



Article

# Self-Supported Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> Nanowire Array on Nickel Foam as an Efficient and Durable Electrocatalyst for Alkaline Hydrogen Evolution

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**Abstract:** Earth-abundant and low-cost catalysts with excellent electrocatalytic hydrogen evolution reaction (HER) activity in alkaline solution play an important role in the sustainable production of hydrogen energy. In this work, a catalyst of Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> nanowire array on nickel foam has been prepared via a facile route for efficient alkaline HER. Benefiting from the collaborative advantages of Ni(P, O)<sub>x</sub> and amorphous MoO<sub>x</sub>, as well as three-dimensional porous conductive nickel scaffold, the hybrid electrocatalyst shows high catalytic activity in 1 M KOH aqueous solution, including a small overpotential of 59 mV at 10 mA cm<sup>-2</sup>, a low Tafel slope of 54 mV dec<sup>-1</sup>, and excellent cycling stability.

**Keywords:** Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> nanowire array; synergistic effect; electrocatalyst; alkaline hydrogen evolution reaction

## 1. Introduction

Exploring new sustainable energy resources and clean energy carriers to replace the traditional fossil fuels is one of the most important challenges of the 21st century. Hydrogen is considered as the most promising energy carrier for sustainable energy applications due to its outstanding energy storage density, environmental friendliness, and renewability [1–3]. Electrochemical water splitting is an important component of several hydrogen generation strategies. However, an efficient catalyst is required to reduce the energy barrier of the hydrogen evolution reaction (HER) [4]. So far, the most effective electrocatalysts for HER are Pt group materials, but the scarcity and high cost of these noble metals significantly limit their wide utilization [5]. Herein, the development of low-cost and efficient HER electrocatalysts based on earth-abundant species is of great importance [6–8].

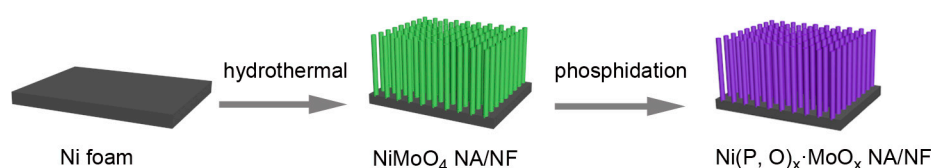
Up to now, various non-precious metal-based materials (e.g., Ni, Co, Fe, Cu, W, and Mo) have been intensively synthesized as promising HER catalysts with high performance [9–12]. Among these alternatives, crystalline MoO<sub>2</sub> has been identified as an excellent candidate owing to its good electric conductivity and high electrocatalytic activity [13–15]. It is noted that most of the reported non-precious electrocatalysts are based on crystalline compounds. In recent years, a growing class of amorphous materials have emerged as more efficient electrocatalysts compared with their crystalline counterparts [16–20]. However, the amorphous catalysts suffer from poor cycling stability caused by slow dissolution of the catalyst components during long-term test, thus resulting in easy degradation in the electro-activity [21,22]. To mitigate this critical problem, a large number of studies have shown that coupling different functional species can generate a strong synergistic effect to significantly improve

the performance [22]. It is important to note that Ni-based electrocatalysts exhibit excellent HER catalytic activity in alkaline media due to the appropriate OH-Ni<sup>2+δ</sup> ( $0 \leq \delta \leq 1.5$ ) bond strength [23,24]. Therefore, it is highly desirable and interesting to combine Ni-species with MoO<sub>2</sub> to synergistically achieve substantial improvements in electro-activity and durability.

Herein, we highlight a Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> nanowire array which grows directly on a nickel foam support (Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF) for a highly efficient electrocatalyst which exhibits preferable HER activity. The direct integration of nanowire array onto the Ni foam not only simplifies the electrode preparation processes, but also ensures the tight connection between electrode framework and active species, resulting in enhanced mechanical stability. In addition, the commercial nickel foam acts as a three-dimensional (3D) macroporous conductive substrate that facilitates facile charge transfer, electrolyte diffusion, and gas bubble release. Consequently, benefiting from the collaborative advantages of Ni-species and amorphous MoO<sub>x</sub>, the as-prepared Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF electrode shows a remarkable electrocatalytic activity with a low overpotential of 59 mV to attain a current density of 10 mA cm<sup>-2</sup> and superior stability for at least 24 h in an alkaline environment, thereby demonstrating a highly-efficient HER catalyst.

## 2. Results and Discussion

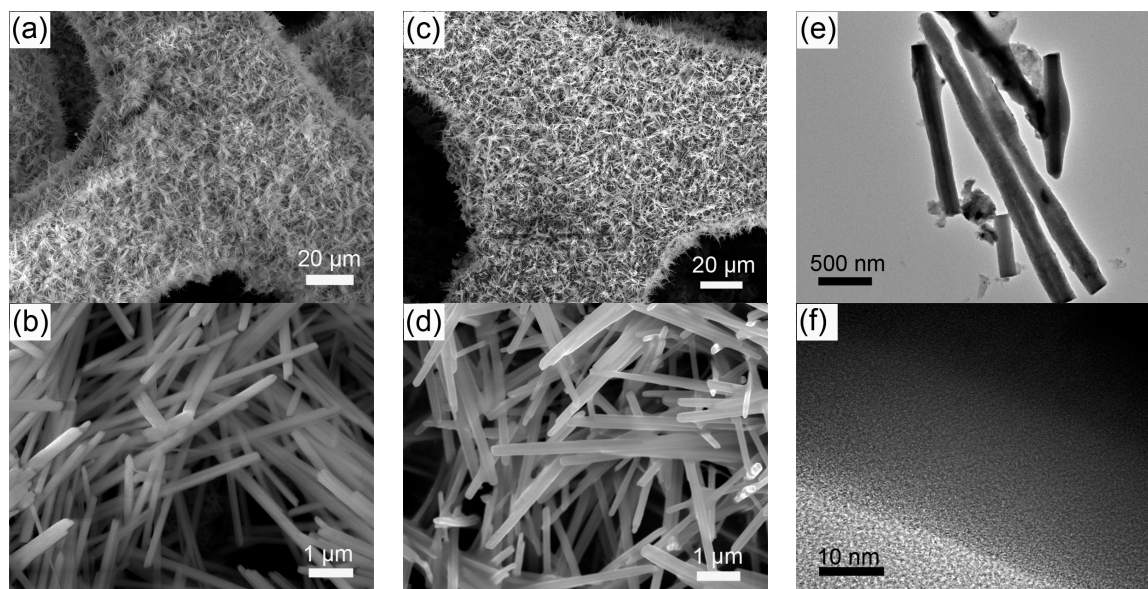
The self-supported Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> nanowire array is fabricated on a commercial Ni foam by a facile template-free hydrothermal process in combination with a subsequent in situ phosphorization treatment. Figure 1 schematically illustrates the typical two-step preparation process. In the first step, the Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> precursor (i.e., NiMoO<sub>4</sub>·xH<sub>2</sub>O) is grown on the 3D porous skeletons of the nickel foam by a hydrothermal reaction. In the second step, the Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> catalyst is obtained through a solid-state phosphorization process between the NiMoO<sub>4</sub>·xH<sub>2</sub>O precursor and NaH<sub>2</sub>PO<sub>2</sub>. The precursor is thermally transformed to crystalline NiMoO<sub>4</sub> nanowire array supported on the Ni foam (NiMoO<sub>4</sub> NA/NF), during which a simple dehydration reaction occurs. As can be seen from the scanning electron microscopy (SEM) image (Figure 2a), high-density NiMoO<sub>4</sub> NA spreads uniformly over the nickel foam skeletons. A closer observation (Figure 2b) indicates that the diameter of the nanowire is about 210 nm, and the length is more than 6 μm. After phosphidation, the 1D nanowire array is maintained well from the precursors (Figure 2c,d), and the diameter of the Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> nanowires is similar to the NiMoO<sub>4</sub>. Transmission electron microscopy (TEM) was employed to further depict the as-prepared Ni(P, O)<sub>x</sub>·MoO<sub>x</sub>. Figure 2e shows the corresponding TEM image of Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF, further identifying the preservation of the 1D morphology after phosphidation. The high-resolution TEM (HRTEM) image (Figure 2f) shows no obvious evidence of lattice fringes, suggesting that the as-synthesized Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> is amorphous or of poor crystallinity.



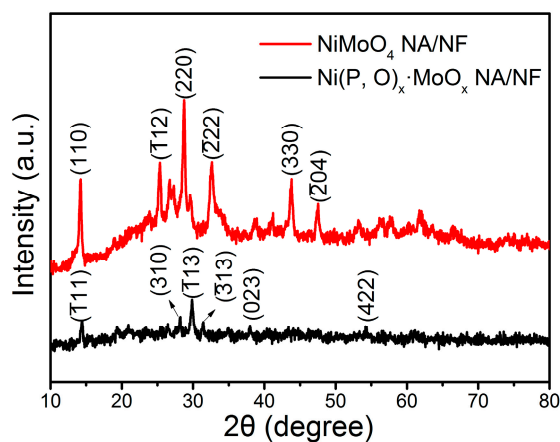
**Figure 1.** Schematic illustration of the formation processes of the Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> nanowire array which grows directly on a nickel foam support (Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF).

The phase structure of the as-prepared samples was examined by X-ray diffraction (XRD) analysis. As shown in Figure 3, the distinct diffraction peaks with 2θ at around 14.3°, 25.4°, 28.9°, 32.6°, 43.9°, and 47.5° correspond to the (002), ( $\bar{1}12$ ), (220), ( $\bar{2}22$ ), (330), and ( $\bar{2}04$ ) crystal planes of NiMoO<sub>4</sub>, respectively (JCPDS No. 86-0361) [25,26]. Compared with the crystalline NiMoO<sub>4</sub>, the Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> sample exhibits weak diffraction peaks, indicating that the phosphidation process results in a significant decrease in the crystallinity. The main peaks can be assigned to nickel phosphates (Ni<sub>2</sub>P<sub>4</sub>O<sub>12</sub>, JCPDS

No. 76-1557). The absence of Mo-related peaks demonstrates that the Mo-based species are amorphous in the as-synthesized  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF [27].

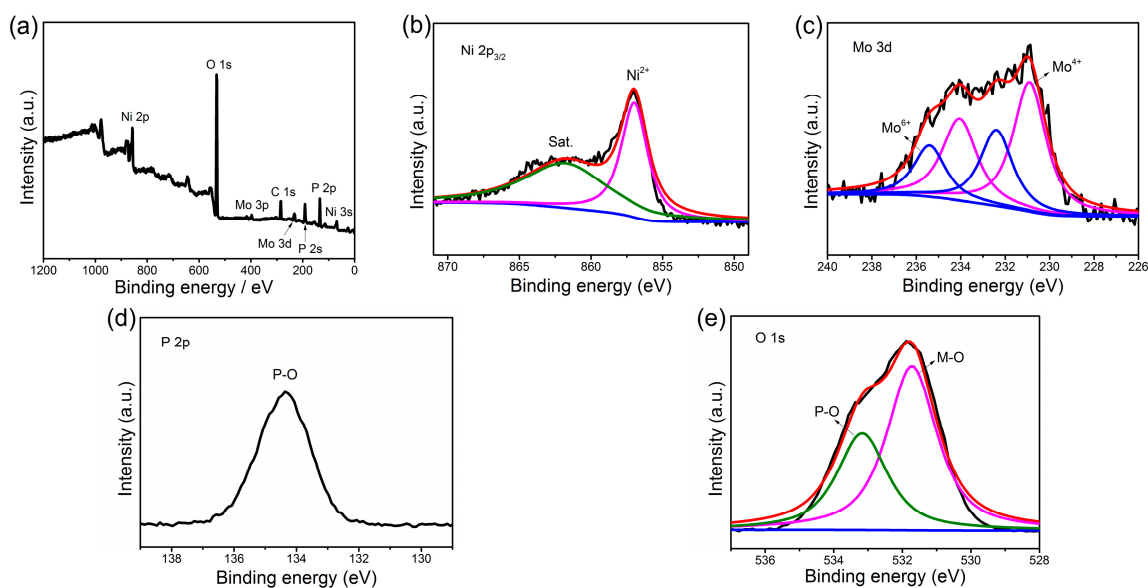


**Figure 2.** SEM images of (a,b)  $\text{NiMoO}_4$  NA/NF and (c,d)  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF; (e) TEM and (f) HRTEM images of  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF.



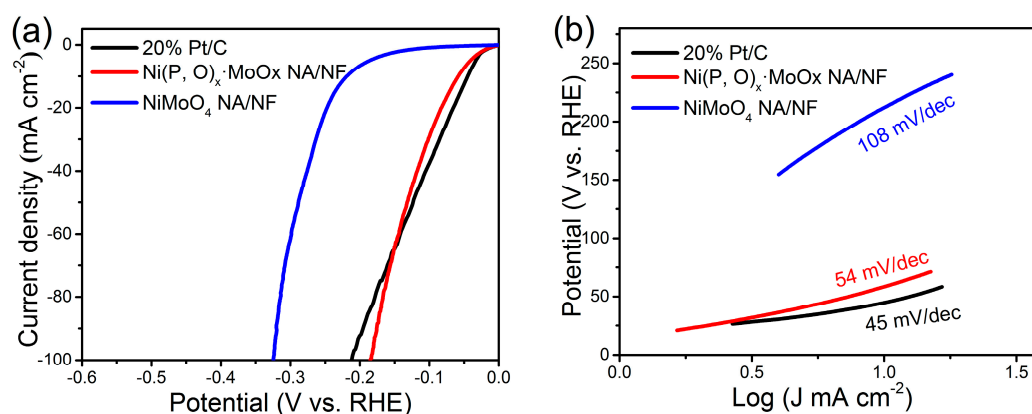
**Figure 3.** XRD pattern of the  $\text{NiMoO}_4$  NA/NF and  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF.

X-ray photoelectron spectroscopy (XPS) measurement was carried out to investigate the surface composition and the oxidation state of the  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF. The survey spectra show that the  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF is composed of Mo, Ni, P, O elements (Figure 4a) and the atomic percentage of P in the product is 17.16%. The Ni  $2p_{3/2}$  high-resolution spectrum (Figure 4b) exhibits two main peaks at binding energies of 856.9 and 862.1 eV, which can be assigned to the Ni–O bond and the satellite peak, respectively [28]. The Mo 3d spectrum of the  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF (Figure 4c) can be resolved into two sets of peaks corresponding to the  $\text{Mo}^{6+}$  and  $\text{Mo}^{4+}$  species, and the ratio between the  $\text{Mo}^{6+}$  and  $\text{Mo}^{4+}$  in the composite is 0.57. The presence of  $\text{Mo}^{4+}$  species is probably attributed to the reduction of the  $\text{Mo}^{6+}$  precursor during phosphidation process [29]. For the profile of P 2p, the sample (Figure 4d) only shows a peak at a binding energy of 134.4 eV, which represents the P–O bond [30]. The high-resolution O 1s spectrum (Figure 4e) can be fitted into two peaks at 531.7 and 533.1 eV, which can be ascribed to the metal–oxygen (M–O) and P–O bonds, respectively [28].



**Figure 4.** X-ray photoelectron spectroscopy (XPS) spectra of  $\text{Ni}(\text{P}, \text{O})_x \cdot \text{MoO}_x$  NA/NF: (a) full scan; (b)  $\text{Ni } 2p_{3/2}$ ; (c)  $\text{Mo } 3d$ ; (d)  $\text{P } 2p$ ; and (e)  $\text{O } 1s$ . M–O: metal–oxygen.

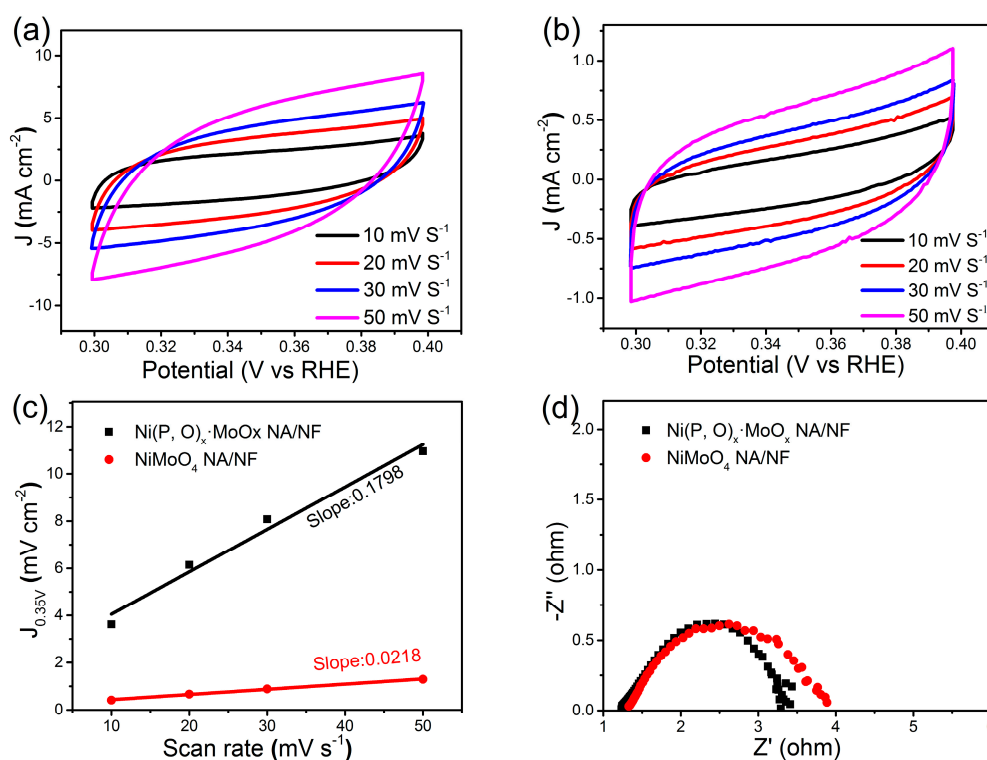
The HER performance of the  $\text{Ni}(\text{P}, \text{O})_x \cdot \text{MoO}_x$  NA/NF was examined in 1 M KOH aqueous solution. For comparison, commercial Pt/C (20 wt % Pt/XC-72) and  $\text{NiMoO}_4$  NA/NF were also evaluated. Figure 5a shows the IR-corrected linear sweep voltammetry (LSV) curves. The  $\text{Ni}(\text{P}, \text{O})_x \cdot \text{MoO}_x$  NA/NF electrode exhibits a low overpotential of 59 and 185 mV to reach a current density of 10 and 100  $\text{mA cm}^{-2}$ , respectively. In sharp contrast, the control  $\text{NiMoO}_4$  NA/NF electrode requires much higher overpotentials of 219 and 324 mV to achieve the same current densities. The lower overpotential of the  $\text{Ni}(\text{P}, \text{O})_x \cdot \text{MoO}_x$  NA/NF electrode indicates a significant improvement in the HER catalytic property. Impressively, the overpotential is almost comparable to the commercial Pt/C electrode, demonstrating that the present electrode material may serve as a practical cathode for the high-efficiency production of hydrogen.



**Figure 5.** (a) Linear sweep voltammetry (LSV) polarization curves for hydrogen evolution reaction (HER) and (b) corresponding Tafel plots.

Figure 5b shows the corresponding Tafel plots. It is worth noting that the Tafel slope of  $\text{Ni}(\text{P}, \text{O})_x \cdot \text{MoO}_x$  NA/NF is about 54  $\text{mV dec}^{-1}$ , which is only half of the control  $\text{NiMoO}_4$  NA/NF electrode (108  $\text{mV dec}^{-1}$ ). This low Tafel slope indicates that the HER occurs on the  $\text{Ni}(\text{P}, \text{O})_x \cdot \text{MoO}_x$  NA/NF electrode following the Volmer–Heyrovsky mechanism, and the rate-limiting step is the electrochemical recombination with an additional proton [9]. More importantly, the  $\text{Ni}(\text{P}, \text{O})_x \cdot \text{MoO}_x$

NA/NF catalytic activity is superior to most Mo-based HER electrocatalysts reported so far (Table 1). In addition, the amount of catalytically active surface area on NiMoO<sub>4</sub> NA/NF and Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF electrodes are roughly estimated from the electrochemical double-layer capacitance (C<sub>dl</sub>) by measuring cyclic voltammetry (CV) curves at different scanning rates (Figure 6a,b). The determined C<sub>dl</sub> for Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF (89.9 mF cm<sup>-2</sup>) is much higher than NiMoO<sub>4</sub> NA/NF (10.9 mF cm<sup>-2</sup>) (Figure 6c), suggesting a larger surface active area and more exposed active sites [10]. Figure 6d shows that the charge-transfer resistance of the Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF electrode (3.4 Ω) is smaller than that of the NiMoO<sub>4</sub> NA/NF (3.9 Ω), indicating rapid charge transfer. The large electro-active surface area along with the enhanced charge transfer kinetics of the Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF are believed to be responsible for the associated higher HER catalytic activity.



**Figure 6.** Cyclic voltammograms of (a) Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF and (b) NiMoO<sub>4</sub> NA/NF; (c) Scan rate-dependent current densities at 0.35 V (vs. reversible hydrogen electrode, RHE); and (d) Nyquist plots of Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF and NiMoO<sub>4</sub> NA/NF.

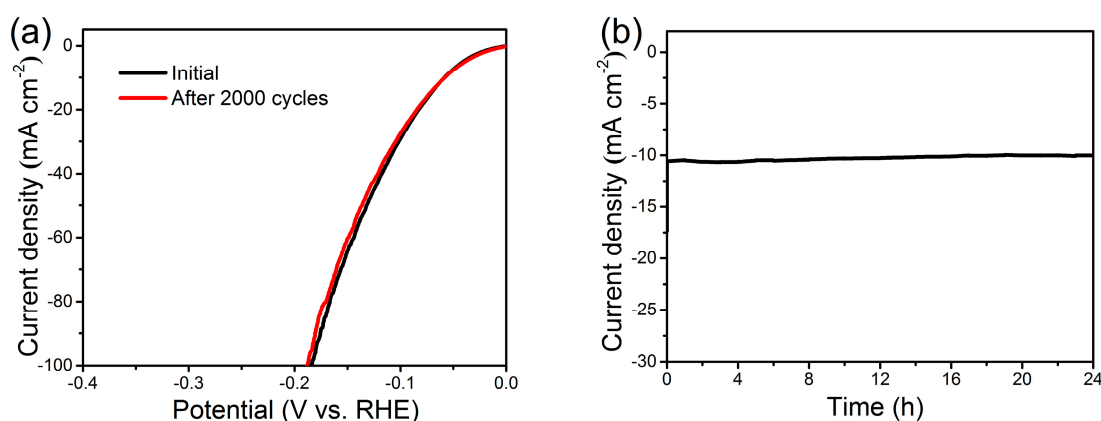
**Table 1.** Comparison of HER performance for Ni(P, O)<sub>x</sub>·MoO<sub>x</sub> NA/NF with Mo-based electrocatalysts.

Catalyst <sup>[a]</sup>	Overpotential at $j = 10 \text{ mA cm}^{-2}$ (mV)	Tafel Slope (mV dec <sup>-1</sup> )	Electrolyte	Reference
MoO <sub>2</sub> @PC-RGO	64	41	0.5 M H <sub>2</sub> SO <sub>4</sub>	[15]
MoP/Ni <sub>2</sub> P/NF	75	100	1 M KOH	[28]
Ni/Mo <sub>2</sub> C	179	101	1 M KOH	[31]
NiMoN-550	89	79	1 M KOH	[32]
Mo <sub>2</sub> C@NC	60	60	1 M KOH	[33]
MoP NA/CC	80	83	1 M KOH	[34]
MoS <sub>2</sub> /MoO <sub>2</sub>	240	76	0.5 M H <sub>2</sub> SO <sub>4</sub>	[35]
MoO <sub>2</sub> /RGO	—	68	0.5 M H <sub>2</sub> SO <sub>4</sub>	[36]
MoP S	64	50	0.5 M H <sub>2</sub> SO <sub>4</sub>	[37]
Ni(P, O) <sub>x</sub> ·MoO <sub>x</sub> NA/NF	59	54	1 M KOH	This work

<sup>[a]</sup> PC-RGO: phosphorus-doped carbon-reduced graphene oxide; NC: nitrogen-rich carbon; CC: carbon cloth.

Good catalytic stability is of critical significance for an electrocatalyst when it comes to potentially practical implementation, particularly considering that the HER catalysts work in harsh environments [8]. The  $\text{Ni(P, O)}_x\cdot\text{MoO}_x$  NA/NF electrocatalyst was first evaluated via a recycling test using LSV method. As shown in Figure 7a, the LSV curves are almost overlapped with a slight loss of the cathodic current densities, indicating a negligible active degradation before and after 2000 scanning cycles. The excellent cycling stability is further validated by the time dependence of the current density curve at a constant overpotential of 60 mV (Figure 7b). The  $\text{Ni(P, O)}_x\cdot\text{MoO}_x$  NA/NF manifests a stable catalytic current over 24 h, confirming the long-term durability of the electrocatalytic activity.

It is believed that the high alkaline HER performance of the  $\text{Ni(P, O)}_x\cdot\text{MoO}_x$  NA/NF can be attributed to the combination of compositional and geometric advantages: (1)  $\text{Ni}^{2+}$  is of great benefit for the adsorption of hydroxyl species, while amorphous  $\text{MoO}_x$  provides catalytically active sites for the adsorption of  $\text{H}^*$  intermediate and further facilitates the subsequent formation of  $\text{H}_2$ . The synergistic cooperation greatly reduces the energy barriers of the initial water decomposition and the subsequent step of  $\text{H}_2$  generation. (2) Coupling Ni-species and amorphous  $\text{MoO}_x$  generates a strong synergistic effect to significantly improve the stability. (3) The nanowire array offers a larger surface active area with more exposed active sites. (4) The 3D porous and conductive nickel foam not only effectively increases the contact area between active catalyst and electrolyte, but also serves as a robust skeleton to provide strong mechanical adhesion and electric connection to the nanowire array, thereby ensuring facile charge and mass transport, gas bubble release, and good electrode structure for long-term test.



**Figure 7.** (a) Polarization curves for  $\text{Ni(P, O)}_x\cdot\text{MoO}_x$  NA/NF in 1 M KOH initially and after 2000 cycles at a scan rate of  $50 \text{ mV s}^{-1}$ ; (b) Time-dependent current density of  $\text{Ni(P, O)}_x\cdot\text{MoO}_x$  NA/NF at 60 mV (vs. RHE).

### 3. Materials and Methods

#### 3.1. Synthesis of $\text{Ni(P, O)}_x\cdot\text{MoO}_x$ NA/NF

All chemical reagents were of analytical grade and used as received without further purification. The Ni foam with a thickness of 1.6 mm and dimensions of  $2 \times 4 \text{ cm}^2$  was sonicated in diluted hydrochloric acid (1 M), acetone, deionized water, and ethanol for 10 min, respectively. In a typical synthetic process, 1 mmol  $\text{Ni(NO}_3)_2\cdot 6\text{H}_2\text{O}$  and 1 mmol  $\text{Na}_2\text{MoO}_4\cdot 2\text{H}_2\text{O}$  were dissolved in 30 mL  $\text{H}_2\text{O}$  to form a clear solution. Subsequently, the solution and purified Ni foam were transferred into a 50 mL Teflon-lined stainless autoclave, which was sealed and heated at  $160 \text{ }^\circ\text{C}$  for 6 h in an oven. After the reaction, the resulting light-green Ni foam was rinsed with deionized water and ethanol, then the sample was dried at  $60 \text{ }^\circ\text{C}$  for overnight.

In the next step, the obtained  $\text{NiMoO}_4\cdot x\text{H}_2\text{O}$  NA/NF precursor and 10 mmol  $\text{NaH}_2\text{PO}_2$  were placed at two separate positions of the tube furnace with the  $\text{NaH}_2\text{PO}_2$  at the upstream side. Subsequently, the samples were heated at  $400 \text{ }^\circ\text{C}$  for 120 min with a ramp rate of  $2 \text{ }^\circ\text{C min}^{-1}$  under

flowing nitrogen. After cooling to room temperature naturally, the resulting electrode was obtained. The mass loading of the as-prepared  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF on the Ni foam was  $\sim 5.4 \text{ mg cm}^{-2}$ . The synthesis of  $\text{NiMoO}_4$  NA/NF was the same as  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF, just without  $\text{NaH}_2\text{PO}_2$ .

### 3.2. Material Characterization

The crystallographic phase of the products was examined by X-ray diffraction (XRD) with  $\text{Cu K}\alpha$  radiation ( $\lambda = 0.15418 \text{ nm}$ ) (X'Pert Pro MPD, Philips, Almelo, The Netherlands). The morphology was characterized by field emission scanning electron microscopy (FE-SEM, FEI Nano SEM 450, FEI, Portland, OR, USA) and transmission electron microscopy (TEM, FEI Tecnai F30G2, FEI, Portland, OR, USA). The surface chemistry and elemental analysis of the sample were characterized by X-ray photoelectron spectroscopy (XPS, ESCALAB 250Xi, Thermo Scientific, Waltham, MA, USA).

### 3.3. Electrochemical Measurements

The catalytic performances of the electrocatalysts were investigated by using an electrochemical workstation (Solartron 1260 + 1287, Bognor Regis, West Sussex, UK) in a three-electrode system. The  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  NA/NF was used as the working electrode; a graphite rod and the saturated calomel electrode (SCE) were used as the counter and reference electrode, respectively. All of the final potentials were calibrated to a reversible hydrogen electrode (RHE). The polarization curves were corrected with IR compensation. The working electrodes were activated before the measurement by cyclic voltammetric scans with a scan rate of  $50 \text{ mV s}^{-1}$ . The HER performances of the obtained electrocatalysts were tested from 0.2 to  $-0.4 \text{ V}$  (vs. RHE) in 1 M KOH aqueous solution by LSV with a scanning rate of  $2 \text{ mV s}^{-1}$ . Electrochemical impedance spectroscopy (EIS) was carried out at  $-0.2 \text{ V}$  (vs. RHE) over a frequency range from 100 kHz to 0.01 Hz with a 10 mV AC dither. To determine the catalytically active surface area of the products, the electrochemical double-layer capacitance ( $C_{dl}$ ) of the electrodes was estimated by using CV method in a non-Faradaic range of 0.3–0.4 V (vs. RHE) at various scan rates. A linear relationship between the current densities at 0.35 V (vs. RHE) and scan rate can be plotted to obtain  $C_{dl}$ , the value of which is half of the resulting slope. The catalytically active surface area of different electrocatalysts can be directly compared by the  $C_{dl}$  values, because the  $C_{dl}$  is in proportion to the active surface area [28].

## 4. Conclusions

In summary, a novel  $\text{Ni(P, O)}_x \cdot \text{MoO}_x$  nanowire array supported on a Ni foam was prepared via a facile approach. Because of the synergistic effect of the Ni-species and amorphous  $\text{MoO}_x$ , the as-prepared catalyst exhibits excellent electrocatalytic performance in an alkaline media, including a low overpotential of 59 mV at  $10 \text{ mA cm}^{-2}$ , a small Tafel slope of  $54 \text{ mV dec}^{-1}$ , and long-term stability. The enhanced electrocatalytic performance demonstrates the advantageous combination of compositional and geometric factors. The present work also provides an avenue to fabricating low-cost alkaline electrocatalysts for practical implementation.

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**Author Contributions:** Jian-Gan Wang and Wei Hua conceived and designed the experiments; Wei Hua and Huanyan Liu performed the experiments; Jian-Gan Wang, Wei Hua and Bingqing Wei analyzed the results and co-wrote this paper.

**Conflicts of Interest:** The authors declare no conflict of interest. The founding sponsors had no role in the design of the study; in the collection, analyses, or interpretation of data; in the writing of the manuscript, and in the decision to publish the results.

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