



# *Article* **Modeling kV X-ray-Induced Coloration in Radiochromic Films**

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**Abstract:** Commercially available radiochromic films are primarily designed for clinical X-ray dosimetry. These films change color upon exposures to radiation as a result of solid-state polymerization (SSP). Built on a previous model developed for SSP upon exposures to ultraviolet (UV) radiation, a new model was developed in the present work for X-ray-induced coloration in Gafchromic EBT3 films. Monte Carlo simulations using the Monte Carlo N-Particle (MCNP) code were employed to model the transport and interaction of photons and the generated secondary electrons within the film active layer. The films were exposed to continuous-energy photon beams. The dose  $D<sub>E</sub>$  in the external radiation detector (i.e., ionization chamber) was determined and the realistic dose *D<sup>A</sup>* in the film active layer was then obtained using the calibration coefficient  $R (=D<sub>A</sub>/D<sub>E</sub>)$ . The finite element method (FEM) was used to solve the classical steady-state Helmholtz equation using the multifrontal massively parallel sparse direct solver (MUMPS). An extensive grid independence test was carried out and the numerical stability of the present model was ensured. The reflected light intensity from the film surface was used to theoretically obtain the net reflective optical density of the film exposed to X-ray. Good agreement was obtained between the experimental and theoretical results of the net reflective optical density of the film. For X-ray doses >~600 cGy, due to the already formed densely cross-linked structure in the active layer of the EBT3 film so further bond formation was less likely, the reflected light intensity from the film surface increased at a relatively lower rate when compared to those for dose values <~600 cGy.

**Keywords:** radiochromic film; EBT3 film; finite element method; parallel computation

## **1. Introduction**

Commercially available radiochromic films were designed for X-ray dosimetry [\[1\]](#page-6-0), where coloration of the films was used to obtain the dose–response relationship. The energy transferred to the receptive parts of the monomers in a radiochromic film could lead to color changes [\[1\]](#page-6-0) through solid-state polymerization (SSP) [\[2\]](#page-6-1). A model for the SSP process in Gafchromic EBT3 films was previously developed [\[3\]](#page-6-2). Since the model aimed at explicitly illustrating the SSP process, coloration upon exposures to ultraviolet (UV) radiation instead of X-rays was modeled to avoid the complication of the need to study the secondary electrons generated by X-ray photons. Built on that developed model [\[3\]](#page-6-2), a new model was developed in the present work for color changes in Gafchromic EBT3 films upon exposures to X-rays. To the best of our knowledge, no theoretical models are currently available for X-ray-induced coloration in EBT3 films. In the present work, Monte Carlo simulation was employed to model the interactions of X-ray photons and the generated secondary electrons in the EBT3 film active layer using the Monte Carlo N-Particle (MCNP) code package [\[4\]](#page-6-3). In our Monte Carlo model, the interaction and the transport of the generated secondary electrons were implemented, and dose dispersion [\[5\]](#page-6-4) in the film active layer was also considered. The realistic amount of energy transferred

to the EBT3 film active layer was obtained through Monte Carlo simulation and determination of the calibration coefficient R (= $D_A/D_E$ ) [\[6,](#page-6-5)[7\]](#page-6-6), where  $D_A$  was the actual dose delivered to the EBT3 film active layer and  $D_E$  was the dose recorded by an external ionization chamber.

Atomic scale and macroscopic scale models were employed to model the coloration of EBT3 films, as in the previous model [\[3\]](#page-6-2). The classical Helmholtz equation was used to model light reflection from<br>was analysis was directed to the direct solver (MUMPS) in the solver analysis was directed to model light reflection fr and transmittance through the modeled EBT3 film active layer, and the present system was solved not the finite element method (FEM) technique by employing the multifrontal massively parallel<br>using the finite element method (FEM) technique by employing the multifrontal massively parallel sparse direct solver (MUMPS) [\[8\]](#page-6-7). A grid independence test analysis was performed for different grid sizes to ensure the numerical stability of the present model. The theoretical and experimental results were compared.

# 2. Materials and Methods **The atomic scale model for diacetylene (DA)** pentacosa-10,12-divnoate monomers **and Methods** monomers in the EBT3 film active layer  $\mathbb{R}$  film active layer  $\mathbb{R}$  film active changes of the EBT3 film active contro

The atomic scale model for diacetylene  $(DA)$  pentacosa-10,12-diynoate monomers  $[3]$ , i.e., monomers in the EBT3 film active layer [\[9\]](#page-6-8), was used to obtain the density changes of the EBT3  $\frac{1}{2}$ film active layer. The density of the unexposed film active layer was used to further refine the results through an inverse problem approach. As in [\[3\]](#page-6-2), changes in the density of the monomer clusters were examined by carefully relaxing the system at the desired exposed energy and temperature (room temperature here) at a specific pressure, which correlated with the energy irradiating the films.

The experimental data employed in the present work for comparison with the modeled data were obtained by Chan et al. [10] who used the X-RAD 320 Irradiator (Precision X-ray (PXi), Inc., North Branford, CT, USA) at operating tube voltage of 150 kV for irradiation, with beam conditioning achieved with a filter consisting of 1.5 mm Al + 0.25 mm Cu + 0.75 mm Sn (and inherent filtration with 0.8 mm beryllium [\[7\]](#page-6-6)), and a surface-to-target distance (STD) of 50 cm. The photon energy distributi[on](#page-1-0) from this beam conditioning is shown in Figure 1, which is similar to the one obtained in [\[7\]](#page-6-6). The administrated dose was determined by the ionization chamber (IOC) (UNIDOS<sup>®</sup> E Universal Dosemeter, PTW, Freiburg, Germany) and ranged from ~11 to 1500 cGy [\[10\]](#page-6-9).

<span id="page-1-0"></span>

**Figure 1.** Photon energy distribution of the X-ray beam used to intervals the EBT3 film  $\frac{1}{2}$ Figure 1. Photon energy distribution of the X-ray beam used to irradiate the EBT3 film [\[10\]](#page-6-9).

#### *2.1. Monte Carlo Model 2.1. Monte Carlo Model*

between the incident photons with (i) the EBT3 film active layer and (ii) the IOC sensitive volume. The setups for irradiating the EBT3 film and the IOC are schematically shown in Figures 2 and 3, respectively. The energies deposited from photons and the generated secondary electrons were scored. The thickness of the EBT3 film active layer was 28  $\mu$ m and the surface area was  $1 \times 1$  cm<sup>2</sup>. The present Monte Carlo model performed two separate tasks, namely, to model the interaction

<span id="page-2-0"></span>

<span id="page-2-1"></span>Figure 2. Schematic diagram showing the irradiation of the EBT3 film (film active layer was 28  $\mu$ m; surface area was  $1 \times 1$  cm<sup>2</sup>).



Chamber, UNIDOS<sup>®</sup> E Universal Dosemeter, PTW, Freiburg, Germany) (IOC: ionization chamber). **Figure 3.** Schematic diagram showing the irradiation of the IOC (model TN 30 013, PTW Farmer® **Figure 3.** Schematic diagram showing the irradiation of the IOC (model TN 30 013, PTW Farmer®

The photon doses absorbed in the targeted domains (i.e., EBT3 film active layer and IOCO). The volume) were determined using the track length estimate of energy deposition (tally F6:P). The mass The mass energy-absorption coefficients (*µen*(*E*)*/ρ*)*m* of the EBT3 film active layer and air were used to determine the absorbed photon dose (*D*) as The photon doses absorbed in the targeted domains (i.e., EBT3 film active layer and IOC The photon doses absorbed in the targeted domains (i.e., EBT3 film active layer and IOC sensitive volume) were determined using the track length estimate of energy deposition (tally F6:P). The mass energy-absorption coefficients  $(\mu_{en}(E)/\rho)_m$  of the EBT3 film active layer and air were used to determine the absorbed photon dose  $(D)$  as

$$
D = \int \psi(E) E\left(\frac{\mu_{en}(E)}{\rho}\right)_m dE \tag{1}
$$

where *ψ*(*E*) was the differential photon fluence for photon energy *E*, and integration from minimum to  $t$  maximum energies in the photon energy spectrum was needed. In the present MCNPX computations, the mass energy-absorption coefficients in the MCPLIB04 database were employed. Data for  $\sigma$  cross-sections, form factors, and fluorescence were all derived from the ENDF/B-VI.8 data fibrary. The photon-coupled-electron transport mode was used to track the propagation of the incluent incident photons and the generated secondary electrons in both EBT3 film active layer and IOC sensitive. The coupled transport mode was essential due to the dose deposited by the secondary secondary electrons. For the pure photon transport mode, local energy deposition would be assumed at their creation sites, which was not desirable for precise determination of energy deposition in the deposition in the EBT3 film active layer and IOC sensitive volume. where  $\psi(E)$  was the differential photon fluence for photon energy  $E$ , and integration from minimum to  $\frac{1}{1}$  maximum energies in the photon energy spectrum was needed. In the present MCNPX computations, the mass energy-absorption coefficients in the MCPLIB04 database were employed. Data for cross-sections, form factors, and fluorescence were all derived from the  $ENDF/B-VI.8$  data library. The photon-coupled-electron transport mode was used to track the propagation of the incident photons and the generated secondary electrons in both EBT3 film active layer and IOC sensitive sensitive volume. The coupled transport mode was essential due to the dose deposited by the volume. The coupled transport mode was essential due to the dose deposited by the secondary electrons. For the pure photon transport mode, local energy deposition would be assumed for electrons at their creation sites, which was not desirable for precise determination of energy deposition in the  $EBT3$  film active layer and IOC sensitive volume.

#### *2.2. Finite Element Method (FEM) 2.2. Finite Element Method (FEM) 2.2. Finite Element Method (FEM)*

Light propagation through the EBT3 film active layer was modeled using the classical steady-state state Helmholtz equation as Light propagation through the EBT3 film active layer was modeled using the classical steady-Light propagation through the EBT3 film active layer was modeled using the classical steady-state Helmholtz equation as

$$
\nabla \cdot (D\nabla \phi) + \mu_{\alpha} \phi = S \tag{2}
$$

where  $\phi$  was the intensity, S was the source term, and  $D = 1/3(\mu_{\alpha} + \mu_s)$ , where  $\mu_{\alpha}$  and  $\mu_s$  were absorption and reduced scattering coefficients, respectively. On further expansion, Equation (2) could absorption and reduced scattering coefficients, respectively. On further expansion, Equation (2) could be expressed as be expressed as

$$
\frac{\partial}{\partial x}\left(D\frac{\partial\phi}{\partial x}\right) + \frac{\partial}{\partial y}\left(D\frac{\partial\phi}{\partial y}\right) + \frac{\partial}{\partial z}\left(D\frac{\partial\phi}{\partial z}\right) + \mu_{\alpha}\phi = S.
$$
\n(3)

Detailed derivation of the model was given in [\[3\]](#page-6-2) and thus would not be repeated here. In the FEM technique, the modeled domain would be broken down into smaller sub-domains using a seamless in the model was given in the repeated here. In the repeated mesh. In [\[3\]](#page-6-2), 4-noded tetrahedron elements were used to divide the simulated EBT3 film active layer.<br>-In the present work, a grid independence test was performed for the FEM model by varying the grid in the simulated EBT3 film the simulated to divide the simulated EBT3 film the simulated EBT3 film the simulated EBT3 film t size and determining the final *Net ROD* value at different X-ray doses. Here, the grid sizes were refined by the present work, a grid sizes were refined through the grid refinement factor (*r*) defined as

$$
r = h_{\text{coarse}} / h_{\text{fine}} \tag{4}
$$

where *hcoarse* was the mesh or grid size before refinement, while *hfine* was the mesh or grid size after refin[em](#page-6-10)ent. In practice,  $r$  should be larger than 1.3 [11], so here we chose three different average grid sizes of 7.68, 5.15, and 3.73  $\mu$ m, which gave *r* values of  $\sim$ 1.49 (=7.68/5.15) and  $\sim$ 1.38 (=5.15/3.73). The absorption coefficients were obtained through  $\mu_{\alpha} = \rho \times (\mu_{\alpha})_{m}$ , where  $\mu_{\alpha}$  was the absorption coefficient,  $(\mu_{\alpha})_m$  the mass-energy absorption coefficient (set to be at 100 cm<sup>2</sup>/g [\[3\]](#page-6-2)), and  $\rho$  the density of the atomic scale system obtained from our current model for X-ray irradiation. The FEM model was tuned so that the input absorption coefficient for the small-scale atomic model could provide *Net ROD* values that could be compared with those obtained experimentally.

#### *2.3. Computation Scheme*

Following [\[3\]](#page-6-2), a monoenergetic light source was used to expose the film, and the reflected light intensity was determine[d](#page-3-0) (schematically shown in Figure 4). The *Net ROD* was defined as

$$
Net\,ROD = \log\left(\frac{\phi_{unexposed}}{\phi_{exposed}}\right) \tag{5}
$$

<span id="page-3-0"></span>where  $\phi_{unexposed}$  and  $\phi_{exposed}$  were the reflected light intensities from the unexposed and exposed  $\Xi$ film to X-ray, respectively. The numerical simulations presented in this paper were executed on a min to X-ray, respectively. The numerical simulations presented in this paper were executed on a<br>supercomputer with Intel® Xeon E5-2630 v3 2.40 GHz using 32 physical cores and hyper-threaded to 64. 64. supercomputer with Inter webit Is 2630 v3 2.40 GHz using 32 physical cores and hyper-threaded



**Figure 4.** Exposure of EBT3 film to the light source and reflection from the film surface. **Figure 4.** Exposure of EBT3 film to the light source and reflection from the film surface.

## **3. Results and Discussions 3. Results and Discussions**

The dose *DE* delivered to the IOC sensitive volume by photons and the secondary electrons The dose *D<sup>E</sup>* delivered to the IOC sensitive volume by photons and the secondary electrons generated by the incident photons was determined to be 8.373 × 10−16 Gy/photon. The dose *DA* generated by the incident photons was determined to be 8.373 × 10−<sup>16</sup> Gy/photon. The dose *D<sup>A</sup>* delivered to the EBT3 film active layer was determined to be 8.695  $\times$  10<sup>-16</sup> Gy/photon with an associated relative error (i.e., the statistical uncertainty) of 0.007. The administrated doses given to associated relative error (i.e., the statistical uncertainty) of 0.007. The administrated doses given to the EBT3 film in the experiments were calibrated using  $R = D_A/D_E = 8.695 \times 10^{-16}/8.373 \times 10^{-16} = 1.038$ . As R was close to unity,  $D_A$  and  $D_E$  were not very different when the photon beam with the continuous energy distribution shown in Figure 1 was employed.

The absorption coefficient of the EBT3 film active layer at different administrated X-ray doses obtained from our model is shown in Figure [5.](#page-4-0) The absorption coefficient increased noticeably for X-ray doses <~600 cGy, but did not change significantly for larger X-ray doses. The insignificant changes in the absorption coefficient for large X-ray doses were due to the formation of a densely cross-linked structure in the exposed EBT3 film active layer, in which significantly more bonds would not form even with a further supply of energy to the monomers. For illustration purposes, snapshots from atomic configurations of the DA monomers [\[9\]](#page-6-8) before and after the addition of energy from X-ray photons are shown in Figure [6a](#page-4-1),b, respectively, which revealed the formation of a densely polymerized system upon exposure of EBT3 films to X-ray.

<span id="page-4-0"></span>

**Figure 5.** Calculated absorption coefficients of EBT3 film active layer versus X-ray dose.

<span id="page-4-1"></span>

Figure 6. Atomic scale configuration of DA monomers: (a) before and (b) after exposure to X-ray. (Green spheres: carbon atoms; purple spheres: hydrogen atoms; lines: bonding among different atoms in the monomer chains, with more lines representing formation of a denser cluster and thus a denser polymerization.) (Note: A 3D model of polymerization was considered. However, 2D representation of the atomic scale system was used for better visualization) (DA: diacetylene).

The obtained absorption coefficients were input into the FEM model and the reflected light intensities  $(log(\phi))$  from the surface of the EBT3 film were obtained as shown in Figure [7.](#page-5-0) The reflected light intensity decreased with increasing X-ray dose absorbed by the film active layer. Similar to the film absorption coefficient (Figure [5\)](#page-4-0), the reflected light intensity from the EBT3 film surface decreased noticeably for X-ray doses <~600 cGy, but did not change significantly for larger X-ray doses. The latter was explained by the insignificant changes in the absorption coefficients for larger X-ray doses as a result of formation of a densely cross-linked structure in the exposed EBT3 film active layer. Since the  $log(\phi)$  values were obtained from the FEM model, a grid independence test analysis was performed. analysis was performed. For the three three three three distributions was performed.<br>For the three different grid sizes, namely, 7.68, 5.15, and 3.73 µm, *log*( $\phi$ ) did not vary significantly with the X-ray doses. For the three different grid sizes, namely, 7.68, 5.15, and 3.73  $\mu$ m, *log*( $\phi$ ) did not vary significantly with the X-ray doses.

Similar to [\[3\]](#page-6-2), the *Net ROD* in our model was defined as *Net ROD* =  $log(\phi_{unexposed}/\phi_{exposed})$ , where  $\phi_{unexposed}$  and  $\phi_{exposed}$  were reflected light intensities from the films unexposed and exposed to X-ray, respectively. The *Net ROD* values from the present model were scored over the entire surface of the radiochromic films. The average reflected intensity was determined as the arithmetic mean of the  $\ddot{\phantom{1}}$  internal data obtained domain  $\ddot{\phantom{1}}$ radiochromic films. The average reflected intensity was determined as the arithmetic mean of the radiochromic films. The average reflected intensity was determined as the arithmetic mean of the

[int](#page-6-11)ensities from the elements in the modeled domain [12–14]. The experimental data obtained by The experimental data of the contents in the included abilitary  $\frac{1}{2}$  and  $\frac{1}{2}$ . The experimental data of diffuse  $\frac{1}{2}$ . that the theoretical predictions were very close to the experimental data. The uncertainties shown as shades around the theoretical prediction curves were introduced by the Monte Carlo model. These<br>we setting is a general and deviations abiding from the multiplication of the validing uncertainties represented one standard deviations obtained from the multiplication of the relative error estimated using the (F6:P) tally by the tally mean value [\[15\]](#page-6-13). The slope of the calibration curve in Figure 8 was larger at [sm](#page-5-1)aller doses and tended to saturate at larger doses. In other words, the degree of polymerization (surrogated by the *Net ROD*) increased at a faster rate with the dose when the dose was smaller. Increase in the X-ray dose beyond  $~600$  cGy did not further darken the film due to saturated polymerization, which prevented a significant additional formation of bonds. The trend<br>caused with the identification of increases in mochanical stresses between the management and the agreed with the identification of increases in mechanical stresses between the monomers and the polymer units with the degree of polymerization [\[16\]](#page-7-0). Kim et al. [\[17\]](#page-7-1) also remarked that the mobility of monomer chains decreased during SSP because the crystallinity increased, and the crystalline structure became more stable. **7.70** to saturated using the (Fo.F) tany by the tany mean value [15]. The slope of the canonal formation of bonds. The trend

<span id="page-5-0"></span>

<span id="page-5-1"></span> $\frac{60}{2}$  from the EBT3 from the EBT3 film surface versus  $\frac{60}{2}$  from the EBT3 film surface grid sizes, namely, 7.68, 5.15 and 3.73  $\mu$ m. **Figure 7.** Reflected light intensity ( $log(\phi)$ ) from the EBT3 film surface versus X-ray dose for three



**Figure 8.** Comparison of *Net ROD* values obtained experimentally by Chan et al. [10] and theoretically in the present work (for three average grid sizes of 7.68, 5.15, and 3.73 µm) for different administrated Figure 8. Comparison of *Net ROD* values obtained experimentally by Chan et al. [\[10\]](#page-6-9) and theoretically in the present work (for three average grid sizes of 7.68, 5.15, and 3.73 µm) for different administrated X-ray doses. The error bars for the experimental data are smaller than the symbols. The shades around the theoretical curves represent the uncertainties introduced by the Monte Carlo model.

#### X-ray doses. The error bars for the experimental data are smaller than the symbols. The shades around **4. Conclusions** the theoretical curves represent the uncertainties introduced by the Monte Carlo model.

In the present work, we built on a previous model developed for SSP to develop a new model **4. Conclusions**  for X-ray-induced coloration in EBT3 films. The Monte Carlo method was employed to model the

transport of photons and the generated secondary electrons within the film active layer. The calibration coefficient *R* was used to determine the exact amount of energy deposited in the film active layer. A good agreement between the theoretically determined *Net ROD* with those obtained experimentally was obtained. In general, the slope of the calibration curve was larger at smaller doses and tended to saturate at doses beyond ~600 cGy. The experimental uncertainties could also be contributed by oxygenation, UV contamination, and variation in film coloration with time. These uncertainties were not simulated by the Monte Carlo method and therefore might lead to discrepancies between the data obtained theoretically and experimentally.

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**Author Contributions:** M.S.B. and K.N.Y. conceived and designed the experiments; M.S.B. and D.K. performed the experiments; M.S.B., D.K., and D.N. analyzed the data; D.N. and K.N.Y. contributed reagents/materials/analysis tools; and M.S.B., D.N., and K.N.Y. wrote the paper.

**Conflicts of Interest:** The authors declare no conflict of interest.

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