdot SiO_2$ - $3CaO \cdot P_2O_5$  $(C_2S-C_3P)$  [\[6,](#page-21-5)[7\]](#page-21-6). Kitamura et al. [\[8\]](#page-21-7) found that the mass transfer of CaO and SiO<sub>2</sub> occurs simultaneously with the mass transfer of  $P_2O_5$ . Xie et al. [\[9\]](#page-21-8) found that the diffusion of phosphorus from the  $C_2S-C_3P$  solid solution reaction layer to the internal encapsulated  $C_2$ S is the rate-controlling step of phosphorus transport. Kakimoto et al. [\[10\]](#page-21-9) found that the thickness of the FeO-CaO layer is related to the thickness of the  $C_2S$  phase. Wu et al. [\[11\]](#page-21-10) found that almost all the phosphorus in CaO-FeO-SiO<sub>2</sub>-MgO-MnO-Al<sub>2</sub>O<sub>3</sub>-P<sub>2</sub>O<sub>5</sub>-TiO<sub>2</sub> slag with the basicity close to 3 is combined with  $C_2S$  but not with  $3CaO·SiO2$  ( $C_3S$ ), and C2S crystal grows uniformly into a spherical shape. Pahlevani et al. [\[12\]](#page-21-11) found that the distribution ratio of phosphorus in solid solution and liquid slag is determined by the CaO content in liquid slag and activity coefficient of  $P_2O_5$  in solid solution.

The effect of temperature on dephosphorization in the double slag converter steelmaking process was one of the most important points for a highly efficient implementation of this process. Tian et al. [\[13\]](#page-21-12) found that when the  $Fe<sup>3+</sup>$  content in slag is between 2.33 and 5.98%, the basicity is between 1.72 and 3.14, and the temperature is between 1600 and 1700  $\degree$ C, the dephosphorization ratio gradually decreases with increasing temperature. Yang et al. [\[14\]](#page-21-13) found that in the temperature range of 1600–1660 °C, with increasing temperature, the phosphorus distribution ratio gradually decreases, the endpoint phosphorus content increases, and the dephosphorization ratio decreases. Zhou et al. [\[15\]](#page-21-14) found that in the temperature range of 1590–1640  $°C$ , the total rephosphorization content of molten steel first decreases and then increases. From the above literature, it can be seen that the effect of temperature on converter dephosphorization mainly focused on the endpoint temperature of converter decarburization, ranging from 1590 to 1700 ◦C; there were few reports on the temperature range of 1350–1450 ◦C.

The effect of the oxygen potentials at the different reaction interfaces on dephosphorization was also studied. Han et al. [\[16\]](#page-21-15) concluded that when the dissolved oxygen content in the endpoint molten steel is between 0.03% and 0.045%, the [P]-[O] equilibrium phosphorus content is 21–38 times that of [P]-(FeO) equilibrium phosphorus content. Kitamura et al. [\[17\]](#page-21-16) found that the oxygen activity of the FeO phase in dephosphorization slag is two orders of magnitude larger than that of equilibrium oxygen activity in hot metal containing more than 4% of carbon.

In addition, some scholars have studied the optimum deslagging temperature of dephosphorization slag in the double slag converter steelmaking process. Based on the selective oxidation reaction of carbon and phosphorus, Wu et al. [\[4\]](#page-21-3) calculated the carbon– phosphorus selective oxidation transition temperature to be 1342 ◦C according to the dephosphorization slag compositions and semi-steel control target, and they took it as the guiding temperature for dephosphorization slag deslagging. Chaogang Zhou et al. [\[18\]](#page-21-17) calculated the optimal deslagging temperature of dephosphorization slag and obtained that the equilibrium temperature of dephosphorization in the early stage of converter blowing is 1435 ◦C.

According to the above literatures, it can be seen that in the research of the double slag converter steelmaking process, the effect of dephosphorization endpoint temperature in the range of 1350–1450 °C on hot metal dephosphorization is rarely studied. Since the lower temperature range of 1350–1450 °C and the lower basicity of range about 1.5–1.6 are the main process range for the dephosphorization in the double slag converter steelmaking process, it is necessary to clarify the effect of the dephosphorization endpoint temperature on hot metal dephosphorization.

In this paper, through the 180 t double slag converter steelmaking industrial experiments, the effect of the dephosphorization endpoint temperature on the dephosphorization was studied at the low basicity slag in the relatively low temperature range of 1350–1450 °C. In addition, XRD and SEM-EDS were used to study the phase species and phase compositions of dephosphorization slag under the optimal dephosphorization ratio, and the original temperature on enrichment path of phosphorus in dephosphorization slag was reasonably speculated. The phosphorus capacity and phosphorus capacity and phosphorus capacity and phosphorus distribution of distribution and phosphorus distribu effect of dephosphorization endpoint temperature on the oxygen activity at different reaction interfaces was analyzed. The phosphorus capacity and phosphorus distribution ratio tion interfaces was analyzed. The phosphorus capacity and phosphorus distribution ratio of slag were calculated and compared with the empirical formula of the previous scholars. The optimum deslagging temperature of dephosphorization slag was further analyzed. analyzed. ments, the effect of the dephosphorization endpoint temperature on the dephosphorizathe low basic through the 100 t double slag converter steemaking moustrial experi-

# **2. Experimental Procedure 2. Experimental Procedure**

ature on hot metal dephosphorization.

# *2.1. Double Slag Converter Steelmaking Experiments 2.1. Double Slag Converter Steelmaking Experiments*

The industrial experiment was carried out with a 180 t converter in the steelmaking The industrial experiment was carried out with a 180 t converter in the steelmaking plant of Ningbo Iron and Steel Co. Ltd., Ningbo, China. Figure [1](#page-2-0) shows the flow chart of the double slag converter steelmaking process, and the double slag converter steelmaking the double slag converter steelmaking process, and the double slag converter steelmaking process is mainly divided into eight steps. process is mainly divided into eight steps.

<span id="page-2-0"></span>

**Figure 1.** Flow chart of double slag converter steelmaking process. **Figure 1.** Flow chart of double slag converter steelmaking process.

After adding light burned dolomite as a slag-adjusting agent, slag splashing was carried out to protect the furnace wall. The added auxiliary materials included lime, magnesite, sinter, OG (Oxygen converter Gas recovery) slag pellets, etc. The OG slag pellets contained 55% T.Fe, 12.4% CaO, and 3.8%  $SiO<sub>2</sub>$ , and they mainly include the mud produced by oxygen converter gas recovery. "T.Fe" refers to the total content of iron element of iron element of its and the total content of its and the total content of its and the total content of its and the total cont element in slag, (mass%). In addition, about 20 mass% of iron oxide scale, 13 mass% of lime, 2 mass is a mass of lime, 2 mass of lime, 2 mass of lime  $\frac{1}{2}$  mass of lime  $\frac{1}{2}$  mass of lime  $\frac{1}{2}$  mass of lime  $\$ lime, 2 mass% of binder, and 65 mass% of OG mud are molded by a high-pressure ball pressure ball pressing machine and then dried at low temperature to form OG slag pellets, which can regulate converter temperature and promote rapid slag formation.

After confirming the solidification of the slag, the active lime and the scrap steel is added at first; then, the hot metal is poured into the furnace. Finally, an oxygen lance is ded at first; then, the hot metal is poured into the furnace. Finally, an oxygen lance is ignited for dephosphorization and a low–high–low lance position is used in the dephosphorization stage. The initial low–lance position aims to promote the ignition of the oxygen profits the subsequent high lance position aims to promote the ignition of the  $\omega$ , gentled blowing. The subsequent high lance position is to promote the formation of molten slag in the early stage of dephosphorization and to increase the T.Fe content in slag. At the later molten slag in the early stage of dephosphorization and to increase the T.Fe content in stage of dephosphorization, the low lance position is used to promote dephosphorization. After the dephosphorization, stage is completed, the dephosphorization slag is poured out After confirming the solidification of the slag, the active lime and the scrap steel is by tilting the furnace, and then lime, dolomite, and other auxiliary materials are added for decarburization. After the decarburization, the molten steel is tapped out, and the decarburization slag is left in the furnace for the dephosphorization in the next charge.

#### *2.2. Compositions of the Initial Hot Metal, the Hot Metal at the Endpoint of Dephosphorization, and Dephosphorization Slag*

The compositions and temperatures of the initial hot metal in the converter are shown in Table [1.](#page-3-0) The number of experimental heats is T1350~T1450, which is used as the number of subsequent hot metal samples at the endpoint of dephosphorization and the dephosphorization slag samples. The hot metal after desulfurization is used for dephosphorization, and the temperature of the initial hot metal charged into the furnace is in the range of 1326–1366 ◦C. The added amount of the initial hot metal is 190–195 t, and the added amount of scrap steel is 20–30 t.

| Sample            | C    | Si   | Mn   | S    | P     | $T$ / $\subset$ |
|-------------------|------|------|------|------|-------|-----------------|
| T <sub>1350</sub> | 4.65 | 0.29 | 0.18 | 0.03 | 0.145 | 1330            |
| T <sub>1360</sub> | 4.65 | 0.36 | 0.25 | 0.01 | 0.162 | 1326            |
| T <sub>1370</sub> | 4.65 | 0.34 | 0.29 | 0.03 | 0.148 | 1350            |
| T1385             | 4.65 | 0.32 | 0.29 | 0.01 | 0.157 | 1350            |
| T1395             | 4.65 | 0.41 | 0.28 | 0.01 | 0.141 | 1365            |
| T <sub>1405</sub> | 4.65 | 0.29 | 0.27 | 0.01 | 0.129 | 1366            |
| T <sub>1410</sub> | 4.65 | 0.31 | 0.25 | 0.02 | 0.136 | 1357            |
| T1420             | 4.65 | 0.40 | 0.31 | 0.04 | 0.135 | 1355            |
| T1450             | 4.65 | 0.39 | 0.29 | 0.03 | 0.158 | 1365            |

<span id="page-3-0"></span>**Table 1.** The compositions and temperatures of the initial hot metal in converter (mass%).

After the dephosphorization reaction, the slag sample is taken by sticking onto an iron rod and cooling in air. After grinding, the slag sample was passed through a 200-mesh sieve to remove the residual iron particles remaining in the slag. The compositions of slag samples were analyzed with an M4 TORNADO fluorescence spectrometer of Bruker Company in Germany. The temperature was measured, the hot metal was sampled with a sub-gun, and the compositions of hot metal were analyzed by SPECTROMAXx direct reading spectrometer of German SPECTRO Company. Before the analyses, the spectrometer was calibrated with the standard samples of the similar compositions.

Table [2](#page-4-0) shows hot metal compositions, dephosphorization ratios, and temperatures at the endpoint of dephosphorization. Table [3](#page-4-1) shows dephosphorization slag compositions and basicities at the endpoint of dephosphorization. It can be seen from Table [2](#page-4-0) that the temperature of the dephosphorization endpoint is in the range of 1350–1450 ◦C. Basicity B is calculated with Equation (1), and  $\eta_P$  represents the dephosphorization ratio, which is calculated by Equation (2) [\[19\]](#page-21-18):

$$
B = \frac{(\%CaO)}{(\%SiO2)}\tag{1}
$$

$$
\eta_P = \frac{[\%P]_0 - [\%P]_e}{[\%P]_0} \times 100\% \tag{2}
$$

(%CaO) and (%SiO<sub>2</sub>) are the mass percentages of CaO and SiO<sub>2</sub> in the dephosphorization slag respectively,  $[\%P]_0$  is the initial phosphorus content in initial hot metal, and  $[\%P]_e$  is the phosphorus content in the hot metal at the end of dephosphorization.

The phase analysis of dephosphorization slag was carried out by a D8 Advance X-ray powder diffractometer (XRD) of Bruker company in Germany. In the range of 2θ = 10–90◦ and step size of  $0.04^{\circ} s^{-1}$ , XRD data were collected by Cu-K $\alpha$  radiation. A small amount of massive dephosphorization slag was embedded in epoxy resin. After solidification, it was ground and polished by an automatic grinding and polishing machine of the PRESI company in France. Then, the surface was carbonized by a magnetron sputtering MC1000 of the HITACHI company in Japan. The morphology of the P-rich phase of dephosphorization slag was observed by a Zeiss EVO 18 electronic scanning microscope of Zeiss company of Germany, and the chemical composition of the P-rich phase in dephosphorization slag was

analyzed by an X-MaxN large area energy-dispersive spectrometer (SEM-EDS) of Oxford Instruments Company in the UK.

<span id="page-4-0"></span>**Table 2.** Hot metal compositions, dephosphorization ratios, and temperatures at the end of dephosphorization (mass%).

| Sample            | C     | Si    | Mn    | S     | P     | $\eta_P$ /% | $T$ / $\rm ^{\circ}$ C |
|-------------------|-------|-------|-------|-------|-------|-------------|------------------------|
| T <sub>1350</sub> | 3.076 | 0.010 | 0.060 | 0.018 | 0.073 | 49.7        | 1350                   |
| T1360             | 3.081 | 0.025 | 0.089 | 0.010 | 0.080 | 50.6        | 1360                   |
| T <sub>1370</sub> | 3.241 | 0.015 | 0.097 | 0.009 | 0.071 | 52.0        | 1370                   |
| T <sub>1385</sub> | 3.039 | 0.010 | 0.092 | 0.010 | 0.069 | 56.1        | 1385                   |
| T <sub>1395</sub> | 3.069 | 0.022 | 0.117 | 0.018 | 0.050 | 64.5        | 1395                   |
| T1405             | 2.942 | 0.009 | 0.167 | 0.018 | 0.034 | 73.6        | 1405                   |
| T <sub>1410</sub> | 2.943 | 0.012 | 0.143 | 0.017 | 0.051 | 62.5        | 1410                   |
| T <sub>1420</sub> | 3.253 | 0.017 | 0.138 | 0.012 | 0.069 | 48.9        | 1420                   |
| T <sub>1450</sub> | 3.488 | 0.015 | 0.147 | 0.019 | 0.103 | 34.8        | 1450                   |

<span id="page-4-1"></span>**Table 3.** Compositions and basicities of dephosphorization slag (mass%).



#### **3. Results**

#### *3.1. Effect of Dephosphorization Endpoint Temperature on Liquidus Region of Dephosphorization Slag*

Figure [2](#page-5-0) shows the effect of dephosphorization endpoint temperature on liquidus regions of the CaO-SiO<sub>2</sub>-FeO-10%MgO-10%MnO-5% $P_2O_5$  dephosphorization slag. The liquidus projection section of dephosphorization slag at 1340–1460  $\degree$ C was drawn by a phase diagram in Factsage7.3. Figure [2](#page-5-0) shows that with increasing dephosphorization endpoint temperature, the liquid phase area of dephosphorization slag increases continuously, and the slag can be melted better. At the same time, the dephosphorization slag selected in the present work is marked in Figure [2](#page-5-0) by using the ternary phase diagram of  $CaO, SiO<sub>2</sub>$ , and FeO. Since the selected dephosphorization slags are from the different heats of the double slag converter steelmaking experiment, the basicities of dephosphorization slag are slightly different. Nine groups of dephosphorization slag components all located in red quadrilateral area indicate that the difference in the basicities of these dephosphorization slags is quite small. Therefore, it can be considered that the experimental results are mainly affected by the dephosphorization endpoint temperature.

#### *3.2. Effect of Dephosphorization Endpoint Temperature on Dephosphorization*

In industrial experiments, it is difficult to ensure that the other process parameters are the same to investigate the effect of dephosphorization endpoint temperature on hot metal dephosphorization. In the heats selected in the present work, the slag basicities ranged from 1.48 to 1.64, and the FeO contents in slag were from 16.38% to 18.97%. The MgO contents in slag from 7.24% to 9.35% increased with the increasing dephosphorization endpoint temperature, which was mainly caused by the erosion of the converter wall. The contents of CaO, SiO<sub>2</sub>, MnO, and  $Al_2O_3$  in the dephosphorization slag changed very little.

<span id="page-5-0"></span>

**Figure 2.** Effect of dephosphorization endpoint temperature on liquidus regions of the CaO-SiO<sub>2</sub>-FeO-10%MgO-10%MnO-5%P2O5 dephosphorization slag. FeO-10%MgO-10%MnO-5%P2O<sup>5</sup> dephosphorization slag.

Figure 3 shows the effect of dephosphorization endpoint temperature on the dephosprocrimation ratio, prospressive content in not metal, and  $\frac{1}{2}\sigma_3$  content in dephosphorization.<br>Slag at the endpoint of dephosphorization. When the dephosphorization endpoint temperature is 1350–1385 °C, the dephosphorization ratio is between 49.7 and 56.1%. The phosphorus content and dephosphorization ratio in hot metal have no obvious change in phosphorus content and dephosphorization ratio in hot metal have no obvious change in prospected content and dephosphorization endpoint temperature is increased to 1385–1405 °C, the dephosphorization ratio increases significantly. At 1405 °C, the dephosphorization ratio  $\frac{1}{2}$ reaches the highest value of 73.64% and the phosphorus content in hot metal reduces to the lowest value of 0.034%. This is because when the dephosphorization endpoint temperature is relatively low in the temperature range of 1350–1385 °C, although the thermodynamic condition is beneficial to dephosphorization, the lime cannot be well melted during the very short time period of about 5 min during the dephosphorization period in the double slag converter steelmaking process. Therefore, the poor kinetic conditions hindered the dephosphorization reaction. When the temperature is in the range of 1385–1405 °C, the auxiliary materials in the converter are well melted, the slagging is improved, and the dephosphorization thermodynamic conditions can still promote the dephosphorization of the converter, so the dephosphorization ratio is significantly increased in this range. phorization ratio, phosphorus content in hot metal, and  $P_2O_5$  content in dephosphorization

In the dephosphorization endpoint temperature range of 1350–1405 °C, the content of  $P_2O_5$  in the dephosphorization slag increases with the increasing dephosphorization endpoint temperature and the content of  $P_2O_5$  in the dephosphorization slag is relatively high, which is from 3.17% to 4.17%. This is because with the increasing dephosphorization endpoint temperature, the lime begins to melt, and the kinetic conditions of the reaction between molten slag and hot metal are improved, so that the initial dephosphorization capacity increases. When the dephosphorization endpoint temperature is  $1405 °C$ , the  $P_2O_5$  content reaches the maximum value of 4.17%.

When the dephosphorization endpoint temperature is further increased to higher than 1405 °C, the dephosphorization ratio of hot metal is decreased, the phosphorus content at the endpoint of dephosphorization is increased, and the content of  $P_2O_5$  in the dephosphorization slag decreases. The reason is that the dephosphorization reaction is exothermic, and the high temperature will inhibit the dephosphorization reaction. Although the slagging is promoted and the kinetic conditions of dephosphorization in the converter are improved, too high temperature has a greatly adverse effect on the thermodynamic conditions of dephosphorization, leading to a great decrease in the equilibrium constant of dephosphorization reaction. Therefore, the thermodynamic and kinetic conditions of dephosphorization jointly determine the dephosphorization ratio.

<span id="page-6-0"></span>

**Figure 3.** Effect of dephosphorization endpoint temperature on the dephosphorization ratio, phosphorus content in hot metal, and  $P_2O_5$  content in dephosphorization slag at the endpoint of dephosphorization.

According to the results of Figure [3,](#page-6-0) when the dephosphorization endpoint temperature is between 1385 and 1410 ◦C, the dephosphorization ratio at the endpoint of dephosphorization rises to be above 55%, and the content of  $P_2O_5$  in the dephosphorization slag is above 3.90%, with the excellent results of dephosphorization of hot metal.

## *3.3. Analysis of Dephosphorization Slag at the Optimum Dephosphorization Temperature of 1405* ◦*C*

Figure [4](#page-7-0) shows the XRD analysis results of dephosphorization slag at 140[5](#page-7-1) °C. Figure 5 shows the images of dephosphorization slag by SEM-EDS at 1405  $°C$ , with magnification of (a) 200 times and (b) 500 times, respectively. As shown in Figure [4,](#page-7-0) the dephosphorization slag is composed of dicalcium silicate  $Ca_2SiO<sub>4</sub>$ , tricalcium silicate  $Ca_3SiO<sub>4</sub>$ , phosphorus containing solid solution  $6C_2S-C_3P$ , silicate phase  $Ca_3Mg(SiO_4)_2$ , calcium ferrite phase  $Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub>$ , and oxide phase RO. It can be seen from Figure [5a](#page-7-1) that there is dark gray phase 1, gray phase 2, light gray phase 3, white phase 4, gray phase 5 and black phase 6 in the dephosphorization slag at the dephosphorization endpoint temperature of 1405 ◦C with the basicity of 1.64. Furthermore, SEM-EDS point analysis and map scanning were used to analyze the phase composition. Table [4](#page-8-0) shows the elements compositions of different phases in dephosphorization slag by SEM-EDS point analysis in Figure [5a](#page-7-1), and Figure [6](#page-7-2) shows the map scanning results of dephosphorization slag at 1405 °C under the magnification of 200 times.

It can be seen from Table [4](#page-8-0) that the phosphorus content in the dark gray phase 1-1, 1-2, and 1-3 is the highest, with an average value of 4.34%, followed by the phosphorus content in the gray phase 2-1, 2-2, and 2-3, with an average value of 3.44%, and the phosphorus content in the light gray phase 3, with the value of 2.20%. The contents of phosphorus in white phase 4, gray white phase 5, and black phase 6 were less than 1.5%. According to the XRD analysis results in Figure [4](#page-7-0) and the map scanning results in Figure [6,](#page-7-2) it can be judged that the dark gray phase 1, the gray phase 2, and the light gray phase 3 are all phosphorus rich phases (P-rich phase), but their phosphorus enrichment degrees are different, and the order of phosphorus enrichment degree is the dark gray phase  $1$  > the gray phase  $2$  > the light gray phase 3. The content of Fe in the white phase 4 is very high, so it can be judged that the white phase 4 is the mixed iron particles. The gray white phase 5 has high contents

of Fe and Ca, which can be identified as  $Ca_2Fe_2O_5$  by XRD analysis, which is named the Fe-rich phase. The black phase 6 contains a high content of Ca, Si, and a small amount of phosphorus. Therefore, the black phase 6 is considered as the matrix phase, which is known as the calcium silicate phase. From the SEM image of dephosphorization slag in Figure [5b](#page-7-1), it is found that there are two kinds of Fe-rich phases in the dephosphorization slag at the dephosphorization endpoint temperature of 1405  $\degree$ C with the basicity of 1.64, which are petal-like and strip-like respectively. There are large P-rich phases in the slag with the size of more than  $40 \mu m$ , and the calcium silicate phases are dispersed in small pieces between the P-rich phase and the Fe-rich phase.

<span id="page-7-0"></span>

**Figure 4.** XRD analysis result of dephosphorization slag at 1405 ◦C. **Figure 4.** XRD analysis result of dephosphorization slag at 1405 °C.

<span id="page-7-1"></span>

Figure 5. Images of dephosphorization slag by SEM-EDS at 1405  $^{\circ}$ C of (a) 200 times; (b) 500 times.  $\frac{1}{2}$  is the P-rich phase phase and the P-rich phase and the Fe-rich phase and  $\frac{1}{2}$  is the Fe-rich phase.

<span id="page-7-2"></span>

 $\mu$  $\frac{1}{2}$   $\frac{1}{2}$  **Figure 6.** Element distribution results of dephosphorization slag at 1405 °C. **Figure 6.** Element distribution results of dephosphorization slag at 1405 ◦C.

| Position | Ca    | Si    | Mn   | Mg   | P    | Al   | Fe    | O     |
|----------|-------|-------|------|------|------|------|-------|-------|
| $1 - 1$  | 34.77 | 12.94 | 2.70 | 2.03 | 4.34 | 0.16 | 1.62  | 41.44 |
| $1-2$    | 35.98 | 12.37 | 2.90 | 2.08 | 4.25 | 0.12 | 1.26  | 41.04 |
| $1 - 3$  | 37.00 | 13.93 | 3.63 | 1.36 | 4.42 | 0.01 | 2.22  | 37.43 |
| $2 - 1$  | 29.68 | 13.23 | 3.39 | 2.94 | 3.61 | 1.65 | 2.65  | 42.85 |
| $2 - 2$  | 35.25 | 13.80 | 2.18 | 1.99 | 3.51 | 0.14 | 1.58  | 41.55 |
| $2 - 3$  | 29.06 | 13.27 | 3.78 | 2.19 | 3.20 | 2.13 | 3.16  | 43.21 |
| 3        | 30.56 | 12.55 | 6.02 | 2.49 | 2.20 | 0.85 | 5.80  | 39.53 |
| 4        | 2.32  | 0.92  | 3.55 | 1.35 | 0.17 | 0.43 | 82.14 | 9.12  |
| 5        | 20.36 | 10.22 | 7.33 | 2.92 | 1.14 | 2.03 | 24.06 | 31.94 |
| 6        | 27.08 | 17.79 | 3.19 | 3.33 | 1.44 | 2.56 | 4.15  | 40.46 |

<span id="page-8-0"></span>**Table 4.** Element compositions of different phases in dephosphorization slag by SEM-EDS point analysis (mass%).

## *3.4. Analysis of the P-Rich Phase in Dephosphorization Slag at 1405* ◦*C*

SEM-EDS was used to further analyze the boundary of the P-rich phase. Figure [7a](#page-8-1),b show the P-rich phase in dephosphorization slag at 1405 ◦C with the magnifications of 1000 times and 2000 times, Figure [7c](#page-8-1) shows the schematic diagram of the P-rich phase boundary, and Figure [7d](#page-8-1) shows the schematic diagram of formation process of the P-rich phase. In Figure  $7a,b$  $7a,b$ , the gray massive phase is the P-rich phase, the gray white phase is the Fe-rich phase and the black phase is calcium silicate in dephosphorization slag. At the same time, it is found that the gray P-rich phase is surrounded by the black calcium silicate phase, and the outermost part is surrounded by a layer of the gray white Fe-rich phase. Furthermore, the P-rich phase of dephosphorization slag was observed at 2000 times magnification, and the size of different phases was measured by Image Pro Plus image software. It can be seen that the size of the P-rich phase is  $45.31 \mu m$ , the width of the black calcium silicate phase wrapped outside is about 1.15  $\mu$ m, and the width of the outermost gray white Fe-rich phase is about  $0.74 \mu m$ .

<span id="page-8-1"></span>

Figure 7. Images of the P-rich phase in dephosphorization slag at  $1405\,^{\circ}\text{C}$  of (a)  $1000$  times; (b)  $2000$  times; (c) Schematic diagram of P-rich phase boundary; (d) Schematic diagram of formation process of the P diagram of P-rich phase boundary; (**d**) Schematic diagram of formation process of the P-rich phase.

From Table [4,](#page-8-0) the content of phosphorus in the P-rich phase is higher than that in the calcium silicate phase, and the content of phosphorus in the calcium silicate phase is higher than that in the Fe-rich phase. Therefore, the enrichment path of phosphorus in hot metal can be reasonably inferred from the content gradient of phosphorus in the different phases, and the formation process of the P-rich phase was speculated, as shown in Figure [7d](#page-8-1). The process from Figure [7\(](#page-8-1)d-1) to (d-2) shows that the solid CaO particles in matrix phase will first form  $C_2$ S at the surface of CaO [\[20–](#page-21-19)[22\]](#page-21-20). The process from Figure [7\(](#page-8-1)d-2) to (d-3) shows that the phosphorus in hot metal is oxidized by iron oxide to form  $P_2O_5$  and enters the slag, and FeO and CaO react with  $P_2O_5$  to form 3FeO $\cdot P_2O_5(F_3P)$  and  $C_3P$ , respectively, which initially fixes phosphorus in dephosphorization slag. Since the  $F_3P$  is extremely unstable especially in the high temperature, CaO in slag will further react with  $F_3P$  to form the stable  $C_3P$  phase [\[23\]](#page-21-21). Finally, phosphorus exists in the P-rich phase in the form of  $nC_2S-C_3P$ solid solution. Since the dephosphorization slag is a multiphase slag with solid–liquid coexistence between 1350 and 1450  $^{\circ}$ C, there is undissolved calcium silicate solid phase in the slag. With the progress of the dephosphorization reaction, the phosphorus containing solid solution grows into the massive P-rich phase. The calcium silicate phase wrapped outside shrinks, and the corresponding FeO-CaO layer shrinks, finally forming the P-rich phase boundary, as shown in Figure [7c](#page-8-1). Therefore, the path of phosphorus in hot metal entering the P-rich phase of dephosphorization slag can be reasonably inferred as: hot metal  $\rightarrow$  Fe-rich phase  $\rightarrow$  P-rich phase in dephosphorization slag.

#### **4. Discussion**

# *4.1. Effect of Dephosphorization Endpoint Temperature on aFeO and PP*−*<sup>C</sup> CO*

In the dephosphorization reaction, the iron oxide in slag acts as the main oxidant for dephosphorization. Therefore, the content of iron oxide in dephosphorization slag has a significant effect on the dephosphorization ability of dephosphorization slag. In this study, it is assumed that all iron oxides in slag exist in the form of FeO, so as to analyze the effect of dephosphorization endpoint temperature on  $a_{\text{FeO}}$ . It should be noted that the activity of each substance in the dephosphorization slag adopts pure substances as the standard state, and  $a_{\text{FeO}}$  in the dephosphorization slag is expressed as Equation (3) [\[24\]](#page-21-22).

$$
a_{\text{FeO}} = \gamma_{\text{FeO}} \times X_{\text{FeO}} \tag{3}
$$

 $X_i$  is the mole fraction of oxide *i* in dephosphorization slag, and  $\gamma_i$  is the activity coefficient of oxide *i*. The percent work uses the regular solution model to calculate  $γ_{FeO}$ , because this model can avoid the problem that the structure of the silicate anion changes with the number and type of cations and the temperature. In this model, it is assumed that various cations, such as  $Ca^{2+}$ ,  $Fe^{2+}$ ,  $Si^{4+}$ , and  $P^{5+}$ , are randomly distributed in the  $O<sup>2</sup>$  matrix, and  $O<sup>2</sup>$  is the common anion of various cations in the melt. According to the interaction energy between cations summarized by Ban-ya [\[25\]](#page-21-23), γ<sub>FeO</sub> can be expressed as Equation (4).

RT In 
$$
\gamma_{FeO} = 7110X_{MnO}^2 - 31,380X_{CaO}^2 + 33,470X_{MgO}^2 - 41,840X_{SiO2}^2 - 31,380X_{P2O5}^2
$$
  
\n $-41,000X_{A12O3}^2 + 67,780X_{MnO}X_{CaO} - 21,410X_{MnO}X_{MgO} + 40,580X_{MnO}X_{SiO2}$   
\n $+60,670X_{MnO}X_{P2O5} + 49,790X_{MnO}X_{A12O3} + 102,510X_{CaO}X_{MgO} + 60,670X_{CaO}X_{SiO2}$   
\n $+188,280X_{CaO}X_{P2O5} + 82,430X_{CaO}X_{A12O3} + 58,570X_{MgO}X_{SiO2} + 39,750X_{MgO}X_{P2O5}$   
\n $+63,600X_{MgO}X_{A12O3} - 156,900X_{SiO2}X_{P2O5} + 45,130X_{SiO2}X_{A12O3} + 189,120X_{P2O5}X_{A12O3}$  (4)

During the dephosphorization process of the converter, as the temperature of the hot metal rises and the dephosphorization reaction proceeds, carbon and phosphorus will undergo a selective oxidation transformation. The selective oxidation reaction of carbon and phosphorus is shown in Equation (5). Namely, when the temperature is lower than the selective oxidation temperature of carbon and phosphorus, the phosphorus in hot metal is preferentially oxidized. At the higher temperature, carbon is preferentially oxidized. The value of *P<sub>CO</sub>* in a selective oxidation reaction of carbon and phosphorus is directly related to the selective oxidation temperature of carbon and phosphorus. Therefore, it is significant to study the relationship between the partial pressure of CO and dephosphorization endpoint temperature in the selective oxidation of carbon and phosphorus for the dephosphorization of hot metal. In this paper,  $P_{\text{CO}}^{\text{P}-\text{C}}$  is used to represent the partial pressure of CO in the selective oxidation of carbon and phosphorus.

$$
2[P] + 5CO = (P2O5) + 5[C]
$$
 (5)

$$
\Delta G^{\theta} = -594,505 + 748.25T\tag{6}
$$

$$
K^{\Theta} = \frac{a_{P2O5} \times a_{[C]}^{5}}{P_{CO}^{5} \times a_{[P]}^{2}}
$$
 (7)

Equation (6) is the Gibbs free energy of the carbon and phosphorus selective oxidation reaction (J). Equation (7) is the equilibrium constant expression of the reaction.  $P_2O_5$  in dephosphorization slag adopts pure substance as the standard state.  $a_{\text{[C]}}$  and  $a_{\text{[P]}}$  represent the activities of carbon and phosphorus in hot metal, respectively. The standard state of the activity of  $a_{\text{[C]}}$  and  $a_{\text{[P]}}$  in hot metal is 1% mass fraction solution, which conforms to Henry's law. Therefore,  $a_{\text{[C]}}$  and  $a_{\text{[P]}}$  can be expressed by Equations (8) and (9).  $a_{\text{P2O5}}$  is expressed as Equation (10).

$$
a_{\left[C\right]} = f_{\left[C\right]} \times \left[ \% \text{C} \right] \tag{8}
$$

$$
a_{\left[\mathrm{P}\right]} = f_{\left[\mathrm{P}\right]} \times \left[\% \mathrm{P}\right] \tag{9}
$$

$$
a_{P2O5} = \gamma_{P2O5} \times X_{P2O5} \tag{10}
$$

It should be noted that *γ*P2O5 is calculated by empirical Equation (11) [\[26\]](#page-21-24). The reason for not using regular solution calculation is that the value of  $a_{P2O5}$  in the actual slag is too small, usually smaller than  $10^{-17}$ . [\[27\]](#page-21-25) There is an error in the process of converting *a*<sub>PO2.5</sub> to *a*<sub>P2O5</sub> obtained by the regular solution model, and the obtained *a*<sub>P2O5</sub> is far from the calculated values in previous studies. [\[28–](#page-21-26)[30\]](#page-22-0) Based on the above reasons, the empirical formula summarized by Turkdogan and Pearson is selected to calculate *γ*P2O5 [\[26\]](#page-21-24), and the activity coefficients of phosphorus and carbon in hot metal can be calculated by Equations (12) and (13).

$$
\log \gamma_{P2O5} = -1.12(22X_{CaO} + 15X_{MgO} + 13X_{MnO} + 12X_{FeO} - 2X_{SiO_2}) -\frac{42,000}{T} + 23.58
$$
\n(11)

$$
\log f_{\rm [P]} = e_{\rm P}^{\rm C} [\%{\rm C}] + e_{\rm P}^{\rm Si} [\%{\rm Si}] + e_{\rm P}^{\rm Mn} [\%{\rm Mn}] + e_{\rm P}^{\rm S} [\%{\rm S}] + e_{\rm P}^{\rm P} [\%{\rm P}] \tag{12}
$$

$$
\log f_{\rm [C]} = e_{\rm C}^{\rm C} [\%{\rm C}] + e_{\rm C}^{\rm Si} [\%{\rm Si}] + e_{\rm C}^{\rm Mn} [\%{\rm Mn}] + e_{\rm C}^{\rm S} [\%{\rm S}] + e_{\rm C}^{\rm P} [\%{\rm P}] \tag{13}
$$

 $e_P^i$  and  $e_C^i$  are the interaction coefficients of solute element *i* to phosphorus and carbon dissolved in hot metal, respectively. With the progress of the dephosphorization reaction, a part of phosphorus in hot metal is fixed in dephosphorization slag and removed from hot metal. At the endpoint of dephosphorization stage in the double slag converter steelmaking process, it can be regarded that the dephosphorization quasi-equilibrium state is reached under the conditions of lower temperature and lower basicity. The value of *a*<sub>FeO</sub> can be calculated by Equations (3) and (4), and the value of  $P_{\text{CO}}^{\text{P}-\text{C}}$  can be calculated by Equations (6)–(13). Considering that  $a_{FeO}$  is mainly affected by the basic oxide contents in slag and the dephosphorization endpoint temperature, through the multiple linear regression on  $a_{\text{FeO}}$ , Equation (14) can be obtained with the regression coefficient  $\mathbb{R}^2$  of is 0.97. Figure [8a](#page-11-0) shows the comparison between the predicted value based on multiple regression and the calculated value of  $a_{\text{FeO}}$  from the regular solution model of Equations (3) and (4). The predicted value is well consistent with the calculated result.

$$
a_{\text{FeO}} = 0.0096 \times (\% \text{FeO}) - 0.004 \times (\% \text{CaO}) + 0.0026 \times (\% \text{MnO}) + 0.0134 \times (\% \text{MgO}) - 2.8923 \times 10^{-4} \times T + 0.5442
$$
\n(14)

<span id="page-11-0"></span>

**Figure 8.** (a) Comparison between the predicted value based on multiple regression and the calculated value of  $a_{FeO}$  from the regular solution model; (**b**) Effect of dephosphorization endpoint temperature on  $P_{\rm CO}^{\rm P-C}$  of selective oxidation reaction of carbon and phosphorus.

Figure [8b](#page-11-0) shows the effect of dephosphorization endpoint temperature on  $P_{\rm CO}^{\rm P-C}$  of a selective oxidation reaction of carbon and phosphorus. In Figure [8b](#page-11-0),  $P_{\rm CO}^{\rm P-C}$  data were fitted by an exponential curve, and the regression coefficient,  $\mathbb{R}^2$ , is 0.98, which indicates that there is a good exponential relationship between the endpoint temperature of dephosphorization and  $P_{\rm CO}^{\rm P-C}$ ; the fitting equation is shown in Equation (15). With increasing dephosphorization endpoint temperature,  $P_{\text{CO}}^{\text{P}−\text{C}}$  shows an exponential upward trend.

$$
P_{\rm CO}^{\rm P-C} = 4.11 \times \exp(\frac{T - 576.82}{211.98}) - 126.41\tag{15}
$$

The blue dashed line area in Figure [8b](#page-11-0) shows that the dephosphorization ratio is greater than 55%, and the temperature range is 1385–1410 °C. In this range, the dephosphorization result is preferential, the  $P_{\rm CO}^{\rm P-C}$  value is 53–80 Pa, and  $a_{\rm FeO}$  is 0.284–0.312. These two ranges are conducive to the dephosphorization of the converter.

#### <span id="page-11-1"></span>*4.2. Effect of Dephosphorization Endpoint Temperature on the Oxygen Activity on Hot Metal Surface, the Phosphorus Oxidation Equilibrium Oxygen Activity, and the Iron Oxidation Equilibrium Oxygen Activity at Slag–Hot Metal Interface*

In the converter dephosphorization process, the oxygen in hot metal is mainly used to remove elements such as silicon, manganese, phosphorus, and carbon. Before dephosphorization reaches quasi equilibrium, with the decarburization reaction going on, the carbon in hot metal decreases continuously, and the carbon–oxygen reaction is in the state of excess oxygen. The equilibrium oxygen content in hot metal mainly depends on the carbon content in hot metal [\[16\]](#page-21-15). The carbon–oxygen reaction in the hot metal is shown in Equation (16), the Gibbs free energy of the reaction is expressed by Equation (17), and the equilibrium constant is given by Equation (18). It should be noted that *P*<sub>CO</sub> in Equation (16) is the partial pressure of CO in the converter, and it is assumed to the value of 1 atm. In this paper,  $P_{\rm CO}^{\rm C-O}$  is used to represent the partial pressure of CO in the converter.

$$
[C] + [O] = CO \tag{16}
$$

$$
\Delta G^{\theta} = -22,200 - 38.34T\tag{17}
$$

$$
K^{\Theta} = \frac{P_{\text{CO}}^{\text{C}-\text{O}}}{a_{\text{[C]}} \times a_{\text{[O]}}}
$$
(18)

Combined with Equations (16)–(18), the theoretical oxygen activity on the surface of hot metal of  $a_{[\mathrm{O}]}^{\mathrm{iron}}$  can be deduced as shown in Equation (19).

$$
\log a_{\text{[O]}}^{\text{iron}} = -\frac{1160}{T} - 2.002 + \log P_{\text{CO}}^{\text{C-O}} - \log a_{\text{[C]}} \tag{19}
$$

At the slag–hot metal reaction interface, phosphorus in hot metal is oxidized to  $P_2O_5$ , which enters slag and exists in the form of  $C_2S-C_3P$  solid solution. The oxidative dephosphorization reaction occurs as Equation (20). Equation (21) is the equilibrium constant of dephosphorization reaction obtained by Turkdogan and Pearson [\[26\]](#page-21-24), and Equation (22) is the expression of the equilibrium constant.

$$
2[P] + 5[O] = (P2O5)
$$
 (20)

$$
\log K = \frac{37,160}{T} - 29.67\tag{21}
$$

$$
K^{\Theta} = \frac{a_{\text{P2O5}}}{a_{\text{[P]}}^2 \times a_{\text{[O]}}^5}
$$
(22)

When the dephosphorization reaction of slag and hot metal reaches quasi equilibrium, by combining Equations (20)–(22), the oxygen activity of phosphorus oxidation at the slag–hot metal interface of  $a_{\rm [O]}^{\rm de [P]}$  $\frac{u_{e[1]}}{[O]}$  is derived as Equation (23).

$$
\log a_{\text{[O]}}^{\text{de[P]}} = \left(-\frac{37,160}{T} + 29.67 + \log a_{\text{P2O5}} - 2\log a_{\text{[P]}}\right) / 5\tag{23}
$$

The oxygen activity in the slag is mainly determined by the oxygen in the form of iron oxides. This article assumes that all iron oxides exist in the form of FeO, and the slag–hot metal interface reaches the quasi-equilibrium state of oxidation. Therefore, the oxygen activity of iron oxidation at the slag–hot metal interface can be expressed by the iron oxide reaction, as shown in Equation (24). The Gibbs free energy of reaction is shown in Equation (25) [\[24\]](#page-21-22), and the equilibrium constant is shown in Equation (26).

$$
[Fe] + [O] = (FeO)
$$
\n
$$
(24)
$$

$$
\Delta G^{\theta} = -128,090 + 57.990T\tag{25}
$$

$$
K^{\Theta} = \frac{a_{\text{FeO}}}{a_{\text{[O]}} \times a_{\text{[Fe]}}}
$$
 (26)

The concentration of [Fe] in hot metal is very high, and the value of  $a_{[Fe]}$  is taken as 1. When the iron oxidation reaction at the slag–hot metal interface reaches quasi equilibrium, by combining Equations (3), (4), and (26), the oxygen activity of the iron oxidation at the slag–hot metal interface of *a*<sup>[Fe]–[O]</sup>  $[0]$  is obtained as shown in Equation (27).

$$
\log a_{\text{[O]}}^{\text{[Fe]}-[\text{O}]} = -\frac{6689.77}{T} + 3.029 + \log a_{\text{FeO}} \tag{27}
$$

The effect of the dephosphorization endpoint temperature on oxygen activity at different reaction interfaces in the dephosphorization stage of the converter is calculated by using Equations (19), (23), and (27), as shown in Figure [9a](#page-13-0). It can be seen from Figure [9a](#page-13-0) that  $a_{\text{LO}}^{\text{de[P]}}$ <sup>de[P]</sup> and *a*<sup>[Fe]−[O]</sup><br>[O]  $\mathcal{L}^{[1] \times [O]}$  both increase with increasing dephosphorization endpoint temperature as a whole, while  $a_{[\text{O}]}^{\text{iron}}$  first increases and then decreases. These results can be explained by the thermodynamic principle. The main decarburization reaction in steelmaking is Equation (16), which is an exothermic reaction. With dephosphorization endpoint temperature from 1350 to 1410 ℃, the carbon-oxygen product increases; under the condition of little change in carbon content, the  $a_{[O]}^{\text{iron}}$  and  $a_{[O]}^{\text{de}[P]}$  $[O]$  increase rapidly [\[31\]](#page-22-1). When the

dephosphorization endpoint temperature is  $1410-1450$  °C, the dephosphorization reaction is weakened, the oxygen content of reaction with phosphorus is reduced, the phosphorus content increases from 0.051% to 0.103%, and the carbon content increases from 2.94% to 3.49% in hot metal, so the  $a_{[\text{O}]}^{\text{iron}}$  decreases, and  $a_{[\text{O}]}^{\text{de}[\text{P}]}$ <sup>de[P]</sup> increases slowly. Since *a*<sup>[Fe]−[O]</sup>  $\prod_{[0]}^{[16]-[0]}$  is mainly affected by the iron oxide content in slag and temperature,  $a_{\text{[O]}}^{\text{[Fe]}-\text{[O]}}$  $[O]$  increases with the increase of temperature when the iron oxide content does not change obviously.

<span id="page-13-0"></span>

**Figure 9. (a)** Effect of dephosphorization endpoint temperature on  $a_{[O]}^{\text{iron}}, a_{[O]}^{\text{de[P]}}$ <sup>de[P]</sup>, and *a*<sup>[Fe]−[O]</sup><br>[O]  $[O]$  ; Comparison between predicted value and calculated value based on multiple regression, (**b**)  $a_{[O]}^{\text{iron}}$ ; (**c**)  $a_{[O]}^{\text{de}[P]}$ <sup>de[P]</sup>; (**d**) *a*<sup>[Fe]–[O]<br>[O] ′′ (**d**) *a*<sup>[</sup>O]</sup>  $\begin{bmatrix} [e] - [S] \\ [O] \end{bmatrix}$ 

According to the different reaction interfaces of dephosphorization in the converter, the relationships between  $a_{[\text{O}]}^{\text{iron}}$ ,  $a_{[\text{O}]}^{\text{de}[\text{P}]}$ de[P] *, a* [Fe]−[O]<br>[O] ′ <sup>*a*</sup>[O]  $[0]$   $[0]$ , the dephosphorization endpoint temperature, and the compositions of hot metal and slag are fitted by multiple linear regression, which are expressed as Equations (28)–(30), respectively. It is found that the values of  $a_{\text{[O]}}$ calculated by Equations (19), (23), and (27) all have a good multiple linear relationship with the regression coefficients,  $\mathbb{R}^2$ , of 0.99, 0.97, and 0.96, respectively.

$$
a_{\text{[O]}}^{\text{iron}} = -1.529 \times 10^{-4} \times [ \%C] + 2.7657 \times 10^{-7} \times T + 3.258 \times 10^{-4}
$$
 (28)

$$
a_{\text{[O]}}^{\text{de[P]}} = -0.00211 \times (\% \text{P2O5}) - 0.131 \times [\% \text{P}] + 2.423 \times 10^{-4} \times T - 0.3017 \tag{29}
$$

$$
a_{\text{[O]}}^{\text{[Fe]-[O]}} = 1.7046 \times 10^{-4} \times T + 0.00125 \times (\% \text{FeO}) - 0.2274 \tag{30}
$$

It can be seen from Figure [9a](#page-13-0) that the increase trend of  $a_{\text{[O]}}^{\text{de[P]}}$  $[O]$  is slightly faster, and the growth trends of  $a_{\text{[O]}}^{\text{iron}}$  and  $a_{\text{[O]}}^{\text{[Fe]}-\text{[O]}}$  $[O]$  are slightly slower in the temperature range of 1350–1405 ◦C, which indicates that the effect of temperature on the phosphorus oxygen equilibrium of the slag–hot metal interface is relatively large in the temperature range of 1350–1405 °C. The increase trend of  $a_{\text{[O]}}^{\text{[Fe]-[O]}}$  $[O]$  is slightly faster in the temperature range of 1405–1450 ◦C, which indicates that the temperature has a greater effect on the iron oxygen equilibrium in the slag in temperature range of 1405–1450  $^{\circ} \textrm{C}.$ 

In the range of 1350–1450 °C,  $a_{\rm [O]}^{\rm [Fe]-[O]}$  ${}_{[O]}^{[Fe]-[O]} > a_{[O]}^{de[P]}$  $\mathbf{p}_{[\mathrm{O}]}^{\mathrm{de[P]}}>a_{[\mathrm{O}]}^{\mathrm{iron}}.$  The results show that  $a_{[\mathrm{O}]}^{\mathrm{[Fe]-[O]}}$  $[O]$ is the highest, while  $a_{\text{[O]}}^{\text{iron}}$  is the lowest, which is consistent with the research results of Kitamura. [\[17\]](#page-21-16) Since there is a large amount of Fe in the dephosphorization process, the oxygen activity at the slag–hot metal interface mainly depends on the oxygen activity in equilibrium with iron. According to the analysis in Figure [9,](#page-13-0)  $a_{\text{[O]}}^{\text{[Fe]}-\text{[O]}}$  $\begin{bmatrix} [ie] - [O] \\ [O] \end{bmatrix}$  plays a major role in the dephosphorization reaction in the range of 1350–1450 °C. Since the temperature range of the high dephosphorization ratio is 1385–1410 °C,  $a_{\rm [O]}^{\rm [Fe]-[O]}$  $[O]$  should be between 0.031 and 0.034.

# *4.3. Effect of Dephosphorization Endpoint Temperature on Phosphorus Capacity and Phosphorus Distribution Ratio of Dephosphorization Slag*

The solubility of phosphorus in molten slag can be expressed by phosphorus capacity, which is an important indicator of the potential dephosphorization capacity of slag. During the dephosphorization process of hot metal, the phosphorus that is oxidized and transferred to slag mainly exists in the form of phosphate ion  $PO_4^{3-}$ . According to the theory of ion dephosphorization, the formation reaction of  $PO<sub>4</sub><sup>3−</sup>$  is Equation (31) [\[32\]](#page-22-2).

$$
[P] + \frac{5}{2}[O] + \frac{3}{2}(O^{2-}) = (PO_4^{3-})(PO_2 > 10^{-13}Pa)
$$
\n(31)

Wagner proposed the concept of phosphate capacity C<sub>PO</sub>3− based on the gas–slag equi-librium; the reaction expression is Equation (32), which is expressed as Equation (33) [\[33\]](#page-22-3). Yang et al. [\[32,](#page-22-2)[34\]](#page-22-4) defined the phosphorus capacity index*, C<sub>PO</sub>3−* <sub>index</sub>, which is based on slag–metal equilibrium. The reaction expression is Equation (31), with  $C_{PO_4^{3-}}$  index expressed as Equation (34). The deduced relationship between  $C_{PO_4^{3-}}$  and  $C_{PO_4^{3-}}$  index can be expressed as Equation (35), and it can be applicable to any slag system [\[34\]](#page-22-4).

$$
\frac{1}{2}P_2(g) + \frac{5}{4}O_2(g) + \frac{3}{2}(O^{2-}) = (PO_4^{3-})
$$
\n(32)

$$
C_{\rm PO_4^{3-}} = \frac{\left(^{\circ}\!\!/\!\circ \text{PO}_4^{3-}\right)}{P_{\rm P2}^{1/2} P_{\rm O_2}^{5/4}}\tag{33}
$$

$$
C_{\text{PO}_4^{3-\text{index}}} = \frac{(\% \text{PO}_4^{3-})}{a_{\text{[P]}} \times (a_{\text{[O]}})^{5/2}}
$$
(34)

$$
\log C_{\text{PO}_4^{3-}} = \log C_{\text{PO}_4^{3-} \text{index}} + \frac{23,531.25}{T} + 0.1606 \tag{35}
$$

In Equations (33) and (34), (% $PO_4^{3-}$ ) is the phosphate content in the slag, and the relationship between  $(\%PO_4^{3-})$  and  $(\%P_2O_5)$  is shown in Equation (36) [\[32](#page-22-2)[,34\]](#page-22-4).

$$
(\%PO_4^{3-}) = \frac{2MPO_4^{3-}}{MP_2O_5} (\%P_2O_5) = 1.3382(\%P_2O_5)
$$
 (36)

*M*<sub>*i*</sub> represents the molecular weight of oxide *i*. The expression of log *C*<sub>PO</sub><sub>3</sub>− is obtained by combining Equations (27) and (33)–(36), as shown in Equation (37).

$$
\log C_{\text{PO}_4^{3-}} = \log \frac{1.3382(\text{N}P_2\text{O}_5)}{a_{\text{[P]}} \times (a_{\text{[O]}}^{\text{[Fe]-[O]}})^{5/2}} + \frac{23,531.25}{T} + 0.1606\tag{37}
$$

In Equation (37),  $a_{[P]}$  is calculated by Equation (11). Iron oxide is the main oxidant in the dephosphorization process. According to the discussion in Section [4.2,](#page-11-1) the oxygen activity in the slag is mainly determined by the oxygen activity of the iron oxidation equilibrium at slag-hot metal interface. Therefore,  $a_{\text{[O]}}$  in Equation (37) is calculated by using  $a_{\text{[O]}}^{\text{[Fe]}-\text{[O]}}$  $[0]$ <sup>[C]– $[0]$ </sup> in Equation (27).

The empirical formulas for calculating log *C*<sub>PO<sup>3−</sup></sub> are listed in Table [5,](#page-15-0) where Λ represents the optical basicity of the slag, which is expressed as Equation  $(38)$ .  $x_i$  is the mole fraction of cations in oxide *i*, which is expressed as Equation (39). *n<sup>i</sup>* is the number of oxygen atoms in oxide *i* and *x*'*<sup>i</sup>* is the mole fraction of oxide *i*. Table [6](#page-15-1) lists the optical basicities of various oxides [\[28\]](#page-21-26).

$$
\Lambda = \sum x_i \Lambda_i \tag{38}
$$

$$
x_i = \frac{n_i(\mathcal{O})x_i'}{\sum n_i(\mathcal{O})x_i'}\tag{39}
$$

<span id="page-15-0"></span>

**Table 5.** Empirical formulas of  $\log C_{\text{PO}_4^3}$ .

<span id="page-15-1"></span>**Table 6.** Optical basicities of different oxides.

.



Figure [10](#page-16-0) shows the comparison of  $\log C_{\text{PO}_4^{3-}}$  between the calculated value of the empirical formulas and the calculated value in this paper at different dephosphorization endpoint temperatures. log *C*<sub>PO<sup>3</sup><sup>−</sup></sub> in Figure [10](#page-16-0) is calculated by Equation (37) and different empirical formulas in Table [5.](#page-15-0) It can be concluded that the calculated values of log C<sub>PO3</sub>− 4 from the empirical formulas of Maruoka et al. [\[35\]](#page-22-5), Selin et al. [\[36\]](#page-22-6), and Sobandi et al. [\[37\]](#page-22-7) all decrease with increasing dephosphorization endpoint temperature, and the changing trend is similar, which is consistent with the thermodynamic result that the increase in temperature is not conducive to the dephosphorization of hot metal. The calculated values of log C<sub>PO</sub><sub>3</sub>− from Yang's empirical formula do not change with the increasing 4 dephosphorization endpoint temperature significantly. This is because Yang et al. [\[28\]](#page-21-26) mainly studied the effect of dephosphorization slag basicity on phosphorus capacity, and

the optical basicity plays a decisive role in their empirical formula. The calculated values of log C<sub>PO</sub><sub>3</sub>− in the present work show a trend of first decreasing, then increasing, and then decreasing again. The value of the calculated results at 1385–1410 °C in the present work are consistent to those of Yang et al. [\[24\]](#page-21-22). This is because the experimental temperature, dephosphorization slag basicity, and composition in the present work are similar to the experimental temperature of 1380 °C, slag basicity of 1.40–1.83, and slag compositions of CaO-MnO-MgO-SiO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub>-Fe<sub>t</sub>O-Al<sub>2</sub>O<sub>3</sub> in their paper, and they did the laboratorial experiments based on the double slag steelmaking process.

<span id="page-16-0"></span>

**Figure 10.** Comparison of log C<sub>PO4</sub><sup>-</sup> between the calculated value of the empirical formulas and the calculated value in this paper at different dephosphorization endpoint temperatures.

The calculated results in the present work are smaller than those calculated by the empirical formulas, which become smaller when the temperature is increased. This is because most of the formulas in Table [6](#page-15-1) are the results of laboratory equilibrium under ideal experimental conditions. However, the dephosphorization reaction has not reached equilibrium value due to the short time of dephosphorization in the industrial experiment, which is only about 5 min. Therefore*,* the log C<sub>PO</sub>3− value calculated in the percent work is smaller than those calculated by empirical formulas.

From Equations (12), (27), and (37), Equation (40) of the phosphorus distribution ratio log<sub>L</sub> based on the oxygen activity of iron oxidation equilibrium at the slag-hot metal interface can be deduced as follows:

$$
\log L_{\rm P} = \log C_{\rm PO_4^{3-}} + \log f_{\rm P} + \frac{5}{2} \log a_{\rm [O]}^{\rm [Fe]-[O]} - 23{,}531.25/T - 0.287\tag{40}
$$

Figure [11](#page-17-0) shows the comparison of  $log L<sub>P</sub>$  between the calculated values of the empirical formulas and the calculated values in this paper at different dephosphorization endpoint temperatures. The values of  $log L<sub>P</sub>$  in Figure [11](#page-17-0) are calculated by Equation (40) and the different empirical formulas in Table [7.](#page-17-1) According to Figure [11,](#page-17-0) all empirical formulas indicate that the values of log<sub>LP</sub> show a downward trend at first, then an upward trend around 1405 ◦C, and then a downward trend again with increasing dephosphorization endpoint temperature. The  $log L<sub>P</sub>$  value calculated by the Ogawa formula is the closest to the calculated results in the present paper. It is noticed that between 1395 and 1410  $\degree$ C, the log<sub>L</sub> values calculated by the Ogawa formula are consistent with the calculated values in

the present work. This is because the Ogawa formula is based on the formula summarized in the dephosphorization stage of the typical double slag converter steelmaking process the MURC process. Therefore, the temperature and basicity used in the Ogawa formula are similar to those in the present paper. In addition, the effect of the higher [%C] content in hot metal on *L*<sup>P</sup> is taken into consideration, so that it is suitable for the dephosphorization of hot metal with the higher carbon content in the dephosphorization stage of the present work. This is consistent with our previous results on the double slag converter steelmaking process [\[4\]](#page-21-3).

<span id="page-17-0"></span>

Figure 11. Comparison of log<sub>LP</sub> between the calculated value of the empirical formulas and the calculated values in this paper at different dephosphorization endpoint temperatures.

<span id="page-17-1"></span>

| Scholar              | Slag  | <b>Empirical Formulas</b>   |
|----------------------|---|---|
| Healy [38]           | CaO-SiO <sub>2</sub> -Fe <sub>t</sub> O-P <sub>2</sub> O <sub>5</sub>   | $\log L_{\rm P} = \log \frac{(\% {\rm P})}{\sqrt{(\% {\rm P})}} = 22,350/T - 24.0 + 7 \log (\% {\rm CaO}) + 2.5 \log (\% {\rm T \ Fe})$<br>$((\%CaO) > 30)$   |
|                      |   | $\log L_{\rm P} = \log \frac{(\% {\rm P})}{\frac{10}{2} \times 10^{9}} = 22{,}350/T - 16.0 + 0.08 \ (\% {\rm CaO}) + 2.5 \log (\% {\rm T}.{\rm Fe})$<br>$((\%CaO) < 30)$  |
| Suito [39]           | $CaO-SiO2-FetO-P2O5-MgO-MnO$  | $\log L_P = \log \frac{(\%P)}{(\%P)(\%T \cdot \text{Fe})^{5/2}} = 0.072[(\%CaO) + 0.3 (\%MgO)]$<br>$+0.6(^{\circ}\!\!_{0}P2O5)+0.6(^{\circ}\!\!_{0}MnO)]+2.5\lg(^{\circ}\!\!_{0}T.Fe)+11,570/T-10.52$   |
| Kawai [40]           | $CaO-SiO2-FetO-P2O5$  | $\log L_P = \log \frac{\binom{6}{6}P}{\binom{6}{6}P} = 22,350/T - 21.876 + 5.6\log(\%CaO) + 2.5\log(\%T \cdot Fe)$  |
| T.Usui&K.Yamada [41] | CaO-Fe <sub>t</sub> O-SiO <sub>2</sub> -P <sub>2</sub> O <sub>5</sub> -MgO-Al <sub>2</sub> O <sub>3</sub> -TiO <sub>2</sub> | $\log L_P = 5.60 \log[(\% \text{CaO}) + 0.3(\text{MgO}) + 0.05(\% \text{FetO})] + 14,800/T$<br>$18.038 + 0.5\log(\%P2O5) + 2.5\log(\%Fet)$  |
| Ogawa $[1]$          | $CaO-SiO2-FetO-P2O5-MgO-MnO$  | $\log L_P = \log \frac{\binom{9}{6}P}{\binom{9}{6}P} = 2.5 \log(\%T \cdot Fe) + 0.0715\{(\%CaO) + 0.25(\%MgO)\}\$<br>$+7710/T - 8.55 + (105.1/T + 0.0723)$ [%C]   |
| $\chi$ hang $[42]$   | CaO-SiO <sub>2</sub> -FeO-P <sub>2</sub> O <sub>5</sub> -MgO-MnO  | $\log L_{\rm P} = \log \frac{(\% \text{P2O5})}{ \% \text{P} } = [162 (\% \text{CaO}) + 127.5 (\% \text{MgO}) + 28.5 (\% \text{MnO})]/\text{T}$<br>$+2.5\lg(\sqrt[6]{6}FeO) - 6.287 \times 10^{-4} \times \log(\sqrt[6]{6}SiO2)^{2} + 11,000/T - 10.4$ |

Table 7. Empirical formulas of log<sub>Lp</sub>.

It can be seen from Figure [11](#page-17-0) that with the increasing dephosphorization endpoint temperature, the calculated results in the present work show log<sub>LP</sub> first rising and then decreasing obviously, which is consistent with the changing trend of the dephosphorization ratio in Figure [3.](#page-6-0) When the dephosphorization endpoint temperature is in the range of 1350– 1405  $\degree$ C, with increasing temperature, the auxiliary materials such as lime and dolomite are easy to melt, so that the kinetic conditions of dephosphorization are improved. Although increasing temperature is unfavorable to the thermodynamic conditions of dephosphorization, the overall *L*<sup>P</sup> still increases, and the dephosphorization result shows an upward trend. When the endpoint temperature of dephosphorization exceeds 1405 ◦C, with further increasing the temperature, the equilibrium constant of the dephosphorization reaction decreases greatly, and the dephosphorization ratio decreases, resulting in a decrease in *L*P.

The calculated values of  $log L<sub>P</sub>$  in the present work are closer to the empirical formula values of Ogawa et al. [\[1\]](#page-21-0), but they are lower than the empirical formula values of the others, because the empirical formulas of Healy et al. [\[38\]](#page-22-8), Suito et al. [\[39\]](#page-22-9), Usui and Yamada et al. [\[41\]](#page-22-11), and Zhang et al. [\[42\]](#page-22-12) were from an equilibrium dephosphorization reaction under laboratorial experiments and not for the double slag process. The empirical formula of Kawai et al. [\[40\]](#page-22-10) was from the industrial experiments but not for the double slag process. The empirical formula of Ogawa et al. [\[1\]](#page-21-0) was for the double slag process. However, their dephosphorization times were up to 8 min, and their experiments were carried out in an 8 t converter so that the dephosphorization reaction could be carried out close to the equilibrium results. According to the results in Figure [11,](#page-17-0) the optimum dephosphorization temperature is in the range of 1385–1410 ◦C and the optimum log*L*<sup>P</sup> value is in the range of 1.76–2.09.

#### *4.4. Effect of Dephosphorization Endpoint Temperature on Dephosphorization, Rephosphorization, and the Optimum Temperature of Deslagging*

According to previous scholars on the double slag converter steelmaking industrial experimental research, it is necessary to study the optimum temperature of intermediate deslagging theoretically [\[18,](#page-21-17)[43,](#page-22-13)[44\]](#page-22-14). As the desiliconization and dephosphorization reactions proceed, the converter temperature increases gradually. Since phosphorus in hot metal is oxidized by FeO and reduced by carbon at the same time, combining the dephosphorization reaction Equation (41) of oxidation of P by FeO and the rephosphorization reaction Equation (42) of the reduction of  $P_2O_5$  by C, the coupling dephosphorization reaction among FeO, C, and P can be obtained as Equation (43). The Gibbs free energy expression of the reaction is taken from the literature [\[18\]](#page-21-17). Equation (46) is obtained from Equations (44) and (45).

$$
2[P] + 5(FeO) = (P2O5) + 5[Fe]
$$
\n(41)

$$
5[C] + (P2O5) = 2[P] + 5CO
$$
 (42)

$$
4[P] + 5(FeO) + 5CO = 2(P2O5) + 5[Fe] + 5[C]
$$
\n(43)

$$
\Delta G_{\text{FeO-P}} = -27,763.36 + 255.23T + RT \ln(\frac{a_{\text{P}_2\text{O}_5}}{a_{\text{FeO}}^5 \times a_{\text{[P]}}^2})
$$
(44)

$$
\Delta G_{\rm C-P_2O_5} = 521,758.34 - 709.02T + RTln(\frac{\left(P_{\rm CO}^{\rm P-C}\right)^5 \times a_{\rm [P]}^2}{a_{\rm P_2O_5} \times a_{\rm [C]}^5})
$$
(45)

$$
\Delta G_{\text{FeO}-\text{CO}-\text{P}} = -549,521.71 + 964.26T + RTln(\frac{a_{\text{P}_2\text{O}_5}^2 \times a_{\text{[C]}}^5}{P_{\text{CO}}^5 \times a_{\text{[P]}}^4 \times a_{\text{FeO}}^5})
$$
(46)

Figure [12](#page-19-0) shows the effect of dephosphorization endpoint temperature on the actual Gibbs free energy (∆*G*) changes of FeO-P, C-P2O5, and FeO-CO-P reactions. The ∆*G* values of the three reactions of FeO-P, C-P<sub>2</sub>O<sub>5</sub>, and FeO-CO-P are calculated from Equations (44)–(46), where  $a_{P_2O_5}$  is calculated from Equations (10) and (11),  $P_{CO}^{P-C}$  is calculated by Equations (6)–(13)*, a<sub>[C]</sub>, a<sub>[P]</sub> and a<sub>FeO</sub> are calculated by Equations (8) and (9), and* Equations (3) and (4), respectively.

<span id="page-19-0"></span>

**Figure 12.** Effect of dephosphorization endpoint temperature on the actual Gibbs free energy changes of FeO-P,  $C$ - $P_2O_5$ , and FeO-CO-P reactions.

It can be seen from Figure [12](#page-19-0) that the ∆*G* values of FeO-P and C-P<sub>2</sub>O<sub>5</sub> are both smaller than 0, which means that in the dephosphorization stage with the temperature in the range of 1350–1405  $\degree$ C, both the dephosphorization reaction of phosphorus oxidation by FeO and rephosphorization reaction of reduction of  $P_2O_5$  by carbon occur. With the increasing dephosphorization endpoint temperature, the ∆*G* value of the C-P<sub>2</sub>O<sub>5</sub> reaction does not change significantly, which indicates that temperature has little effect on rephosphorization in 1350–1405 ◦C.

The ∆*G* values of the three reactions are linearly fitted with the dephosphorization endpoint temperature, with the regression coefficients,  $R^2$ , are 0.92, 0.99, and 0.92, respectively. The fitting line of the C-P2O<sup>5</sup> ∆*G* value and the fitting line of the FeO-P ∆*G* value cross when  $T_1$  is about 1413 °C, which means the oxidation of phosphorus by FeO and the reduction of  $P_2O_5$  by carbon are in dynamic equilibrium at this temperature. At the temperature lower than 1413 ◦C, dephosphorization and rephosphorization reactions are carried out simultaneously in the converter, and the trend of dephosphorization reaction is greater than that of rephosphorization. When the temperature is higher than  $T_1$ , the trend of the rephosphorization reaction in the converter will be greater than that of the dephosphorization reaction, resulting in the rephosphorization of hot metal.

Figure [12](#page-19-0) also shows the result of Equation (46) obtained by coupling Equations (44) and (45), as shown by the blue line in the Figure [12.](#page-19-0) The ∆*G* value of the FeO-CO-P reaction increases with the increasing dephosphorization endpoint temperature. When the temperature *T*<sup>2</sup> exceeds 1413 ◦C, the ∆*G* is greater than 0. The trend of rephosphorization in the FeO-CO-P coupling dephosphorization reaction is greater than that of dephosphorization.

It can be seen from Figure [12](#page-19-0) that the temperature  $T_2$  at which the  $\Delta G$  value of the FeO-CO-P reaction is 0 is the same as  $T_1$ . It shows that the FeO-CO-P coupled dephosphorization reaction is equivalent to combining the dephosphorization reaction and the rephosphorization reaction. Considering the experimental results and thermodynamic calculation results of industrial experiments by the double slag dephosphorization process, the optimal temperature range for intermediate deslagging is about 1400–1420 ◦C.

### **5. Conclusions**

In the present work, the industrial experiments of dephosphorization in the double slag converter steelmaking process under the conditions of low temperature and low basicity were carried out. The effect of dephosphorization endpoint temperature on the dephosphorization of hot metal was studied, and the following conclusions were drawn:

- (1) In the temperature range of 1350–1450 °C, with increasing dephosphorization endpoint temperature, the dephosphorization ratio and phosphorus distribution ratio first increase and then decrease. The phosphorus content in hot metal at the end of dephosphorization first decreases and then increases. The optimum dephosphorization temperature is in the range of 1385–1410  $\degree$ C, with the dephosphorization ratio higher than 55%, the  $P_2O_5$  content in the dephosphorization slag of 3.93–4.17%, and the log<sub>L</sub><sub>P</sub> value of 1.76–2.09.
- (2) Dephosphorization slag is mainly composed of the gray massive P-rich phase, gray white Fe-rich phase, and black calcium silicate phase. The path of phosphorus in hot metal entering the P-rich phase of dephosphorization slag can be reasonably inferred as: hot metal  $\rightarrow$  Fe-rich phase  $\rightarrow$  P-rich phase in dephosphorization slag.
- (3)  $a_{\text{FeO}}$  is decided by the dephosphorization endpoint temperature and basic oxide content in dephosphorization slag, and  $a_{\text{FeO}}$  has multiple linear correlation with them. The  $P_{\rm CO}^{\rm P-}$  of selective oxidation reaction of carbon and phosphorus has a good exponential relationship with the endpoint temperature of dephosphorization. It is beneficial to converter dephosphorization when the temperature is 1385–1410 ◦C, the value of  $P_{\rm CO}^{\rm P-C}$  is 53–80 Pa, and  $a_{\rm FeO}$  value is 0.284–0.312.
- (4) The oxygen activity on the surface of hot metal of  $a_{[O]}^{\text{iron}}$ , the oxygen activity of the phosphorus oxidation at the slag–hot metal interface of  $a_{\text{[O]}}^{\text{de[P]}}$  $\begin{bmatrix} \alpha e_{1} \\ \beta \end{bmatrix}$ , and the oxygen activity of the iron oxidation at the slag–hot metal interface of  $a_{\text{[O]}}^{\text{[Fe]}-\text{[O]}}$  $[O]$ <sup>[re<sub>]</sub>- $[O]$  all increase</sup> with increasing the dephosphorization endpoint temperature in the range of 1350– 1405  $\degree$ C. The temperature has the greater effect on the phosphorus oxygen equilibrium of the slag–hot metal interface in the temperature range of 1350–1405  $°C$ , and the temperature has a greater effect on the iron oxygen equilibrium in the slag in the temperature range of 1405–1450 °C. In the temperature range of 1350–1450 °C, owing to  $a_{\text{[O]}}^{\text{[Fe]}-\text{[O]}}$  ${}_{[O]}^{[Fe]-[O]} > a_{[O]}^{de[P]}$ <sup>de[P]</sup> > *a*<sup>iron</sup>, *a*<sup>[Fe]−[O]</sup><br>[O] , *a*<sup>[O]</sup> [*O*] plays the most important role on the dephosphorization in the double slag steelmaking process.
- (5) Under the present industrial experimental conditions, when the temperature is 1413 ◦C, the dephosphorization and rephosphorization reactions are in dynamic equilibrium. Considering the experimental results and thermodynamic calculation results of industrial experiments by the double slag dephosphorization process, the optimal temperature range for intermediate deslagging is about  $1400-1420$  °C.

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