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Impact of Dispersion Methods on Mechanical Properties of Carbon Nanotube (CNT)/Iron Oxide (Fe₃O₄)/Epoxy Composites

Zulfiqar Ali ^{1,2}, Saba Yaqoob ^{1,2} and Alberto D'Amore ^{2,*} 

¹ Department of Mathematics and Physics, Università degli Studi della Campania "Luigi Vanvitelli", 81100 Caserta, Italy; zulfiqar.ali@unicampania.it (Z.A.); saba.yaqoob@unicampania.it (S.Y.)

² Department of Engineering, Università degli Studi della Campania "Luigi Vanvitelli", Via Roma 29, 81031 Aversa, Italy

* Correspondence: alberto.damore@unicampania.it; Tel.: +39-081-501-0291

Abstract: Integrating nanomaterials like carbon nanotubes (CNTs) and iron oxide (Fe₃O₄) into epoxy composites has attracted significant interest due to their potential to enhance mechanical properties. This study evaluates the impact of dispersion quality on the mechanical performance of CNT/Fe₃O₄/epoxy composites, comparing stirring and sonication methods at three different loadings: 0.1, 0.3, and 0.5 wt.%. Tensile testing revealed that sonicated composites consistently outperformed stirred composites, with a significant increase in the elastic modulus and ultimate tensile strength (UTS). However, fracture strain decreased in both composite types compared to pure epoxy, with sonicated composites experiencing a more significant reduction than stirred composites. These results underscore the importance of high-quality dispersion for optimizing mechanical properties.

Keywords: carbon nanotubes; iron oxide; hybrid composites; dispersion; mechanical properties



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1. Introduction

In recent years, multi-functional composite materials have gained significant attention due to the combination of desired properties. Owing to their unique structure, excellent tensile strength, and high electrical/thermal conductivity, carbon nanotubes (CNTs) are promising candidates to incorporate throughout polymer matrices. However, CNTs possess a hydrophobic nature, and their greater affinity to agglomerate tends to pose serious challenges to attaining a homogeneous dispersion throughout polymers, causing limitations to fully harnessing the inherent potential of CNTs [1,2]. On the other hand, the structural integrity of CNTs allows them to interlink with other filler materials [3]. Iron oxide (Fe₃O₄) nanoparticles, known for their magnetic responsiveness, significant mechanical strength, electrical conductivity, and thermal stability, present comprehensive properties when introduced throughout polymers [4–6]. Due to their large surface area, crystalline nature, and surface characteristics, the introduction of Fe₃O₄ nanoparticles accompanied with CNTs as a hybrid filler in polymer matrices highly assists in achieving a CNT uniform dispersion and interfacial bonding and thus synergistically improves the desired properties for different applications like electromagnetic interference shielding and data storage [7–11]. While incorporating epoxy composites, the presence of Fe₃O₄ nanoparticles plays a critical role in improving thermal and mechanical properties. The alignment of Fe₃O₄ nanoparticles in epoxy via magnetic field strength results in outstanding anisotropic properties [12].

Tong et al. [6] prepared CNTs/Fe₃O₄ hybrids by a modified co-precipitation method and used a magnetic field to induce the alignment of these hybrids in carbon fabric-reinforced epoxy composites (CFRPCs). It was observed that shear strength and hardness were greatly improved due to the alignment of the hybrids. In another study, Wu et al. [7] reported CNTs-Fe₃O₄/poly(lactic acid) (PLA) composites via melt-blending and investigated the electrical and electromagnetic interference (EMI) properties. With a CNT and nano-Fe₃O₄ ratio of (50:50), electrical conductivity was observed to be 90.6 S·m^{−1} while an

EMI shielding effectiveness of ~ 40.5 dB was reported. However, a CNT and nano- Fe_3O_4 ratio of (25:75) resulted in an EMI shielding effectiveness of 24.6 dB with a peak absorptivity of 40.3%. Liu et al. [8] synthesized MWCNTs/nano- Fe_3O_4 /PLA composite films with an optimized filler ratio to minimize agglomeration. The characterization results showed an increase in EMI shielding effectiveness ~ 22 dB. It was observed that the mechanical properties and conductivity notably increased by increasing the MWCNT content, while Fe_3O_4 increased the magnetic properties of the PLA composite films.

In this study, CNTs/ Fe_3O_4 /epoxy hybrid composites were successfully fabricated by using mechanical stirring and sonication approaches, and by varying the weight percentages of the hybrid filler, highlighting the effectiveness of dispersion and synergistically improving the mechanical properties of the epoxy composites. Later, the composite samples were characterized, and comparison results are discussed in detail.

2. Materials and Methods

2.1. Materials

Single-walled carbon nanotubes (SWCNTs), with an outer diameter of 1.6 ± 0.4 nm, a length of ≥ 5 μm , a surface area of $300 \text{ m}^2/\text{g}$, and a carbon nanotube content of ≥ 80 wt.%, were obtained from Tuball. Iron oxide (Fe_3O_4) powder with a predominant size of 200 nm was purchased from INOXIA Ltd. Pure acetone ($\geq 99.9\%$) was purchased from KEMIPOL SRL. IN2 Epoxy resin (Viscosity 325 mPa.s) and AT30 slow epoxy hardener (95–115 min pot-life) were purchased from Easy Composites. Non-toxic silicon rubber liquid and hardener were acquired from Reschimica for mold making.

2.2. Preparation of CNT/ Fe_3O_4 /Epoxy Hybrid Composites

The fabrication of CNT/ Fe_3O_4 /epoxy hybrid composites was performed using probe sonication and mechanical stirring to investigate the effectiveness of uniform dispersion on mechanical properties. The composites were prepared with different weight percentages, i.e., (0.1, 0.3, and 0.5 wt.%) of the CNT/ Fe_3O_4 hybrid filler, and three samples were prepared for each weight percentage for both methods. Figure 1 shows the complete schematic process of preparation.

In the sonication method, a certain amount of CNTs was initially dispersed in 100 g of acetone followed by a 30 min sonication. Afterward, a specified amount of Fe_3O_4 powder was added and sonicated for another half hour. Lastly, epoxy resin was introduced with gradual mixing and the whole solution was further sonicated for another half hour (Figure 1a).

Later, the subsequent solution was kept in an oven for 24 h at 80°C to evaporate the acetone. Subsequently, the resultant solution was placed in a vacuum oven for 5 h at room temperature to remove air bubbles and voids. Afterward, a slow hardener was introduced and mixed, and the solution was returned to the vacuum for 1 h. Then, this homogeneous solution was cast into silicon molds for curing at room temperature for 48 h. Finally, the composite specimens were de-molded and post-cured for 6 h at 100°C . However, while processing the dispersion quality of CNTs and Fe_3O_4 was monitored under a microscope after each sonication duration and thus sonication durations were extended proportionally with increased weight percentages.

Conversely, comparison composite samples with the same wt.%s were prepared by mechanically stirring the CNTs in a specific amount of epoxy, followed by adding and stirring Fe_3O_4 . The mechanical stirring was conducted for 10 min at a rotation speed of 1000 rpm, utilizing a propeller stirrer tip geometry. The stirring was performed in an Erlenmeyer flask to avoid the evaporation of acetone and to prevent solution splash, as shown in Figure 1b. This process was repeated for all three weight percentages, and the rest of the procedure was like previous practice. Moreover, pure epoxy samples were also produced for comparison results.

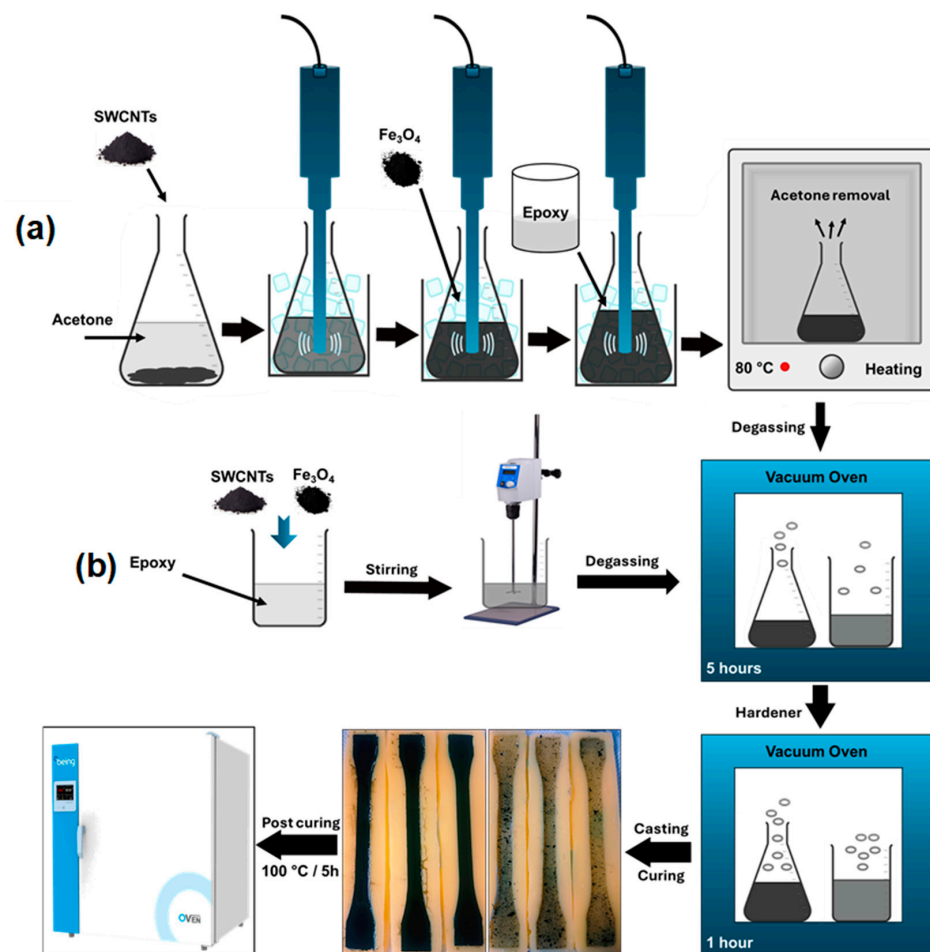


Figure 1. Schematic preparation of (a) fully dispersed CNT-Fe₃O₄/epoxy composites obtained by sonication method and (b) partially dispersed CNT-Fe₃O₄/epoxy composites obtained by mechanical stirring.

2.3. Characterization

A Zeiss AxioVert.A1 optical microscope (Carl Zeiss Microscopy GmbH, Jena, Germany) was used to observe the dispersion quality of the composite solution. The mechanical properties of the composite were analyzed using a Zwick-Roell Z010 universal testing machine (ZwickRoell GmbH & Co. KG, Ulm, Germany) at room temperature, with a load cell of 10 kN, 50 mm gauge length, and 5 mm/min crosshead speed. The obtained stress–strain curves were used to calculate the elastic modulus, ultimate tensile strength (UTS), and strain at rupture.

3. Results and Discussion

3.1. Dispersion Analysis

This study was meant to disperse CNT/Fe₃O₄/epoxy composites by two distinct approaches, i.e., sonication and stirring, and carry out performance evaluation of the composites. To better observe and compare the dispersion quality between the two fabrication techniques with a gradually increasing content of CNT/Fe₃O₄, the composite was observed from time to time using an optical microscope. Microscopic observations indicated that the dispersion quality was much improved with a homogeneously dispersed CNT/Fe₃O₄ throughout the epoxy matrix by the sonication method, while at the same time, the composite solution prepared by the stirring method showed poor dispersion with noticeable agglomerations of CNTs. The employed dispersion time was adjusted by varying the weight percentages of CNT/Fe₃O₄. However, with increasing filler loading, the disparity

in dispersion quality between the sonication and stirring techniques became more evident. The microscopic analysis provided valuable insights and exposed clear differences in the dispersion quality of the composites based on the fabrication method employed. Figure 2 shows the microscopic images captured during the observation of the dispersion features with an escalating filler weight percentage (wt.%). Figure 2 represents the dispersion results of the CNT/Fe₃O₄/epoxy composites that were (a, a1) 0.1 wt.% sonicated, (b, b1) 0.1 wt.% stirred, (c, c1) 0.3 wt.% sonicated, (d, d1) 0.3 wt.% stirred, (e, e1) 0.5 wt.% sonicated, and (f, f1) 0.5 wt.% stirred. It can be perceived that even at higher filler loadings the dispersion phenomena are more prominent with the sonication method, while the presence of huge clusters and agglomerations of CNTs evident bad dispersion in the composites prepared by the stirring method. Overall, the visual observations underscored valuable information on the morphology and structural integrity of the manufactured composite materials.

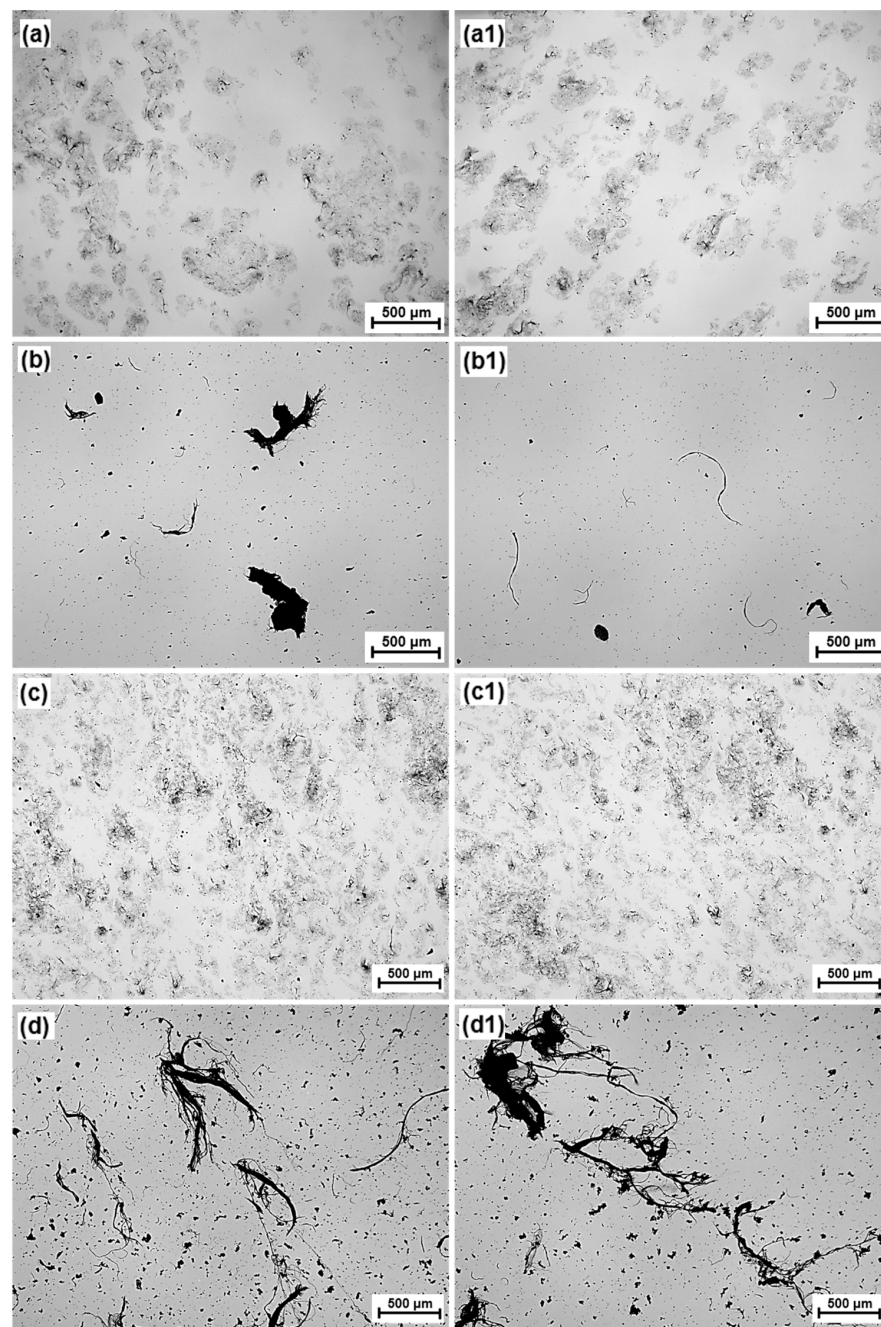


Figure 2. Cont.

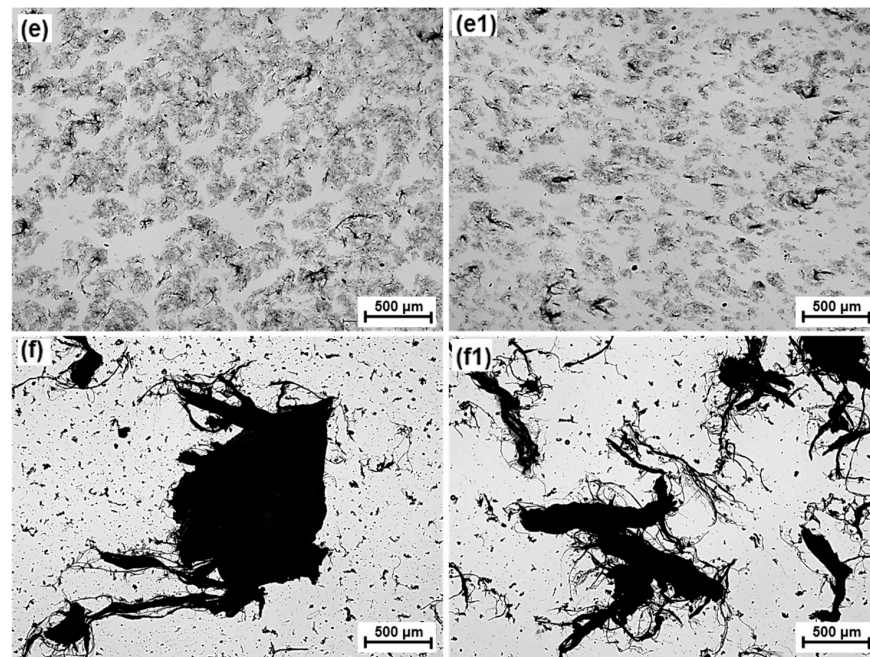


Figure 2. Microstructural observations of CNT/Fe₃O₄/epoxy composites with varying content: (a,a1) 0.1 wt.% sonicated, (b,b1) 0.1 wt.% stirred, (c,c1) 0.3 wt.% sonicated, (d,d1) 0.3 wt.% stirred, (e,e1) 0.5 wt.% sonicated, and (f,f1) 0.5 wt.% stirred.

Furthermore, in Figure 3a–c the optical photographs visually compare the distribution quality of cured CNT/Fe₃O₄/epoxy composites prepared by stirring (left slightly dark samples) and sonication (right dense black samples) showcasing the influence on CNT/Fe₃O₄ distribution at various loadings (0.1 wt.%, 0.3 wt.%, and 0.5 wt.%). The increasing agglomeration in the stirred composites corresponds to poor distribution as compared to the more uniform dispersion in the sonicated composites. This visual narrative focuses on the critical role of production methods in determining the filler dispersion and structural integrity, highlighting the importance of Fe₃O₄ inclusion in enhancing composite performance.

The addition of Fe₃O₄ was intended to improve the dispersion of the CNTs. To further explore this, the sonicated composite samples were analyzed using Scanning Electron Microscopy (SEM) to visualize the dispersion and interaction of the components within the composite. It was observed that the presence of Fe₂O₃ particles facilitated the breaking up of nanotube bundles, promoting a more even distribution within the epoxy matrix [6]. Figure 4 presents the SEM images of the composite materials, divided into three sections, (a), (b), and (c), representing the composites with 0.1 wt.%, 0.3 wt.%, and 0.5 wt.% CNTs, respectively. In Figure 4a, the SEM image of the composite with 0.1 wt.% CNTs is shown. The inset image provides a higher-resolution view, indicating the presence of Fe₃O₄ particles (highlighted by arrows) and their interaction with the CNTs. Due to the lower amount of CNTs, there are insufficient nanotubes to form a network structure, although their presence is evident in the inset image. This interaction suggests that Fe₃O₄ particles achieve a more uniform distribution of CNTs within the epoxy matrix. Moving to Figure 4b, the SEM image corresponds to the composite with 0.3 wt.% CNTs. The inset image reveals a more homogeneous structure with well-dispersed CNTs forming interconnected pathways. The arrows point to the Fe₃O₄ particles, which are crucial in facilitating the improved dispersion of CNTs. The fully dispersed CNTs create significant interconnected pathways, notably enhancing the composite's mechanical, electrical, and thermal properties [8]. Lastly, Figure 4c shows the SEM image of the composite with 0.5 wt.% CNTs. The inset image highlights the dense networks of CNTs, with the arrows indicating the presence of Fe₃O₄ particles. However, at this higher concentration, the dispersion of the CNTs is not as effective, leading to noticeable agglomeration of the

nanotubes. This agglomeration adversely affects the mechanical properties compared to the composite with 0.3 wt.% CNTs.



Figure 3. Optical images of cured CNT/Fe₃O₄/epoxy composites obtained by stirring (left samples) and sonication (right samples) with (a) 0.1 wt.%, (b) 0.3 wt.%, and (c) 0.5 wt.%.

3.2. Tensile Testing

The fabrication method plays a fundamental role in shaping the mechanical properties of composite materials. The dispersion of the fillers contributes to the creation of the network structure; thus, a good dispersion of fillers throughout the polymer media enhances the mechanical properties, whereas a bad dispersion leads to deteriorated properties [13]. Moreover, filler loading also significantly influences the mechanical properties of polymer composites. For example, the elastic modulus, measuring the stiffness of a material, and tensile strength increases with increasing aspect ratios as the reinforcing effect elevates the connected structure. Conversely, the fracture strain decreases with increasing loading due to the reduced flexibility of the composites [14].

Herein, both types of CNT/Fe₃O₄/epoxy composites (stirred/sonicated) were examined by tensile testing to assess the mechanical performance of the composites and evaluate the role of dispersion quality on the tensile properties of these composites. Finally, the results were compared with pure epoxy.

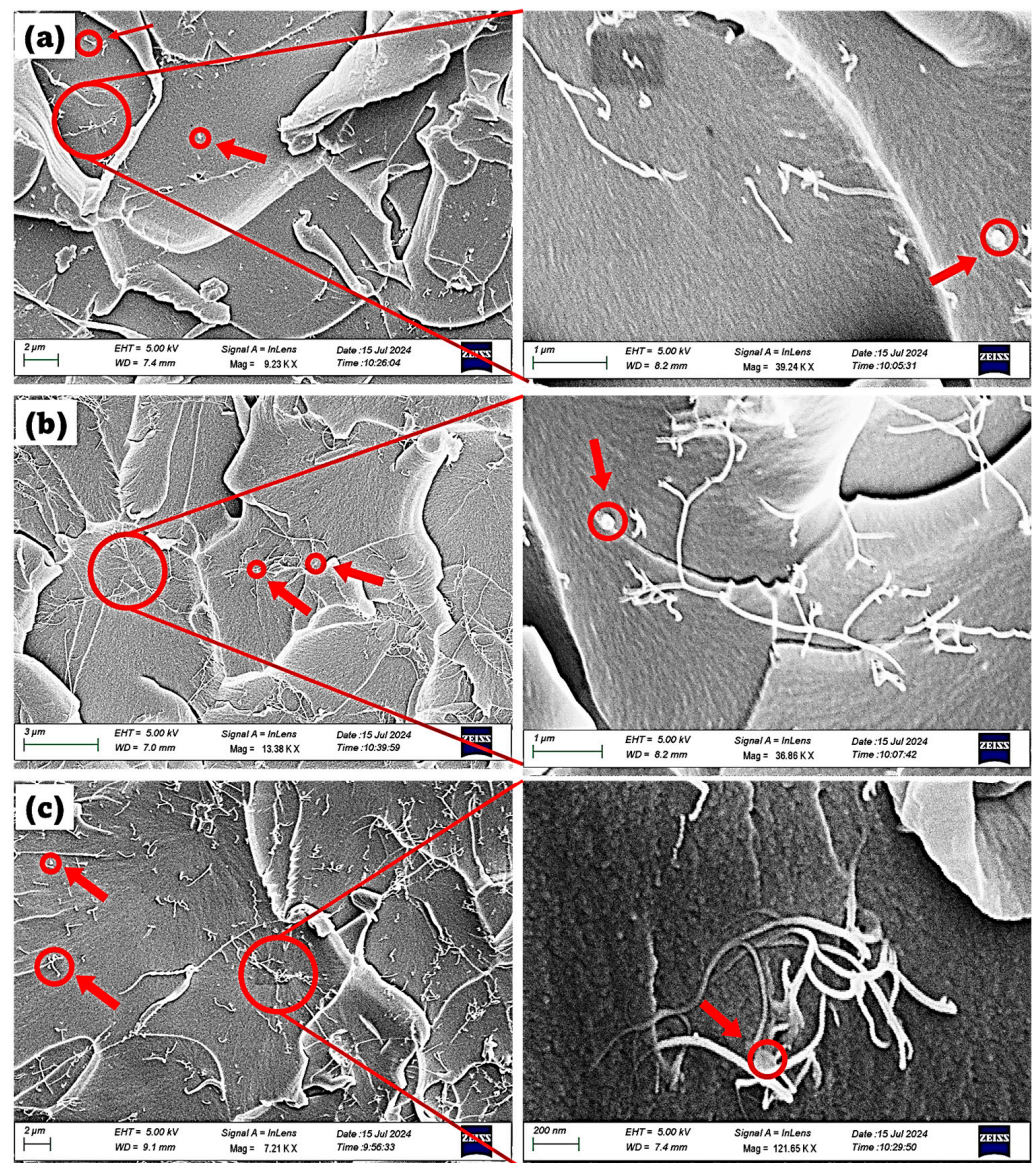


Figure 4. SEM images of sonicated CNT/Fe₃O₄/epoxy composites with (a) 0.1 wt.%, (b) 0.3 wt.%, and (c) 0.5 wt.% showing the dispersion effects. Insets highlight higher-resolution views. Arrows point out the presence and interaction of Fe₃O₄ with CNTs.

Figure 5a presents a comparative illustration of stress–strain curves of epoxy enhanced with varying concentrations of CNT/Fe₃O₄, subjected to two distinct preparation methods: stirring and sonication.

A critical examination of the graph reveals that the integration of CNT/Fe₃O₄ significantly increases the mechanical properties of the epoxy. CNTs and Fe₃O₄ nanoparticles are known to enhance the mechanical properties of epoxy composites due to their high aspect ratio and strong interfacial bonding with the epoxy matrix [6,15]. Notably, at lower concentrations (0.1 wt.%), nanoparticles are more effectively dispersed within the matrix, which leads to a discernible increase in stress resistance compared to pure epoxy. This enhancement becomes more pronounced with increased concentrations, indicating a positive correlation between CNT/Fe₃O₄ concentration and the material's ability to withstand stress. The graph demonstrates that the preparation method is crucial in optimizing these mechanical properties. For each given concentration, samples prepared via sonication exhibit superior performance over their stirred counterparts. This behavior could be attributed to

the more effective dispersion of nanoparticles within the matrix during sonication, leading to enhanced interfacial bonding and load transfer efficiency.

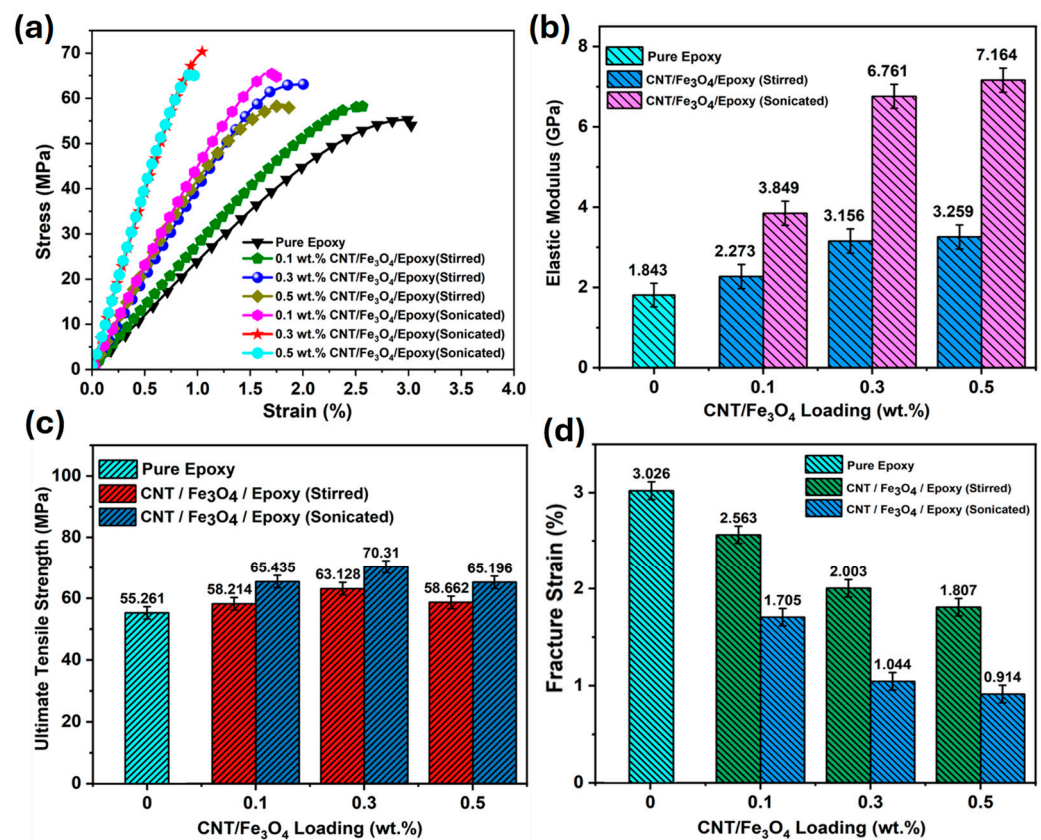


Figure 5. Comparative results for tensile properties of pure epoxy and CNT/Fe₃O₄ epoxy composites prepared by two different stirring and sonication methods at various filler loadings: (a) stress–strain curves, (b) elastic modulus, (c) ultimate tensile strength, and (d) fracture strain.

However, as the wt.% increases, the effect becomes less pronounced, likely due to nanoparticle agglomeration at higher concentrations [16]. When nanoparticles agglomerate, they form clusters that do not interact with the matrix as effectively as well-dispersed individual nanoparticles. This results in a reduction in the effective surface area for stress transfer and diminishes the reinforcing effect of the nanoparticles. Consequently, the increase in stress is less at a higher wt.% compared to a lower wt.%. It is important to ensure the optimal dispersion of nanoparticles within the composite to maximize the enhancement of the mechanical properties.

Figure 5b provides a comprehensive illustration of the elastic modulus of various compositions of epoxy, specifically pure epoxy, CNT/Fe₃O₄/epoxy (stirred), and CNT/Fe₃O₄/epoxy (sonicated), at different loadings of CNT/Fe₃O₄ (wt.%). The graph discloses that the sonicated mixtures exhibit a significant increase in stiffness with higher loadings of CNT/Fe₃O₄. At the 0.5 wt.% loading, the sonicated mixture attains an elastic modulus of 7.164 GPa, markedly higher than pure epoxy's 1.843 GPa at the 0 wt.% loading. Furthermore, the stirred mixtures also exhibit an increase in stiffness (3.259 GPa at 0.5 wt.%), which is consistently lower than their sonicated counterparts at equivalent loadings. This could be attributed to the dispersion of the CNTs in the epoxy matrix as stirring may cause more agglomeration, limiting the effective stress transfer, while sonication disrupts agglomerates and thus sonicated CNT/Fe₃O₄/epoxy composites likely have a larger effective surface area which enhances the interaction between nanoparticles and the matrix, resulting in better dispersion and a higher elastic modulus.

Figure 5c underscores the effectiveness of CNT/Fe₃O₄ incorporation in significantly improving the UTS of epoxy composites and provides a comparison of the ultimate tensile strength (UTS), the maximum stress a material can withstand while being stretched or pulled before breaking, of pure epoxy, stirred CNT/Fe₃O₄/epoxy composites, and sonicated CNT/Fe₃O₄/epoxy, at different weight percentages (wt.%) of CNT/Fe₃O₄ loading.

Notably, at all weight fractions (0.1 wt.%, 0.3 wt.%, and 0.5 wt.%), both composite types show a higher UTS compared to pure epoxy. This enhancement can be attributed to two key mechanisms: improved stress transfer and enhanced packing density. The incorporation of CNTs likely facilitates the formation of a stress transfer network within the epoxy matrix, effectively distributing the applied load and leveraging the superior strength of the CNTs. Furthermore, Fe₃O₄ nanoparticles could enhance interfacial bonding between the CNTs and the epoxy matrix. This improved interfacial adhesion could further contribute to more efficient stress transfer within the composite. This effect is potentially amplified in sonicated composites due to a more uniform CNT dispersion compared to the stirred method, as evidenced by their consistently higher UTS values. Additionally, introducing CNTs and Fe₃O₄ nanoparticles contributes to a denser composite microstructure by effectively occupying voids within the epoxy matrix. This reduction in porosity minimizes stress concentrations and consequently enhances the overall load-bearing capacity of the composite.

However, the observed decrease in the UTS at a loading of 0.5 wt.% for both composite types suggests potential limitations to these mechanisms at higher CNT and Fe₃O₄ concentrations. Excessive nanoparticle loading might lead to agglomeration phenomena, where individual nanotubes and nanoparticles clump together. These agglomerates can act as detrimental stress concentrators, negating the benefits of improved packing density and potentially leading to a decline in the UTS.

Figure 5d provides a comparison graph of the fracture strain (the amount of deformation a material can withstand before breaking) of pure epoxy, stirred CNT/Fe₃O₄/epoxy composites, and sonicated CNT/Fe₃O₄/epoxy composites. The graph shows that pure epoxy has the highest fracture strain, followed by the stirred CNT/Fe₃O₄/epoxy composite, and then the sonicated CNT/Fe₃O₄/epoxy composite. For example, at a 0.1 wt.% loading, pure epoxy has a fracture strain of 3.026%, while the stirred and sonicated composites have a fracture strain of 2.563% and 1.705%, respectively. This trend continues at higher weight percentages as well, i.e., the fracture strain of the composites decreases as the weight percentage of the CNT/Fe₃O₄ filler material increases. For instance, the fracture strain of the stirred CNT/Fe₃O₄/epoxy composite goes from 2.563% at a 0.1 wt.% loading to 1.807% at a 0.5 wt.% loading. Similarly, for the sonicated CNT/Fe₃O₄/epoxy composite, the fracture strain varies from 1.705% at a 0.1 wt.% loading to 0.914% at a 0.5 wt.% loading. This suggests that a higher concentration of CNT/Fe₃O₄ makes the epoxy composite more brittle. This behavior of composites could be due to the following two possible reasons: (1) Stress concentration points, i.e., as the filler content increases, the particles tend to clump together or agglomerate. These agglomerates create weak spots in the composite material. When stress is applied, these areas experience a higher stress concentration compared to the matrix. This localized stress can initiate cracks more easily and propagate faster, leading to brittle failure at a lower strain [17]. (2) Reduced matrix domination, i.e., the epoxy matrix dominates the composite's properties at lower filler concentrations. The epoxy, being more ductile, allows for some deformation before breaking. However, with increasing filler content, the influence of the rigid filler particles becomes more significant by restricting the mobility of the epoxy chains and hindering their ability to absorb stress and deform. The composite becomes stiffer and less able to bend or stretch, leading to a drop in fracture strain and increased brittleness [18–20].

In Figure 6a, the graph compares the % enhancement in the elastic modulus of the CNT/Fe₃O₄/epoxy composites prepared by two different methods, stirring and sonication, compared to pure epoxy. The composites prepared by the sonication method exhibit a

greater elastic modulus enhancement than those prepared by stirring when measured against pure epoxy.

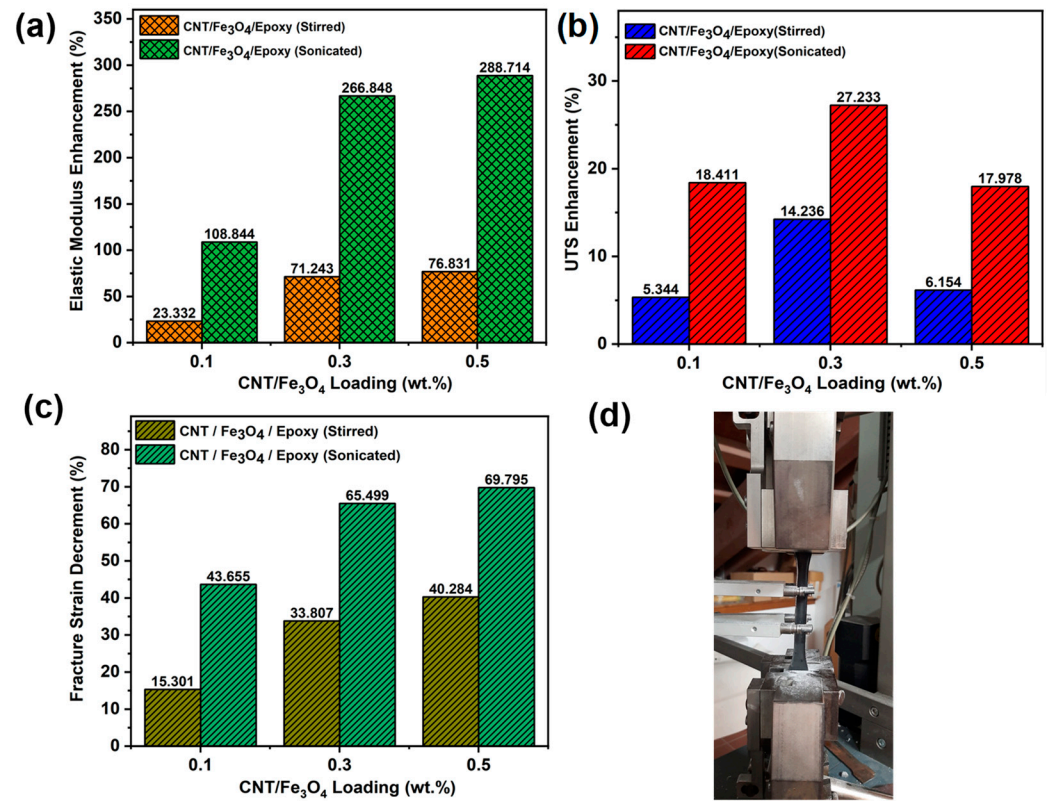


Figure 6. Comparison of CNT/Fe₃O₄ epoxy composites with increasing filler loading to pure epoxy for (a) elastic modulus enhancement (%), (b) ultimate tensile stress (UTS) enhancement (%), (c) fractural strain decrement (%), and (d) tensile testing setup.

The percentage (%) enhancement in the elastic modulus is calculated using the formula:

$$\%Modulus\ enhancement = \frac{E_c - E_e}{E_e} \%$$

where E_c is the modulus of composite and E_e is the modulus of neat epoxy.

The elastic modulus of the nanocomposite is enhanced with increasing CNT/Fe₃O₄ loading. The data for sonicated samples are consistently higher than stirred samples, verifying that better dispersion leads to more effective stress transfer between the CNTs and the epoxy matrix, resulting in a greater enhancement of the elastic modulus. For example, at a 0.5 wt.% loading, the stirred CNT/Fe₃O₄/epoxy composites showed an elastic modulus enhancement of 76.831%, while the sonicated CNT/Fe₃O₄/epoxy composites achieved a 288.714% enhancement as compared to pure epoxy.

On the other hand, Figure 6b compares the % enhancement in the UTS of the composites. At the lowest loading (0.1 wt.%) of CNT/Fe₃O₄, the composites prepared by the stirring method show a modest increase in strength, with a 5.344% enhancement in UTS. In contrast, the sonication method significantly outperforms stirring at the same loading, with an 18.411% enhancement, suggesting that the sonication method is more effective at this concentration for improving the mechanical properties of the composite. As the loading increases to 0.3 wt.%, both methods show improved UTS enhancements. The stirring method's UTS enhancement more than doubled to 14.2%; however, the sonication method still leads with a remarkable 27.2% enhancement, maintaining its superior performance over stirring.

Interestingly, at the highest loading of 0.5 wt.%, there is a decrease in UTS enhancement for both methods. The stirring method drops to 6.1%, and the sonication method decreases to 18.0%. This reduction could be due to agglomeration or poor dispersion at higher loadings, which can negatively affect the mechanical properties. Figure 6c shows the % decrement in fracture strain of the composites compared to pure epoxy. The following formula measures the decrement:

$$\% \text{fracture strain decrement} = \frac{\varepsilon_c - \varepsilon_e}{\varepsilon_e} \%$$

where ε_c and ε_e are the strain to fracture of composite and neat epoxy, respectively.

The stirred method resulted in a 15.3% reduction in fracture strain for the composites with a 0.1 wt.% CNT/Fe₃O₄ loading, while the sonication method led to a much higher reduction of 43.6%. This suggests that at low CNT/Fe₃O₄ loadings, the sonication method significantly affects the ductility of the composite.

The difference between the two methods became more pronounced as the CNT/Fe₃O₄ loading increased to 0.3 wt.%. The stirred composites showed a 33.8% decrement, whereas the sonicated composites exhibited a 65.5% reduction. This occurrence indicates that higher loadings of CNT/Fe₃O₄, when treated with sonication, greatly diminish the material's ability to undergo strain before fracturing. At the highest loading (0.5 wt.%), both methods resulted in similar decrements, with the stirred method at 40.3% and the sonicated method at 69.8%. Overall, across all CNT/Fe₃O₄ loading levels, the sonicated composites consistently demonstrated a higher % decrement in fracture strain than those prepared by stirring. The more uniform dispersion and better bonding of CNT/Fe₃O₄ within the epoxy matrix achieved through sonication might lead to less flexibility and a greater reduction in fracture strain. The graph clearly illustrates the impact of the preparation method and CNT/Fe₃O₄ loading on the mechanical properties of the composites. Figure 6d shows the tensile testing setup with a composite sample clamped and attached with an extensometer.

4. Conclusions

Our study assessed the impact of dispersion quality on the mechanical properties of CNT/Fe₃O₄/epoxy composites, comparing composites fabricated through stirring and sonication at varying filler loadings (0.1, 0.3, and 0.5 wt.%). The critical role of Fe₃O₄ in overcoming the dispersion challenges associated with CNTs highlights its importance as a key enabler for developing high-performance CNT-based epoxy composites. Sonication provided superior dispersion, leading to greater enhancements in the elastic modulus and ultimate tensile strength (UTS) compared to stirring. It was observed that Fe₃O₄ particles play a key role in achieving a homogeneous dispersion of CNTs in epoxy composites. SEM analysis revealed that Fe₃O₄ effectively dispersed CNTs, particularly at 0.3 wt.%, resulting in well-formed interconnected pathways that enhanced the composite's mechanical properties. However, at 0.5 wt.% CNTs, agglomeration occurred, underscoring the need for an optimal concentration. The results showed that at a 0.5 wt.% CNT loading, sonicated composites exhibited a 288.714% increase in the elastic modulus, and a 27.233% increase in the UTS at a 0.3 wt.% loading, while stirred composites showed a 76.831% increase in the elastic modulus at 0.5 wt.% and a 14.236% increase in the UTS at a 0.3 wt.% loading. However, the fracture strain decreased for both composites at a 0.5 wt.% loading. The sonicated composites experienced a pronounced reduction (69.795%) while the stirred composites had a lower reduction (40.284%) compared to pure epoxy. These findings suggest that sonication improves stiffness and strength and can also lead to increased brittleness. Future research should explore strategies to maintain enhanced mechanical properties while mitigating the loss of ductility in these composite materials.

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