



Review

Microstructured All-Optical Switching Based on Two-Dimensional Material

Jiao Xu ¹, Yuxiang Peng ², Shengyou Qian ^{1,*} and Leyong Jiang ^{1,*}

- School of Physics and Electronics, Hunan Normal University, Changsha 410081, China
- Institute of Mathematics and Physics, Central South University of Forestry and Technology, Changsha 410004, China
- * Correspondence: qiansy@hunnu.edu.cn (S.Q.); jiangly28@hunnu.edu.cn (L.J.)

Abstract: Microstructured all-optical switching, possessing the unique function of light controlling light, is an important part of the on-chip ultra-fast optical connectivity network and integrated logic computing chip. Microstructured all-optical switching has attracted extensive research interest, the latest great developments of which have also yielded progress in nanophotonics, nonlinear optics, optical communications, and integrated optics, etc. The emergence of two-dimensional materials with good third-order optical nonlinearity provides an important driving force for the improvement of all-optical switches. This paper reviews the implementation principles, novel configurations, improved performance indexes, and research progress based on different two-dimensional materials for micro/nano all-optical switching. Not only is a systematic discussion of the current state provided, but also, a brief outlook is afforded on the remaining challenges in the pursuit of the application of practical on-chip microstructured all-optical switching that is based on two-dimensional materials.

Keywords: all-optical switching; graphene; black phosphorus; transition metal dichalcogenide; layered material heterostructures



Citation: Xu, J.; Peng, Y.; Qian, S.; Jiang, L. Microstructured All-Optical Switching Based on Two-Dimensional Material. *Coatings* 2023, 13, 876. https://doi.org/ 10.3390/coatings13050876

Academic Editor: Piotr Potera

Received: 28 February 2023 Revised: 11 March 2023 Accepted: 20 March 2023 Published: 6 May 2023



Copyright: © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https://creativecommons.org/licenses/by/4.0/).

1. Introduction

Micro/nano-scale all-optical devices are considered the core of next-generation ultrafast communication and signal processing systems. In the current phase of photoelectric hybrids, all-optical switching technology must be developed to break through the bottleneck of information exchange. As the core component, all-optical switches play a crucial role in building on-chip ultrafast all-optical switch networks [1]. All-optical switching is a device based on the light-controlled-by-light concept, which can convert or logically operate the optical signals in an optical transmission line or integrated optical circuit. The all-optical switch uses fast optical nonlinearity to achieve a fast and reversible transition from one state to another for a parameter of the optical signal, such as power, phase, frequency, and polarization, etc. [2]. This lays the foundation for optical computing systems, optical communication networks, and quantum information processing chips [3]. Therefore, controlling photons on the micro–nano scale and developing all-optical devices with a smaller size and better performance have become a hot spot in the international research frontier, as well as in regard to competition with new technologies.

Because of the scientific importance and practical application of all-optical switching, it has been conducted for a long time, even dating back to the study of optical bistability in 1965 and also the experimental demonstration of optical bistability in 1976 [4]. However, the nonlinear refractive index of conventional materials is small and the ohmic loss of metallic materials is large. These material bottleneck limitations constitute a major obstacle to all-optical switching. Although optical nonlinearities are related to optical matter interactions enhanced by microstructures (including microcavities) and optical nonlinearities enhanced by slow-light effects, it is still difficult to achieve nanoscale all-optical switching with ultra-high speed, ultra-low energy consumption, and ultra-high switching efficiency is

Coatings **2023**, 13, 876 2 of 25

still difficult to implement [1]. In addition, large nonlinear magnetization rates and ultrafast response times are difficult to achieve simultaneously for conventional third-order optically nonlinear materials, including semiconductors, polymers, and liquid crystals, even in previously reported microphotonic devices, where light–matter interactions and subsequent optical nonlinearities are typically weak [5–7]. This also limits the use of optical third-order nonlinear materials in the context of all-optical switching.

In fact, in addition to many traditional third-order nonlinear materials, many new materials have been investigated. Recent studies have shown that two-dimensional materials have great potential for achieving ultrafast all-optical switching. The remarkable optical properties of two-dimensional materials make them promising for important applications in nanophotonics. In this review, we discuss recent advances in microstructured all-optical switching based on two-dimensional materials, including the special strategies to address inherent barriers and fundamental challenges for ultrafast nanoscale on-chip triggered all-optical switching. The basic principles, performance indicators, and novel two-dimensional materials for realizing ultrafast all-optical switches are reviewed. In addition to this, the current research challenges are briefly anticipated, as well as the development directions and future prospects of microstructured all-optical switching that are based on two-dimensional materials.

2. Basic Concepts and Realization Principles

All-optical switching is based on nonlinear optics, which generally occur when the optical field that is associated with one or more laser beams propagating in a material are large enough to produce a polarization field that is directly proportional to the product of two or more incident optical fields [8,9]. All-optical switching based on a nonlinear principle operates directly in the optical domain without photoelectric conversion [10], resulting in high efficiency, low switching noise, low loss [11,12], and switching speeds that are of nanoseconds, picoseconds, or even less [13]. However, since photons are not charged, it is not possible to control light by light directly, as the case in electronics; only nonlinear optical indirect methods can be used. With an external beam of strong light (controllable pump), or by the intense light of the signal light itself (self-pumped), and the propagation of the signal light-medium interaction, only then does its optical properties undergo nonlinear changes. The changes, such as absorption or refraction of the medium (the imaginary or real part of susceptibilities), occur depending on the change in optical power, so as to control the signal light wave vector, phase, or polarization, thus resulting in finally achieving control of the intensity (amplitude) or propagation direction of the signal light [14]. For instance, the phase shift is given by Equation (1):

$$\Delta \varphi = 2\pi \Delta n F L / \lambda \tag{1}$$

where F accounts for any cavity resonance enhancement, light travels a distance L, and the change in the refractive index of light is Δn . All-optical switches are based on a variety of different nonlinear mechanisms. The main nonlinear mechanisms and corresponding switching principles are briefly listed in the Table 1 [15]:

Table 1. Nonlinear mechanisms and corresponding switching principles.

Nonlinear Mechanisms	Switching Principles
Nonlinear refraction	Optical Kerr effect, self-focusing, self-defocusing, and the two-photon refraction effect
Nonlinear absorption	Saturated and reverse saturable absorption, two-photon absorption, and dichroism
Nonlinear reflection	Reflection at the nonlinear interface or on the surface of two prisms containing nonlinear liquid

Coatings **2023**, 13, 876 3 of 25

Table 1. Cont.

Nonlinear Mechanisms	Switching Principles
Nonlinear polarization	Nonlinear optical rotation effect of liquid crystal and chiral materials caused by strong light
Nonlinear frequency conversion	Frequency doubling, sum frequency, parametric process, four-wave mixing, stimulated Raman scattering, etc.
Nonlinear phase transition	Photo-induced material phase transitions causes a change in the refractive index or absorption coefficient of the medium
Nonlinear gratings	Single nonlinear grating and the grating pair are connected by nonlinear waveguides
Nonlinear couplers	The phase difference between the two arms of light are caused by a strong light incident nonlinear symmetric optical coupler
Nonlinear amplifier	Strong light saturates the semiconductor optical amplifier and changes the phase of light
Nonlinear interferometer	The change in the refractive index of the interferometer material is caused by the strong light, which causes the phase difference between the two beams to be π

There are many kinds of all-optical switches and ways in which to realize them. These include the optical bistable all-optical switching [4,16,17], the optical limiting all-optical switch [18], the nonlinear interferometer all-optical switching (UNI) [19], the nonlinear grating all-optical switching [20], the nanophotonics all-optical switch [2,21], and other all-optical switching. In conclusion, all-optical switching is a device that changes the intensity or propagation direction of controlled light by controlling the nonlinear optical effect of light on the medium. Furthermore, it is a key technology and important device for future optical computing systems, as well as high-speed all-optical communication networks.

All-optical switching is based on the spatial or temporal nonlinear response of the material [18,22]. Practical all-optical switching devices can use photonic (or plasmonic) micro/nano-structures as a matrix, combined with optical nonlinear materials. Nonlinear optical microstructures (including nonlinear photonic crystals (PhCs), PhC nanocavities [23,24], nonlinear dielectric ring nanocavities side-coupled bus signal waveguides [25], etc.), nonlinear plasmonic structure-like nonlinear plasmonic nanocavities [26], and nonlinear metamaterials/super surfaces [27], etc., are widely used to construct all-optical switching. The basic implementation mechanism of all-optical switching is the nonlinear Kerr effect, which can be defined as a light-induced double refraction or the intensity-dependent refractive index. The refractive index of nonlinear material can be obtained by using the nonlinear Kerr effect [28]:

$$n = n_0 + n_2 I \tag{2}$$

where n_0 is the linear refractive index, n_2 is the nonlinear part, and I is the intensity of the control light.

2.1. Photonic Crystal All-Optical Switching

In the past few decades, photonic crystals have become a rapidly growing field in quantum physics and micro-optics. One of its major characteristics is that it can control the propagation of photons and is the host of various new optical nonlinear effects [29]. PhCs are artificial microstructures with a spatially periodic dielectric distribution. The periodic distribution of the refractive index can produce a photonic band gap (also called the stop band) in photonic crystals. No photon can pass through the photonic band gap, and the incident signal light is completely reflected [30]. If the periodic characteristics of PhCs are destroyed by introducing defects, corresponding defect energy levels will

Coatings **2023**, 13, 876 4 of 25

be formed in the photonic band gap. Only light with a specific frequency can appear in this defective energy level. Changes in the permittivity of nonlinear materials change the spatially periodic refractive index distribution of photonic crystals, leading to changes in the photonic crystal band structure. Switching photonic bandgap crystals provides dynamic control of the density of states, which will allow switching on or off of the light source in the bandgap [31]. In addition, the photonic crystal structure can have an infinite number of arrangement forms, which makes it possible to realize the all-off switch that is based on photonic crystals [32]. All-optical switching based on a silicon chip is realized in photonic crystal nanocavities (~50 ps) [33]. For example, Takasumi et al. proved that all-optical switches on a silicon chip were realized by photonic crystal nanocavities at a high speed (~50 ps). D. Nau et al. discussed the optical switching of metallic photonic crystal slabs with a photochromic polymer [34]. Kengo et al. demonstrated low-energy all-optical switching by using a combination of photonic crystal nanocavities and strong carrier-induced nonlinearity in InGaAsP [2]. Zhao et al. demonstrated an ultrafast optical switching based on photonic molecule-waveguide coupling in a photonic crystal plate structure [35].

2.2. All-Optical Switching Based on Dielectric Ring Nanocavities

By coupling the bus signal waveguide on the side, the dielectric ring nanocavity can be used to build all-optical switching devices [25]. As we all know, the resonant mode is very sensitive to the dielectric constant. Due to the optical Kerr effect of nonlinear materials, the dielectric constant can be changed by the pump light. In addition, the signal light can be selected at the nanocavity mode center. Then, the incident light is coupled into the nanocavity through the coupling aisle. No signal light can be transmitted through the bus signal waveguide, which corresponds to the OFF state of the all-optical switching. The nanocavity mode of the controlled optical pumping deviates from the signal light wavelength, and the energy coupling between the signal light and the nanocavity mode is cut off, corresponding to the ON state of the all-optical switching. Therefore, the all-optical switch function is formed by coupling a dielectric ring nano-cavity and a side bus signal waveguide.

2.3. Plasmonic All-Optical Switching

Recently, surface plasmon polaritons (SPPs) have been found to provide an alternative way by which to achieve strong nonlinear optical effects, as well as to minimize all-optical components due to its significantly enhanced optical field strength and light manipulation in a nanoscale domain [36–38]. Surface plasmonpolaritons (SPPs), a concept of surface waves, are formed by the combined action of electron oscillations and incident photons, which are located at the interface of two different media with opposite signs of the real part of the dielectric constant [39]. For extended metal surfaces, this means that SPPs are surface electromagnetic wave-propagation modes occurring between metal dielectric interfaces. For metals in nanoparticles, the response yields local electromagnetic wave oscillation modes (surface electromagnetic wave propagation modes), which indicate the presence of local SPPs. In this context, many new physical concepts have been proposed, including metamaterials, plasmonic nanoparticles, and plasmonic PhCs, etc. Plasmonic nanostructures have small feature sizes in the order of hundreds of nanometers, as well as possess the unique property of limiting light intensity to the subwave length range, and have tremendous near-field enhancement [40].

In recent years, several nonlinear optical devices based on SPPs have been investigated [41–45]. Compared with dielectric nanostructures, plasmonic nanostructures have stronger optical-confinement and field-enhancement effects, which are suitable for achieving all-optical switching. Three factors make SPPs an excellent candidate for achieving ultra-low and ultra-fast all-optical switching. First, SPPs can enhance the interaction between light and matter due to its powerful local enhancement effect, as this reduces the pump intensity. Second, changes in the dielectric constant of the ambient medium can greatly affect the resonant response of SPPs. A small change in the surrounding material

Coatings **2023**, 13, 876 5 of 25

index can produce a large change in the resonance properties [46], which can also reduce the pump energy consumptions. Finally, the relaxation time of the plasmon resonance can reach the sub-picoseconds level, which can ensure ultra-fast all-optical switching. This makes it feasible to build all-optical switching devices using plasmonic nanostructures. In the presence of controlled light, the refractive index of the ambient medium material changes, which leads to a shift in the resonance wavelength of the plasma mode. Compared with other types of all-optical switching, SPP all-optical switching require more third-order nonlinear optical materials due to their low excitation characteristics. In summary, typical plasma all-optical switching devices are based on the nonlinear Kerr effect by using a control-light-induced plasmonic mode shift or window shifts of plasmone-induced transparency (PIT).

3. Significance Performance Indexes

According to the requirements of practical applications, the five most basic and important indicators of all-optical switch are as per the following [47]:

- Ultrafast switching time (transition time between "ON" and "OFF" states). Switching
 time is the time at an output port of a switch when the switch is turned on and off,
 measured from the moment the switching energy is applied to or removed from the
 switch, respectively;
- 2. Ultralow threshold control power. The time required for the optical power to change from the initial state to the on or off state; the switching power is the minimum input control power required to turn on the output signal light in the optical switch and the control optical power required for the all-optical switch, which can be controlled in two ways: self-control and cross-control;
- 3. Ultrahigh switching efficiency. The transmission or reflection contrast between "ON" and "OFF" states;
- 4. Nanoscale feature size. The nonlinear coefficients of the nonlinear materials used in conventional optical switching devices are low, thus requiring large dimensions that make it difficult to accommodate the requirements of small size and low power consumption in integrated optics. Therefore, it is also crucial to realize nano-sized optical switches;
- 5. Manufacturability. In addition, all-optical switches need to be reliable, low cost to produce, and simple to process.

The most important of these switching characteristics are switching time and control power. The emergence of a new optical switch begins with the consideration of these two characteristic parameters. However, these two parameters are often contradictory, e.g., liquid crystals and semiconductors have low switching power, but it is difficult to make the switches all become fast-responding [48]. Kerr dielectrics and organic materials have fast switching speeds [12], but the required switching power is too high. Due to these contradictions, all-optical switches cannot, as of yet, be practically applied thus far. In order to solve the practical problems of all-optical switches, it is necessary to select the appropriate parameters and to optimize the design. While the most widely used conventional 1×2 and 2×2 mechanical optical switches are not affected by polarization and wavelength and have millisecond switching speeds, they are still large in size—which limits their large-scale application. Additionally, when metal elements are used for nanoscale light confinement, there are problems of low efficiency and losses [49]. Therefore, in recent years, the research and development of optical switches has adopted many new techniques, mechanisms, and materials; in addition, the scale of optical switches is also increasing. Various approaches have been proposed, and three methods have been widely used: synthesizing novel thirdorder nonlinear optical materials; constructing novel structures and mechanisms to enhance nonlinear interactions; and seeking novel physics and ideas to achieve all-optical switches. Thus far, it is still a challenge to realize the ultra-low power consumption, broadband operation, and high switching efficiencies of ultra-high speed on-chip-triggered all-optical switches at the same time.

Coatings **2023**, 13, 876 6 of 25

Since the switching time depends mostly on the response time of the nonlinear material and the switching efficiency depends on the size of the nonlinear sensitivity of the nonlinear material, for conventional third-order nonlinear materials, it is difficult to obtain ultra-fast responses and huge nonlinear refractive indexes at the same time. In addition, resonant excitation enhances the optical nonlinearity while reducing the temporal response of the nonlinear material. Typical pump power and response times are GW cm⁻² and several hundred picoseconds, respectively. The high power consumption and low response time of all-optical switches severely limit their practical applications. In integrated photonic circuits, in addition to these key important performance metrics, there are many performance metrics, such as on-chip trigger and wideband (or multiple) wavelengths. These excellent metrics are becoming a key concern for researchers. Thus far, it is still a challenge to achieve the ultra-low power consumption, broadband operation, and high switching efficiencies of ultra-high speed on-chip-triggered all-optical switches simultaneously.

For the all-optical switching based on nonlinear optical effects, the main performance of the optical switch depends on the material quality. Additionally, for optical bistable switching devices based on the Kerr effect, the nonlinear refraction coefficient n_2 and linear absorption coefficient α_0 of the material are important parameters. It can be shown that the critical input light intensity of the device is $I_c \propto \lambda \alpha_0/n_2$ (λ is wavelength). This means that the smaller α_0 and larger n_2 , then the lower the switching power. Therefore, $n_2/\lambda\alpha_0$ is a quality factor to measure the goodness of the optical switch material. Materials with large nonlinear refraction coefficients and small absorption coefficients are necessary to design the optical switch in order to obtain a low switching power. If the response time of the material is also considered an important factor for a fixed operating wavelength, the quality factor of the optical switch can be written as per the following:

$$Q = \frac{n_2}{\alpha_0 \tau} \tag{3}$$

Typical nonlinear optical material characteristic parameters are shown in Table 2.

Nonlinear Materials	$n_2/(\mathrm{cm}^2/\mathrm{W})$	α_0 /cm ⁻¹	τ/s	$Q/[cm^3/(s{\cdot}W)]$
PleLiquid crystal (ferroelectric liquid crystal, etc.)	10^{-7}	10 ³	10^{-3}	10^{-7}
Semiconductors (GaAs, etc.)	10^{-8}	10^{4}	10^{-8}	10^{-4}
Organic materials (PTS, etc.)	10^{-16}	10^{-1}	10^{-12}	10^{-3}
Glass (Si, SiO2, etc.)	10^{-20}	10^{-5}	10^{-14}	10^{-1}

Table 2. The characteristic parameters of several nonlinear optical materials.

4. Two-Dimensional Material

In order to meet the application requirements of ultra-fast all-optical devices, it is necessary to combine new two-dimensional materials with strong nonlinearity. Two-dimensional (2D) materials exhibit a variety of electronic properties, from insulating hexagonal boron nitride and semiconducting transition metal dichalcogenides (TMDs, such as molybdenum disulfide) to semi-metallic graphene, as shown in Figure 1 [50]. The successful preparation of two-dimensional materials has completely released the huge potential of two-dimensional materials in the research field of all-optical devices. In recent years, 2D nanomaterials have been favored and sought after by researchers due to their unique structural characteristics and novel physical properties. Additionally, a large number of graphene-like structures have been predicted theoretically or synthesized experimentally, such as hexagonal nitride sheds, black phosphorus, transition metal-sulfur compounds, and germanene, etc. Currently, these two-dimensional materials form a particularly complete class of low-dimensional nanomaterial systems, with a wealth of properties and functional attributes to meet the needs of almost all research areas. Based on the characteristics of the layered structure of two-dimensional materials, different two-dimensional materials can

Coatings 2023, 13, 876 7 of 25

form heterogeneous structures, which is extremely important for the realization of various all-optical optoelectronic devices.

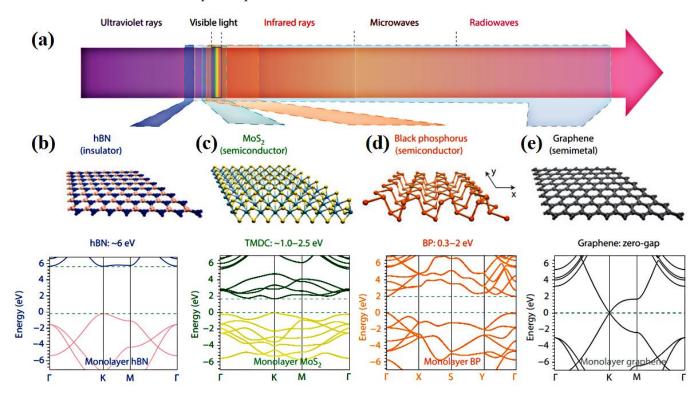


Figure 1. Examples of 2D materials covering a broad spectral range [50]. (a) Electromagnetic spectrum. (b–e) Are the atomic structure and the band structures of (b) hBN, (c) MoS₂, (d) BP, and (e) graphene.

The dimensionality of a material not only defines its atomic composition, but also determines, to a great extent, the properties of the material itself. Among them, the most famous material in the field of two-dimensional material technology is carbon-based material. The discovery of and research on carbon materials have promoted the progress of modern science and technology. After the early research boom of 0D fullerenes (C60) [51] and 1D single-walled carbon nanotubes [52], Geim and Novoselov et al. successfully exfoliated 2D graphene monolayers in 2004, for which they were awarded the 2010 Nobel Prize in Physics [53]. It is this hexagonal monolayer of sp² hybridized carbon atoms that has particularly striking mechanical and electronic properties, such as a linear dispersion relation in its energy band at the K-high symmetry point in the Brillouin zone. The special energy band structure has led to many novel physical phenomena, such as the anomalous quantum Hall effect. In addition, it has opened up a whole new field of research in "Fermi-Dirac" physics. Not only that, graphene has a high transparency of 97.7% [54], has excellent mechanical properties [55], and is also an excellent conductor of electricity and heat [56,57]. Currently, graphitic materials have been successfully applied to build light-transmitting conductors, thermal interface materials, and potential barrier transistors, etc. [58]. In addition, the density of the states of graphene is near the Dirac point, and the Fermi energy level can change significantly with the carrier density shift. Thus, four-wave mixing can be enhanced by a slow light photonic crystal waveguide that is coated with a single layer of graphene. The energy gap in the patterned nanoribbons can be tuned by a proper selection of the bandwidth of the preparation, which shows the flexibility of graphene in all-optical switching [59]. Overall, the discovery of graphene can be said to have truly opened a new chapter in the study of two-dimensional materials. The great success of graphene has also encouraged researchers to actively search for other two-dimensional materials with novel physical properties. Many theoretically proposed two-dimensional monolayers have

Coatings **2023**, 13, 876 8 of 25

also been synthesized, including the derivatives of graphene, such as black phosphorus, transition metal carbides, and hexagonal nitrides [60,61].

Unlike graphene, the two-dimensional hexagonal boron nitride (hBN) monolayer structure is a wide-band insulator material [62]. Single-layer hBN nanosheets can be considered graphene-like materials, equivalent to graphene in which half of the carbon atoms are replaced by nitrogen atoms and the other half by boron atoms; moreover, they are white in color, such that they are often called white graphene. Compared to graphene—which possesses excellent electrical conductivity—hexagonal boron nitride is a wide band gap insulator material, which has especially good thermal and chemical conductivity, as well as neutron-absorption capabilities. Two-dimensional hexagonal boron nitride can also change its band gap width by introducing impurity atoms or atomic site lattice defects to obtain semiconductor properties. Secondly, the hBN monolayer structure has a near-flat-band-energy band signature throughout the Brillouin zone, while graphene has a linearly dispersive energy band near the Fermi energy level and thus forms a perfect Dirac cone, which is a zero-bandgap semimetal [61,63,64].

Two-dimensional material black phosphorus (BP), ever since it was reported by Prof. Zhang and Prof. Chen in 2014 [65], has been a hot topic of research in the field of twodimensional materials. Black phosphorus is one of the three main isomers of phosphorus and is the most thermodynamically stable compared to the other two (i.e., red phosphorus and white phosphorus). Black phosphorus has four crystal structures, orthogonal, rhombic, simple cubic, and amorphous (which is orthogonal under atmospheric and room temperature conditions). Similar to graphite, bulk black phosphorus is a layered material consisting of layers of individual atoms superimposed by van der Waals interactions. However, unlike graphite, the atoms within the same layer within black phosphorus are not in the same plane. In addition, the phosphorus atoms in the same layer are in two planes above and below, showing a honeycomb-like folded structure. It is this folded structure that makes the crystal structure of black phosphorus asymmetric, resulting in properties that are very different from other two-dimensional materials, such as graphene, which exhibits a very pronounced anisotropy. First of all, black phosphorus has anisotropic mechanical properties, the asymmetry of which will lead to different mechanical properties in different directions. The photon–electron interaction is more pronounced when the incident light is polarized along the armchair edge direction of the BP nanosheet when compared to the zig-zag edge direction. Thus, the optical absorbance, transmittance, and reflectance properties provide a means to indicate the orientation of the BP crystal. Optical anisotropy further provides opportunities for the application of polarization-dependent linear and nonlinear optical devices, based on stripped BP nanosheet materials. Meanwhile, the photoelectric properties of BP are directly related to its layer number. BP is 1.51 eV, which can emit red light via excitation, and the band gap of the sample becomes smaller as the thickness increases.

In subsequent explorations, researchers have gradually shifted their focus to the monolayer material systems of transition metal dichalcogenides (TMDs) [63]. Compared to the traditional two-dimensional materials, two-dimensional transition metal materials are more abundant, and research on these materials accounts for half of the research field of two-dimensional materials [66]. In current research progress, two-dimensional transition metal materials generally have numerous advantages over graphene, making them very promising for applications in electronic devices, such as field-effect transistors, lithium-ion batteries, sensors, and photocatalysis, etc. [67]. The TMDs also exhibit a number of low-temperature-specific phenomena, including superconductivity, charge density waves, and Mott phase transitions. These unique properties are often retained when these materials are stripped into a single layer or a few layers. Due to the local effect, additional interesting properties can also be added [68,69]. Similar to graphene, when two-dimensional TMDs are thinned from a macroscopic bulk to a few layers or a single layer, its two-dimensional or even two-dimensional thickness exhibits particularly different physical properties from the bulk material. In contrast to the semi-metallic character of

Coatings **2023**, 13, 876 9 of 25

graphene, the TMDs show more potential for all-optical switching research due to their rich band gap variety. Therefore, TMDs are expected to surpass graphene and open up completely new research opportunities for inorganic two-dimensional materials.

In addition, the TMDs obtained by inserting carbon elements into the lattice of metallic materials are a diverse group of material systems with many interesting properties and promising applications. These materials usually have good metal conductivity and catalytic activity comparable to those of noble metals. Recently, the development of transition metal carbides has entered the research field of two-dimensional materials; for example, Duan et al. successfully prepared large-area, high-quality two-dimensional α -Mo2C ultrathin layer structures via the process of chemical vapor deposition (CVD) [70,71]. Moreover, this general synthetic method can be used to obtain more two-dimensional materials with unique properties. Such stable, high-quality two-dimensional crystalline materials not only provide a new material platform for research in the field of superconductivity physics, but also exhibit the characteristics required for an ideal material in the preparation of all-optical open-light devices.

5. All-Optical Switching Based on Two-Dimensional Materials

Although all-optical switching has seen significant progress in the field of plasma and photonic crystals, it is still a great challenge to achieve all the required indicators of all-optical switches. The nonlinear refractive index of traditional materials is small, and the ohmic loss of metals is still large. Additionally, it is difficult to achieve large nonlinear susceptibility and an ultra-fast response time at the same time for traditional nonlinear optical materials (as shown in Table 2), which also limits the application of optical third-order nonlinear materials in all-optical switches. In recent years, two-dimensional (2D) materials, represented by graphene, have been especially investigated in the field of new photonic devices (including photodetectors, optical modulators, optical absorbers, fiber lasers, etc.). Through numerous photonic studies, these materials have gradually developed into emerging candidate materials with the ability to move the current research from the nanometer scale to the two-dimensional scale. Two-dimensional materials can be tuned from insulator to metal based on their layer thickness band gap, which can be used in the application of broadband photonic devices. Two-dimensional materials similar to graphene, such as topological insulators (Bi₂Se₃, Bi₂Te₃, and Sb₂Te₃) and transition metal bihalogenated hydrocarbons (MoS₂, WS₂, WSe₂, etc.), are widely studied.

5.1. Graphene

Graphene plasma provides a distinct and outstanding platform for nonlinear all-optical switching due to its high nonlinear electrical conductivity and tight optical constraints. Graphene is widely applied in optical devices, such as fiber lasers, broadband modulators, polarizers, and optical switching, etc., because of its high optical nonlinearity and widewavelength-range non-dispersive nonlinear optical response.

Melinda et al. first demonstrated lithographically patterned graphene ribbon structures in 2007 and found that the energy gap was closely related to the ribbon width [59]. However, in order to facilitate the peak performance of classical nonlinear devices, based on the strong deterministic interaction between single photons in single-photon nonlinear optics, Gulans et al. proved that this field confinement causes the deterministic interaction between two plasma polaritons (i.e., photons) on the picosecond time scale, as shown in Figure 2a [72]. This also shows that the single photon switch and the nonclassical light can be realized through this interaction. Without optimized phase matching, Gullans achieved a 17 nm bandwidth and -23 dB conversion efficiency. At the same time, M. Takahashi et al. proposed an optical switch controlled by a saturable absorption of graphene that was vertically inserted or loaded on the waveguide, as shown in Figure 2b [73]. The saturated absorption of graphene-inserted fiber and graphene-loaded waveguides show that the measured absorption modulation depth can meet the switching requirements experimentally. The optical saturable absorption of single-layer graphene that was ver-

Coatings 2023, 13, 876 10 of 25

tically inserted between optical fibers and covered on ion-exchanged waveguides was measured for picosecond switching applications. In the proposed switching, the complete switching was realized by controlling the amplitude of signal light through the structure of two cascaded Mach–Zehnder interferometers. In order to control the amplitude of signal light, M. Takahashi et al. used saturated absorption in graphene. When switching, the modulation depth of the amplitude is required to be 2.414, and the corresponding intensity is 7.65 dB. In the graphene-loaded waveguide with a length of 7 mm, the modulation absorption depth of more than 10 dB was obtained experimentally. To achieve the switching and modulation of optical interactions with deeply sub-wavelength films, Shraddha et al. reported experimental observations of the all-optical modulation of light in a graphene film [74]. However, conventionally synthesized graphene relies on high-temperature and vacuum devices, which can present detrimental transfer steps that degrade the quality of graphene, thereby affecting the efficiency of nonlinear optical operations and lacking a customized pattern with a minimal footprint, as well as lacking a convenient fabrication process. Kelvin J. A. Ooi et al. demonstrated the laser-assisted direct in situ synthesis of multilayer graphene on a side-polished fiber plane under ambient conditions. Four-wave mixing (FWM) with grown graphene is used to elucidate ultrafast all-optical switching near 1550 nm to confirm nonlinear response improvements. It was demonstrated that impressive switching performances can be obtained at sub-MW/cm² optical pump intensities with phase extinction modulations of graphene plasma waveguides [75]. In addition, the large surface-induced nonlinear enhancement caused by the tight confinement effect can potentially make the propagating plasma pump power drop to the pW range.

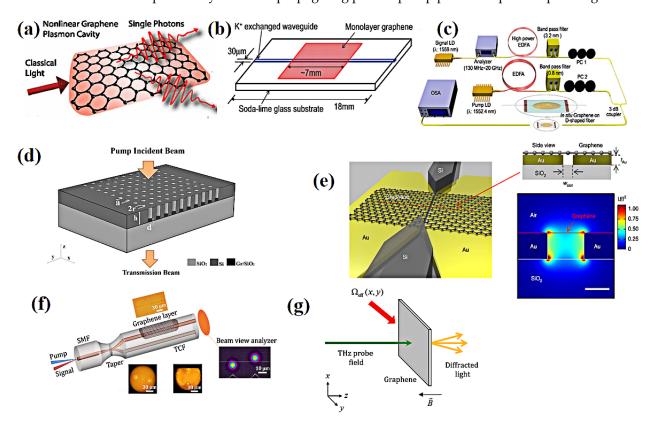


Figure 2. (a) A doped graphene disk confines photons [72]. (b) Waveguide structure with an overlaid graphene sheet [73]. (c) Ultrafast all-optical switching based on FWM mixing in situ synthesized graphene [76]. (d) Nonlinear transmission variation in all-optical switching for Fano resonance in two-dimensional photonic crystals [77]. (e) Schematic diagram of an all-optical control of graphene-supported metal-insulator-metal WG and the electric field distribution of characteristic modes $|E|^2$ [78]. (f) The schematic diagram of the side-polished two-core fiber Michelson interferometer [79]. (g) The schematic diagram of laser-induced graphene-based diffraction grating [80].

Coatings **2023**, 13, 876 11 of 25

Further, Debnath et al. used laser irradiation in ambient conditions to synthesize multilayer graphene directly along the polished surface of a D-shaped fiber, as shown in Figure 2c [76]. They conducted ultrafast all-optical switching experiments on grown graphene at nearly 1550 nm by using the four-wave mixing (FWM) technique. They confirmed that 58.5% of the improvement in the nonlinear response came from graphene. Mehrdad Hoseini et al. achieved all-optical switching using nonlinear transport variations of the Fano resonance of a two-dimensional photonic crystal (PhC), as shown in Figure 2d. The light intensity enhancement due to Fano resonance and focus was added together to provide greater refractive index variation at the prescribed pump intensity. With a modulation depth of 90%, this reduces the threshold input intensity of the pump for conversion to ~MW/cm². In addition, they investigated the transient state of the structure, and an optical bistable-based all-optical switch was proposed to achieve a bistable switch with a modulation depth of 50% and a switching time of about 180 fs [77]. Xiang et al. designed the light path of an all-optical switch using spatial cross-phase modulation. The results showed that the control beam can modulate the phase change in the signal light (532 nm) to achieve the on/off function [81]. This predicts that the SXPM-based ultra-sensitive switching device may provide new ideas for all-optical switching. In 2020, Masaaki et al. proposed an ultrafast all-optical switch using graphene-loaded deep-subwavelength plasmonic waveguides ($30 \times 20 \text{ nm}^2$) with a switching energy of 35 fJ and a switching time of 260 fs, as shown in Figure 2e [78]. In the same year, Chu et al. investigated an all-optical phase shifter and switching for a Michelson interferometer (MI) graphene-decorated sidepolished twin-core fiber (TCF), as shown in Figure 2f. An optical switching extinction ratio of 7 dB was obtained with a rise (fall) time of 55.8 ms (15.5 ms) [79]. In 2022, Popkova et al. used Bloch surface waves (BSWs) to enhance graphene-light interactions in order to achieve all-optical switching, whereby the magnitude of the reflection change reaches 0.3% at the fluence of 40 μ J cm⁻² [82]. In 2023, Taghizadeh et al. investigated the effect of terahertz fields on the optical response of Landau quantized graphene monolayers and discussed the possibility of achieving ultrafast all-optical switching using terahertz field strength in the near infrared, as shown in Figure 2g [80].

In recent years, all-optical switches based on graphene and Mach–Zehnder interferometers have been widely studied. In 2019, Hao et al. proposed and experimentally demonstrated an all-optical switch based on a graphene-coated fiber Mach–Zehnder interferometer (MZI) (as shown in Figure 3a) with rise and fall times of 30 ms and 50 ms, respectively [83]. In the same year, Armaghani et al. also proposed an all-optical graphene switch based on the MZI, as shown in Figure 3b, where the switch operation can be accomplished by generating a 180-ohm phase difference between the two arms of the graphene MZI [84]. In 2021, Qiu et al. proposed and experimentally demonstrated a graphene-on-silicon nitride (Si₃N₄) all-optical switching based on an MZI (as shown in Figure 3c), with rise and fall time constants of 571 ns and 1.29 μ s, respectively [85]. Recently, Jiang et al. proposed a graphene-buried balanced MZI structure, as shown in Figure 3d, for a low-power all-optical switch with an actual pump power absorbed by the graphene-activated switch of around 2.2 mW; the rise and fall times of the switch were 1.0 and 2.7 ms, respectively [86]. Table 3 summarizes the rise time, fall time, and switching energy of graphene based all-optical switching with different structures.

Table 3. Characteristic parameters of several all optical switching based on graphene.

Structure	Rise Time	Fall Time	Switching Energy	Reference
graphene-loaded deep-subwavelength plasmonic waveguides	260 fs	260 fs	35 fJ	[76]
graphene decorated side-polished twin-core fiber Michelson interferometer	55.8 ms	15.5 ms	/	[77]

Coatings 2023, 13, 876 12 of 25

TET 1	1 1		•		
13	n	Δ	- 4	. Co	u+

Structure	Rise Time	Fall Time	Switching Energy	Reference
graphene-coated fiber Mach-Zehnder interferometer	30 ms	50 ms	4.8 mW	[83]
graphene-on-silicon nitride (Si ₃ N ₄) all-optical switch based on Mach–Zehnder interferometer	571 ns	1.29 μs	/	[85]
graphene-buried balanced Mach-Zehnder interferometer	1.0 ms	2.7 ms	2.2 mW	[86]

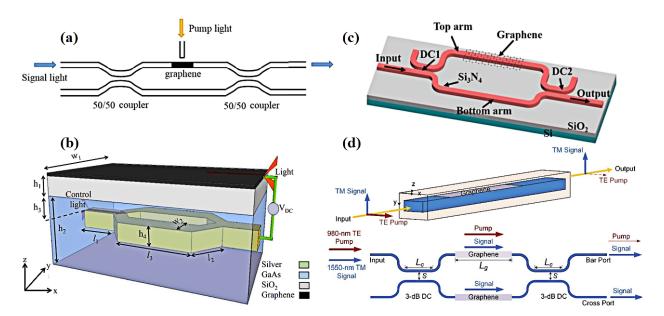


Figure 3. (a) Graphene-based fiber Mach–Zehnder all-optical switches [83]. (b) All-optical switches based on graphene and insulated metal plasma waveguides [84]. (c) Graphene-on-silicon nitride all-optical switching [85]. (d) Graphene-buried balanced Mach–Zehnder interferometer [86].

5.2. Black Phosphorus

Black Phosphorus (BP) has anisotropic nonlinear optical responses and varies with film thickness, which is quite different from other typical two-dimensional layered materials (such as graphene and transition metal dihalides). The most promising feature, thus far, of BP may be that it fills a gap in the 2D family [87,88]. Firstly, the band gap of BP is layer-dependent and covers 0.3 to 2 eV [89]. Due to its relatively strong out-ofplane quantum confinement, its bandgap-tuning range is much larger than that of other two-dimensional materials. This tunable bandgap property offers multiple advantages, with the large tuning range leading to the flexibility required in electronic or photonic applications. Next, BP transistors offer a compromise between carrier mobility and the switching ratio. Mobility and the switching ratio are the simplest metrics to characterize transistor performance. BP is now attracting a great deal of research interest. Normally, BP is a layered orthorhombic crystal with phosphorus atoms arranged in a puckered honeycomb lattice, where in a separate layer each phosphorus atom is bound to three neighboring atoms via sp³-hybridized orbitals, which is unlike the sp²-hybridized carbon atoms in graphene. In the base plane, the folding of the lattice leads to two unequal directions: the x-direction (i.e., the armchair edge) and the y-direction (i.e., the zig-zag edge).

In 2015, Li et al. generated ultrafast (i.e., a pulse duration down to 786 fs at mode locking) and large energy (Q-switched pulses up to >18 nj) pulses in a fiber laser in the \sim 1.5 μ m NIR communication band based on the nonlinear optical properties of BP [90]. Similarly, the Sotor research group successfully obtained an ultrashort pulse with a bandwidth of 10.2 nm and a duration of 272 fs in the 1550 nm communication band using a BP-based SA [91]. All of these properties make it possible to achieve nonlinear and ultrafast all-optical applications, based on BP.

Coatings 2023, 13, 876 13 of 25

In 2016, Markus et al. designed a SiO₂/black phosphorus/SiO₂ heterostructure, as shown in Figure 4a, to achieve ultra-fast measures of BP at the nanoscale and to achieve switchable interface polaritons [92]. Where the surface phonon mode of the SiO₂ layer hybridizes with the mode of the BP, it can then be activated by photo-induced interband excitation. In the Reststrahlen band of SiO₂, the hybrid interface polariton has surface-phonon-like properties. In addition, it also possesses a well-defined frequency and momentum, as well as an excellent coherence. The results show that the surface mode activates in 50 fs and disappears in 5 ps with the recombination of electron-hole pairs in BP. Debnath's group not only performed ultrafast all-optical switching experiments on grown graphene at nearly 1550 nm using the four-wave mixing (FWM) technique, as shown in Figure 4b, but also exploited the evanescent field interaction between propagating light and BP to achieve FWM [93]. They performed FWM-based wavelength conversion experiments on modulated signals in the frequency range up to 20 GHz by using a BP-based nonlinear optical device consisting of a side polished fiber or a D-shape that was deposited with BP.

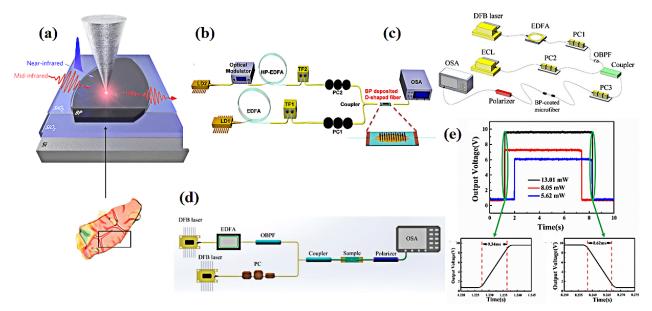


Figure 4. (a) The SiO₂/BP/SiO₂ heterostructure [92]. (b) A schematic diagram of the experimental apparatus for FWM observation in a BP-deposited apparatus [93]. (c) A Kerr switch based on black phosphorus [94]. (d) The Kerr switcher in a BPQD-deposited nonlinear fiber optical device [95]. (e) The response time of black-phosphorus-functionalized microfiber coil resonators with different pump powers [96].

The refractive index of black phosphorus is affected by high-power laser light, which is also known as the optical Kerr effect. This phenomenon enables all-optical exchange in all-optical network components. To evaluate the applicability of metal-ion-doped layer-less BP in all-optical signal processing under a nonlinear Kerr effect, a deposited sample placed onto a microfiber with a swift field interaction scheme of transmitted light and BP was used by Zheng et al. [94]. The basic operating principle and experimental setup of the Kerr switch are shown in Figure 4c. This BP-clad microfiber device can be used both as an optical Kerr switch with an extinction ratio of up to 26 dB and an operation range of more than 18 nm and an FWM-based wavelength converter with a maximum conversion efficiency of −59.15 dB. The results show that, thanks to the high nonlinear optical index and protection of several layers of BP that were doped with metal ions, the new nonlinear optical device can be used as a broadband all-optical processing device with high stability and durability, which may constitute the first prototype of BP-based optical Kerr exchange devices. In 2018, Wang et al. demonstrated, for the first time, experimentally that the sub-band structure of ultrathin black phosphorus (BP) nanosheets is caused by strong many-body effects and unique quasi-one-dimensional band dispersions. The photoexcitation processes of two different

Coatings **2023**, 13, 876 14 of 25

sets of BP nanosheets were demonstrated through careful analysis of photoluminescence spectra and femtosecond time-resolved transient absorption spectra. The BP nanosheets based on the energy band structure exhibit a significant optical switching effect during the generation of photocatalytic reactive oxygen species (ROS) [97]. Wang et al. demonstrated an optical Kerr switch with an extinction ratio of 20 dB with BP quantum dots deposited on the microfiber as shown in Figure 4d. The conversion efficiency of this wavelength converter in the RF band is about -40 dB [95]. In 2021, Yin et al. performed all-optical modulation experiments in a cavity of black-phosphorus-functionalized microfiber coil resonators (MCR). The resonance wavelength and extinction ratio responses to the incident pump power were ~ 0.033 nm mW $^{-1}$ and ~ 0.180 dBm mW $^{-1}$, respectively, with an average rise/fall time of $\sim 9.58/8.53$ ms. The device can be applied to all-optical modulators, optical switches, or tunable optical filters [96].

Table 4 summarizes the rise time, fall time, and extinction ratio of BP based all-optical switching with different structures. In summary, BP is still in its beginning, although it has recently attracted much research attention and has seen great progress. From a practical point of view, fundamental and technical issues are raised. These issues pose great challenges, as well as chances, for upcoming research in BP-based optics and photonics.

Structure	Rise Time	Fall Time	Extinction Ratio	Reference
SiO ₂ /BP/SiO ₂ heterostructure	50 fs	5 ps	/	[92]
BP-clad microfiber device	/	/	26 dB	[94]
BP quantum dots deposited on the microfiber	/	/	20 dB	[96]
BP functionalized microfibre coil resonator	9.58 ms	8.53 ms	$\sim 0.180 \; \mathrm{dBm} \; \mathrm{mW}^{-1}$	[97]

Table 4. Characteristic parameters of several all-optical switching based on BP.

5.3. Transition Metal Dichalcogenide (TMDs)

Two-dimensional transition metal dichalcogenides (TMDs), which are denoted by the formula MX₂ (where M denotes a transition metal element and X denotes a chalcogen, such as S, Se, or Te), are considered highly anticipated candidates for the next generation of electronics and photonics, with three advantages [98–107]: firstly, the weak van der Waals interactions and the lack of surface dangling bonds make the construction of heterostructures easy, thus eliminating the strict thermal and lattice matching requirements [107,108]. Secondly, the coupling of broken inversion symmetry and spin orbit in certain single-layer TMDs provides us with a new platform for studying valley electronics and fabricating spintronic devices [64,107,109–111]. Thirdly, certain TMDs (e.g., 2H MoS₂, 2H WSe₂) exhibit indirect–direct bandgaps as the number of layers is reduced to a single layer, which can stimulate various optoelectronic applications, such as photodetectors, photovoltaic devices, and light-emitting devices [112–116].

Unlike graphene, certain members of the TMD family are true semiconductors with a significant band gap. In particular, molybdenum and tungsten-based dihalides exhibit optical forbidden bands in the 1–2 eV range and are suitable for near-infrared absorption and emissions. Although these materials, interestingly, have an indirect band gap in the bulk and layer-less form, they have become direct gap semiconductors with strong photoluminescence at the monolayer level [112,117]. TMDs have been shown to have direct energy gaps in the near infrared to the visible spectral region, and thereby, a new window has been opened for photonics and optoelectronics. TMDs normally show resonant absorption in visible light, enabling pulsed fiber lasers in the visible region. Symmetry breaking in the TMDs leads to finite second-order optical nonlinearities. The d-electrons in layered TMDs can give rise to a variety of physical properties. Furthermore, the photoluminescence increases with decreasing coating thickness. When using two different laser power densities, a broad-range switch between the exciton and trion of the monolayer WS₂ can be reversibly achieved, thus resulting in a continuous modulation of its photon absorption [118]. Seyler et al. reported a mechanism for electrically controlling second-order optical nonlinearities in a monolayer WSe2, thereby demonstrating that monolayer TMD

Coatings **2023**, 13, 876 15 of 25

field-effect transistors represent a new class of an electrically tunable nonlinear optical device [119]. Wu et al. demonstrated a fiber optic all-optical phase shifter, as shown in Figure 5c. It uses a layer-less two-dimensional material, such as tungsten disulfide (WS₂), which is deposited on a tapered fiber. The WS₂ absorbs the injected 980 nm pump light (control light) and generates heat. A maximum phase shift of 6.1π is achieved near 1550 nm due to the thermo-optical effect, which changes the refractive index of the WS₂ and the tapered fiber. The device has a loss of 3.7 dB. In addition, it also has an all-optical switch with an extinction ratio of 15 dB and a rise time of 7.3 ms, which was obtained by constructing a single-arm MZI with a WS₂ phase shifter [120]. This all-fiber, low-cost, and compact optical phase shifter and switch demonstrates the potential of two-dimensional transition metal dihalide compounds for all-optical signal processing devices. In addition, the nonlinear refractive index and third-order nonlinear magnetization of SnS were characterized, for the first time, by Wu et al. In addition, its powder-related nonlinear refractive index was also experimentally verified. On this basis, a new type of nonlinear all-optical switch was proposed as shown in Figure 5a. The results show that the phase change in the signal light (633 nm) can be controlled by another control light (532 nm), thus realizing the on/off mode of the all-optical switch [121]. In 2022, Xu et al. experimentally designed MoTe2 quantum dots-based all-optical switching as shown in Figure 5g in order to manipulate light ($\lambda = 457$ nm) against signal light ($\lambda = 532$ nm) using the spatial cross-phase modulation (SXPM) method [122].

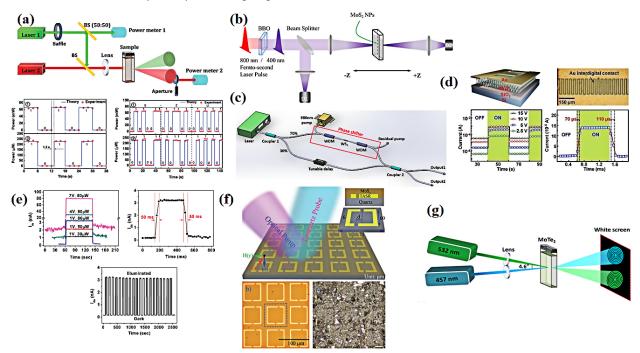


Figure 5. (a) Schematic of SXPM in a few-layer SnS [121]. (b) Schematic diagram of the Z—scan experimental setup [123]. (c) Experimental setup of the all-optical phase shifter and switching, based on a WS₂-tapered fiber [120]. (d) Schematic of the optical microscopic top-view image and switching characteristic of the few-layer MoS₂ MSM PDs [124]. (e) Optical switching characteristics and stability of monolayer MoS₂ phototransistors (in room temperature) [125]. (f) MoS₂ drop casted on a terahertz asymmetric split ring resonator structure [126]. (g) MoTe₂ quantum dots, based all-optical switching [122].

The most studied material in TMDs, MoS_2 , is a novel two-dimensional nanomaterial with broadband high optical gain and fast optical switching. As such, it is promising to be the next generation of optical devices. Yin et al. made the first phototransistor based on a mechanically exfoliated monolayer MoS_2 , as shown in Figure 5e. Furthermore, the observed switching duration of the current rise (from OFF to ON) or the decay (from ON

Coatings **2023**, 13, 876 16 of 25

to OFF) process was only around 50 ms [125]. Tsai et al. reported, for the first time, on a new-layered MoS₂ as Schottkymetal-semiconductor-metal photodetectors (MSM PDs) that are capable of broadband photodetection from visible to UV wavelengths at operating temperatures, in harsh conditions, of up to 200 °C. At a 10 V bias, photogain up to 13.3 leads to ultrahigh responsivity and detectivity for MoS₂ up to 0.57 A/W and ~1010 cm Hz1/2/W, which was previously not obtainable in pristine graphene and MoS2-based PDs. Moreover, MoS₂ MSM PDs show a very fast and stable optical response with a rise time of around 70 μs and a fall time of approximately 110 μs, as shown in Figure 5d. Although the thickness of several layers of MoS₂ is only 1.9 nm, their light absorption in the visible region is as high as 10% [124]. In 2014, Zhang et al. investigated the nonlinear optical properties of MoS₂ nanosheets from the visible to the near-infrared band using the hydrothermal exfoliation method, as shown in Figure 5b. The laser experiments showed that the mode-locker, based on MoS₂ with few layers, can generate stable mode-locked laser pulses with a central wavelength of 1054.3 nm, as well as a 3 dB spectral bandwidth of 2.7 nm and a pulse duration of 800 ps [123]. In 2017, Yogesh et al. demonstrated that sharp Fano resonances in molybdenum disulfide-coated metamaterials, consisting of asymmetric split ring resonator arrays, have ultrasensitive active switching and modulation, thus allowing for ultrafast switching of Fano resonances on a 100 ps time scale, as shown in Figure 5f [126]. Table 5 summarizes the rise time, fall time, and extinction ratio of TMDs based on all-optical switching with different structures.

Table 5. Characteristic parameters of several all-optical switches based on TMDs.

Structure	Rise Time	Fall Time	Extinction Ratio	Reference
WS ₂ deposited tapered fiber	7.3 ms	/	15 dB	[120]
Few-layered MoS ₂ as Schottky metal semiconductor metal photodetectors	70 μs	110 μs	/	[124]
phototransistor based on mechanically exfoliated monolayer MoS ₂	50 ms	50 ms	/	[125]

5.4. Layered Material Heterostructures

Two-dimensional materials have attracted much attention from researchers due to their unique optical band gap structure, extremely strong light-matter interactions, and large specific surface areas. Graphene, hexagonal boron nitride (h-BN), and certain other transition metal dihalides (TMDs) have emerged as promising new materials. They are composed of strong covalent bonds transversely, which provide great in-plane stability. The recent development of vapor-phase growth techniques for two-dimensional materials has further paved the way for directly synthesizing vertical and lateral heterojunctions. Each layer in a two-dimensional layered material consists of a covalently bonded, danglingbond-free lattice and is weakly bonded to neighboring layers via van der Waals interactions. This makes it possible to separate, mix, and match highly different atomic layers in order to create a wide range of van der Waals heterostructures without the limitations of lattice matching and processing compatibility [110]. Integrating two-dimensional materials with a variety of nanoscale materials can create a wide range of van der Waals heterogeneous structures with previously impossible functions. The exfoliation and restacking of various 2D materials opens up a flexible path for the design and creation of a new generation of optical switches. As a result, the heterojunctions of 2D-layered materials have become a hot new topic, opening up new areas of material science and device applications [127–129]. For example, it has been shown that laminates of graphene and hexagonal boron nitride (hBN), molybdenum disulfide (MoS₂), or tungsten disulfide (WS2) can run tunneling transistors [130,131]. This is in addition to phenomena, such as Coulomb drag resistance [132] and the fractional quantum Hall effect [133], which can also be observed.

Raman scattering photo luminescence and absorption spectra can be used to characterize interlayer interactions. In addition, exciton dynamics can be studied using transient absorption spectra. This allows electronic and optical properties to be elucidated

Coatings 2023, 13, 876 17 of 25

non-destructively, accurately, and efficiently, thereby optimizing the fabrication of these stacked junctions and improving the design of new structures [134–143]. Ferrari et al. first demonstrated that the profile of the 2D Raman peak evolves with an increasing number of graphene layers. Lui et al. found a significant difference in the Raman 2D peak shape between ABA and ABC stacked tri- and quad-layer graphene [144]. Cong and Zhang et al. also found that the M-band profile at \sim 1750 cm $^{-1}$ and the peak position in the G-band were very sensitive to the stacking order of the few layers of graphene [145,146].

Isolated atomically layered two-dimensional materials can be combined into connections with precisely defined configurations, such as lateral junctions and vertical junctions [147–149]. Early studies have shown that van der Waals stacks of 2D materials work especially well and greatly enrich the optoelectronic applications of 2D materials.

The weak van der Waals inter-layer force is able to isolate two-dimensional monolayers and restack them into arbitrary stacking heterojunctions without regard to atomic commensurability. Two-dimensional heterostructures are van der Waals-bonded heterostructures, which offer greater flexibility than conventional covalently bonded heterostructures. Without the constraint of atomically precise commensurability, it is possible to create high-quality semiconductor heterointerfaces. Two-dimensional heterostructures that have been studied include graphene-BP, TMD-TMD, TMD-graphene, and TMD-hBN combinations [127,128,150,151]. Heterogeneous structures involving TMD have strong light-matter interactions and unique optical properties that are essential for all-optical modulation applications [152]. It has been demonstrated that graphene-semiconducting TMD-graphene heterostructures exhibit strong photo-matter interactions, which can lead to large photon absorption and photocurrent production [153]. Another experiment shows that hole transfer from the MoS₂ layer to the WS₂ layer is very fast, within 50 fs of photoexcitation. This opens up the possibility of new types of two-dimensional light trapping devices [140]. Optoelectronic devices based on an MoS₂/WSe₂ vertical heterojunction have also recently been demonstrated, as shown in Figure 6b [154]. In 2016, large nonlinear refraction ($n_2 = 0.2 \times 10^{-7} \text{ cm}^2/\text{W}$) and ultrafast dynamic relaxation times (intraband relaxation time τ_1 = 270 \pm 20 fs; inter-band relaxation time τ_2 = 3.6 \pm 0.2 ps) of the graphene–Bi₂Te₃ heterostructure were reported [155]. Recently, an anisotropic infrared plasmonic broadband absorber based on graphene-BP multilayers was proposed by Cai et al.

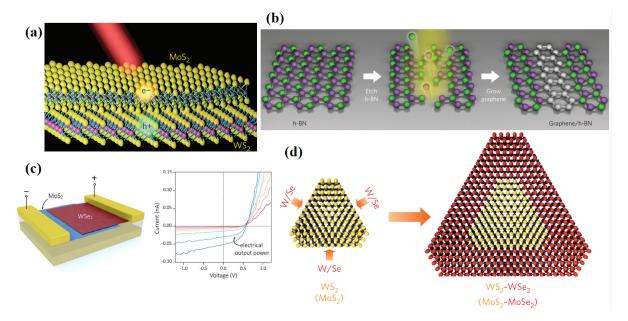


Figure 6. (a) Schematic diagram of the MoS₂/WS₂ heterostructure [139]. (b) MoS₂/WSe₂ hetero—diode and current—voltage characteristics of the MoS₂/WSe₂ vertical heterojunction under light [154]. (c) Epitaxial growth lateral graphene—hBN heterostructures [156]. (d) Schematic diagram of the transverse epitaxial growth of WS₂–WSe₂ and MoS₂–MoSe₂ heterostructures [71].

Coatings 2023, 13, 876 18 of 25

The vertical junctions of two-dimensional atomic-layered materials can be achieved via simple mechanical transfer without lattice mismatch [157]. However, lateral heterojunctions cannot be achieved through mechanical transfer, and lateral epitaxial growth is the best method for preparing atomic-layered lateral heterojunctions. Unlike vertical Van-der-Waals-stacked heterojunctions, lattice mismatch is a key issue for seamless transverse 2D junctions. In addition, lattice mismatches between MoS₂ and WS₂, MoSe₂ and WSe2, MoS2 and MoSe2, WS2 and WSe2, and graphene and hBN are relatively small (<4%) [156,158–163]. The Van der Waals heterostructures can be achieved via transfer or direct epitaxial growth by using these two-dimensional layered materials. Xu et al. systematically investigated photocurrents in monolayer-bilayer graphene homojunctions [164]. Howell and colleagues studied photocurrent generation in both monolayer and multilayer MoS₂ film interfaces [165]. Graphene–boron nitride(BN)-based two-dimensional lateral heterostructures have been reported [156,165–167]. It has also been successfully demonstrated that the lateral heteroepitaxial junctions of two-dimensional layered semiconductors can be grown using the lateral heteroepitaxial approach. Graphene and hBN have a tight lattice match (1.7%); therefore, a near-perfect transverse graphene hBN junction can be achieved. In addition, Levendorf and colleagues reported the versatile, scalable regeneration of hBN monolayers, which can be performed from the edges of pre-patterned graphene monolayers [162]. Liu et al. proposed that growing graphene in a flat hBN atomic layer can form a transverse graphene-hBN heterostructure, as shown in Figure 5c [156]. Duan et al. prepared the transverse heterostructures of MoS₂, MoSe₂ and WS₂, and WSe₂ through the in situ modulation of the vapor-phase reactants. Moreover, Raman and photoluminescence mapping studies showed that the resulting heterostructured domains exhibit clear structural and optical modulations [71].

6. Summary and Outlook

In recent years, profound achievements have been made in the field of all-optical switches in terms of nonlinear materials, device structures, triggering methods, advanced performance indicators, and chip integration compatibility. Materials with excellent third-order nonlinear optical response properties are an important basis for improving the performance index of all-optical switches. Two-dimensional materials are represented by graphene, black phosphorus, and TMDs. The emergence of new materials with good third-order optical nonlinearity provides an important impetus for the improvement of all-optical switches. This not only provides a way to build ultrafast nonlinear media, overcoming the inherent material limitations that constrain the development of integrated photonic devices, but also provides a lab-on-chip platform for the study of novel physical effects.

The integration of 2D materials into photonic platforms is currently limited. Although 2D-material all-optical switches are not yet well established, they may be better suited for on-chip integration by using simple, inexpensive, and scalable post-processing techniques. Within the rich family of 2D materials, there are many more candidates to explore. In addition, the bottlenecks with respect to their use for large-scale applications may soon be replaced by recent breakthroughs in wafer-scale synthesis methods and manufacturing processes [168,169]. In the last few years, certain breakthroughs in switching times and energy consumption have been made in integrated optical switches, aided by, for example, two-dimensional materials. However, it is still hard to further reduce the energy consumption to the attojoule-level, which is necessary for large-scale photonic-integrated circuits in the future. This requires a detailed, systematic, and in-depth exploration of the mechanisms that enhance light-matter interactions, i.e., the mechanisms and methods that control the interaction of multiple physical fields (i.e., light, heat, and electric fields) within the medium. The ultra-fast switching of attojoule-level energy consumption can be achieved based on emerging nanomaterials, as well as the properties of plasma, nanophotonics, and hybrid integration. More efforts must be made in this area to further improve performance. Coatings **2023**, 13, 876 19 of 25

Author Contributions: Conceptualization, formal analysis, and writing—original draft, J.X., Y.P., L.J. and S.Q.; writing—original draft preparation, J.X. and Y.P.; data curation and funding acquisition, Y.P., L.J. and S.Q.; project administration and supervision, L.J. and S.Q.; data analysis, writing—review, and editing, J.X., Y.P., L.J. and S.Q. All authors have read and agreed to the published version of the manuscript.

Funding: This work was supported by the National Natural Science Foundation of China (Grant Nos.11774088, 11704119), Scientific Research Fund of Hunan Provincial Education Department (Grant No. 21B0048), the Hunan Provincial Natural Science Foundation of China (Grant Nos. 2021JJ30149), and the Changsha Natural Science Foundation (Grant No. kq2202236).

Data Availability Statement: The datasets used and/or analyzed during the current study are available from the corresponding author upon reasonable request.

Conflicts of Interest: The authors declare no conflict of interest.

References

- 1. Chai, Z.; Hu, X.; Wang, F.; Niu, X.; Xie, J.; Gong, Q. Ultrafast All-Optical Switching. Adv. Opt. Mater. 2017, 5, 1600665. [CrossRef]
- 2. Nozaki, K.; Tanabe, T.; Shinya, A.; Matsuo, S.; Sato, T.; Taniyama, H.; Notomi, M. Sub-femtojoule all-optical switching using a photonic-crystal nanocavity. *Nat. Photonics* **2010**, *4*, 477–483. [CrossRef]
- 3. Altug, H.; Englund, D.; Vučković, J. Ultrafast photonic crystal nanocavity laser. Nat. Phys. 2006, 2, 484–488. [CrossRef]
- 4. Venkatesan, T.; Lemaire, P.J.; Wilkens, B.; Soto, L.; Gossard, A.C.; Wiegmann, W.; Jewell, J.L.; Gibbs, H.M.; Tarng, S.S. All-optical data switching in an optical-fiber link using a GaAs optical bistable device. *Opt. Lett.* **1984**, *9*, 297–299. [CrossRef] [PubMed]
- 5. Van, V.; Ibrahim, T.; Ritter, K.; Absil, P.; Johnson, F.; Grover, R.; Goldhar, J.; Ho, P.-T. All-optical nonlinear switching in GaAs-AlGaAs microring resonators. *IEEE Photonics Tech. L.* **2002**, *14*, 74–76. [CrossRef]
- 6. Almeida, V.R.; Barrios, C.A.; Panepucci, R.R.; Lipson, M. All-optical control of light on a silicon chip. *Nature* **2004**, 431, 1081–1084. [CrossRef]
- 7. Koos, C.; Vorreau, P.; Vallaitis, T.; Dumon, P.; Bogaerts, W.; Baets, R.; Esembeson, B.; Biaggio, I.; Michinobu, T.; Diederich, F. All-optical high-speed signal processing with silicon–organic hybrid slot waveguides. *Nat. Photonics* **2009**, *3*, 216. [CrossRef]
- 8. Nie, W. Optical Nonlinearity:Phenomena, applications, and materials. Adv. Mater. 1993, 5, 520–545. [CrossRef]
- 9. Leuthold, J.; Koos, C.; Freude, W. Nonlinear silicon photonics. *Nat. Photonics* **2010**, *4*, 535–544. [CrossRef]
- 10. Dobbelaere, P.D.; Falta, K.; Gloeckner, S.; Patra, S. Digital MEMS for optical switching. *IEEE Commun. Mag.* **2002**, 40, 88–95. [CrossRef]
- 11. Ikeda, K.; Saperstein, R.E.; Alic, N.; Fainman, Y. Thermal and Kerr nonlinear properties of plasma-deposited silicon nitride/silicon dioxide waveguides. *Opt. Express* **2008**, *16*, 12987–12994. [CrossRef] [PubMed]
- 12. Lu, H.; Liu, X.; Wang, L.; Gong, Y.; Mao, D. Ultrafast all-optical switching in nanoplasmonic waveguide with Kerr nonlinear resonator. *Opt. Express* **2011**, *19*, 2910–2915. [CrossRef] [PubMed]
- 13. Waldow, M.; Plötzing, T.; Gottheil, M.; Först, M.; Bolten, J.; Wahlbrink, T.; Kurz, H. 25ps all-optical switching in oxygen implanted silicon-on-insulator microring resonator. *Opt. Express* **2008**, *16*, 7693–7702. [CrossRef] [PubMed]
- 14. Li, N.; Xu, J.; Song, G.; Zhu, C.; Xie, S.; Yang, Y.; Zubairy, M.S.; Zhu, S.-Y. Nonlinear all-optical switch based on a white-light cavity. *Phys. Rev. A* **2016**, *93*, 043819. [CrossRef]
- 15. Haché, A.; Bourgeois, M. Ultrafast all-optical switching in a silicon-based photonic crystal. *Appl. Phys. Lett.* **2000**, *77*, 4089–4091. [CrossRef]
- 16. Almeida, V.R.; Lipson, M. Optical bistability on a silicon chip. Opt. Lett. 2004, 29, 2387–2389. [CrossRef]
- 17. Li, S.; Cai, X. High-contrast all optical bistable switching in coupled nonlinear photonic crystal microcavities. *Appl. Phys. Lett.* **2010**, *96*, 131114. [CrossRef]
- 18. Scalora, M.; Dowling, J.P.; Bowden, C.M.; Bloemer, M.J. Optical Limiting and Switching of Ultrashort Pulses in Nonlinear Photonic Band Gap Materials. *Phys. Rev. Lett.* **1994**, 73, 1368–1371. [CrossRef]
- 19. Kim, C.; May-Arrioja, D.A.; LiKamWa, P.; Newman, P.; Pamulapati, J. Ultrafast all-optical multiple quantum well integrated optic switch. *Electron. Lett.* **2000**, *36*, 1929–1930. [CrossRef]
- 20. Delgado-Pinar, M.; Zalvidea, D.; Díez, A.; Pérez-Millán, P.; Andrés, M.V. Q-switching of an all-fiber laser by acousto-optic modulation of a fiber Bragg grating. *Opt. Express* **2006**, *14*, 1106–1112. [CrossRef]
- 21. Ren, M.; Jia, B.; Ou, J.-Y.; Plum, E.; Zhang, J.; MacDonald, K.F.; Nikolaenko, A.E.; Xu, J.; Gu, M.; Zheludev, N.I. Nanostructured Plasmonic Medium for Terahertz Bandwidth All-Optical Switching. *Adv. Mater.* **2011**, *23*, 5540–5544. [CrossRef] [PubMed]
- 22. Tran, P. Optical limiting and switching of short pulses by use of a nonlinear photonic bandgap structure with a defect. *J. Opt. Soc. Am. B* **1997**, *14*, 2589–2595. [CrossRef]
- 23. Fan, S. Sharp asymmetric line shapes in side-coupled waveguide-cavity systems. Appl. Phys. Lett. 2002, 80, 908–910. [CrossRef]
- 24. Yanik, M.F.; Fan, S.; Soljačić, M. High-contrast all-optical bistable switching in photonic crystal microcavities. *Appl. Phys. Lett.* **2003**, *83*, 2739–2741. [CrossRef]

Coatings **2023**, 13, 876 20 of 25

25. Wada, O. Femtosecond all-optical devices for ultrafast communication and signal processing. New J. Phys. 2004, 6, 183. [CrossRef]

- 26. Pasquazi, A.; Stivala, S.; Assanto, G.; Amendola, V.; Meneghetti, M.; Cucini, M.; Comoretto, D. In situ tuning of a photonic band gap with laser pulses. *Appl. Phys. Lett.* **2008**, *93*, 091111. [CrossRef]
- 27. Dani, K.M.; Ku, Z.; Upadhya, P.C.; Prasankumar, R.P.; Brueck, S.R.J.; Taylor, A.J. Subpicosecond Optical Switching with a Negative Index Metamaterial. *Nano Lett.* **2009**, *9*, 3565–3569. [CrossRef]
- 28. Boyd, R. Nonlinear Optics; Academic Pres Inc.: Boston, MA, USA, 1992.
- 29. Slusher, R.E. Nonlinear Photonic Crystals; Springer Science & Business Media: Berlin, Germany, 2003; Volume 10.
- 30. Sakoda, K. Optical Properties of Photonic Crystals; Springer Science & Business Media: Berlin, Germany, 2004; Volume 80.
- 31. Euser, T.G.; Molenaar, A.J.; Fleming, J.G.; Gralak, B.; Polman, A.; Vos, W.L. All-optical octave-broad ultrafast switching of Si woodpile photonic band gap crystals. *Phys. Rev. B* **2008**, 77, 115214. [CrossRef]
- 32. Yablonovitch, E. Photonic crystals: What's in a name? Opt. Photonics News 2007, 18, 12–13.
- 33. Tanabe, T.; Notomi, M.; Mitsugi, S.; Shinya, A.; Kuramochi, E. All-optical switches on a silicon chip realized using photonic crystal nanocavities. *Appl. Phys. Lett.* **2005**, *87*, 151112. [CrossRef]
- 34. Nau, D.; Bertram, R.; Buse, K.; Zentgraf, T.; Kuhl, J.; Tikhodeev, S.; Gippius, N.; Giessen, H. Optical switching in metallic photonic crystal slabs with photoaddressable polymers. *Appl. Phys. B* **2006**, *82*, 543–547. [CrossRef]
- 35. Zhao, Y.; Qian, C.; Qiu, K.; Gao, Y.; Xu, X. Ultrafast optical switching using photonic molecules in photonic crystal waveguides. *Opt. Express* **2015**, *23*, 9211–9220. [CrossRef] [PubMed]
- 36. Enoch, S.; Quidant, R.; Badenes, G. Optical sensing based on plasmon coupling in nanoparticle arrays. *Opt. Express* **2004**, *12*, 3422–3427. [CrossRef]
- 37. Drachev, V.P.; Chettiar, U.K.; Kildishev, A.V.; Yuan, H.-K.; Cai, W.; Shalaev, V.M. The Ag dielectric function in plasmonic metamaterials. *Opt. Express* **2008**, *16*, 1186–1195. [CrossRef] [PubMed]
- 38. Krasavin, A.; Zayats, A. All-optical active components for dielectric-loaded plasmonic waveguides. *Opt. Commun.* **2010**, 283, 1581–1584. [CrossRef]
- 39. Barnes, W.L.; Dereux, A.; Ebbesen, T.W. Surface plasmon subwavelength optics. Nature 2003, 424, 824–830. [CrossRef]
- 40. Gramotnev, D.K.; Bozhevolnyi, S.I. Plasmonics beyond the diffraction limit. Nat. Photonics 2010, 4, 83–91. [CrossRef]
- 41. Porto, J.; Martin-Moreno, L.; Garcia-Vidal, F. Optical bistability in subwavelength slit apertures containing nonlinear media. *Phys. Rev. B* **2004**, *70*, 081402. [CrossRef]
- 42. Smolyaninov, I.I. Quantum fluctuations of the refractive index near the interface between a metal and a nonlinear dielectric. *Phys. Rev. Lett.* **2005**, *94*, 057403. [CrossRef]
- 43. Min, C.; Wang, P.; Chen, C.; Deng, Y.; Lu, Y.; Ming, H.; Ning, T.; Zhou, Y.; Yang, G. All-optical switching in subwavelength metallic grating structure containing nonlinear optical materials. *Opt. Lett.* **2008**, *33*, 869–871. [CrossRef]
- 44. Min, C.; Veronis, G. Absorption switches in metal-dielectric-metal plasmonic waveguides. *Opt. Express* **2009**, *17*, 10757–10766. [CrossRef] [PubMed]
- 45. Zhong, Z.-J.; Xu, Y.; Lan, S.; Dai, Q.-F.; Wu, L.-J. Sharp and asymmetric transmission response in metal-dielectric-metal plasmonic waveguides containing Kerr nonlinear media. *Opt. Express* **2010**, *18*, 79–86. [CrossRef] [PubMed]
- 46. Chai, Z.; Hu, X.; Zhu, Y.; Sun, S.; Yang, H.; Gong, Q. Ultracompact Chip-Integrated Electromagnetically Induced Transparency in a Single Plasmonic Composite Nanocavity. *Adv. Opt. Mater.* **2014**, *2*, 320–325. [CrossRef]
- 47. Nazarathy, M.; Zalevsky, Z.; Rudnitsky, A.; Larom, B.; Nevet, A.; Orenstein, M.; Fischer, B. All-optical linear reconfigurable logic with nonlinear phase erasure. *J. Opt. Soc. Am. A* **2009**, *26*, A21–A39. [CrossRef]
- 48. Alexe-Ionescu, A.-L.; Ionescu, A.T.; Barna, E.S.; Barna, V.; Scaramuzza, N. Fast electro-optic switching in nematic liquid crystals. *Appl. Phys. Lett.* **2003**, *84*, 40–42. [CrossRef]
- 49. Shcherbakov, M.R.; Vabishchevich, P.P.; Shorokhov, A.S.; Chong, K.E.; Choi, D.-Y.; Staude, I.; Miroshnichenko, A.E.; Neshev, D.N.; Fedyanin, A.A.; Kivshar, Y.S. Ultrafast All-Optical Switching with Magnetic Resonances in Nonlinear Dielectric Nanostructures. *Nano Lett.* **2015**, *15*, 6985–6990. [CrossRef]
- 50. Xia, F.; Wang, H.; Xiao, D.; Dubey, M.; Ramasubramaniam, A. Two-dimensional material nanophotonics. *Nat. Photonics* **2014**, *8*, 899–907. [CrossRef]
- 51. Kroto, H.; Heath, J.; O'brien, S.; Curl, R.; Smalley, R. This Week's Citation Classic[®]. Nature 1985, 318, 162–163. [CrossRef]
- 52. Iijima, S. Helical microtubules of graphitic carbon. *Nature* **1991**, 354, 56–58. [CrossRef]
- 53. Novoselov, K.S.; Geim, A.K.; Morozov, S.V.; Jiang, D.; Zhang, Y.; Dubonos, S.V.; Grigorieva, I.V.; Firsov, A.A. Electric field effect in atomically thin carbon films. *Science* **2004**, *306*, 666–669. [CrossRef]
- 54. Nair, R.R.; Blake, P.; Grigorenko, A.N.; Novoselov, K.S.; Booth, T.J.; Stauber, T.; Peres, N.M.; Geim, A.K. Fine structure constant defines visual transparency of graphene. *Science* **2008**, 320, 1308. [CrossRef] [PubMed]
- 55. Lee, C.; Wei, X.; Kysar, J.W.; Hone, J. Measurement of the elastic properties and intrinsic strength of monolayer graphene. *Science* **2008**, 321, 385–388. [CrossRef] [PubMed]
- 56. Balandin, A.A. Thermal properties of graphene and nanostructured carbon materials. Nat. Mater. 2011, 10, 569–581. [CrossRef]
- 57. Morozov, S.; Novoselov, K.; Katsnelson, M.; Schedin, F.; Elias, D.; Jaszczak, J.A.; Geim, A. Giant intrinsic carrier mobilities in graphene and its bilayer. *Phys. Rev. Lett.* **2008**, *100*, 016602. [CrossRef] [PubMed]

Coatings **2023**, 13, 876 21 of 25

58. Butler, S.Z.; Hollen, S.M.; Cao, L.; Cui, Y.; Gupta, J.A.; Gutiérrez, H.R.; Heinz, T.F.; Hong, S.S.; Huang, J.; Goldberger, J.E.; et al. Progress, Challenges, and Opportunities in Two-Dimensional Materials Beyond Graphene. *ACS Nano* 2013, 7, 2898–2926. [CrossRef] [PubMed]

- 59. Han, M.Y.; Özyilmaz, B.; Zhang, Y.; Kim, P. Energy Band-Gap Engineering of Graphene Nanoribbons. *Phys. Rev. Lett.* **2007**, 98, 206805. [CrossRef] [PubMed]
- 60. Kim, K.K.; Hsu, A.; Jia, X.; Kim, S.M.; Shi, Y.; Hofmann, M.; Nezich, D.; Rodriguez-Nieva, J.F.; Dresselhaus, M.; Palacios, T. Synthesis of monolayer hexagonal boron nitride on Cu foil using chemical vapor deposition. *Nano Lett.* **2012**, *12*, 161–166. [CrossRef]
- 61. Zhi, C.; Bando, Y.; Tang, C.; Kuwahara, H.; Golberg, D. Large-scale fabrication of boron nitride nanosheets and their utilization in polymeric composites with improved thermal and mechanical properties. *Adv. Mater.* **2009**, *21*, 2889–2893. [CrossRef]
- 62. Giovannetti, G.; Khomyakov, P.A.; Brocks, G.; Kelly, P.J.; Van Den Brink, J. Substrate-induced band gap in graphene on hexagonal boron nitride: Ab initio density functional calculations. *Phys. Rev. B* **2007**, *76*, 073103. [CrossRef]
- 63. Han, J.H.; Lee, S.; Cheon, J. Synthesis and structural transformations of colloidal 2D layered metal chalcogenide nanocrystals. *Chem. Soc. Rev.* **2013**, 42, 2581–2591. [CrossRef]
- 64. Chhowalla, M.; Shin, H.S.; Eda, G.; Li, L.-J.; Loh, K.P.; Zhang, H. The chemistry of two-dimensional layered transition metal dichalcogenide nanosheets. *Nat. Chem.* **2013**, *5*, 263. [CrossRef] [PubMed]
- 65. Li, L.; Yu, Y.; Ye, G.J.; Ge, Q.; Ou, X.; Wu, H.; Feng, D.; Chen, X.H.; Zhang, Y. Black phosphorus field-effect transistors. *Nat. Nanotechnol.* **2014**, *9*, 372–377. [CrossRef] [PubMed]
- 66. Choi, W.; Choudhary, N.; Han, G.H.; Park, J.; Akinwande, D.; Lee, Y.H. Recent development of two-dimensional transition metal dichalcogenides and their applications. *Mater. Today* **2017**, *20*, 116–130. [CrossRef]
- 67. Wang, Q.H.; Kalantar-Zadeh, K.; Kis, A.; Coleman, J.N.; Strano, M.S. Electronics and optoelectronics of two-dimensional transition metal dichalcogenides. *Nat. Nanotechnol.* **2012**, *7*, 699. [CrossRef] [PubMed]
- 68. Zeng, H.; Dai, J.; Yao, W.; Xiao, D.; Cui, X. Valley polarization in MoS₂ monolayers by optical pumping. *Nat. Nanotechnol.* **2012**, *7*, 490. [CrossRef]
- 69. Cao, T.; Wang, G.; Han, W.; Ye, H.; Zhu, C.; Shi, J.; Niu, Q.; Tan, P.; Wang, E.; Liu, B.; et al. Valleyselective circular dichroism of monolayer molybdenum disulphide. *Nat. Commun.* **2012**, *3*, 887. [CrossRef]
- Guimarães, M.H.; Gao, H.; Han, Y.; Kang, K.; Xie, S.; Kim, C.-J.; Muller, D.A.; Ralph, D.C.; Park, J. Atomically thin ohmic edge contacts between two-dimensional materials. ACS Nano 2016, 10, 6392–6399. [CrossRef]
- 71. Duan, X.; Wang, C.; Shaw, J.C.; Cheng, R.; Chen, Y.; Li, H.; Wu, X.; Tang, Y.; Zhang, Q.; Pan, A. Lateral epitaxial growth of two-dimensional layered semiconductor heterojunctions. *Nat. Nanotechnol.* **2014**, *9*, 1024. [CrossRef]
- 72. Gullans, M.; Chang, D.E.; Koppens, F.H.L.; de Abajo, F.J.G.; Lukin, M.D. Single-Photon Nonlinear Optics with Graphene Plasmons. *Phys. Rev. Lett.* **2013**, *111*, 247401. [CrossRef]
- 73. Takahashi, M.; Ueda, W.; Goto, N.; Yanagiya, S. Saturable Absorption by Vertically Inserted or Overlaid Monolayer Graphene in Optical Waveguide for All-Optical Switching Circuit. *IEEE Photonics J.* **2013**, *5*, 6602109. [CrossRef]
- 74. Rao, S.M.; Heitz, J.J.F.; Roger, T.; Westerberg, N.; Faccio, D. Coherent control of light interaction with graphene. *Opt. Lett.* **2014**, *39*, 5345–5347. [CrossRef]
- 75. Ooi, K.J.A.; Cheng, J.L.; Sipe, J.E.; Ang, L.K.; Tan, D.T.H. Ultrafast, broadband, and configurable midinfrared all-optical switching in nonlinear graphene plasmonic waveguides. *APL Photonics* **2016**, *1*, 046101. [CrossRef]
- 76. Debnath, P.C.; Uddin, S.; Song, Y.-W. Ultrafast All-Optical Switching Incorporatingin SituGraphene Grown along an Optical Fiber by the Evanescent Field of a Laser. *ACS Photonics* **2017**, *5*, 445–455. [CrossRef]
- 77. Hoseini, M.; Malekmohammad, M. All-optical high performance graphene-photonic crystal switch. *Opt. Commun.* **2017**, *383*, 159–164. [CrossRef]
- 78. Ono, M.; Hata, M.; Tsunekawa, M.; Nozaki, K.; Sumikura, H.; Chiba, H.; Notomi, M. Ultrafast and energy-efficient all-optical switching with graphene-loaded deep-subwavelength plasmonic waveguides. *Nat. Photonics* **2020**, *14*, 37–43. [CrossRef]
- 79. Chu, R.; Guan, C.; Bo, Y.; Liu, J.; Shi, J.; Yang, J.; Ye, P.; Li, P.; Yang, J.; Yuan, L. Graphene decorated twin-core fiber Michelson interferometer for all-optical phase shifter and switch. *Opt. Lett.* **2020**, *45*, 177–180. [CrossRef]
- 80. Taghizadeh, M.; Bozorgzadeh, F.; Bordbar, G.H. All-optical diffraction and ultrafast switching in a terahertz-driven quantized graphene system. *Opt. Laser Technol.* **2023**, *159*, 108969. [CrossRef]
- 81. Shan, Y.; Tang, J.; Wu, L.; Lu, S.; Dai, X.; Xiang, Y. Spatial self-phase modulation and all-optical switching of graphene oxide dispersions. *J. Alloys Compd.* **2019**, 771, 900–904. [CrossRef]
- 82. Popkova, A.A.; Chezhegov, A.A.; Rybin, M.G.; Soboleva, I.V.; Obraztsova, E.D.; Bessonov, V.O.; Fedyanin, A.A. Bloch Surface Wave-Assisted Ultrafast All-Optical Switching in Graphene. *Adv. Opt. Mater.* **2022**, *10*, 2101937. [CrossRef]
- 83. Hao, T.; Chang, Z.; Chiang, K.S. Externally pumped low-loss graphene-based fiber Mach-Zehnder all-optical switches with mW switching powers. *Opt. Express* **2019**, 27, 4216–4225. [CrossRef]
- 84. Armaghani, S.; Khani, S.; Danaie, M. Design of all-optical graphene switches based on a Mach-Zehnder interferometer employing optical Kerr effect. *Superlattices Microstruct.* **2019**, 135, 106244. [CrossRef]
- 85. Qiu, C.; Zhang, C.; Zeng, H.; Guo, T. High-Performance Graphene-on-Silicon Nitride All-Optical Switch Based on a Mach–Zehnder Interformeter. J. Light. Technol. 2021, 39, 2099–2105. [CrossRef]

Coatings **2023**, 13, 876 22 of 25

86. Jiang, L.; Huang, Q.; Chiang, K.S. Low-power all-optical switch based on a graphene-buried polymer waveguide Mach-Zehnder interferometer. *Opt. Express* **2022**, *30*, 6786–6797. [CrossRef]

- 87. Ling, X.; Wang, H.; Huang, S.; Xia, F.; Dresselhaus, M.S. The renaissance of black phosphorus. *Proc. Natl. Acad. Sci. USA* **2015**, 112, 4523. [CrossRef]
- 88. Castellanos-Gomez, A. Black Phosphorus: Narrow Gap, Wide Applications. J. Phys. Chem. Lett. 2015, 6, 4280–4291. [CrossRef]
- 89. Tran, V.; Soklaski, R.; Liang, Y.; Yang, L. Layer-controlled band gap and anisotropic excitons in few-layer black phosphorus. *Phys. Rev. B* **2014**, *89*, 235319. [CrossRef]
- 90. Li, D.; Jussila, H.; Karvonen, L.; Ye, G.; Lipsanen, H.; Chen, X.; Sun, Z. Polarization and Thickness Dependent Absorption Properties of Black Phosphorus: New Saturable Absorber for Ultrafast Pulse Generation. *Sci. Rep.* **2015**, *5*, 15899. [CrossRef]
- 91. Sotor, J.; Sobon, G.; Macherzynski, W.; Paletko, P.; Abramski, K.M. Black phosphorus saturable absorber for ultrashort pulse generation. *Appl. Phys. Lett.* **2015**, 107, 051108. [CrossRef]
- 92. Huber, M.A.; Mooshammer, F.; Plankl, M.; Viti, L.; Sandner, F.; Kastner, L.Z.; Frank, T.; Fabian, J.; Vitiello, M.S.; Cocker, T.L.; et al. Femtosecond photo-switching of interface polaritons in black phosphorus heterostructures. *Nat. Nanotechnol.* **2017**, *12*, 207–211. [CrossRef] [PubMed]
- 93. Uddin, S.; Debnath, P.C.; Park, K.; Song, Y.-W. Nonlinear Black Phosphorus for Ultrafast Optical Switching. *Sci. Rep.* **2017**, *7*, 43371. [CrossRef]
- 94. Zheng, J.; Yang, Z.; Si, C.; Liang, Z.; Chen, X.; Cao, R.; Guo, Z.; Wang, K.; Zhang, Y.; Ji, J.; et al. Black Phosphorus Based All-Optical-Signal-Processing: Toward High Performances and Enhanced Stability. ACS Photonics 2017, 4, 1466–1476. [CrossRef]
- 95. Wang, K.; Chen, Y.; Zheng, J.; Ge, Y.; Ji, J.; Song, Y.; Zhang, H. Black phosphorus quantum dot based all-optical signal processing: Ultrafast optical switching and wavelength converting. *Nanotechnology* **2019**, *30*, 415202. [CrossRef]
- 96. Yin, Y.; Li, S.; Ren, J.; Du, Y.; Farrell, G.; Brambilla, G.; Wang, P. All-optical modulation in Black Phosphorus functionalized microfibre coil resonator. *Meas. Sci. Technol.* **2020**, 32, 015202. [CrossRef]
- 97. Wang, H.; Jiang, S.; Shao, W.; Zhang, X.; Chen, S.; Sun, X.; Zhang, Q.; Luo, Y.; Xie, Y. Optically Switchable Photocatalysis in Ultrathin Black Phosphorus Nanosheets. *J. Am. Chem. Soc.* **2018**, *140*, 3474–3480. [CrossRef] [PubMed]
- 98. Shim, J.; Jang, S.W.; Lim, J.-H.; Kim, H.; Kang, D.-H.; Kim, K.-H.; Seo, S.; Heo, K.; Shin, C.; Yu, H.-Y.; et al. Polarity control in a single transition metal dichalcogenide (TMD) transistor for homogeneous complementary logic circuits. *Nanoscale* **2019**, *11*, 12871–12877. [CrossRef] [PubMed]
- 99. Tan, C.; Cao, X.; Wu, X.-J.; He, Q.; Yang, J.; Zhang, X.; Chen, J.; Zhao, W.; Han, S.; Nam, G.-H.; et al. Recent Advances in Ultrathin Two-Dimensional Nanomaterials. *Chem. Rev.* **2017**, *117*, 6225–6331. [CrossRef]
- 100. Hu, Z.; Wu, Z.; Han, C.; He, J.; Ni, Z.; Chen, W. Two-dimensional transition metal dichalcogenides: Interface and defect engineering. *Chem. Soc. Rev.* **2018**, *47*, 3100–3128. [CrossRef]
- 101. Long, M.; Wang, P.; Fang, H.; Hu, W. Progress, Challenges, and Opportunities for 2D Material Based Photodetectors. *Adv. Funct. Mater.* **2019**, 29, 1803807. [CrossRef]
- 102. Wang, G.; Li, L.; Fan, W.; Wang, R.; Zhou, S.; Lü, J.-T.; Gan, L.; Zhai, T. Interlayer Coupling Induced Infrared Response in WS₂/MoS₂ Heterostructures Enhanced by Surface Plasmon Resonance. *Adv. Funct. Mater.* **2018**, *28*, 1800339. [CrossRef]
- 103. Li, L.; Wang, W.; Chai, Y.; Li, H.; Tian, M.; Zhai, T. Few-Layered PtS2 Phototransistor on h-BN with High Gain. *Adv. Funct. Mater.* **2017**, 27, 1701011. [CrossRef]
- 104. Li, L.; Han, W.; Pi, L.; Niu, P.; Han, J.; Wang, C.; Su, B.; Li, H.; Xiong, J.; Bando, Y.; et al. Emerging in-plane anisotropic two-dimensional materials. *InfoMat* **2019**, *1*, 54–73. [CrossRef]
- 105. Wang, R.; Zhou, F.; Lv, L.; Zhou, S.; Yu, Y.; Zhuge, F.; Li, H.; Gan, L.; Zhai, T. Modulation of the Anisotropic Electronic Properties in ReS₂ via Ferroelectric Film. *CCS Chem.* **2019**, *1*, 268–277.
- 106. You, H.; Zhuo, Z.; Lu, X.; Liu, Y.; Guo, Y.; Wang, W.; Yang, H.; Wu, X.; Li, H.; Zhai, T. 1T'-MoTe2-Based On-Chip Electrocatalytic Microdevice: A Platform to Unravel Oxidation-Dependent Electrocatalysis. *CCS Chem.* **2019**, *1*, 396–406. [CrossRef]
- 107. Zhang, W.; Wang, Q.; Chen, Y.; Wang, Z.; Wee, A.T.S. Van der Waals stacked 2D layered materials for optoelectronics. 2D Mater. **2016**, *3*, 022001. [CrossRef]
- 108. Liu, Y.; Weiss, N.O.; Duan, X.; Cheng, H.-C.; Huang, Y.; Duan, X. Van der Waals heterostructures and devices. *Nat. Rev. Mater.* **2016**, *1*, 16042. [CrossRef]
- 109. Mak, K.F.; He, K.; Shan, J.; Heinz, T.F. Control of valley polarization in monolayer MoS₂ by optical helicity. *Nat. Nanotechnol.* **2012**, 7, 494–498. [CrossRef] [PubMed]
- 110. Xiao, D.; Liu, G.-B.; Feng, W.; Xu, X.; Yao, W. Coupled Spin and Valley Physics in Monolayers of MoS₂ and Other Group-VI Dichalcogenides. *Phys. Rev. Lett.* **2012**, *108*, 196802. [CrossRef]
- 111. Splendiani, A.; Sun, L.; Zhang, Y.; Li, T.; Kim, J.; Chim, C.-Y.; Galli, G.; Wang, F. Emerging Photoluminescence in Monolayer MoS₂. *Nano Lett.* **2010**, *10*, 1271–1275. [CrossRef]
- 112. Huang, X.; Zeng, Z.; Zhang, H. Metal dichalcogenide nanosheets: Preparation, properties and applications. *Chem. Soc. Rev.* **2013**, 42, 1934–1946. [CrossRef]
- 113. Mak, K.F.; Lee, C.; Hone, J.; Shan, J.; Heinz, T.F. Atomically Thin MoS₂: A New Direct-Gap Semiconductor. *Phys. Rev. Lett.* **2010**, 105, 136805. [CrossRef]

Coatings **2023**, 13, 876 23 of 25

114. Eda, G.; Yamaguchi, H.; Voiry, D.; Fujita, T.; Chen, M.; Chhowalla, M. Photoluminescence from Chemically Exfoliated MoS₂. *Nano Lett.* **2011**, *11*, 5111–5116. [CrossRef] [PubMed]

- 115. Tongay, S.; Zhou, J.; Ataca, C.; Lo, K.; Matthews, T.S.; Li, J.; Grossman, J.C.; Wu, J. Thermally Driven Crossover from Indirect toward Direct Bandgap in 2D Semiconductors: MoSe₂ versus MoS₂. *Nano Lett.* **2012**, 12, 5576–5580. [CrossRef] [PubMed]
- 116. Zhao, W.; Ghorannevis, Z.; Chu, L.; Toh, M.; Kloc, C.; Tan, P.-H.; Eda, G. Evolution of Electronic Structure in Atomically Thin Sheets of WS2 and WSe₂. ACS Nano **2013**, 7, 791–797. [CrossRef] [PubMed]
- 117. Chowdhury, T.; Erick, C.S.; Thomas, J.K. Progress and prospects in transition-metal dichalcogenide research beyond 2D. *Chem. Rev.* **2020**, *120*, 12563–12591. [CrossRef]
- 118. Yang, C.; Gao, Y.; Qin, C.; Liang, X.; Han, S.; Zhang, G.; Chen, R.; Hu, J.; Xiao, L.; Jia, S. All-Optical Reversible Manipulation of Exciton and Trion Emissions in Monolayer WS₂. *Nanomaterials* **2020**, *10*, 23. [CrossRef]
- 119. Seyler, K.L.; Schaibley, J.R.; Gong, P.; Rivera, P.; Jones, A.M.; Wu, S.; Yan, J.; Mandrus, D.G.; Yao, W.; Xu, X. Electrical control of second-harmonic generation in a WSe₂ monolayer transistor. *Nat. Nanotechnol.* **2015**, *10*, 407–411. [CrossRef]
- 120. Wu, K.; Guo, C.; Wang, H.; Zhang, X.; Wang, J.; Chen, J. All-optical phase shifter and switch near 1550 nm using tungsten disulfide (WS₂) deposited tapered fiber. *Opt. Express* **2017**, 25, 17639–17649. [CrossRef]
- 121. Wu, L.; Xie, Z.; Lu, L.; Zhao, J.; Wang, Y.; Jiang, X.; Ge, Y.; Zhang, F.; Lu, S.; Guo, Z.; et al. Few-Layer Tin Sulfide: A Promising Black-Phosphorus-Analogue 2D Material with Exceptionally Large Nonlinear Optical Response, High Stability, and Applications in All-Optical Switching and Wavelength Conversion. *Adv. Opt. Mater.* 2018, 6, 1700985. [CrossRef]
- 122. Xu, H.; Lin, Z.; Dai, X. MoTe₂ quantum dots-based all-optical switching. Opt. Commun. 2022, 506, 127573. [CrossRef]
- 123. Zhang, H.; Lu, S.B.; Zheng, J.; Du, J.; Wen, S.C.; Tang, D.Y.; Loh, K.P. Molybdenum disulfide (MoS₂) as a broadband saturable absorber for ultra-fast photonics. *Opt. Express* **2014**, 22, 7249–7260. [CrossRef]
- 124. Tsai, D.-S.; Liu, K.-K.; Lien, D.-H.; Tsai, M.-L.; Kang, C.-F.; Lin, C.-A.; Li, L.-J.; He, J.-H. Few-Layer MoS₂ with High Broadband Photogain and Fast Optical Switching for Use in Harsh Environments. *ACS Nano* **2013**, *7*, 3905–3911. [CrossRef] [PubMed]
- 125. Yin, Z.; Li, H.; Li, H.; Jiang, L.; Shi, Y.; Sun, Y.; Lu, G.; Zhang, Q.; Chen, X.; Zhang, H. Single-Layer MoS₂ Phototransistors. *ACS Nano* 2012, 6, 74–80. [CrossRef]
- 126. Srivastava, Y.K.; Chaturvedi, A.; Manjappa, M.; Kumar, A.; Dayal, G.; Kloc, C.; Singh, R. MoS₂ for Ultrafast All-Optical Switching and Modulation of THz Fano Metaphotonic Devices. *Adv. Opt. Mater.* **2017**, *5*, 1700762. [CrossRef]
- 127. Geim, A.K.; Grigorieva, I.V. Van der Waals heterostructures. Nature 2013, 499, 419–425. [CrossRef] [PubMed]
- 128. Wang, H.; Liu, F.; Fu, W.; Fang, Z.; Zhou, W.; Liu, Z. Two-dimensional heterostructures: Fabrication, characterization, and application. *Nanoscale* **2014**, *6*, 12250–12272. [CrossRef]
- 129. Buscema, M.; Island, J.O.; Groenendijk, D.J.; Blanter, S.I.; Steele, G.A.; van der Zant, H.S.J.; Castellanos-Gomez, A. Photocurrent generation with two-dimensional van der Waals semiconductors. *Chem. Soc. Rev.* **2015**, *44*, 3691–3718. [CrossRef]
- 130. Britnell, L.; Gorbachev, R.V.; Jalil, R.; Belle, B.D.; Schedin, F.; Mishchenko, A.; Georgiou, T.; Katsnelson, M.I.; Eaves, L.; Morozov, S.V.; et al. Field-Effect Tunneling Transistor Based on Vertical Graphene Heterostructures. *Science* 2012, 335, 947. [CrossRef]
- 131. Georgiou, T.; Jalil, R.; Belle, B.D.; Britnell, L.; Gorbachev, R.V.; Morozov, S.V.; Kim, Y.-J.; Gholinia, A.; Haigh, S.J.; Makarovsky, O.; et al. Vertical field-effect transistor based on graphene–WS₂ heterostructures for flexible and transparent electronics. *Nat. Nanotechnol.* **2013**, *8*, 100–103. [CrossRef]
- 132. Gorbachev, R.V.; Geim, A.K.; Katsnelson, M.I.; Novoselov, K.S.; Tudorovskiy, T.; Grigorieva, I.V.; MacDonald, A.H.; Morozov, S.V.; Watanabe, K.; Taniguchi, T.; et al. Strong Coulomb drag and broken symmetry in double-layer graphene. *Nat. Phys.* **2012**, *8*, 896–901. [CrossRef]
- 133. Dean, C.R.; Young, A.F.; Cadden-Zimansky, P.; Wang, L.; Ren, H.; Watanabe, K.; Taniguchi, T.; Kim, P.; Hone, J.; Shepard, K.L. Multicomponent fractional quantum Hall effect in graphene. *Nat. Phys.* **2011**, *7*, 693–696. [CrossRef]
- 134. Rivera, P.; Schaibley, J.R.; Jones, A.M.; Ross, J.S.; Wu, S.; Aivazian, G.; Klement, P.; Seyler, K.; Clark, G.; Ghimire, N.J.; et al. Observation of long-lived interlayer excitons in monolayer MoSe₂–WSe₂ heterostructures. *Nat. Commun.* **2015**, *6*, 6242. [CrossRef] [PubMed]
- 135. Chiu, M.-H.; Li, M.-Y.; Zhang, W.; Hsu, W.-T.; Chang, W.-H.; Terrones, M.; Terrones, H.; Li, L.-J. Spectroscopic Signatures for Interlayer Coupling in MoS2–WSe2 van der Waals Stacking. *ACS Nano* **2014**, *8*, 9649–9656. [CrossRef] [PubMed]
- 136. Fang, H.; Battaglia, C.; Carraro, C.; Nemsak, S.; Ozdol, B.; Kang Jeong, S.; Bechtel Hans, A.; Desai Sujay, B.; Kronast, F.; Unal Ahmet, A.; et al. Strong interlayer coupling in van der Waals heterostructures built from single-layer chalcogenides. *Proc. Natl. Acad. Sci. USA* **2014**, *111*, 6198–6202. [CrossRef] [PubMed]
- 137. Tongay, S.; Fan, W.; Kang, J.; Park, J.; Koldemir, U.; Suh, J.; Narang, D.S.; Liu, K.; Ji, J.; Li, J.; et al. Tuning Interlayer Coupling in Large-Area Heterostructures with CVD-Grown MoS₂ and WS₂ Monolayers. *Nano Lett.* **2014**, *14*, 3185–3190. [CrossRef] [PubMed]
- 138. Ceballos, F.; Bellus, M.Z.; Chiu, H.-Y.; Zhao, H. Ultrafast Charge Separation and Indirect Exciton Formation in a MoS₂–MoSe₂ van der Waals Heterostructure. *ACS Nano* **2014**, *8*, 12717–12724. [CrossRef]
- 139. Hong, X.; Kim, J.; Shi, S.-F.; Zhang, Y.; Jin, C.; Sun, Y.; Tongay, S.; Wu, J.; Zhang, Y.; Wang, F. Ultrafast charge transfer in atomically thin MoS2/WS2 heterostructures. *Nat. Nanotechnol.* **2014**, *9*, 682–686. [CrossRef]
- 140. Zhou, K.-G.; Withers, F.; Cao, Y.; Hu, S.; Yu, G.; Casiraghi, C. Raman Modes of MoS2 Used as Fingerprint of van der Waals Interactions in 2-D Crystal-Based Heterostructures. *ACS Nano* **2014**, *8*, 9914–9924. [CrossRef]
- 141. Wu, M.; Zeng, X.C. Intrinsic Ferroelasticity and/or Multiferroicity in Two-Dimensional Phosphorene and Phosphorene Analogues. *Nano Lett.* **2016**, *16*, 3236–3241. [CrossRef]

Coatings **2023**, 13, 876 24 of 25

142. Castellanos-Gomez, A.; van der Zant, H.S.J.; Steele, G.A. Folded MoS₂ layers with reduced interlayer coupling. *Nano Res.* **2014**, 7, 572–578. [CrossRef]

- 143. Crowne, F.J.; Amani, M.; Birdwell, A.G.; Chin, M.L.; O'Regan, T.P.; Najmaei, S.; Liu, Z.; Ajayan, P.M.; Lou, J.; Dubey, M. Blueshift of the \$A\$-exciton peak in folded monolayer MoS₂. *Phys. Rev. B* **2013**, *88*, 235302. [CrossRef]
- 144. Lui, C.H.; Li, Z.; Chen, Z.; Klimov, P.V.; Brus, L.E.; Heinz, T.F. Imaging Stacking Order in Few-Layer Graphene. *Nano Lett.* **2011**, 11, 164–169. [CrossRef] [PubMed]
- 145. Cong, C.; Yu, T.; Sato, K.; Shang, J.; Saito, R.; Dresselhaus, G.F.; Dresselhaus, M.S. Raman Characterization of ABA- and ABC-Stacked Trilayer Graphene. *ACS Nano* **2011**, *5*, 8760–8768. [CrossRef] [PubMed]
- 146. Zhang, W.; Yan, J.; Chen, C.-H.; Lei, L.; Kuo, J.-L.; Shen, Z.; Li, L.-J. Molecular adsorption induces the transformation of rhombohedral- to Bernal-stacking order in trilayer graphene. *Nat. Commun.* **2013**, *4*, 2074. [CrossRef] [PubMed]
- 147. Cui, X.; Lee, G.-H.; Kim, Y.D.; Arefe, G.; Huang, P.Y.; Lee, C.-H.; Chenet, D.A.; Zhang, X.; Wang, L.; Ye, F.; et al. Multi-terminal transport measurements of MoS2 using a van der Waals heterostructure device platform. *Nat. Nanotechnol.* **2015**, *10*, 534–540. [CrossRef]
- 148. Kang, J.; Tongay, S.; Zhou, J.; Li, J.; Wu, J. Band offsets and heterostructures of two-dimensional semiconductors. *Appl. Phys. Lett.* **2013**, *102*, 012111. [CrossRef]
- 149. Huang, S.; Ling, X.; Liang, L.; Kong, J.; Terrones, H.; Meunier, V.; Dresselhaus, M.S. Probing the Interlayer Coupling of Twisted Bilayer MoS2 Using Photoluminescence Spectroscopy. *Nano Lett.* **2014**, *14*, 5500–5508. [CrossRef]
- 150. Jariwala, D.; Sangwan, V.K.; Lauhon, L.J.; Marks, T.J.; Hersam, M.C. Emerging Device Applications for Semiconducting Two-Dimensional Transition Metal Dichalcogenides. *ACS Nano* **2014**, *8*, 1102–1120. [CrossRef]
- 151. Wang, H.; Yuan, H.; Sae Hong, S.; Li, Y.; Cui, Y. Physical and chemical tuning of two-dimensional transition metal dichalcogenides. *Chem. Soc. Rev.* **2015**, *44*, 2664–2680. [CrossRef]
- 152. Huo, N.; Kang, J.; Wei, Z.; Li, S.-S.; Li, J.; Wei, S.-H. Novel and Enhanced Optoelectronic Performances of Multilayer MoS2–WS2 Heterostructure Transistors. *Adv. Funct. Mater.* **2014**, *24*, 7025–7031. [CrossRef]
- 153. Britnell, L.; Ribeiro, R.M.; Eckmann, A.; Jalil, R.; Belle, B.D.; Mishchenko, A.; Kim, Y.J.; Gorbachev, R.V.; Georgiou, T.; Morozov, S.V.; et al. Strong Light-Matter Interactions in Heterostructures of Atomically Thin Films. *Science* **2013**, *340*, 1311. [CrossRef]
- 154. Furchi, M.M.; Pospischil, A.; Libisch, F.; Burgdörfer, J.; Mueller, T. Photovoltaic Effect in an Electrically Tunable van der Waals Heterojunction. *Nano Lett.* **2014**, *14*, 4785–4791. [CrossRef] [PubMed]
- 155. Wang, Y.; Mu, H.; Li, X.; Yuan, J.; Chen, J.; Xiao, S.; Bao, Q.; Gao, Y.; He, J. Observation of large nonlinear responses in a graphene-Bi₂Te₃ heterostructure at a telecommunication wavelength. *Appl. Phys. Lett.* **2016**, *108*, 221901. [CrossRef]
- 156. Liu, Z.; Ma, L.; Shi, G.; Zhou, W.; Gong, Y.; Lei, S.; Yang, X.; Zhang, J.; Yu, J.; Hackenberg, K.P.; et al. In-plane heterostructures of graphene and hexagonal boron nitride with controlled domain sizes. *Nat. Nanotechnol.* **2013**, *8*, 119–124. [CrossRef]
- 157. Huang, C.; Wu, S.; Sanchez, A.M.; Peters, J.J.P.; Beanland, R.; Ross, J.S.; Rivera, P.; Yao, W.; Cobden, D.H.; Xu, X. Lateral heterojunctions within monolayer MoSe₂–WSe₂ semiconductors. *Nat. Mater.* **2014**, *13*, 1096–1101. [CrossRef]
- 158. Gong, Y.; Lei, S.; Ye, G.; Li, B.; He, Y.; Keyshar, K.; Zhang, X.; Wang, Q.; Lou, J.; Liu, Z.; et al. Two-Step Growth of Two-Dimensional WSe₂/MoSe₂ Heterostructures. *Nano Lett.* **2015**, *15*, 6135–6141. [CrossRef] [PubMed]
- 159. Gong, Y.; Lin, J.; Wang, X.; Shi, G.; Lei, S.; Lin, Z.; Zou, X.; Ye, G.; Vajtai, R.; Yakobson, B.I.; et al. Vertical and in-plane heterostructures from WS2/MoS2 monolayers. *Nat. Mater.* **2014**, *13*, 1135–1142. [CrossRef]
- 160. Tang, M.; Zhang, D.; Wang, D.; Deng, J.; Kong, D.; Zhang, H. Performance prediction of 2D vertically stacked MoS₂-WS₂ heterostructures base on first-principles theory and Pearson correlation coefficient. *Appl. Surf. Sci.* **2022**, *596*, 153498. [CrossRef]
- 161. Heo, H.; Sung, J.H.; Jin, G.; Ahn, J.-H.; Kim, K.; Lee, M.-J.; Cha, S.; Choi, H.; Jo, M.-H. Rotation-Misfit-Free Heteroepitaxial Stacking and Stitching Growth of Hexagonal Transition-Metal Dichalcogenide Monolayers by Nucleation Kinetics Controls. *Adv. Mater* 2015, 27, 3803–3810. [CrossRef]
- 162. Levendorf, M.P.; Kim, C.-J.; Brown, L.; Huang, P.Y.; Havener, R.W.; Muller, D.A.; Park, J. Graphene and boron nitride lateral heterostructures for atomically thin circuitry. *Nature* **2012**, *488*, 627–632. [CrossRef]
- 163. Haigh, S.J.; Gholinia, A.; Jalil, R.; Romani, S.; Britnell, L.; Elias, D.C.; Novoselov, K.S.; Ponomarenko, L.A.; Geim, A.K.; Gorbachev, R. Cross-sectional imaging of individual layers and buried interfaces of graphene-based heterostructures and superlattices. *Nat. Mater.* 2012, 11, 764–767. [CrossRef]
- 164. Xu, X.; Gabor, N.M.; Alden, J.S.; van der Zande, A.M.; McEuen, P.L. Photo-Thermoelectric Effect at a Graphene Interface Junction. *Nano Lett.* **2010**, *10*, 562–566. [CrossRef] [PubMed]
- 165. Howell, S.L.; Jariwala, D.; Wu, C.-C.; Chen, K.-S.; Sangwan, V.K.; Kang, J.; Marks, T.J.; Hersam, M.C.; Lauhon, L.J. Investigation of Band-Offsets at Monolayer–Multilayer MoS₂ Junctions by Scanning Photocurrent Microscopy. *Nano Lett.* **2015**, *15*, 2278–2284. [CrossRef] [PubMed]
- 166. Liu, L.; Park, J.; Siegel David, A.; McCarty Kevin, F.; Clark Kendal, W.; Deng, W.; Basile, L.; Idrobo Juan, C.; Li, A.-P.; Gu, G. Heteroepitaxial Growth of Two-Dimensional Hexagonal Boron Nitride Templated by Graphene Edges. *Science* 2014, 343, 163–167. [CrossRef] [PubMed]
- 167. Lu, J.; Zhang, K.; Feng Liu, X.; Zhang, H.; Chien Sum, T.; Castro Neto, A.H.; Loh, K.P. Order–disorder transition in a two-dimensional boron–carbon–nitride alloy. *Nat. Commun.* **2013**, *4*, 2681. [CrossRef]

Coatings 2023, 13, 876 25 of 25

168. Liu, Y.; Huang, Y.; Duan, X. Van der Waals integration before and beyond two-dimensional materials. *Nature* **2019**, *567*, 323–333. [CrossRef]

169. Bae, S.-H.; Kum, H.; Kong, W.; Kim, Y.; Choi, C.; Lee, B.; Lin, P.; Park, Y.; Kim, J. Integration of bulk materials with two-dimensional materials for physical coupling and applications. *Nat. Mater.* **2019**, *18*, 550–560. [CrossRef]

Disclaimer/Publisher's Note: The statements, opinions and data contained in all publications are solely those of the individual author(s) and contributor(s) and not of MDPI and/or the editor(s). MDPI and/or the editor(s) disclaim responsibility for any injury to people or property resulting from any ideas, methods, instructions or products referred to in the content.