



Article Tunable Effect of the Calcination of the Silanol Groups of KIT-6 and SBA-15 Mesoporous Materials

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Received: 19 December 2019; Accepted: 22 January 2020; Published: 2 February 2020



Abstract: The calcination process is a crucial step during SBA-15 and KIT-6 synthesis. It is used to completely remove the organic template and condense silanol groups, and it allows the determination of the textural and physical properties of these materials, depending on the adopted conditions. Moreover, calcination influences the number of silanols available on the surface of the material. The concentration of silanols is important if these materials were synthesized for use in adsorption or functionalization. To understand and optimize the silanol groups of SBA-15 and KIT-6, in this study, the temperature and time calcination parameters were varied. The experiments were performed at 300, 400, and 500 °C for 300, 400, and 500 min. The results show that the ideal temperature to preserve the silanol groups is 300 °C, but to optimize the textural properties, it is better to calcine these molecular sieves at 400 °C. A calcination for 10 h did not give better results than a calcination for 5 h, demonstrating that the former duration is excessive for use.

Keywords: mesoporous materials; KIT-6; SBA-15; silanol groups; calcination

1. Introduction

According to the IUPAC (International Union of Pure and Applied Chemistry), porous materials can be grouped into three distinct categories according to their porosities [1]. Materials with pores smaller than 2 nm are classified as microporous, those with pores larger than 50 nm are called macroporous, and those with intermediate pore sizes are mesoporous [2]. In the early 1990s, Mobil Oil Corporation researchers published the first syntheses of mesoporous materials (M41S), which were manufactured using an alkyltriethylammonium halide as a template in an alkaline medium [3,4]. These new compounds presented very characteristic and peculiar properties, such as a high specific surface area (up to 1400 m² g⁻¹), a pore volume greater than 0.5 cm³ g⁻¹ and narrowly distributed cylindrical pores (2–30 nm) [5]. The physicochemical properties of these compounds have been tested and evaluated by many researchers [6–12].

In 1998, the syntheses of a group of mesoporous materials called SBA, using an organic copolymer (mainly PEO-PPO-PEO) as a template in an acid medium, were published [13,14]. In this family of compounds, the best known is SBA-15, which is a highly ordered two-dimensional hexagonal (*P6mm*) thick-walled silica (3 to 6 nm) mesostructured material with uniform, large, adjustable pores up to 30 nm in diameter and high thermal and hydrothermal stability due to the presence of mesoporous interconnected micropores [13]. In 2003, the synthesis of a new mesoporous material called KIT-6 was first published [15]. This material was manufactured in a manner very similar to that used in the production of the SBA-15 but using butanol-1 as a swelling agent. KIT-6 exhibits cubic *la3d* symmetry with excellent thermal/hydrothermal stabilities and consists of two continuous and interpenetrating

chiral channel systems interconnected in a three-dimensional open cylindrical mesoporous network [16]. It has a wall thickness between 4 and 6 nm [17], a specific area of up to 800 m² g⁻¹, and a high pore volume (up to 1.05 cm³ g⁻¹) with pore diameters in the range of 4 to 12 nm [15]. The main pores are interconnected with micropores up to 1.8 nm in diameter at specific points [18,19].

With the discoveries of these two mesoporous nanosilica (MNS) materials, a wide range of possibilities has been opened for the application of both SBA-15 [20–28] and KIT-6 [29–37] in various areas, such as catalysis, adsorption, separation, drug release, optical devices, and sensors. The development of MNS has involved three distinct generations [38]. The first corresponded to the conceptual demonstration of MNS through the discovery of materials such as the M41S and SBA families. The second generation was characterized by the improvement and refinement of the study of these materials in terms of in vitro applications and in situ analysis. Finally, the third and current generation of studies functionalized MNS [39,40]. Pure silica mesoporous materials have intrinsic limitations in both functionality and strength, and controlling the pore surface chemistry is critical to improving their applications [12]. The surface functionalization of these mesoporous compounds with organic and inorganic functional groups leads to the formation of materials with new physicochemical properties [41], which can be used in various areas, such as catalysis, adsorption, and separation [42,43].

The formation of the materials SBA-15 and KIT-6 are made by the sol-gel process [44]. This is a traditional methodology based on the hydrolysis and condensation of metal alkoxides (TEOS) to produce colloidal particles (sol) that are converted into a lattice (gel) [45]. The conditions employed directly affect the properties of the materials produced. The most important parameters in the sol-gel process is the type of silanol group, its concentrations, the pH, the solvent, thermic treatment, aging time, and the additives, such as catalysts and co-solvents [45–50].

SBA-15 and KIT-6 pure silica materials contain two basic types of silicon–oxygen groups. The first is siloxanes, and the second is silanol-type hydroxyls that can be divided into three groups, free silanols, vicinal silanols, and geminal silanols [44], as shown in Figure 1.



Figure 1. Silicon–oxygen group types present in SBA-15 and KIT-6.

Siloxane groups are more hydrophobic, are the most abundant group, and make up the interior of SBA-15 and KIT-6 MNS. Silanol groups attract the most attention in regard to the functionalization of MNS because hydroxyl groups react more easily with other chemical species [51]. Thus, it is of interest to preserve these silanol groups in the highest concentration possible. For this, it is necessary to use low temperatures in the calcination of MNS, since temperature is the main factor leading to a decrease in the silanol concentration [52]. However, there are boundaries to the temperature used in all SBA-15 and KIT-6 synthesis processes. The use of a high temperature is necessary for the formation of mesoporous materials, through the condensation of hydroxyls, to form siloxane bonds, but several studies have demonstrated the effects on the quality of the temperature-related MNS formation used during a static hydrothermal synthesis of SBA-15 [53–55] and KIT-6 [5]. Nevertheless, during the calcination process, these bonds are consolidated, giving rise to the desired material when the largest loss of surface silanol groups occurs; however, too great a loss leaves the material unusable for functionalization.

This study thus proposes calcining samples of SBA-15 and KIT-6 at different temperatures and different calcination times. To determine the efficiency of these thermal processes, the carbon content

was monitored by thermal analysis, the X-ray diffraction (XRD) patterns of the mesoporous material were evaluated, the textural properties were assessed by N_2 sorption, and the availability of silanol and siloxane groups were monitored by nuclear magnetic resonance (NMR) ²⁹Si{¹H} cross-polarization magic-angle spinning (CP-MAS). The choice of these MNS materials was based on two different factors. On the one hand, the SBA-15 is a heavily studied material, with many articles published on its functionalization [25,56–60], and while KIT-6 is a relatively new material, it has already attracted much attention in studies worldwide, with some works about its functionalization [61–68]; on the other hand, the availability of silanol groups in KIT-6 has been less investigated, whereas it has already been reported for SBA-15 [51,56,69–71].

2. Materials and Methods

2.1. Chemicals

Pluronic P123 triblock copolymer ($EO_{20}PO_{70}EO_{20}$, Mn = 5800, Aldrich, St. Louis, MO, USA), tetraethyl orthosilicate (TEOS, Aldrich, 98%), butanol (Aldrich, 99.4%), and hydrochloric acid (HCl, Atrion, 37 wt %) were used to synthesize the mesoporous samples.

2.2. Characterization

Small-angle X-ray diffraction (XRD) patterns at 2 θ angles from 0.5° to 5° were recorded on a Bruker D8 diffractometer using Cu K α radiation (30 kV, 10 mA). Nitrogen adsorption isotherms were collected by a Micromeritics Tristar II instrument at –196 °C (77 K). The samples were degassed under vacuum at 200 °C overnight. The Brunauer–Emmett–Teller (BET) method was used to calculate the specific surface area. The total pore volume was determined at a relative pressure p/p⁰ of 0.98. The mesoporous and microporous volumes were obtained using α s-plots methodology. The pore size distribution was determined my means of the Barrett–Joyner–Halenda (BJH) method, using the correction factor proposed in the Villarroel–Barrera–Sapag (VBS) method. The carbon, hydrogen and nitrogen elemental analyses were performed using a Perkin Elmer Series II 2400. The silicon nuclear magnetic resonance (²⁹Si-NMR) solid-state spectra were registered on an Agilent DD2 500 MHz spectrometer. The method utilized was cross-polarization magic-angle spinning (CP-MAS) at a Larmor frequency of 99.3 MHz, a pulse duration of 2.9 ms, a relaxation time of 5 s, an acquisition time of 17 ms, and a scan number of 12,000.

2.3. Synthesis of SBA-15

This material was prepared by the conventional methodology [72] of dissolving 12.1 g of P123 in an HCl solution (450 mL, 2.0 mol L^{-1}). The system was magnetically stirred at approximately 500 rpm for two hours at 45 °C. When the mixture became a homogeneous solution, 4.3 g of TEOS was added to the reaction container in a dropwise manner. The system was kept at 45 °C for 24 h under stirring. After that, the mixture was transferred to an autoclave, and a hydrothermal treatment was performed for 24 h at 110 °C. The product obtained was filtered and washed with distilled water to a neutral pH. The material was then dried at 60 °C for 26 h. The SBA-15 produced was heat-treated under different conditions.

2.4. Synthesis of KIT-6

The production of this mesoporous material was performed using the partitioned cooperative self-assembly (PCSA) methodology [73]. First, 18.6 g of P123 was dissolved in 669.6 g of distilled water and 36.8 g of concentrated HCl. The system was magnetically stirred at approximately 800 rpm at 35 °C for 90 min. After this period, 18.6 g of 1-butanol was slowly added. After 60 min under the same temperature and stirring conditions, 28 g (70% equivalent) of TEOS was dropped into the synthesis mixture. After 5 h, the remaining 30% TEOS (12 g) was added in the same way. The system was continuously agitated at approximately 800 rpm at 35 °C for 24 h. The material was then transferred to

a Teflon autoclave and placed in a digester block, where it remained for 24 h at 100 $^{\circ}$ C. The formed material was filtered with distilled water to pH 7 and dried for 17 h at 60 $^{\circ}$ C.

2.5. The Methodology of Mesoporous Calcination

The mesoporous material synthesized was calcined through six different methodologies using a muffle furnace. The sample mass used in each method, as well as the calcination conditions, are detailed in Table 1.

Method	Temperature ¹ (°C)	Time (h)	SBA-15 Mass (mg)	KIT-6 Mass (mg)
1	300	5	514	519
2	300	10	515	516
3	400	5	512	556
4	400	10	515	516
5	500	5	508	514
6	500	10	502	503

Table 1. Methodologies used for the calcination of SBA-15 and KIT-6 samples.

¹ The heating ramp of 2° min⁻¹.

The samples were identified by utilizing the letter S to designate SBA-15 and the letter K to designate KIT-6. The calcination methodology used is indicated by the letter M followed by the method number. For example, the identification code SM5 corresponds to the sample of SBA-15 calcinated according to method number 5. The number zero indicates that the material was not calcinated.

3. Results and Discussion

3.1. Effect of Calcination Temperature on the Quantity of Polymer

The effect of calcination on the amount of polymer in the mesoporous material was assessed by measuring the amount of carbon remaining in the material by elemental analysis. The results are shown in Table 2.

Samples	Residual Carbon Content (%)	Samples	Residual Carbon Content (%)
SM0	25.33	KM0	28.86
SM1	0.32	KM1	0.42
SM2	0.19	KM2	0.30
SM3	0.08	KM3	0.09
SM4	0.08	KM4	0.09
SM5	0.03	KM5	0.09
SM6	0.01	KM6	0.07

Table 2. Residual carbon content in the SBA-15 and KIT-6 samples.

Both SBA-15 and KIT-6 lose most of their organic templates at 300 °C in the first two heat treatment methods. After treatment for 5 h, 0.32% and 0.42% of the carbon remains for SBA-15 and KIT-6, respectively. Upon increasing the treatment time to 10 h, the amount of carbon remaining drops to 0.19% for SBA-15 and 0.30% for KIT-6, indicating that the polymer remains in the mesoporous materials. Previous studies [51,55,69] have shown that, at 200 °C, the polymer decomposes and that, by calcining for 4 h at 300 °C or for 4 h at 350 °C, the polymer is removed; however, to the best of our knowledge, no work has quantified the residual polymer content. When comparing the results obtained at a higher temperature (500 °C) with those obtained at an intermediate temperature (400 °C), the difference in the amount of organic material removed is too small. Another factor is the time at higher temperatures; calcining for 10 h does not result in substantial benefits in terms of the removal of organic material, since the values of % C remaining are identical at 400 °C and are almost equal at 500 °C for the two

mesoporous materials. Nonetheless, at 300 °C, there is a large difference between calcination at 5 and 10 h. The results of this analysis are shown in Figure 2.



Figure 2. The residual carbon content for KIT-6 and SBA-15 after each calcination method.

3.2. Effect of Calcination Temperature on the Mesostructure of the MNS

As already stated, the structure of mesoporous materials was monitored by XRD analysis to determine the effects of calcination temperature. The results are presented in Table 3.

Samples	d ₁₁₀ (nm)	a_0 (nm) 1	Samples	d ₂₁₁ (nm)	a ₀ (nm) ²
SM0	11.2	13.0	KM0	10.7	26.2
SM1	10.4	12.0	KM1	9.4	23.0
SM2	10.3	11.9	KM2	9.8	24.0
SM3	10.8	12.5	KM3	9.6	23.5
SM4	10.8	12.5	KM4	9.7	23.8
SM5	10.6	12.2	KM5	9.8	24.0
SM6	10.4	12.0	KM6	9.4	23.3

Table 3. Results of the X-ray diffraction (XRD) analyses of the mesoporous nanosilica (MNS) samples.

¹ calculated by the formula [62] $a_0 = 2.d_{100}.100.\sqrt{3}$, ² calculated by the formula [66] $a_0 = d_{211}.\sqrt{6}$.

All XRD diffractograms show the formation of materials with angles of $0.8^{\circ} < 2\theta < 2^{\circ}$, characteristic of mesoporous materials, and are displayed in Figure 3. An analysis of Figure 3 shows that, among the calcined samples, all SBA samples have good structural organization except that prepared by Method 1. All KIT-6 samples showed good structural organization, indicating that the different calcination methods had little influence on the mesostructure, as expected [13–15]. Moreover, in all calcinated materials, the pores shrink more than in the noncalcinated materials, but among all calcination methods, there is no linear relationship between pore shrinkage and the calcination temperature.



Figure 3. Small-angle XRD pattern of KIT-6 and SBA-15 calcinated under different conditions.

3.3. Effect of the Calcination Temperature on the Textural Properties of the MNS

The porosity of calcined MNS materials under different conditions was analyzed by N_2 sorption. As expected for these materials, isotherms are type IVa (according to the new IUPAC [74] classification) with type I hysteresis. Figure 4 shows the SBA-15 isotherms, and Figure 5 shows the KIT-6 isotherms. The physical properties taken from these isotherms are presented in Table 4.

Samples	$\begin{array}{c} A_{BET} \\ (m^2 \ g^{-1}) \end{array}$	V _T (cm ³ g ⁻¹)	V_{Mi} (cm ³ g ⁻¹)	V _{Me} (cm ³ g ⁻¹)	D _p (nm)	W _t (nm)
SM1	572	0.78	0.02	0.76	7.5	4.5
SM2	632	0.86	0.02	0.84	7.5	4.4
SM3	641	0.84	0.03	0.81	7.8	4.7
SM4	1197	1.52	0.04	1.48	7.5	5.0
SM5	1099	1.69	0.00	1.69	8.9	3.3
SM6	770	0.98	0.06	0.92	7.8	4.2
KM1	695	0.82	0.07	0.75	7.2	15.8
KM2	826	0.98	0.07	0.91	6.1	17.9
KM3	905	1.07	0.07	1.00	7.5	16.0
KM4	852	1.03	0.08	0.95	7.9	15.9
KM5	1174	1.44	0.08	1.36	7.3	16.7
KM6	874	1.10	0.05	1.05	7.4	15.6

Table 4. Physical properties of SBA-15 and KIT-6 samples.

 A_{BET} is the specific area obtained from the Brunauer–Emmett–Teller (BET) method. The V_T is the total pore volume obtained from a relative pressure p/p^0 equal to 0.98. The V_{Mi} is the microporous volume obtained from the α -plots method. The V_{Me} is the mesoporous volume obtained from the α -plots method. The D_p is the pore diameter obtained from the Barrett–Joyner–Halenda (BJH) method corrected from the Villarroel–Barrera–Sapag (VBS) method. The W_t is the wall thickness.

For SBA-15 and KIT-6, the BET area indicates a suitable porosity with material type [13–15]. The samples of SBA-15 presented the maximum value of A_{BET} upon calcination at 400 °C for 10 h. Increasing the temperature to 500 °C for 5 h decreased the value of A_{BET} slightly (from 1197 to 1099 m² g⁻¹), and maintaining the temperature at 500 °C but doubling the calcination time to 10 h considerably decreased the value of the total material area to 770 m² g⁻¹. This decrease in the area must be associated with the process of the dehydroxylation of the material, which begins to intensify under these heat treatment conditions. For KIT-6, a calcination at 300 °C for 5 h results in the lowest total area value. This is quite reasonable since it is under these conditions that the largest amount of organic material exists, as attested by the results of elemental carbon analysis. The largest BET area was obtained by calcining at 500 °C for 5 h. When the heat exposure time was increased from 5 to 10 h without an increase in the temperature, the BET area decreased (1174 to 874 m² g⁻¹). This indicates that

at temperatures above 500 °C, contraction of the structure begins to occur, indicating a dehydroxylation process of KIT-6 [69].

From the measurements made after the capillary condensation step at the beginning of the isotherm plateau, the total pore volume was determined from the adsorbed nitrogen volume values. Micropore and mesopore volumes were calculated according to reference [75]. Since SBA-15 has hexagonal cells and KIT-6 has cubic cells, the calculation of the network parameters a_0 is different, and consequently, the calculation of the pore wall thickness (W_t) will also be performed by two different methods. For the SBA-15 samples, the formula adopted [13] is $W_t = a_0 - D_p$, and for the KIT-6 samples, the thickness is determined by the expression [66] $W_t = \frac{a_0}{2} - D_p$. The pore diameter (D_p) was determined with the adsorption branch by the BJH (Barrett–Joyner–Halenda) method, and the mesoporous correction was done in accordance with to the VBS (Villarroel–Barrera–Sapag) method. This is an improved macroscopic method that modifies the Kelvin equation used in the BJH method and considers the most accepted capillary evaporation and condensation mechanisms for mesoporous materials [76].



Figure 4. N₂ sorption isotherm and pore distribution of SBA-15 calcined under different conditions.



Figure 5. N₂ sorption isotherm and pore distribution of KIT-6 calcined under different conditions.

The results show that, for SBA-15, the mesoporous volume and pore diameter increased as the heat treatment conditions became more drastic, ultimately reaching their maximum values upon calcination at 500 °C for 5 h, showing that, under these conditions, dehydroxylation becomes more intense. The literature reports that calcining at temperatures above 600 °C greatly affects the textural properties of the MNS [77]. For KIT-6, the maximum mesopore volume was obtained under the calcination conditions of 500 °C for 5 h. When the heat exposure time was doubled, a contraction of the mesoporous volume was observed, evidencing the greater tendency to dehydroxylate the material at higher temperatures. Structure shortening under the most extreme conditions was also observed in the XRD analysis. This behavior is attributed to the structure of the interconnected mesopore channels, forming cubic symmetry. This can hinder heat flow within the material, and, as the results of the elemental analysis show, the carbon content in the most extreme methodology is higher in KIT-6 than in SBA-15, showing a greater difficulty for heat flow in those materials.

3.4. Effect of the Calcination Temperature on the Silanol Groups

The results of ²⁹Si solid-state CP-MAS NMR experiments are shown in Figure 6. The literature [69] reports three characteristic resonance peaks for these types of materials at -110 ppm, -101 ppm, and -92 ppm, corresponding to species Q⁴ (siloxane, Si(OSi)₄), Q³ (vicinal silanol, Si(OSi)₃(OH)) and Q² (geminal silanol, Si(OSi)₂(OH)₂).

Table 5 gives an overview of the meaning peaks characteristics, showing the position of the peaks, these intensities and the corresponding integration peaks of each specie, calculated in terms of percentage [78]. Figure 6 is a qualitative analysis and its results must be compared with some quantitative dates. In this manner, the Table 5 gives additional information showing that the integrations percentages areas oscillate between 13.8–16.1%, 69.3–73.4%, and 11.6–15.1% for SBA-15 samples, respectively, in the peaks Q^2 , Q^3 , and Q^4 , considering only the calcinated samples. In the same way, for the KIT-6 samples, the integration percentage areas oscillate between 11.8–16.9%, 67.8–77.3%, and 10.4–16.0%, respectively, in the peaks Q^2 , Q^3 , and Q^4 , considering only the calcinated samples. The position of the peaks is in accordance with literature [69].

Samples	Signal	δ (ppm)	Intensity	Area (%)
	Q ²	-91.3	127	8.0
SM0	Q^3	-101.3	549	65.8
	Q^4	-110.3	228	26.2
	Q ²	-91.2	145	15.9
SM1	Q^3	-100.5	455	72.5
	Q^4	-109.7	76	11.6
	Q^2	-91.4	39	16.1
SM2	Q^3	-100.5	123	72.3
	Q^4	-109.0	25	11.6
	Q ²	-91.6	662	13.8
SM3	Q^3	-100.3	2227	73.4
	Q^4	-109.0	516	12.8
	Q ²	-91.1	740	14.8
SM4	Q^3	-100.2	2308	72.0
	Q^4	-108.1	548	13.2
	Q ²	-91.3	160	16.0
SM5	Q^3	-100.4	449	70.5
	Q^4	-108.1	126	13.5
	Q^2	-91.2	746	15.6
SM6	Q^3	-100.1	1931	69.3
	Q^4	-109.7	439	15.1
	Q^2	-91.2	196	16.9
KM0	Q^3	-100.8	605	67.8
	Q^4	-109.8	142	15.3
	Q ²	-91.4	315	16.1
KM1	Q^3	-100.5	961	72.2
	Q^4	-108.9	188	11.7
	Q ²	-91.2	105	14.6
KM2	Q^3	-100.1	362	75.0
	Q^4	-109.0	64	10.4
	Q ²	-91.5	73	15.0
KM3	Q^3	-100.3	204	73.8
	Q^4	-109.0	39	11.2

Table 5. Parameters of ¹H decoupled ²⁹Si solid-state MAS NMR spectra of SBA-15 and KIT-6 samples.

Samples	Signal	δ (ppm)	Intensity	Area (%)
	Q ²	-91.2	1103	11.8
KM4	Q^3	-100.7	3928	77.3
	Q^4	-111.0	956	11.0
	Q^2	-91.2	1329	15.8
KM5	Q^3	-100.7	3528	69.0
	Q^4	-111.0	1086	15.2
	Q ²	-91.6	113	14.0
KM6	Q^3	-100.3	295	69.9
	Q^4	-108.7	76	16.0

SM0 SM1 SM2 0 O 0 o -100 -105 -110 -115 -120 δ(ppm) δ(ppm) .80 -85 -90 -95 -100 -105 -110 -115 -120 -80 -85 -90 -95 -80 -85 -90 -95 $\delta(\text{ppm})$ SM5 SM4 SM3 Q3 Q 0³ O o O, -80 -85 -90 -95 -100 -105 -110 -115 -120 -85 -90 -95 -100 -105 -110 -115 -120 -80 -85 -90 -95 -100 -105 -110 -115 -120 -80 δ(ppm) δ(ppm) δ(ppm) SM6 KM1 KM0 0 o 0 Q Q -85 -90 -95 -100 -105 -110 -115 -120 -85 -90 -95 -100 -105 -110 -115 -120 -85 -90 -95 -100 -105 -110 -115 -120 ę, -80 -80 δ(ppm) ð(ppm) δ(ppm) KM 2 KM 3 KM4 0 Q 0 -85 -90 -95 -100 -105 -110 -115 -120 -95 -100 -105 -110 -115 -120 -95 -100 -105 -110 -115 -120 -80 -80 -85 -90 -85 -90 -80 ð(ppm) δ(ppm) δ(ppm) C KM6 KM 5 O 0 O -95 -100 -105 -110 -115 -120 δ(ppm) -95 -100 -105 -110 -115 -120 δ(ppm) -85 -90 -85 -90 -80

Figure 6. ¹H decoupled ²⁹Si solid-state magic-angle spinning (MAS) nuclear magnetic resonance (NMR) spectra of the SBA-15 and KIT-6 samples calcined under different conditions.

Table 5. Cont.

The literature reports a tendency to increase the Q^3 and, especially, Q^4 signal, as the calcination temperature increases [69,78]. Molecular sieves such as SBA-15 and KIT-6 have more siloxane groups than vicinal silanols. Thus, the Q^4 signal should be more intense than the Q^3 signal, even in qualitative analyses. However, due to the porosity of these samples and because there is no pretreatment needed for NMR analysis, moisture retained in the pores affects the Q^3 and Q^4 signal intensities; thus, the Q^3 intensity is dominant [78]. A manner for minimizing this effect is a relative analysis the peaks. Table 6 presents the ratios of the percentage area integration of $Q^2:Q^4$, $Q^3:Q^4$, and $Q^2:Q^3$ just for the calcinated samples.

Q ² :Q ⁴	$Q^3:Q^4$	$Q^2:Q^3$
1.37	6.25	0.22
1.38	6.23	0.22
1.07	5.73	0.19
1.12	5.45	0.21
1.19	5.22	0.23
1.03	4.58	0.22
1.37	6.17	0.22
1.40	7.21	0.19
1.33	6.59	0.20
1.07	7.03	0.15
1.03	4.54	0.23
0.88	4.37	0.20
	$\begin{array}{c} \mathbf{Q^2:Q^4} \\ 1.37 \\ 1.38 \\ 1.07 \\ 1.12 \\ 1.19 \\ 1.03 \\ 1.37 \\ 1.40 \\ 1.33 \\ 1.07 \\ 1.03 \\ 0.88 \end{array}$	$\begin{array}{c c c c c c c c c c c c c c c c c c c $

Table 6. Peak ratios of ¹H decoupled ²⁹Si solid-state MAS NMR spectra of SBA-15 and KIT-6 samples.

According to results presented in the Table 6 for SBA-15 and KIT-6, the ratio $Q^2:Q^4$ decreases as the temperature increases, indicating that the geminal silanols are converted in siloxane as the temperature increases. Besides that, in 400 °C, the most accentuated decrease in this ratio begins, indicating that the dehydroxylation had started and that this is the crucial moment for preserving the geminal silanols. This is in accordance with the results obtained from N₂ sorption and XRD because silanol condensation shrinks the structure and decreases the total pore volume. This result indicates that calcination at 300 °C is more beneficial than that at 400 °C in terms of silanol preservation. Nonetheless, the N₂ sorption results show that the pore volume is small upon calcination at 300 °C. The ratio $Q^3:Q^4$ decreases too as the temperature increases; however, it is most accentuated at the most extreme treatment (500 °C during 10 h). In this way, this result indicates a tendency to convert the vicinal silanol in siloxane, as expected. The ratio $Q^2:Q^3$ is practically unchanged.

4. Conclusions

The calcination of mesoporous SBA-15 and KIT-6 at 300 °C for 5 h eliminates almost all the polymer and produces compounds with good structural properties. However, to optimize the textural properties of these materials, such as pore area and pore volume, a calcination at 400 °C for 10 h or at 500 °C for 5 h is the best condition for SBA-15, while for KIT-6, the best performance is obtained upon calcining at 500 °C for 5 h.

The deterioration of the silanol groups by condensation becomes more relevant at temperatures above 400 °C. The results show that geminal silanols are better preserved. Therefore, for the surface modification of SBA-15 and KIT-6 via their hydroxyl groups, it is preferable to calcine at 300 °C for 5 h or, at the most, at 400 °C for 5 h. Although there is a decrease in the pore area and volume, hydroxylated silica groups will be more abundant.

Although most mesoporous synthesis involves calcining for 5 h at 600 $^{\circ}$ C, to the best of our knowledge, this method is unnecessary, resulting in a waste of time and silanol groups without compensatory benefits to justify this method.

Author Contributions: "A.M.B and S.B.C.P. conceived and designed the experiments; A M.B. and B.P.N. performed the experiments; A.M.B. and K.B.-G. analyzed the data; S.B.C.P. contributed reagents/materials/analysis tools; A.M.B. and B.P.N wrote the paper; S.B.C.P. and K.B.-G. corrected the paper". All authors have read and agreed to the published version of the manuscript.

Funding: This research was funded by CAPES, Finance Code 001.

Acknowledgments: All the authors are grateful for support from LABPEMOL and LRC Brazilian laboratories groups, situated, respectively, in the cities of Natal/RN and Porto Alegre/RS.

Conflicts of Interest: The authors declare no conflict of interest. The funders had no role in the design of the study; in the collection, analysis, or interpretation of data; in the writing of the manuscript; or in the decision to publish the results.

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